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Catalytic activities of titania-supported nickel for carbon-dioxide methanation

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Abstract

Titania (TiO₂)-supported nickel (Ni) catalysts (3, 6, 12 and 20wt%; Ni/TiO₂) were prepared by a sol-gel method to be used as a catalyst for carbon-dioxide (CO₂) methanation. The catalytic activity tests were conducted with a fixed-bed reactor in the temperature range of 100–550 °C. The results showed that the CO₂ conversion increased with increasing Ni content and 20wt% Ni/TiO₂ exhibited the highest CO₂ conversion and the methane (CH₄) yield which corresponded to the lowest apparent activation energy. Moreover, 20wt% Ni/TiO₂ also showed high stability under the CO₂ methanation stream for 72 h. The addition of Ni into TiO₂ during the sol-gel process modified the catalyst properties which affected to their catalytic activities, and the influence of Ni addition on the catalyst structure was studied by several techniques. The results on XRD and EXAFS revealed unit cell expansion and lattice distortion which indicated that nickel(II) cation was incorporated into TiO₂ lattice. Upon increasing in Ni content, the nickel(II) oxide (NiO) phase was observed indicating the high crystallinity of NiO and suppression of TiO₂ crystal growth. These effects led to smaller crystalline size of TiO₂ and higher surface area of the catalysts. The oxidation states of nickel and titanium were analyzed to be +2 and +4, respectively for fresh catalysts. After pretreatment by hydrogen (H₂), Ni(II) was converted to Ni(0) and this electronic state was still unchanged during the course of methanation while Ti(IV) was kept unaltered for fresh catalysts and those during the reaction. However, the distortion or defect around Ti(IV) ions in Ni-loaded TiO₂ was observed as increase in Ti K-edge XANES pre-edge peaks. The addition of Ni led to formation of oxygen vacancy in TiO₂ lattice and NiO-phase formation which might act as the sites for CO₂

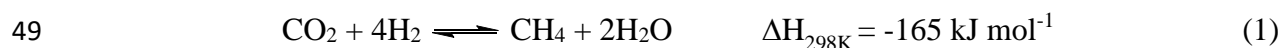
35 and H₂ adsorption, respectively. Therefore, higher amount of adsorbed CO₂ and H₂ could
36 enhance the surface reaction and also speed up the CO₂-methanation.

37 **Keywords:** carbon-dioxide methanation; titania-supported nickel; XANES; EXAFS

38

39 1. Introduction

40 Recently, global warming is the most concerned problem of the world. The main reason
41 for this problem is releasing high level of greenhouse gases (GHG) into the atmosphere.
42 Carbon dioxide (CO₂) has a large proportion of GHG. Enormous CO₂ emission level is a
43 mainly result from human activities such as burning fossil fuel (coal, natural gas, and oil),
44 transportation, deforestation, agriculture, electricity and industrial sectors. Therefore, reduction
45 and utilization of CO₂ to produce value-added raw materials for chemical reactions are
46 interested. CO₂ methanation is one of a powerful approach to utilize the CO₂ as a feedstock for
47 producing CH₄. This reaction is called the Sabatier reaction which is defined as the following
48 equation;



50 The characteristic of this reaction are as follows; i) this reaction is a highly exothermic process,
51 thus, CH₄ formation prefers to occur at low temperature and high pressure [1,2] ii) there is a
52 kinetic barrier with eight-electron reduction process of the CO₂ (4+) to form CH₄ (4-) iii) CO₂
53 molecule is a strong linear structure with two double bond, so high dissociation energy must
54 be consumed and iv) side reaction, reverse water gas shift (CO₂ + 4H₂ \rightleftharpoons CO + 2H₂O) [3],
55 occurs usually at high temperature which can produce an undesired product (CO). Therefore,
56 from these limitations of this reaction, to increase the rate at low temperature, the catalyst must
57 be employed.

58 Normally, the catalyst is consisted of two components; support and active component.
59 The typical support properties are chemical stability, high thermal resistance and high surface
60 area and porosity. The last properties have a significant role in catalytic reaction, since the
61 adsorption and surface reaction processes are occurred on catalyst surface. An active
62 component is another important component which is usually added onto the support with a
63 small amount when compared with support. The function of an active site is selectively toward
64 on adsorption of reactants, thus, an appropriate selectively active component is used for
65 enhancing the adsorption and also speed-up the surface reaction. Consequently, tuning and
66 modifying these two components properties can control and dominate the catalytic
67 performance of the catalyst. For CO₂ methanation, enhancing the adsorption ability of two

68 reactants (CO_2 and H_2) on catalyst surface can possibly improve the surface reaction between
69 these two adsorbed species and then the product formation rate also increases. Therefore,
70 modifying catalyst with appropriate active site for CO_2 and H_2 adsorption can enhance the
71 reaction rate. It is well known that H_2 favors to adsorb on metal, especially transition metal
72 group, various active metals are widely used for this reaction such as Rh, Ru, Pt and Pd [4-7].
73 However, using the noble metals are limited due to their high cost and scarcity [8]. From this
74 reason, nickel is more attractive to be used for CO_2 methanation due to its relatively low price
75 and also give high activity [9-11]. In order to enhance the H_2 adsorption capacity, high
76 dispersion of active metal on catalyst surface must be done. However, Ni can be deactivated
77 by using at high temperature due to sintering process, thus, modifying the catalyst with
78 resistance of this effect is also of interested. For CO_2 adsorption, providing the basic sites on
79 catalyst surface can improve the CO_2 uptake via acid-base interaction. For this aspect, addition
80 of basic metal oxide such as La_2O_3 , CeO_2 and MgO is usually employed. Other method to
81 increase CO_2 uptake is to produce an oxygen vacancy which is reported as an adsorption site
82 for CO_2 molecule [12,13]. Oxygen vacancy can be formed within the metal oxide support by
83 substitution of the host metal by adding metal with lower oxidation state than that of host metal
84 [14,15]. Formation of oxygen vacancy by this way is usually used with a reducible support
85 (TiO_2 , CeO_2 , ZrO_2 , etc.), since the oxygen vacancy can mobile inside the lattice and plays an
86 important role in redox process. Finally, it is seen that tuning and modifying of catalyst
87 properties by considering the characteristic of reaction can produce the catalyst with great
88 catalytic performance. It is reported that the synthesis method is strongly determined the
89 physical and chemical properties of catalyst [11,16]. Many synthesis methods were reported to
90 be used and provided a great catalytic performance [6,17,18]. However, the simple,
91 environment friendly and inexpensive processes were usually preferred. The sol-gel method is
92 one of the simplest and the most widely used due to its low cost of the required instrument,
93 mild synthesis condition and also providing high homogeneous of the catalyst.

