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Mismatched changes of the photoluminescence and crystalline structure of a mechanochromic gold(I) isocyanide complex

Tomohiro Seki, a Kenta Sakurada a and Hajime Ito a*

Mechanochromic compound 1 forms green-emitting crystals (1g) and blue-emitting crystals (1b) that exhibit two intriguing responses to mechanical stimulation. Upon mechanical stimulation, the emission color of 1g does not change but its crystalline structure does. Conversely, 1b shows a clear emission shift, but it retains its molecular arrangement upon grinding.

Studies of luminescent mechanochromic compounds based on organic and organometallic compounds are currently considered important because of their wide possible applicability in sensing and recording devices. In these systems, mechanical force, such as grinding, pressing, shearing, and ball milling, alters molecular arrangements in the solid phase and thus changes intermolecular interaction patterns. As a result, solid-state photoluminescence properties can be switched by mechanical stimulation. Because emission is strongly correlated with solid-state structure, most mechanochromic compounds show changes of both emission behavior and crystalline structure upon mechanical stimulation. In particular cases, mechano-induced emission color changes occur because of minor alterations in crystalline structure in which the overall molecular arrangements of the materials are retained. However, there are few examples of such mismatches in the magnitude of changes between emission profiles and crystalline structures upon grinding.

We previously reported a mechano-triggered single-crystal-to-single-crystal (SCSC) phase transition with a corresponding change in the emission profile of gold(I) isocyanide complex 1, a unique class of mechanochromism. Weak mechanical stimulation on a local area of green-emitting single crystal 1g readily induces phase transition of the local area. This local phase change gradually occurs over the entire crystal spontaneously. As a result, the whole 1g crystal is transformed into weak blue-emitting single crystal 1b in an SCSC manner. This is the first example of a mechanically triggered SCSC phase transition. By taking advantage of the SCSC phase transition of 1, we determined the detailed molecular arrangements of 1g and 1b, and carefully examined the relationship between crystalline structures and emission properties of these materials. However, we have not reported the strong mechanical stimulation-induced phase transition behavior of 1 with a corresponding emission color change, which is strikingly different from the previously reported behavior.

Herein, we present the unusual mechanochromism of 1g and 1b, where the magnitudes of the changes of emission profiles induced by mechanical stimulation are mismatched with the changes in the crystalline structures (Fig. 1). Rapidly crystallized green-emitting 1g (Fig. 1b) does not show an emission color change upon strong mechanical stimulation (Fig. 1c). Despite the absence of an emission color change of 1g induced by mechanical stimulation, powder X-ray diffraction (PXRD) studies indicate that the crystalline structure is completely changed to the other crystalline phase. In contrast, slowly crystallized blue-emitting 1b (Fig. 1b) exhibits a clear emission color change to green upon grinding (Fig. 1d). Despite this mechanochromism of 1b, PXRD analyses reveal that no crystalline structure change is induced by mechanical stimulation. Detailed investigation indicates that these two mismatches in the magnitude of changes of the photoluminescence and crystalline structures of 1g and 1b occur because of energy transfer within ground powder 1g ground and 1b ground, respectively.
Complex 1 shows an intriguing emission color change upon strong mechanical stimulation. Rapid and slow crystallization of 1 from dichloromethane/hexane afforded green- and blue-emitting crystals 1g and 1b, respectively (Fig. 1b). Single-crystal structure analyses indicated that crystallized 1g and 1b are identical to the previously reported single crystals of 1, between which a mechano-triggered SCSC phase transition occurred.4,8 Strong grinding of a 1g crystal with a pestle did not induce a prominent emission color change (Fig. 1c). In contrast, strong grinding of 1b induced a substantial emission color change to green (Fig. 1d). When 1g and 1b were strongly stimulated by ball milling for 30 min at 4600 rpm, the same emission changes were also observed. The resulting ground phases 1g_{ground} and 1b_{ground} exhibited identical emission spectra (Figs. 2a and b and S1 and S2). Remarkably, the changes in the emission profiles of 1 upon strong grinding are completely opposite to that observed for the previously reported mechano-triggered SCSC phase transition from 1g to 1b, in which emission color changed from green to blue.3 In our previous work, a small mechanical stimulus was applied at a local area of a 1g crystal and thus the resulting 1b was also a crystal.5 Meanwhile, in this study, strong mechanical stimulation was applied to the crystals, so the powder forms 1g_{ground} and 1b_{ground} were obtained.

