



Title	Fiber-microsphere laser with a submicrometer sol-gel silica glass layer codoped with erbium, aluminum, and phosphorus
Author(s)	Takashima, Hideaki; Fujiwara, Hideki; Takeuchi, Shigeki; Sasaki, Keiji; Takahashi, Masahide
Citation	Applied Physics Letters, 90(10), 101103 https://doi.org/10.1063/1.2711384
Issue Date	2007-03-05
Doc URL	http://hdl.handle.net/2115/33857
Rights	Copyright 2007 American Institute of Physics. This article may be downloaded for personal use only. Any other use requires prior permission of the author and the American Institute of Physics.
Type	article
File Information	ApplPhysLett_90_101103.pdf



[Instructions for use](#)

Fiber-microsphere laser with a submicrometer sol-gel silica glass layer codoped with erbium, aluminum, and phosphorus

Hideaki Takashima, Hideki Fujiwara, Shigeki Takeuchi,^{a)} and Keiji Sasaki
Research Institute for Electronic Science, Hokkaido University, Sapporo 060-0812, Japan

Masahide Takahashi
Institute for Chemical Research, Kyoto University, Uji, Kyoto 611-0011, Japan

(Received 6 October 2006; accepted 29 January 2007; published online 5 March 2007)

Lasing of a taper-microsphere system with a gain layer of submicrometer thickness is demonstrated. For the gain layer, an Er³⁺-doped P₂O₅-Al₂O₃-SiO₂ thin film with a thickness of 200 nm was fabricated using the sol-gel method on a silica microsphere. The demonstration of single-mode lasing with the thin gain layer suggests the improved dispersion of Er ions in the P codoped gain layer. © 2007 American Institute of Physics. [DOI: 10.1063/1.2711384]

Microsphere resonators coupled with a tapered optical fiber have been attracting considerable attention recently as ideal cavity systems having ultrahigh Q factors and single spatial mode input-outputs. After pioneering demonstrations of coupling between a silica glass microsphere and a tapered fiber,^{1,2} demonstrations of lasing phenomena have been reported.³⁻⁸ One concern has been to improve the coupling parameter (β) between the spontaneous emission of the gain medium and the cavity mode in the microsphere in order to realize ultralow threshold lasers.⁹ Realization of a submicrometer thick gain layer is essential for achieving better coupling between the gain layer and a cavity mode in the microsphere, since the spatial distribution of the mode is about 1 μm .

The first demonstration of a fiber-microsphere laser was reported by Cai *et al.* using an erbium:ytterbium-codoped phosphate glass microsphere.³ Later, demonstrations using tellurite glass microspheres with doped Er ions⁴ and with doped Tm ions⁵ were also reported. In those experiments, however, the whole spheres were filled with the gain media. Yang and Vahala applied erbium-doped sol-gel films to the surface of a silica microsphere.⁶ The thickness of the gain region was varied from 1 μm to more than 5 μm . They observed continuous wave operation for microspheres having thin gain layers (1 μm), while pulse mode operation was observed in microspheres having thicker ($\geq 5 \mu\text{m}$) gain layers. Recently, Hoi *et al.* adopted erbium-doped sol-gel silica-alumina glasses as the gain layer.⁷ They succeeded in increasing the concentration of Er ions upto 15 000 ppm by aluminum codoping. However, the thickness of the gain medium was still more than 1 μm . In earlier studies,^{6,7} the spheres were repeatedly dipped into the coating sol, heated and irradiated by a laser beam even for gain layers having a 1 μm thickness since the buildup rate was 0.3 $\mu\text{m}/\text{cycle}$.⁶ Recently, an ion implantation technique was used to control the position and thickness of the gain region;⁸ however, an ion accelerator is required for such a process.

In this letter, we report single-mode lasing of a taper-microsphere system having a gain layer of a thickness of a few hundred nanometers using phosphorus codoped sol-gel erbium silica-aluminum glass. Our trial of phosphorus dop-

ing is motivated by a recent electron spin resonance study¹⁰ that reported a striking difference between Al and P codopings for the local structure near Er ions, suggesting a better dispersion of highly doped Er ions with P codoping. We consider that the improved dispersion of Er ions in our P codoped gain layer resulted in the realization of the lasing using a gain layer a few hundred nanometers thick. It should be noted that the gain layer was formed by performing dipping, heating, and irradiating with a laser irradiation once; repetition of the coating processes was not necessary for our thin gain layers.