94 In this work, Ni was selected to be used as an active metal and TiO_2 was a support.
95 Since using TiO_2 as a catalyst support for CO_2 methanation is still developed, thus it is
96 challenging to modify TiO_2 catalysts which provide the excellent activity and produce the
97 desired products for this reaction. Sol-gel method was used as a synthesis method to produce
98 Ni/ TiO_2 catalysts with different Ni loading. The structural and surface properties of synthesized
99 samples upon addition of Ni was investigated by X-ray diffraction (XRD), N_2 adsorption-
100 desorption, transmission electron microscopy (TEM), H_2 temperature-programmed reduction
101 (TPR), H_2 chemisorption and thermogravimetric analysis. All strategies were used for studying

102 the effect of Ni addition for CO₂ methanation. Advanced analysis of local structures and
103 oxidation states of probe metals (Ni and Ti) was determined by using X-ray absorption
104 spectroscopy (XAS) which was including with X-ray absorption near-edge structure (XANES)
105 and extended X-ray-absorption fine structure (EXAFS). The effect of Ni addition to enhance
106 the CO₂ methanation was discussed based on the all characterized results.

107

108 **2. Experimental**

109 **2.1 Syntheses of TiO₂ and Ni-loaded TiO₂ catalysts**

110 All synthesized samples were prepared by sol-gel method using titanium(IV)
111 isopropoxide (98%, Acros Organics) as a precursor. For pure TiO₂, Ti precursor was mixed
112 with glacial acetic acid (RCI Labscan) and 2-propanol (Carlo Erba Reagents) and then stirred
113 for 30 min. The mixture was dropped wise with deionized water (DI) and vigorously stirred for
114 starting hydrolysis reaction. Finally, the solution was dried in an oven at 110°C for 24 h and
115 calcined at 500°C for 2 h to obtain bare TiO₂.

116 In preparation of Nickel (Ni) modified TiO₂ catalysts, the titanium(IV)
117 isopropoxide, glacial acetic acid and 2-propanol were mixed with the same manners as the
118 above-mentioned pure-TiO₂ preparation. The desired amount of nickel nitrate (Carlo Erba
119 Reagents) was dissolved with DI water and then dropped wise in the Ti precursor mixture. The
120 solution was dried and calcined at the same conditions as pure TiO₂. The synthesized samples
121 were denoted as Xwt% Ni/TiO₂ where X = 3, 6, 12 and 20wt%.

122 **2.2 Characterization**

123 The structural properties of synthesized samples were characterized by X-ray
124 diffraction (XRD) using Rigaku SmartLab X-ray diffractometer using Cu K_α (λ = 0.154 nm).
125 The XRD pattern was collected in the 2θ range of 10° to 80°. Bragg's equation was used to
126 determine the unit cell parameters of samples by using (101) crystallographic plane of anatase
127 TiO₂. The means of crystallite sizes of TiO₂ and Ni-loaded TiO₂ were calculated by the
128 Scherer's equation:

$$129 \quad D = \frac{K\lambda}{\beta \cos \theta} \quad (2)$$

130 where *D* is the crystallite size, *K* is the shape factor, *λ* is the X-ray wavelength, *β* is the
131 broadening at half maximum and *θ* is the diffraction angle. In order to determine the strain
132 induced by addition of Ni, the Williamson-Hall (W-H) plot method was used which can be
133 expressed by the following equation;

134
$$\beta \cos \theta = \frac{K\lambda}{D} + 4\varepsilon \sin \theta \quad (3)$$

135 where ε is the strain induced from crystal imperfection and distortion. The W-H plot shows a
136 relationship between $\beta \cos \theta$ and $4\sin \theta$. The strain (ε) can be estimated from the slope of linear fit and
137 the crystallite size (D) can also be determined from y-intercept.

138 Nitrogen (N_2) adsorption-desorption isotherms were recorded using a
139 Quantachrome (previously Yuasa Ionics; Autosorb 6AG) surface-area and pore-size analyzer.
140 Prior N_2 adsorption, the sample was degassed at 200 °C for 2 h to remove the moisture on
141 surface sample. Specific surface area, average pore size and pore volume were calculated by
142 Brunnauer-Emmett-Teller (BET) equation. The specific surface area was commonly
143 determined at relative pressure (P/P_0) range between 0.1 to 0.3. Barrett-Joyner-Halenda (BJH)
144 method was used to calculate the pore size distribution. The pore volume was measured at the
145 relative pressure closed to unity ($P/P_0 \sim 1$).

146 Hydrogen (H_2) temperature-programmed reduction (TPR) was investigated by a
147 Belcat B apparatus. A 20-mg portion of sample was pretreated by flowing with helium (He)
148 at 120 °C for 30 min to clean up the surface. After cooling down, the mixture gases of H_2 and
149 argon (Ar) were flowed over the sample while the temperature was rose up to 1000 °C with
150 ramping rate of 10 K min^{-1} . Thermal conductivity detector (TCD) was used to collect the
151 change of thermal conductivity of the gas stream. H_2 consumption was plotted as a function
152 of temperature.