We measured the optical properties of 1g and 1b. Green-emitting 1g showed a broad emission spectrum peaking at 535 nm upon excitation at 370 nm (solid green line in Fig. 2a). The excitation spectrum of 1g exhibited a broad band extending to around 450 nm (solid green line in Fig. 2c). The absolute emission quantum yield (Φ_{em}) and average lifetime τ_{av} = (ΣA_iτ_i)/(ΣA_i) of 1g were 84% and 0.97 μs, respectively (Fig. S3 and Table S1). Meanwhile, blue-emitting 1b showed a structured emission band with maxima at 422, 457, and 489 nm (solid blue line in Fig. 2d).9 The excitation maximum of 1b (λ_{ex} = 445 nm) was located at 312 nm (solid blue line in Fig. 2d). Both emission and excitation spectra of 1b appeared shorter wavelength than those of 1g, and 1b showed τ_{av} of 12.9 μs and Φ_{em} of 33% (Table S1).

We next investigated the optical properties of 1g_{ground} and 1b_{ground} and compared them with those of 1g and 1b, respectively. The emission spectra of 1g_{ground} and 1b_{ground} are identical to each other, consisting of structureless emission bands with maxima at 539 nm (Figs. 2a and b and S1). These maxima are almost identical to that of 1g (535 nm), indicating that the 1g crystal did not show mechanochromism. Despite the absence of emission spectral changes when 1g was transformed to 1g_{ground} (Fig. 2a), their excitation spectra are slightly changed (Fig. 2c), which may indicate that their photoluminescence processes are different. In contrast, emission bands exhibited a red shift of 78 nm upon the phase transition from 1b to 1b_{ground} (Fig. 2d), indicating that 1b is a mechanochromic material. Despite the complete shift in emission when 1b was transformed to 1b_{ground} upon grinding, 1b_{ground} has a remaining excitation shoulder at 313 nm which is corresponding to the excitation maximum of 1b (Fig. 2d).10 Not only the excitation and emission spectra but also other photophysical properties are changed when 1g → 1g_{ground} and 1b → 1b_{ground} transitions occur. For example, the τ_{av} values of 1g_{ground} and 1b_{ground} are around 0.8 μs (Fig. S3 and Table S1), which are similar to that of 1g (0.970 μs) and much shorter than that of 1b (12.9 μs). In addition, Φ_{em} of 1g_{ground} and 1b_{ground} are around 20%, smaller than those of 1g and 1b (Table S1).

Interestingly, PXRD analyses indicated that mechano-induced change profiles of crystalline structures of 1 are mismatched from those of photoluminescence. PXRD patterns of 1g_{ground} and 1b_{ground} are identical to each other and showed intense diffractions (lower lines in Fig. 3a and b).11 The upper lines in Fig. 3a and b are simulated powder patterns of 1g and 1b, respectively, based on the single crystalline structures.4 The PXRD pattern of 1b_{ground} is similar to the simulated pattern of 1b (Fig. 3b). This indicates that molecular arrangements of 1b and 1b_{ground} are similar, which is further supported by their UV/vis absorption spectra (Fig. S4). This suggests that the crystalline structure of 1b is almost unchanged upon grinding (Fig. 3b) despite its marked change in emission properties (Fig. 1d). Conversely, mechanical grinding of 1g induced dramatic crystalline structure changes into a phase similar to that of 1b (Fig. 3a), even though emission color barely changed (Fig. 1d).
The green emission of 1g and 1b is suggested to occur via energy transfer from blue- to green-emitting crystalline domains within the powder (Fig. 4, middle).12 PXRD analyses indicate that the molecular arrangements of both 1g and 1b are similar to that of blue-emitting 1b (Fig. 3); however, both 1g and 1b exhibited green emission similar to that of 1g (Fig. 1). This indicates that both 1g and 1b must contain small amounts of green-emitting crystalline domain 1g within the main 1b crystalline domain (Fig. 4, middle). It is believed that this minor 1g domain within 1g or 1b is responsible for the overall green emission followed by energy transfer from the main 1b crystalline domain. The dominant emission of minor crystalline domains of mechanochromic compounds has been reported previously.3,4 For example, Zhang reported that mechanical stimulation to a mechanochromic compound produces small amount of the new crystalline domain which is responsible for the photoluminescence of the resulting powder because of the energy transfer from the remaining original crystalline domain.3b In the present study, the following several experimental findings indicate that this process occurs in 1. For example, the emission decay profiles of 1g and 1g and 1b are quite similar (Fig. S3), indicating that their excited-state species involved in photoluminescence are identical. The lower Q of 1g and 1b than those of 1g and 1b are caused by quenching of the excited state through energy transfer (Table S1).13

