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Enhanced Photocatalytic Activity by Particle Morphology—Preparation, Characterization and Photocatalytic Activities of Octahedral Anatase Titania Particles

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Titanium(IV) oxide (titania) has been used extensively for many years as a photocatalyst, mainly owing to its redox ability, stability and nonstoichiometry.1–2 Though hundreds of papers on titania photocatalysis have been published, the essence of its photocatalytic activity, e.g., the reason of various levels of photocatalytic activity of samples with various properties and polymorphic composition, has not been clarified.3 Recently, morphology of crystallites, widely studied for reactions occurring on a titania single crystal (mainly rutile (110))4 but having been neglected for particulate photocatalysts, has been considered as one of the crucial factors of photocatalytic activity of particulate titania on the basis of the results of studies using faceted octahedral and decahedral anatase particles. We have recently developed an ultrasonication (US)–hydrothermal reaction (HT) process5 to prepare octahedral anatase particles (OAPs), exposing eight equivalent (101) facets that are thermodynamically the most stable, with enhanced photocatalytic activity, especially for oxidation of acetic acid in aerated aqueous solutions, compared with many commercial titania samples with similar properties. However, proof of higher activity only owing to OAP morphology has not yet been obtained, since particles with the same structural properties except for morphology could not be prepared. In order to prove the effect of particle morphology on photocatalytic activity, we examined various preparation conditions. In this paper, we represent the results for preparation of samples with different OAP contents and with other structural properties remaining almost constant and we show a comparison of their photocatalytic activities.

Samples were fabricated by the US–HT process (for details, see SI) using partially proton-exchanged potassium titanate nanowires (TNW; Earthclean Tohoku Co. Ltd.) prepared by HT reaction of Evonik P25 (Nippon Aerosil) in a potassium hydroxide solution (17 mol L−1) at 383 K for 20 h followed by washing with dilute hydrochloric acid.6,7 Controlled parameters were US time (0–4 h), HT reaction time (3–48 h) and HT temperature (413–473 K). The products were collected by centrifugation and vacuum drying (353 K, 12 h).

Specific surface area of the products was estimated by nitrogen adsorption at 77 K using the Brunauer-Emmett-Teller (BET) equation. Crystallinity (anatase content) as well as crystallite size and morphology were measured by X-ray diffractometry (XRD; Rigaku SmartLab) and scanning electron microscopy (SEM; JEOL JSM-7400F), respectively. Particles in a sample were classified into three groups based on results of SEM analysis as shown in Fig. 1: (a) OAP: an octahedral particle without observable defects, (b) semi-OAP: an octahedral particle with a defect (defects) and (c) others: an irregular shaped non-octahedral particle. Composition of these particles in each sample was measured by counting at least 200 particles in several SEM images of a sample. The chemical formula and atomic composition of TNW and the products were examined by thermogravimetry (TG; ULVAC-RIKO TGD-9700), X-ray photoelectron spectroscopy (XPS; JEOL JPC-9010MC) and SEM with energy dispersive X-ray spectroscopy (SEM/EDS; JEOL JSM-6360LA).

An XRD pattern of TNW resembled a pattern of potassium titanate K2TiO3, and the composition was estimated to be H1.32K0.64Ti5O17·4H2O (see SI). Based on SEM observation, it was found that the wire structure of TNW remained in the product after the US–HT process (see SI), and thus the composition of the final product, a mixture of hydrated titanate with partially proton-exchanged potassium titanate, was estimated to be, e.g., TiO2·0.026(H0.83K0.17Ti5O17)·0.24H2O for the sample prepared by 1-h US and 6-h HT at 433 K. As expected, water content and molar ratio of remaining TNW decreased with increases in HT time and temperature. Thus, the US–HT process produced anatase crystallites with a small amount of titanate.

Influence of HT time on structure of the products was examined with other parameters kept constant, i.e., 1-h US and 433-K HT temperature, as shown in Fig. 2. With increase in HT time, titane wires were converted into anatase crystallites, resulting in crystallinity increase, and specific surface area, water content and K/Ti (data not shown) were decreased, while HT time longer than 9 h did not change the structural properties. This indicates that most of the TNW...
was converted into anatase crystallites at > 9 h. Microscopic observations confirmed the presence of TNW for a short HT time (3 h) and almost complete disappearance of TNW for a longer reaction time.