153 Transmission electron microscopy were performed by TEM, a FEI Tecnai G2 20 S-
154 TWIN), to analyze particles sizes and morphologies of samples. The catalyst particles were
155 dispersed in ethanol by sonication before dropping on copper grids.

156 Inductively coupled plasma-optical emission spectroscopy (ICP-OES) analysis was
157 employed to measure the Ni contents. The powder (0.025 g) of sample was digested by a
158 mixture of concentrated H_2SO_4 (6 mL) and 30% H_2O_2 (2 mL). The calibration curve was in the
159 range of 0-30 ppm. An Optima 100 DVICO-OES (PerkinElmer) instrument was used to
160 analyze Ni contents in a sample solution. The plasma was generated by an RF power with
161 1300W, and plasma flow rate was 15 L/min. Auxillary and neubilizer flow rate was 0.2 L/min
162 and 0.8 L/min, respectively. Emission light signal was detected by a CCD-array detector.

163 **2.3 H_2 chemisorption and CO_2 uptake**

164 H_2 chemisorption experiment was used to investigate the metal active sites on the
165 catalyst surface which play an important role in enhancing the reaction rate. The H_2

166 chemisorbed amount of all-loaded samples (3, 6, 12 and 20wt% Ni/TiO₂) was carried out by
167 Quantachrome Autosorb-1-C instrument. 200 mg of catalyst was first pretreated by flowing
168 with He at 120 °C (ramp rate 20 °C min⁻¹) for 30 min. After pretreatment, H₂ (99.999%, Thai
169 Industrial Gas) was flowed over the sample during rising the temperature up to 300 °C (ramp
170 rate 20 °C min⁻¹) and the temperature was kept constant at this point for 120 min. Next, the
171 sample was cooled down to 40 °C under vacuum and then H₂ was sequentially added to the
172 sample. The amount of H₂ chemisorption was obtained by extrapolating method of the graph
173 between volume of H₂ uptake and pressure. The monolayer of H₂ uptake (N_m ; μmol g⁻¹) can be
174 calculated by the following equation;

$$175 \quad N_m = 44.61 V_m \quad (4)$$

176 where V_m is the volume of monolayer uptake which can be determined by extrapolating method.
177 CO₂ adsorption capacity of sample can be also used to explain the role of catalyst to catalyze
178 the reaction rate. The CO₂ uptake was monitored by thermalgravimetric analysis (TGA, Pyris
179 Diamond, Perkin Elmer Instrument). Al₂O₃ was used as a reference. First, the synthesized
180 sample was separately weighted and put on the aluminum pan and then placed on the sample
181 holder. The maximum different weight between sample and reference was 0.5 mg. The sample
182 was first pretreated by heating at 150 °C (10 °C min⁻¹) for 45 min under N₂ flowing to remove
183 an impurities. After that, CO₂ uptake was started by cooling down the sample to 40 °C under
184 CO₂ flowing (100 mL min⁻¹) for 60 min. Upon decreasing the temperature, the weight of sample
185 was increased due to adsorption of CO₂ on solid surface. The CO₂ uptake can be calculated by
186 differentiate the weight before and after adsorption.

187 **2.4 CO₂ methanation catalytic activity test**

188 The CO₂ methanation-activity test was carried out in a fixed bed reactor with inner
189 diameter of 0.6 mm. 50 mg of sample was placed at middle of pyrex reactor tube and
190 sandwiched with quartz wool. Accurate temperature was monitored by K-type thermocouple
191 which was located on the top of catalyst bed. The flow rates of the reactant gases were precisely
192 controlled by mass flow controllers. Prior activity test, the synthesized catalyst was reduced
193 under H₂ flow (50 mL min⁻¹) at 450°C and maintained for 90 min. After that, the system was
194 cooled down to 150°C to start the reaction. H₂ and CO₂ was used as a feed gas with a ratio of
195 H₂/CO₂ = 4 (balance with He). Total flow rate of feed gas was 40 mL min⁻¹ and GHSV was
196 48,000 mL g⁻¹ min⁻¹. The reaction temperatures were in the range of 100-550 °C. The effluent
197 products and remaining reactants were analyzed by an online gas chromatography (Agilent

198 6890N Series, Agilent Technology) equipped with HEYSEP D Packed Column and TCD
199 detector. Instrumental diagram of CO₂ methanation catalytic activity test set up is shown in
200 Figure 1. CO₂ conversion, CH₄ and CO selectivity and yield were calculated by the following
201 equations:

$$202 \quad X_{\text{CO}_2} = \left(\frac{F_{\text{CO}_2\text{in}} - F_{\text{CO}_2\text{out}}}{F_{\text{CO}_2\text{in}}} \right) \times 100 \quad (5)$$

$$203 \quad S_{\text{CH}_4, \text{CO}} = \left(\frac{F_{\text{CH}_4\text{out}} + F_{\text{COout}}}{F_{\text{CH}_4\text{out}} + F_{\text{COout}}} \right) \times 100 \quad (6)$$

