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Citation: Journal of Applied Physics 122, 135305 (2017); View online: https://doi.org/10.1063/1.4997813
View Table of Contents: http://aip.scitation.org/toc/jap/122/13
Published by the American Institute of Physics

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Solid-phase epitaxial film growth and optical properties of a ferroelectric oxide, Sr$_2$Nb$_2$O$_7$

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(Received 27 July 2017; accepted 19 September 2017; published online 4 October 2017)

I. INTRODUCTION

Epitaxial films with highly ordered atomic arrangements and atomically flat surfaces are extremely important for experiments that aim to extract the fundamental physical properties of functional oxides, in particular, their optical and carrier transport properties, and the photochemical reactivity of their surfaces. Generally, vapor phase epitaxy (VPE) methods, such as pulsed laser deposition (PLD) and molecular beam epitaxy (MBE), are used to fabricate high-quality epitaxial oxide films at high temperatures (~1000 °C). In addition to these VPE methods, the solid phase epitaxy (SPE) method, which involves high temperature crystallization of an amorphous precursor film, is particularly useful for the epitaxial growth of complex oxides with layered crystal structures. Since the melting point of complex oxides usually exceeds 2000 °C, a heating temperature of $T > 1000$ °C, which is ~60% of the melting point, is required to achieve Frank–van der Merwe growth. When using SPE techniques, the chemical composition has proved to be very difficult to control if the valence states of the Nb ions vary from structure to structure, with each containing different ratios of Nb$^{5+}/$Nb$^{4+}$; although the $n=\infty$ phase SrNbO$_3$ (space group $Pm\overline{3}m$, $a = 4.023$ Å) crystallizes with a simple perovskite structure [Fig. 1(c)], the $n=4$ phase Sr$_2$Nb$_2$O$_7$ (space group $Cm\overline{2}1$, $a = 3.933$ Å, $b = 26.726$ Å, and $c = 5.683$ Å) and the $n=5$ phase SrNbO$_{3.4}$ (space group $Pnmm$, $a = 3.995$ Å, $b = 56.740$ Å, and $c = 32.456$ Å) form layered structures composed of distorted (110) SrNbO$_3$ slabs in the $n=4$ phase, four distorted SrNbO$_3$ slabs are stacked along the [010] direction [Fig. 1(a)]; for the $n=5$ phase, five distorted SrNbO$_3$ slabs are stacked along the [001], as shown in [Fig. 1(b)]. The number of SrNbO$_3$ slabs in the layered Sr$_n$NbO$_{3n+1}$ structures depends on the degree of oxidation—the valence states of the Nb ions vary from structure to structure, with each containing different ratios of Nb$^{5+}/$Nb$^{4+}$ (Nb$^{4+}$)).

At the time of writing, there are only two existing reports on the epitaxial growth of Sr$_2$Nb$_2$O$_7$ films: in 1976, Ishitani and Kimura fabricated such films on single crystal (010) Sr$_2$Ta$_2$O$_7$ substrates using an RF sputtering technique, while in 2008, Balasubramaniam et al. reported that (010)-oriented epitaxial Sr$_2$Nb$_2$O$_7$ films could be grown on (110) SrTiO$_3$ by PLD. Therefore, we first tried to fabricate epitaxial films of Sr$_2$Nb$_2$O$_7$ on single crystal (110) LaAlO$_3$ substrates by PLD at high temperatures (~900 °C) under a range of different oxygen partial pressures ($10^{-4}$–10$^1$ Pa).
However, we were not able to successfully deposit the desired epitaxial films (data not shown), and the films synthesized using these conditions were composed of two phases, \( \text{Sr}_2\text{Nb}_2\text{O}_7 \) and \( \text{SrNbO}_3 \). The epitaxial relationships between the films and the substrate were \( \text{Sr}_2\text{Nb}_2\text{O}_7 \) \([110]\)[001] \( \LaAlO_3 \) and \( \text{SrNbO}_3 \) \([110]\)[001] \( \LaAlO_3 \), respectively. The film surfaces were composed of irregularly shaped grains with diameters of \(<100\text{nm} \), as observed by atomic force microscopy (AFM), similar to those reported by Balasubramaniam \textit{et al.}\textsuperscript{20}