The coating sol was prepared using ethanol (EtOH), dimethylformamide (DMF), tetraethoxysilane (TEOS), hydrochloric acid (HCl), aluminum chloride (AlCl₃), erbium chloride hexahydrate (ErCl₃·6H₂O), and triethylphosphate (C₂H₅O)₃PO as follows. A mixture of EtOH, DMF, TEOS, and HCl was stirred at 0 °C for 1 h. At this temperature, AlCl₃ dissolved in EtOH was added to the solution, and it was stirred for another 1 h. Then, ErCl₃·6H₂O dissolved in EtOH was mixed in the same way for AlCl₃. After this, distilled water was slowly added to the solution, and it was then stirred at room temperature for 17 h. This solution was used as the coating sol for reference samples that did not have phosphorus. After stirring, (C₂H₅O)₃PO was added to this solution and was stirred for another 12 h. The obtained coating sol was used to form Er³⁺-doped P₂O₅-Al₂O₃-SiO₂ films.

Silica microspheres with stems were formed by melting the edge of a tapered fiber tip using radiation from a CO₂ laser.^{11,12} The diameters of the microspheres used were from about 40 to 200 μm . The microspheres were immersed in the coating sol for 30 s with the stem attached to a motorized linear stage. After withdrawing the microspheres at a speed of 3 cm/min from the solution, the microspheres were kept with their spheres pointing up (i.e., their stems pointing down) to avoid extra accumulation of the coating sol on the surface of the spheres. The microspheres were then annealed in an electrical furnace at 900 °C for 10 min. Then, the microspheres were irradiated with a CO₂ laser for a few tens of seconds to anneal out microcracking in the sol-gel surface.⁶ The molar composition of the fabricated films was SiO₂:Al₂O₃:P₂O₅:Er₂O₃=1:0.12:0.05:0.01, corresponding to a concentration of Er ions of 10⁴ ppm. For reference samples having thicker gain layers, the above-mentioned

^{a)}Electronic mail: takeuchi@es.hokudai.ac.jp

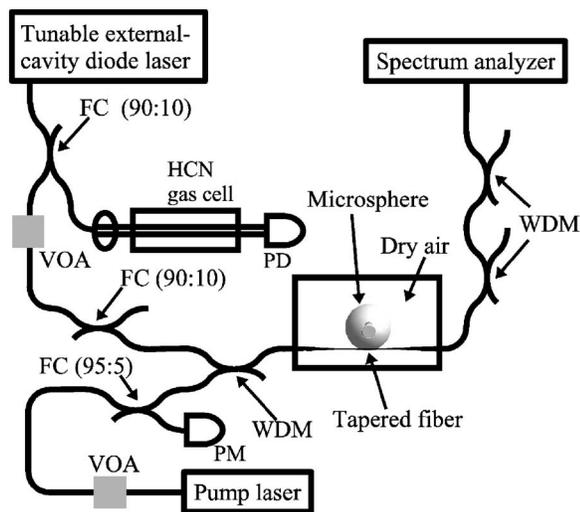


FIG. 1. Schematic of experimental setup. VOA: variable optical attenuator, FC: 95/5 fused fiber coupler, PM: optical power meter, and WDM: 980/1550 nm wavelength division multiplexer.

coating (dipping and annealing) process was repeated for the required number of times, followed by laser irradiation. The thicknesses of the sol-gel films fabricated on the surface oxidized silicon substrate using the same procedures were analyzed using an ellipsometer. The thickness of the sol-gel film having a single coating was estimated to be 200 nm, which is consistent with the measurement result of 570 nm for a film having a triple coating and comparable with a previously reported buildup rate of $\sim 0.3 \mu\text{m}/\text{cycle}$.⁶ We consider that the thicknesses of the films on the silica microsphere are almost same as these values.

A tapered fiber (diameter $2.75 \pm 0.25 \mu\text{m}$) was fabricated by heating a single mode fiber (Thorlabs, 1060XP) with a ceramic heater and stretching the end of the fiber.^{13,14} The transmittance of the fiber was about 90% (when fabricated) and flat for the wavelength region from 1500 to 1620 nm, indicating single mode operation at 1550 nm.¹⁴

The experimental setup for acquiring the lasing spectrum is shown in Fig. 1. A single-mode laser diode operating at 975 nm (Thorlabs, PL980P200) with single-mode optical fiber output was used for pumping. The intensity of the pump light was controlled using a variable optical attenuator. The intensity of the pump power was monitored using an optical power meter (Newport, 2835-C) connected to the output of an unbalanced (95:5) fused fiber coupler. A probe light of a tunable single-frequency diode laser (wavelength 1500 to 1620 nm, Santec TSL-210V) was coupled via a 980/1550 nm wavelength division multiplexer (WDM) in order to check the coupling of the tapered fiber and the microsphere. The output of the WDM was connected to the tapered fiber. The probe light was turned off when during the lasing experiments.