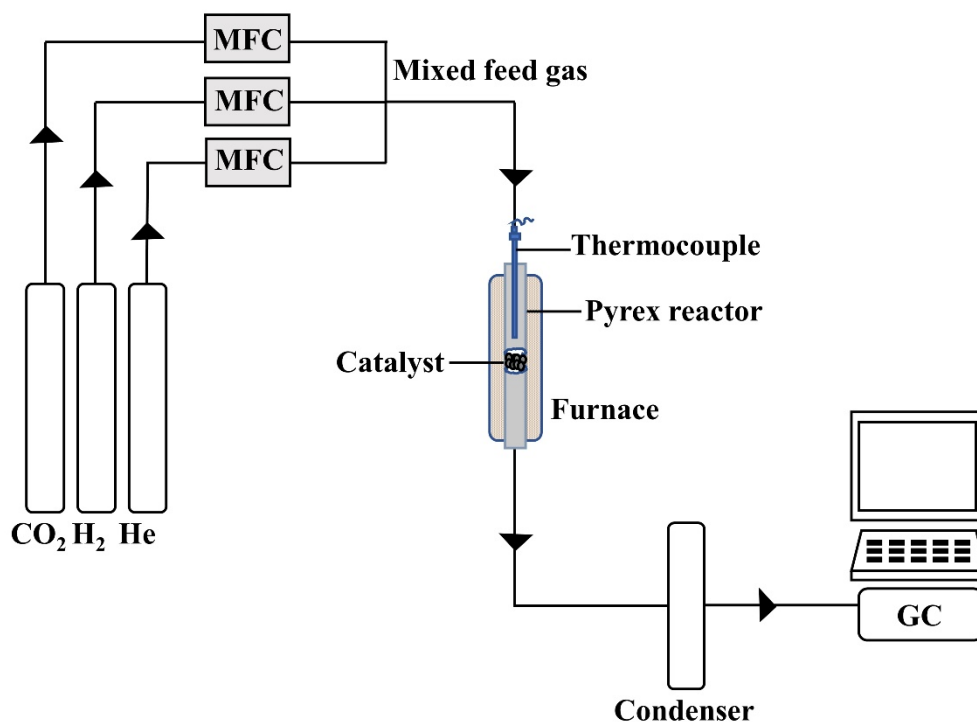
$$204 \quad Y_{\text{CH}_4, \text{CO}} = \left(\frac{X_{\text{CO}_2} \times S_{\text{CH}_4, \text{CO}}}{100} \right) \quad (7)$$

205 where, X_{CO_2} is the CO₂ conversion, S is the selectivity and Y is the yield. F is the molar flow
206 rate of gas (mol s⁻¹).

207 The rate of CO₂ methanation were measured in the temperature range of 240-290 °C
208 to keep below 10% of CO₂ conversion. The rate of reaction was calculated using the following
209 equation [19];

$$210 \quad \text{Rate} = \left(\frac{F_{\text{CO}_2}}{W} \right) \times X_{\text{CO}_2} \quad (8)$$

211 where, F_{CO_2} is the flow rate of CO₂ (mol s⁻¹), W represents the weight of catalyst (g).



212 **Figure 1** Instrumental diagram of CO₂ methanation catalytic activity test.

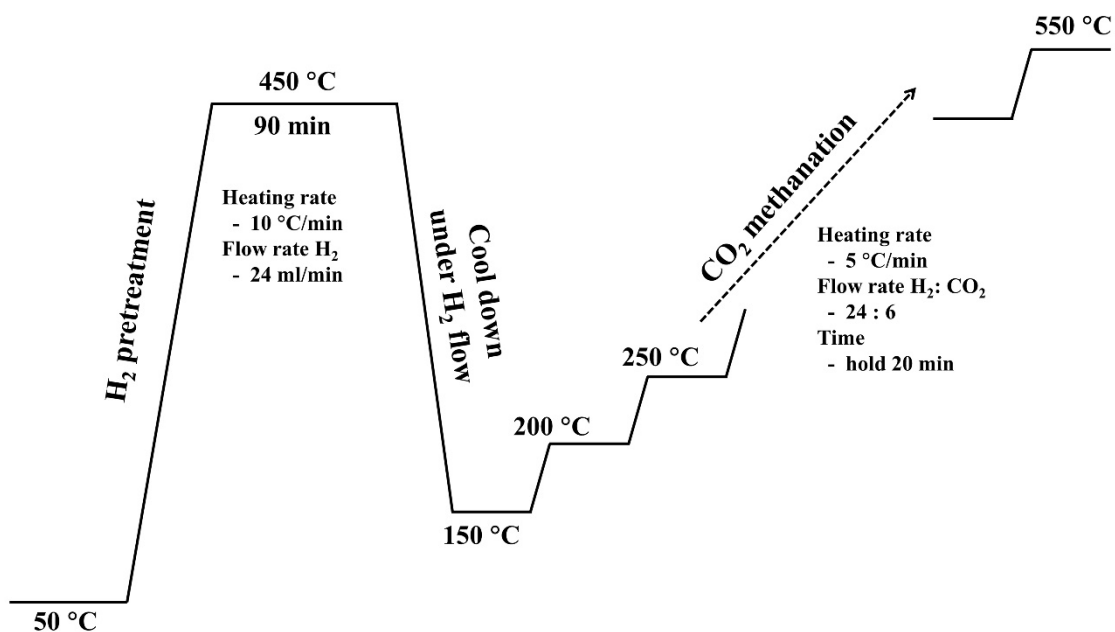
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214 2.5 X-ray Absorption Spectroscopy (XAS)

215 2.5.1 X-ray Absorption Near Edge Structure (XANES)

216 X-ray Absorption Near Edge Structure (XANES) experiment was divided into
 217 two parts; ex-situ and in-situ XANES experiments. The ex-situ XANES experiment was
 218 conducted to investigate the oxidation state of Ni and Ti on the fresh catalysts. While in-situ
 219 experiment was used to monitor the electronic state changing of Ni and Ti during CO₂
 220 methanation. The ex-situ XANES experiment was carried out on beamline 5.2 at Synchrotron
 221 Light Research Institute (SLRI), Nakhon Ratchasima, Thailand [20,21]. The Ge (220) was used
 222 as double crystal monochromator with energy resolution of 2×10^{-4} eV. The in-situ XANES
 223 experiment was carried out on beamline 2.2 at Synchrotron Light Research Institute (SLRI),
 224 Nakhon Ratchasima, Thailand. The Si (111) crystal was used as an Energy Dispersive
 225 Monochromator (EDM) which covered photons energy of 2400-12000 eV. The linear image
 226 sensor was a detector. Samples were prepared by mixing with Boron Nitride (BN) to a
 227 homogeneous powder by using ball milling for 20 minutes. After that, the mixed solid sample