On the one hand, for the 1g → 1g phase transition induced by strong grinding, the emission color is unchanged [Fig. 4, (i)]. On the other hand, emission color is changed to blue as a result of the previously reported 1g → 1b SCSC phase transition by weak mechanical stimulation;5 this difference is likely because the 1g phase exhibits more efficient energy transfer than SCSC-produced 1b. Despite the different emission properties of the 1g powder (green emission) and single crystal 1b obtained by the SCSC phase transition (blue emission), their crystalline structures are identical to each other. As previously reported, even a blue-emitting single crystal 1b prepared by the mechnano-triggered SCSC phase transition of 1g has a residual 1b domain that participates in energy transfer.4 Thus, the emission spectrum of the resulting single crystal 1b exhibited a residual green emission band as well as a newly emerged blue emission band.4 The 1g powder obtained by strong grinding has a less ordered molecular arrangement and smaller particle size than single crystal 1b,4 which may facilitate energy transfer in 1g and thus no residual blue emission band from the main crystalline domain was observed for 1g (Fig. 2a).

The 1b → 1b phase transition with a blue-to-green emission color change induced by strong grinding [Fig. 4, (ii)] is surprising because X-ray diffraction studies revealed there was no marked change in crystalline structure (Fig. 3b).5 We believe this shift of emission is caused by the formation of a small amount of a kinetically trapped 1g crystalline domain within 1b when 1b phase is mechanically stimulated. One possible evidence for this “reverse mechanochromism” in a local area in 1b: however, the appearance of an excitation maximum at 365 nm for 1b obtained by grinding 1b light green line in Fig. 2d) may support the emergence of a small 1g domain within 1b.

In summary, we reported two different mismatched changes in the structure and emission profiles of mechanochromic complex 1, in which 1g exhibited no change of emission color but a prominent structural change upon mechanical stimulation, whereas 1b showed a clear change of emission color but a similar overall crystalline structure. We proposed that these complicated changes of emission and crystalline structures upon mechanical stimulation are caused by energy transfer within ground powders 1g and 1b. Although mechanochromism with retention of the overall crystalline structure has been observed,3 a change in the crystalline structure of the same compound without alteration of emission properties upon grinding is unprecedented.

Notes and references


7 Because the isolated state of complex 1 in CHCl3 solution exhibits photoluminescence properties ($\lambda_{\text{em,max}} = 430$ nm and $\phi_{\text{em}} = 3\%$) different from those in the solid phase 1g and 1b, it is obvious that emission properties of 1 are affected by the intermolecular interactions.

8 See reference 4 for a detailed discussion on the crystalline structures and their relationship with emission. Briefly, single crystalline structure of 1g contains aurophilic interactions whereas that of 1b does not contain aurophilic interactions. In general, the formation of aurophilic interaction resulted in the red-shifted emission band. Therefore, longer wavelength emission of 1g ($\lambda_{\text{em,max}} = 535$ nm) than that of 1b ($\lambda_{\text{em,max}} = 445$ nm) is caused by the formation of the aurophilic interaction in the single crystal structure of 1g.

9 These bands are almost consistent with the emission bands of mechanically obtained single crystal 1b in a shorter wavelength region; see reference 4 for more detail.

10 Excitation spectrum of 1gground also showed excitation shoulder at 313 nm similar to 1bground.


13 The spectral overlap of the emission spectrum of the energy donor (1b domain) and absorption spectrum of the energy acceptor (1g domain) also supports the occurrence of the energy transfer (Fig. S5).