HT time-dependent photocatalytic activities in the systems (for details, see SI) of mineralization of acetic acid in an aerated aqueous solution (CO$_2$ system) and dehydrogenation of methanol in a deaerated aqueous solution with an in-situ platinized (2wt%) photocatalyst (H$_2$ system) are also shown in Fig. 2. Although the photocatalytic activity in the H$_2$ system relative to that of the standard (FP-6) was low (< 40%), being consistent with our previous results, its HT time-dependence was similar to that in the CO$_2$ system, which was higher than that of the standard. The activities reached maximum values (ca. 40% and 150% for H$_2$ and CO$_2$ systems, respectively) at 6-h HT time. In the conventional understanding, balance between high crystallinity and sufficient specific surface area might be responsible for the highest photocatalytic activity. It has been reported that sufficient specific surface area resulting in a large amount of adsorbents increases the rate of capture of electrons and positive holes by them, while high crystallinity induces a smaller amount of defects and thus reduces recombination of electrons and positive holes, resulting in enhancement of the photocatalytic activity. However, it should be noted that the sample with the highest level of photocatalytic activity also possesses the highest content of OAP and semi-OAP, as shown in Fig. 2. Thus, the dependence of photocatalytic activity on structural properties seems to be complex, and the trend could not be interpreted only by crystallinity or specific surface area.

The influence of HT temperature on structural properties and thus on photocatalytic activity was also investigated with US time and HT time kept at 1 h and 6 h, respectively. The data obtained (not shown) are similar to the results for the above-mentioned influence of HT time, i.e., at a lower temperature (413 K), the presence of TNW was observed, and an increase in HT temperature above 433 K (similar to the increase in HT time) resulted in their disappearance. With an increase in HT temperature, specific surface area, water content and K/Ti of the products decreased, while crystallinity increased. As a result, HT temperature influenced photocatalytic activities in the CO$_2$ and H$_2$ systems. The photocatalytic activity in the H$_2$ system was less sensitive to HT temperature than was that in the CO$_2$ system. The optimum photocatalytic activities in the CO$_2$ system reached 150% for the sample prepared at 433 K. The dependence of photocatalytic activity on the properties could also not be interpreted by crystallinity or specific surface area alone, and it could be concluded that the balance between them resulted in the higher photocatalytic activity. This hypothesis is supported by observation that the highest content of OAP and semi-OAP (66%), for the sample prepared at 453 K, correlated with the highest crystallinity (88%) but not with the highest level of photocatalytic activity (at 433 K), and thus not OAP content but balance between crystallinity and specific surface area could be crucial. This conclusion contradicts the previous one, for the influence of HT time, showing that morphology can be responsible for photocatalytic activity. However, as has already been mentioned, to find the real dependence and key factor of photocatalytic activity, a clear relation between only two parameters should be drawn.

As has been shown before, change in the preparation conditions of samples (HT time and temperature) caused change in their photocatalytic activity, morphology, and structural properties (specific surface area, crystallinity and crystallite size). In order to study the correlation between OAP morphology and photocatalytic activity, the structural properties should be kept the same (at least almost constant). It has been found that control of US time results in production of such samples possessing almost the same structural properties and different OAP contents.

US influenced the morphology of the precursor TNW; the length of TNW was shortened by 1-h US (a standard condition) as shown in Fig. 3, indicating that US could also influence the morphology of the final product. Indeed, the
The correlation between photocatalytic activities in the H\textsubscript{2} system is still unknown. It should be noted that the H\textsubscript{2} system requires noble-metal deposition, and it has been reported that reductive deposition as in the case of in-situ platinum deposition of a noble metal occurred preferentially on [101] facets. Further studies on mechanisms of specified redox reactions on OAPs are presently under way.

In conclusion, samples of almost the same structural properties with various OAP (and semi-OAP) contents have been successfully prepared through the US–HT process with various US times and constant HT time and temperature. Photocatalytic activities of these samples suggested that OAP morphology enhanced the activity, especially for the photocatalytic reactions involving oxygen uptake.

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References

Graphical Abstract

Textual Information

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