228 was compressed to a pellet. For ex-situ XANES experiment the pellet was placed on a Kapton
229 tape which was stuck on the sample frame. The XANES spectra were collected in the K
230 absorption edge energy for both Ni and Ti in transition mode. The ionization chamber was
231 employed as a detector which was located in front of and behind the sample. For in-situ
232 XANES experiment, the prepared sample in pellet form was placed in an in-situ cell. The in-
233 situ XANES experiment condition was almost the same as CO₂ methanation catalytic activity
234 test. The sample was pretreated with H₂ (24 mL min⁻¹) during the temperature rose up from
235 room temperature to 450 °C (heating rate of 5 °C min⁻¹) and the XAS data were collected for
236 each 10 °C increment. The reduction temperature (450 °C) was held for 1.5 h and the data was
237 also recorded every 10 minutes. After pretreatment, the system was cooled down to 150 °C
238 while the mixed feed gas between CO₂ (6 mL min⁻¹) and H₂ (24 mL min⁻¹) were purged through
239 the sample to start the CO₂ methanation. At the starting state (150 °C), the reaction temperature
240 was raised with the rate of 5 °C min⁻¹ to 200 °C and the data was recorded every 10 °C
241 increment. Then, the temperature was kept constant at 200 °C for 20 minutes with recording of
242 the data for every 10 minutes. The experiment was continuously conducted with the same
243 procedure as above; i.e. the reaction temperature was also increased for each 50 °C with the
244 heating rate of 5 °C min⁻¹ (recording data for each 10 °C increment) and held at that temperature
245 for 20 minutes (recoding data for each 10 minutes). The reaction temperature of an in-situ
246 experiment was in the range of 150-550 °C. Figure 2 shows the schematic of in-situ experiment
247 for XANES.



248 **Figure 2** Schematic of in-situ experiment for XANES.

249 2.5.2 Extended X-ray Absorption Fine Structure (EXAFS)

250 Extended X-ray Absorption Fine Structure (EXAFS) was carried out to
 251 determine the coordination number, interatomic distance and structural and thermal disorder
 252 around Ni probe. The EXAFS was used to monitor the electronic state changing of Ni during
 253 CO₂ methanation. The experiment was done on Beamline 5.2 at the Synchrotron Light
 254 Research Institute (SLRI). (electron energy of 1.2 GeV; bending magnet; beam current 80-150
 255 mA; 1.1 to 1.7 x 10¹¹ photon s⁻¹). The Ni K-edge experiment was collected in transmission
 256 mode (8333 eV). The X-ray incident and transmitted beam intensities were detected by
 257 ionization chamber which was placed in front of and behind the sample. Samples were mixed
 258 with Boron Nitride (BN) to form a homogeneous powder and compressed to a pellet. The
 259 sample was pretreatment under H₂ flowing at 450 °C (20 mL min⁻¹). The pretreatment
 260 temperature was hold at this point for 90 min and the data were collected during this state.
 261 Afterwards, the system was cooled down to 100 °C and CO₂ was purged at 5 mL min⁻¹ to start
 262 the reaction. In each step, the reaction temperature was increased for each 50 °C and kept
 263 constant for 30 min with collecting the EXAFS spectra. The experiment was continuously run
 264 by increasing reaction temperature from 100 to 550 °C.

265 The fine structure of EXAFS region is expressed in terms; $\chi(E) = \mu(E) -$
 266 $\mu_0(E) / \Delta\mu_0(E)$, where $\mu(E)$ is the absorption coefficient, $\mu_0(E)$ is the smooth background

267 function and $\Delta\mu_0$ is the absorption edge. The wave behavior of photoelectron (k) can be written
 268 by X-ray energy as; $k = \sqrt{2m_e(E-E_0)/h^2}$, where E_0 is the absorption edge energy and m_e is the
 269 electron mass. The oscillation in $\chi(k)$ model of EXAFS region is corresponding to

$$270 \quad \chi(k) = \sum_j \frac{N_j S_0^2 f_j(k) e^{-2R_j/\lambda} e^{-2k^2\sigma_j^2}}{kR_j^2} \sin[2kR_j + \delta_j(k)] \quad (9)$$

271 where N_j is the number of neighboring atoms, R_j is the neighboring distance and σ_j^2 is the
 272 disorder of neighboring distance. $F(k)$ and $\delta(k)$ are the scattering-amplitude and phase-shift.
 273 S_0^2 is the amplitude reduction factor. The data is usually expressed by Fourier transformation
 274 from $\chi(k)$ to R space to study local structure around neighboring atom as

$$275 \quad \chi(R) = \frac{1}{\sqrt{2\pi}} \int_{k_{\min}}^{k_{\max}} k^n \chi(k) W(k) e^{i2kR} dk \quad (10)$$

276 The next step is normalization of the experiment data by ATHENA program. The
 277 spectra were amplified by k^2 weight with window in R space range of 1-5 Å and k range of 3-
 278 10 Å⁻¹. To determine the local structure of Ni-loaded TiO₂ in CO₂ methanation, Ni foil was
 279 used as a model fitting of EXAFS data. The experimental EXAFS data was fitted by ARTEMIS
 280 program.