Rather than continuing with this technique, we opted to seek an alternative synthetic route, deciding to proceed using SPE. Here, amorphous \( \text{Sr–Nb–O} \) films, deposited by PLD at room temperature, were heated at high temperatures. We expected that the most stable phase, \( \text{Sr}_2\text{Nb}_2\text{O}_7 \), would crystallize directly during the heat treatment, without forming the \( \text{SrNbO}_3 \) phase. From the results of this experiment, we confirm that high-quality epitaxial films of \( \text{Sr}_2\text{Nb}_2\text{O}_7 \) can be successfully grown by SPE, possessing both the highly ordered atomic arrangements and flat surfaces. The refractive index of the resultant film was 2.1, indicating that the dielectric permittivity of the film was in between 20 and 80, which corresponds well with that of the single crystals.

II. EXPERIMENTAL

\( \text{Sr}_2\text{Nb}_2\text{O}_7 \) films were fabricated by the SPE technique on a (110)-oriented \( \LaAlO_3 \) single crystal substrate. First, amorphous \( \text{Sr–Nb–O} \) films were deposited by PLD at room temperature, using a KrF excimer laser \( (\lambda = 248\text{nm}, 20\text{ns pulse width, Coherent COMPex Pro 102}) \) to ablate the dense, ceramic target disk consisting of \( \text{Sr}_2\text{Nb}_2\text{O}_7 \). The laser fluence, repetition rate, and deposition rate were \( \sim 1\text{J cm}^{-2} \text{ pulse}^{-1}, 1\text{Hz}, \text{and} \sim 17\text{pm pulse}^{-1} \), respectively. Then, the deposited films were heated at 1400\textdegree C in a tube furnace, which was evacuated during heating by a rotary pump.

The crystallographic orientations of the resultant films were analyzed by high-resolution X-ray diffraction (XRD, ATX-G, Rigaku), using monochromated Cu K\( \alpha_1 \) radiation \( (\lambda = 1.54059 \text{Å}) \). The surface morphology of the resultant films was observed by AFM (Nanocute, Hitachi High-Tech). Analysis of the cross-sectional microstructure and atomic arrangement of the films was conducted by transmission electron microscopy (TEM) and scanning transmission electron microscopy (STEM). Thin-foil specimens for TEM and STEM imaging were prepared using a standard Ar ion-beam thinning process, with an accelerating gun voltage in the range of 1–4 kV. Bright-field TEM images were taken at 200 kV using a JEM-2010F (JEOL) microscope, and high-angle annular dark-field (HAADF) images were obtained using a 200 kV STEM (ARM200FC, JEOL) equipped with a probe corrector (CEOS Gmbh).

The optical transmission and reflection spectra were measured using a UV-VIS-NIR spectrometer (SolidSpec-3700, Shimadzu) at room temperature.

III. RESULTS AND DISCUSSION

Figure 2 summarizes XRD patterns of these resultant films grown by the SPE method. Only intense diffraction peaks of \((0k0) \) \((k = \text{even}) \text{Sr}_2\text{Nb}_2\text{O}_7 \) were seen in the out-of-plane Bragg diffraction pattern [Fig. 2(a)], together with the 110 peaks from the \( \LaAlO_3 \) substrate. The average crystal tilting of \( \text{Sr}_2\text{Nb}_2\text{O}_7 \) was \( \sim 0.1^\circ \), which was evaluated by measuring the out-of-plane X-ray rocking curve of the (080) diffraction peak, as shown in the inset of Fig. 2(a); this indicated that the film was strongly orientated in the \( b \)-axis.
Kiessig fringes\textsuperscript{21} are clearly seen at lower values of $q_{\text{out-of-plane}}/2\pi$, indicating that these films possessed smooth surfaces; additionally, the film thickness was estimated to be $\sim 55$ nm, calculated from the fringes around $0.2\text{ nm}^{-1}$.