One of the prepared microsphere samples was gently touched to the tapered fiber under microscope observation using a two-dimensional piezomanipulator, which gripped the stem of the sphere. The microsphere and the tapered fiber were in a plastic box, which was filled with dry air.¹³ Unabsorbed pump light (975 nm) was removed from the laser emission light (about 1550 nm) using two 980/1550 nm WDM couplers. Then, the laser emission light was analyzed using an optical spectrum analyzer (Anritsu, MS9001A).

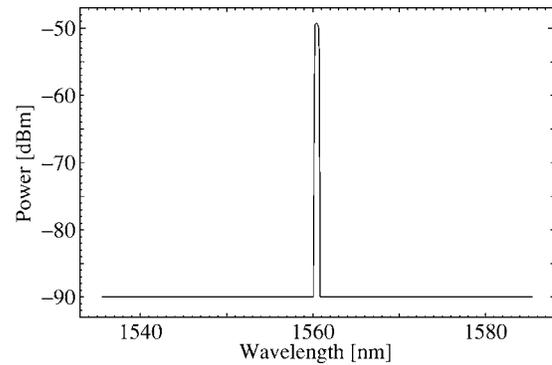


FIG. 2. Single-mode lasing spectrum of a microsphere having a diameter of $38 \pm 1 \mu\text{m}$.

Note that the output powers shown in Figs. 2 and 3 are estimated values at the end of the tapered fiber obtained by compensating for optical losses (19.5%) at the two WDM couplers and the fiber connectors.

Figure 2 shows a single-mode lasing spectrum of a microsphere (diameter of $38 \pm 1 \mu\text{m}$) having a 200-nm-thick P-doped gain layer produced using a single coating process. The pump power launched into the tapered fiber was $670 \mu\text{W}$, which is an estimated value for the input of the tapered fiber after the WDM (Fig. 1) measured using the monitor power meter (PM in Fig. 1). The spectrum shown was measured at a resolution of 0.5 nm. Even when we used a higher resolution of 0.1 nm, the single peak was not discriminated. During the experiment, the microsphere was in contact with the tapered fiber. The wavelength of the pump laser was fine tuned by adjusting the temperature of the laser diode to give the highest output signal. When we adjusted the contact position of the tapered fiber on the microsphere, we observed both single-mode and multimode lasings (not shown), but this behavior was not dependent on the power or the wavelength of the pump laser.

Figure 3 shows the output power versus the pump power launched into the tapered fiber for the same sample and the same conditions as those for Fig. 2. Again, the pump powers are estimated values at the input of the tapered fiber, and the output power is the peak value of the single-mode lasing spectrum (Fig. 2). Above a threshold input pump power, the laser output power increased linearly with the pump power. The threshold was estimated to be about $180 \mu\text{W}$ by fitting the data with a linear function. This threshold value is comparable to that found in a previous study.⁶

We tested another microsphere having a diameter of $60 \mu\text{m}$ and a gain layer having a thickness estimated to be

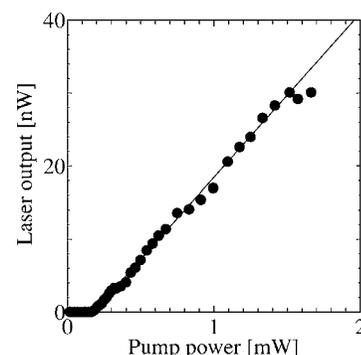


FIG. 3. Laser output power vs pump power at 975 nm.

560 nm, produced using a multiple coating process. We also observed single-mode lasing for low pump powers in this sample. However, for this sample, single-mode lasing became multimode lasing when the pump power was increased.