281

282 3. Results and discussion

283 3.1 Catalytic performances of CO₂ methanation

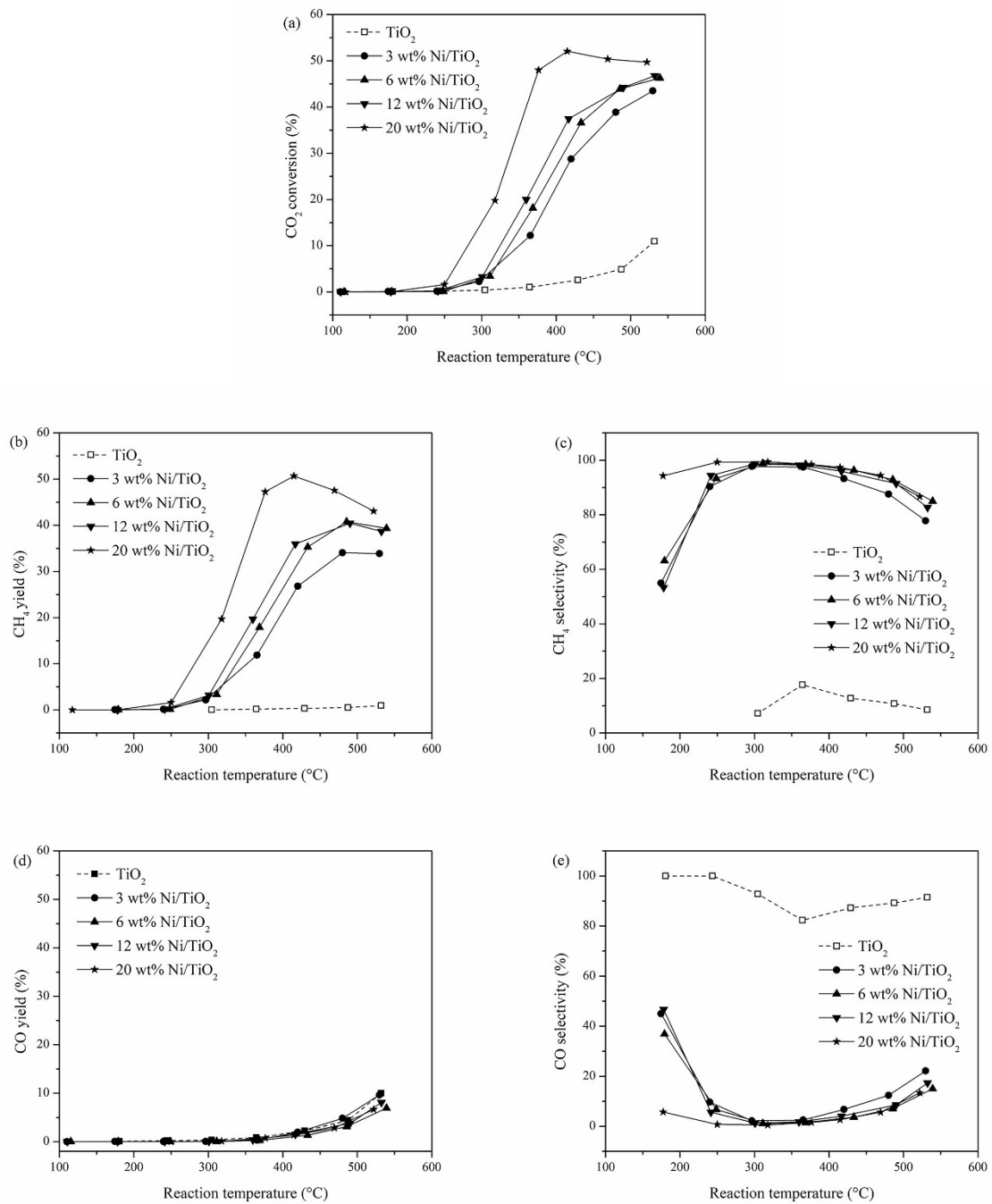
284 The CO₂ methanation was carried out at reaction temperature range of 100-550 °C
 285 with a weight hourly space velocity (WHSV) of 48,000 mL g⁻¹ h⁻¹. Catalytic performances of
 286 TiO₂, 3, 6, 12 and 20wt% of Ni-loaded TiO₂ samples were displayed in term of CO₂ conversion
 287 as a function of temperature. The CO₂ conversion, products (CH₄ and CO) selectivities and
 288 yields are represented in Figure 3(a)-3(e). The CO₂ conversion for all samples increased
 289 according to reaction temperature. However, upon the addition of Ni onto TiO₂, shifting of CO₂
 290 conversion toward lower reaction temperature were observed. For bare TiO₂, the catalytic
 291 performance was almost inactive with illustrating of the lowest CO₂ conversion and low CH₄
 292 yield and selectivity, while CO selectivity was relatively high. In contrast, Ni supported TiO₂
 293 catalysts (3, 6, 12 and 20wt% Ni/TiO₂) exhibited a dramatically enhance in catalytic activities
 294 which can be observed by increasing of CO₂ conversion at lower reaction temperature. Upon
 295 addition of Ni, product selectivities were changed, i.e., CH₄ yield were developed and
 296 selectivity toward CH₄ were almost 100%, while CO yield and selectivity were rapidly dropped
 297 to lower 10%. This result indicated that addition of Ni onto TiO₂ support improved the desired

298 product (CH₄) selectivity and suppressed an unwanted product (CO). Increase of Ni
299 concentration resulted in increasing of CO₂ conversion and CH₄ yield, while CH₄ selectivities
300 were almost nearly the same (~100%) for all samples. However, 20wt% Ni/TiO₂ exhibited the
301 highest CO₂ conversion and CH₄ yield, it is noteworthy that the CH₄ selectivity was almost
302 100% in the wide range of reaction temperature (150-450 °C).