Two-fold symmetry was clearly observed in the in-plane X-ray Bragg diffraction patterns in Fig. 2(b). In the upper figure (azimuth [001] LaAlO$_3$), the 200 Sr$_2$Nb$_2$O$_7$ diffraction peak was seen alongside the 002 LaAlO$_3$ substrate peak, whereas in the lower figure ([110] LaAlO$_3$), the diffraction peaks from the 00$l$ ($l = \text{even}$) were observed. The epitaxial relationship between the film and the substrate was found to be (010)[100] Sr$_2$Nb$_2$O$_7$||(110)[001] LaAlO$_3$. Furthermore, we recorded the reciprocal space mappings of the resultant films around the LaAlO$_3$ 222 diffraction spots [Fig. 2(c)]. Intense 2200 diffraction spots were seen at $(q_{1000}/2\pi, q_{0100}/2\pi = 5.06 \text{ nm}^{-1}, 7.51 \text{ nm}^{-1})$ together with the 222 LaAlO$_3$ (5.27 nm$^{-1}$, 7.48 nm$^{-1}$), clearly demonstrating that incoherent epitaxial growth had occurred in these films. The lateral grain size of the Sr$_2$Nb$_2$O$_7$ film was calculated to exceed 500 nm, although a precise calculation was not possible due to the resolution limitations of the technique.

The surface morphology of the films was observed using AFM, with RHEED patterns also used for complementary surface analysis. Figure 3(a) shows a typical topographic AFM image ($2 \times 2 \mu\text{m}^2$) of a Sr$_2$Nb$_2$O$_7$ film deposited by SPE. Large, hexagonally shaped grains with a stepped surface structure were observed, with step increments of $\sim 1.3\text{ nm}$, which corresponds well with the value of half of the $b$-axis length for Sr$_2$Nb$_2$O$_7$ ($b/2 = 1.34\text{ nm}$) (data not shown). Intense diffraction spots were seen in the RHEED patterns shown in Fig. 3(b), indicating that the surface of each grain was atomically flat and that the grains were of high crystal quality.

Figure 4(a) shows cross-sectional bright-field TEM images of the SPE-derived Sr$_2$Nb$_2$O$_7$ films, acquired using an incident electron beam direction of [001] Sr$_2$Nb$_2$O$_7$ (=[110] LaAlO$_3$). Here, the topmost part of the film was observed to be atomically flat, and the film thickness was $\sim 0.2\text{ nm}^{-1}$.
found to be \(55\) nm, a value that corresponds closely to those calculated from the XRD results. Many stripes were seen in the film, indicating that the \((110)\) \(\text{SrNbO}_3\) slabs were stacked along the \(b\)-axis. A darker layer with a thickness of a few nanometers was observed at the heterointerface; in order to clarify the origin of this layer, we collected data on the atomic arrangements around the heterointerface using HAADF–STEM, with a collection semi-angle of 68–280 mrad—these results are shown in Fig. 4(b). Interfacial steps and misfit dislocations (indicated by the yellow arrows) were seen around the heterointerface, which were considered to lead to the presence of stress concentration at the interface and the dark contrast seen in Fig. 4(a). In the region above the interface, a highly ordered, periodic layered structure with \(\text{SrNbO}_3\) slabs located every \(1.3\) nm can clearly be observed. From these results, we concluded that the resultant \(\text{Sr}_2\text{Nb}_2\text{O}_7\) films were enough to discuss its intrinsic physical properties.