We also tried to observe lasing in microspheres that did not have P-doped (Er^{3+} -doped $\text{Al}_2\text{O}_3\text{-SiO}_2$) gain layers using several samples (diameters of 45 to 150 μm and gain layer thicknesses estimated to be 70 to 510 nm), including a sample (diameter of 45 μm and gain layer thickness of 280 nm) whose size and gain-layer thickness were similar to the one used to obtain the data shown in Figs. 2 and 3. However, we were not able to observe lasing (neither single mode nor multimode) for those P-undoped samples.

According to a recent electron spin resonance study,¹⁰ there is a striking difference between Al and P codopings for the local structure near Er ions; doped P ions preferentially coordinate to the Er ions to form a “solvent shell structure” which results in a better dispersion of highly doped Er ions. We think the improved dispersion of Er ions in our phosphorus codoped gain layer reduced cross relaxation of Er ions and produced a higher efficiency, resulting in the realization of the lasing in a gain layer which is a few hundred nanometers thick.

In conclusion, we have reported single-mode lasing of a taper-microsphere system having a gain layer of a few hundred nanometers thick using P codoped sol-gel Er silica-aluminum glass. We consider that the improved dispersion of Er ions in our P codoped gain layer resulted in the demonstration of the lasing in a gain layer, which was a few hundred nanometers thick. In our experiment, the gain layer was formed using a single coating process.

Numerical calculation based on the Mie scattering theory suggests that the intensity of the resonant-mode field near the surface, where the gain layer of the current sample exists, is just 20% of the highest intensity at about 600–800 nm inside from the surface. Thus, the coupling between the gain layer and the resonant mode will be much improved when the gain layer is adjusted to this position.

The control of the gain layer position may be possible by overcoating of undoped sol-gel silica glass on the precoated gain layer. Although it is difficult to estimate the laser performance precisely, the β parameter will be much improved resulting in the significant decrease of the laser threshold.

The authors would like to thank Eun-Seok Kang and Hidenori Konishi for technical assistance. The current work was partly supported by the program “R&D support scheme for funding selected IT proposals” of the Ministry of Public Management, Home Affairs, Posts and Telecommunications, a Grant-in-Aid from the Japan Society for the Promotion of Science, the 21st Century COE program, CREST-Project and PRESTO-Project of the Japan Science and Technology Agency.

¹J. C. Knight, G. Cheung, F. Jacques, and T. A. Birks, *Opt. Lett.* **22**, 1129 (1997).

²M. Cai and K. Vahala, *Opt. Lett.* **25**, 260 (2000).

³M. Cai, O. Painter, and K. J. Vahala, *Opt. Lett.* **25**, 1430 (2000).

⁴X. Peng, F. Song, S. Jiang, N. Peyghambarian, M. Kuwata-Gonokami, and L. Xu, *Appl. Phys. Lett.* **82**, 1497 (2003).

⁵K. Sasagawa, Z. Yonezawa, R. Iwai, J. Ohta, and M. Nunoshita, *Appl. Phys. Lett.* **85**, 4325 (2004).

⁶L. Yang and K. J. Vahala, *Opt. Lett.* **28**, 592 (2003).

⁷P. V. Hoi, C. T. T. Ha, and H. Q. Hung, *Appl. Phys. Lett.* **87**, 161110 (2005).

⁸J. Kalkman, A. Polman, T. J. Kippenberg, K. J. Vahala and M. L. Brongersma, *Nucl. Instrum. Methods Phys. Res. B* **242**, 182 (2006).

⁹R. J. Horowicz, H. Heitmann, Y. Kadota, and Y. Yamamoto, *Appl. Phys. Lett.* **61**, 393 (1992).

¹⁰A. Saitoh, S. Matsuishi, C. Se-Weon, J. Nishii, M. Oto, M. Hirano, and H. Hosono, *J. Phys. Chem. B* **110**, 7617 (2006).

¹¹A. Chiba, H. Fujiwara, J. Hotta, S. Takeuchi, and K. Sasaki, *Jpn. J. Appl. Phys., Part 1* **28**, 6138 (2004).

¹²H. Takashima, H. Fujiwara, J. Hotta, S. Takeuchi, K. Sasaki, S. Murakami, T. Torimoto, and B. Ohtani, *Jpn. J. Appl. Phys., Part 1* **45**, 6917 (2006).

¹³A. Chiba, H. Fujiwara, J. Hotta, S. Takeuchi, and K. Sasaki, *Appl. Phys. Lett.* **86**, 261106 (2005).

¹⁴H. Konishi, H. Fujiwara, S. Takeuchi, and K. Sasaki, *Appl. Phys. Lett.* **89**, 121107 (2006).