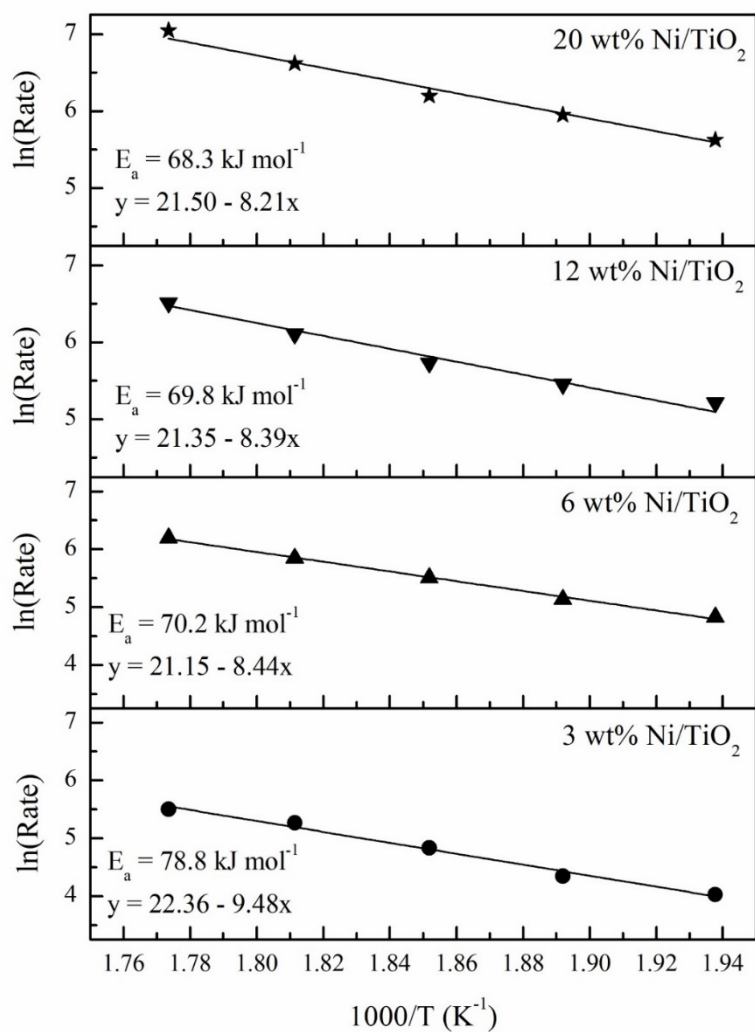
303 In order to study the kinetic parameter (apparent activation energy), heat and mass
304 transfer effects were checked during measurement of reaction rate by the Koros-Nowak test
305 [22]. The catalytic performances were tested by using samples with different of Ni active site
306 loading (6wt% and 20wt% Ni) at 250 °C at feed gas ratio of H₂/CO₂/He = 24:6:10 and WHSV
307 = 48,000 mL g⁻¹ h⁻¹. The results revealed that the CO₂ conversion of 6wt% Ni/TiO₂ and 20wt%
308 Ni/TiO₂ were 1.90 and 4.30 %, respectively, while turn over frequency (TOF) were almost the
309 same values (0.27 s⁻¹ and 0.25 s⁻¹, respectively). Therefore, it was attributed that heat and mass
310 transfer were absence for this experiment condition.

311 The apparent activation energy (E_a) was usually used to elucidate the catalytic
312 performance in enhancing the reaction rate. This value can be obtained by using the Arrhenius
313 equation ($k = Ae^{-E_a/RT}$). The plotting between ln(rate) and reciprocal of absolute temperature
314 (1/T) was constructed. The slope of the linear graph was used to calculate the apparent
315 activation energy. Figure 4 illustrates the Arrhenius plots of 3, 6, 12 and 20wt% Ni/TiO₂. It
316 was seen that the apparent activation energies were decreased with increasing of Ni content (E_a
317 ranges between 68-78 kJ mol⁻¹). Reduction of apparent activation energy upon more Ni
318 addition indicated that lower kinetic barrier for catalyzing toward the reaction, thus, the
319 catalytic activity can be enhanced. The E_a values were corresponded to the catalytic activities
320 results that 20wt% Ni/TiO₂ exhibited the highest CO₂ conversion due to the lowest E_a for this
321 reaction. The long-term stability test is also an important property to determine the efficiency
322 of catalyst in thermal sintering for industrial scale. Figure 5 exhibits the CO₂ conversion and
323 CH₄ yield of 20wt% Ni/TiO₂ in CO₂ methanation for 72 h. The stability test of 20wt% Ni/TiO₂
324 was done by keeping the reaction temperature at 420 °C since this temperature exhibited the
325 highest in CO₂ conversion. The CO₂ conversion was stable during 72 h on stream (decrease

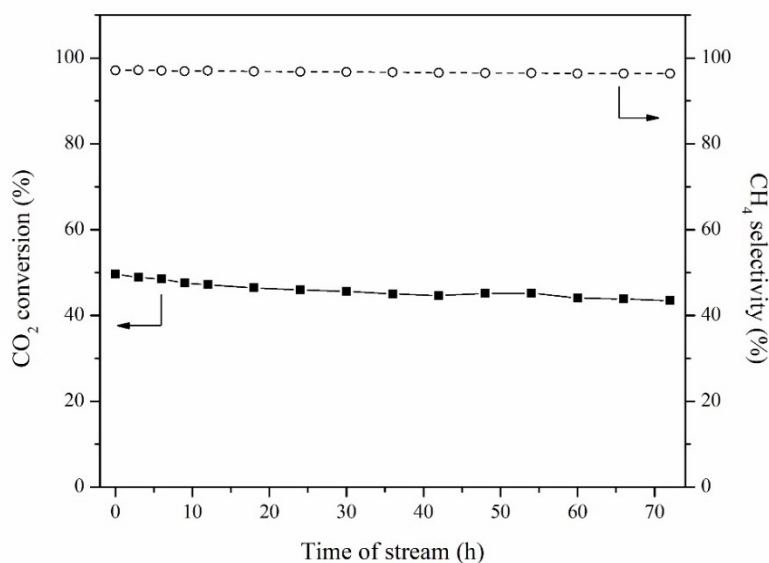
326 not over 10%) which was corresponded to consistency of CH₄ selectivity (almost 100%) for
 327 this sample.