In order to compare the physical properties of the resultant \(\text{Sr}_2\text{Nb}_2\text{O}_7\) films with those of \(\text{Sr}_2\text{Nb}_2\text{O}_7\) single crystals, we measured optical properties including transmission \((T)\), reflection \((R)\), and absorption \((\text{Abs.})\) spectra at room temperature. Figure 5(a) shows \(T\), \(R\), and \(\text{Abs.}\) spectra of a \(\text{Sr}_2\text{Nb}_2\text{O}_7\) film (Thickness: \(60\) nm). The absorption edge is \(320\) nm, indicating that the direct bandgap of the film is \(3.9\) eV. In the whole wavelength region, \(R\) exceeds 20%,

![Image](image_url)
indicating the large refractive index (n) of the film. In fact, we obtained rather large n = 2.1 at 589.3 nm (the sodium D line) from the R spectrum [Fig. 5(b)]. We then speculated the dielectric permittivity at dc limit (\(\varepsilon_r\)) of the Sr2Nb2O7 film. Figure 5(c) shows the relationship between \(\varepsilon_r\) and n of several dielectric oxides. The \(\varepsilon_r\) of the film was speculated to be in between 20 and 80, comparable to that of the single crystal \((\varepsilon_{rz} = 75, \varepsilon_{rb} = 46, \varepsilon_{rz} = 43)\). These optical properties clearly indicate that the resultant Sr2Nb2O7 films are enough high quality to discuss its intrinsic physical properties.

Finally, we discuss the differences between direct VPE and SPE in the formation of Sr2Nb2O7 on (110) LaAlO3 substrates. In the case of VPE (PLD), both (001)-oriented Sr2Nb2O7 and (110)-oriented SrNbO3 phases formed, and the epitaxial relationship between the Sr2Nb2O7 film and the substrate was (001)[100] Sr2Nb2O7 //[(110)[001] LaAlO3, in which the SrNbO3 slabs were stacked along the in-plane direction. We consider that the formation of perovskite SrNbO3 is more favorable in comparison to Sr2Nb2O7 during VPE, due to the insufficient oxidation of the material in the vapor phase. The formation of perovskite SrNbO3 domains may lead to a decrease in the degrees of freedom of the direction of crystal growth in neighboring Sr2Nb2O7 domains; (110) SrNbO3 slabs were not found to be stacked in the out-of-plane direction due to the differences in the out-of-plane atomic arrangement between Sr2Nb2O7 and SrNbO3. Therefore, when depositing the films by PLD, the (110) SrNbO3 slabs were stacked along the in-plane direction; on the other hand, in the case of SPE, Sr2Nb2O7 was able to crystallize directly from the amorphous state into the structure with the most stable epitaxial relationship.

IV. SUMMARY

In summary, high-quality epitaxial films of Sr2Nb2O7 were successfully fabricated by solid phase epitaxy on (110) LaAlO3 single crystal substrates. Amorphous Sr–Nb–O films, deposited by PLD at room temperature, crystallized epitaxially as Sr2Nb2O7 films at a high temperature of 1400 °C, resulting in the formation of a highly ordered atomic arrangement and an atomically flat surface. The epitaxial relationship between the film and the substrate was (010)[100] SrNbO3 //[(110)[001] LaAlO3. The refractive index of the resultant film was 2.1, indicating that the dielectric permittivity of the film was in between 20 and 80, which corresponds well with that of the single crystals. These results clearly demonstrate the effectiveness of the SPE method for the fabrication of high-quality epitaxial films of Sr2Nb2O7.

ACKNOWLEDGMENTS

This research was supported by Grants-in-Aid for Scientific Research on Innovative Areas “Nano Informatics” (25106005, 25106007) from the Japan Society for the Promotion of Science (JSPS). H. O. was supported by Grants-in-Aid for Scientific Research A (17H01314) and B (26287064) from JSPS. Y. Z. thanks China Scholarship Council (CSC) for a scholarship to study in Japan.


**Current Topics in Materials Science**, edited by E. Kaldis (Elsevier Science Ltd, 1980).