328 **Figure 3** (a) CO₂ conversion, (b) CH₄ yield, (c) CH₄ selectivity, (d) CO yield and (e) CO
 329 selectivity of all catalysts; TiO₂, 3, 6, 12 and 20wt% Ni-loaded TiO₂ by using
 330 H₂/CO₂/He = 24:6:10, total flow rate 40 mL min⁻¹, WHSV = 48,000 mL g⁻¹ h⁻¹ at
 331 reaction temperature 100-550 °C.



332 **Figure 4** Arrhenius plots for the CO₂ methanation with different Ni contents on TiO₂; TiO₂,
 333 3, 6, 12 and 20wt% Ni/ TiO₂.



334 **Figure 5** CO₂ conversion and CH₄ selectivity over 20wt% Ni/ TiO₂ during long-term stability
 335 test at reaction temperature of 420 °C, H₂/CO₂/He = 24:6:10 and WHSV = 48,000
 336 mL g⁻¹ h⁻¹.

337 3.2 Standard characterization

338 3.2.1 X-ray diffraction (XRD)

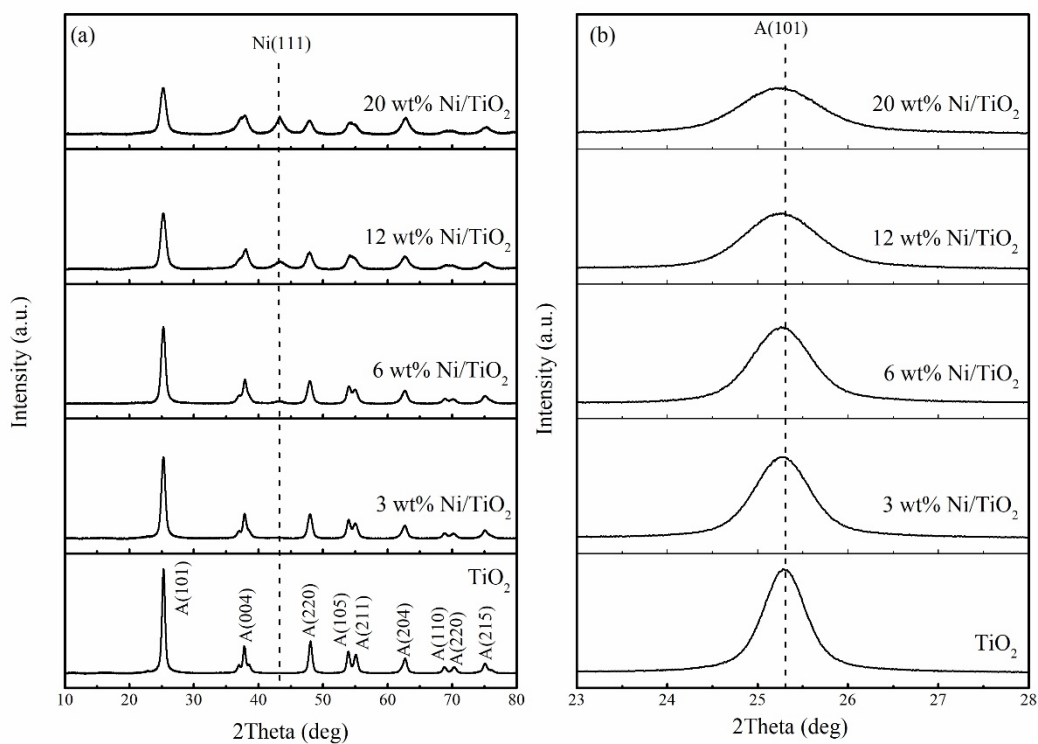
339 XRD patterns of TiO₂ with various Ni contents (3, 6, 12 and 20wt%) are
 340 shown in Figure 6. The XRD peaks of TiO₂ were observed at 25.3°, 37.8°, 48.0°, 53.9°, 55.1,
 341 62.7, 68.9°, 70.3° and 75.1° which were corresponding to (101), (004), (200), (105), (211),
 342 (204), (110), (220) and (215) planes of anatase structure with using JICST database (from The
 343 Crystallographic Society of Japan). For all Ni-loaded catalysts (3, 6, 12 and 20wt%), the
 344 anatase structure of TiO₂ remained unchanged. However, the intense diffraction peaks
 345 decreased when increasing of Ni loading. The lowering of these peaks related to reducing of
 346 crystallinities and crystalline sizes of TiO₂. This indicated that addition of Ni into TiO₂ during
 347 sol-gel process can prevent the crystalline growth of TiO₂ [23]. It was found that the crystalline
 348 sizes of TiO₂ were reduced with more added of Ni content as illustrated in Table 1. In addition,
 349 shifting of diffraction peaks to lower diffraction angles were also observed which indicated to
 350 unit cell expansion. This phenomenon resulted from the substitution of Ti⁴⁺ ion by larger Ni²⁺
 351 ion (ionic radii of Ni²⁺ = 0.72 Å and Ti⁴⁺ = 0.68 Å for 6-fold coordination). The unit cell of
 352 TiO₂ catalyst was enlarged upon increase of Ni content which indicated to more incorporation
 353 of Ni²⁺ ions into TiO₂ lattice [24-26]. To confirm the incorporation of adding metal into host
 354 lattice, the Williamson-Hall (W-H) parameter was used. Upon incorporation of adding metal,

355 the host lattice was distorted and strain was increased. In this case, Ti^{4+} site was replaced by
356 Ni^{2+} divalent cation, thus oxygen vacancies were expected to form. Therefore, the W-H
357 parameter can indicate to the degree of incorporation of adding metal [27,28]. Table 1
358 summarizes the unit cell parameter, unit cell volume and lattice strain of all synthesized
359 samples. It can be seen that the unit cell parameter and unit cell volume were increased with
360 increasing of Ni loading. The lattice strain was also enlarged which indicated to higher amount
361 of incorporated Ni^{2+} into TiO_2 lattice upon increase of Ni content.

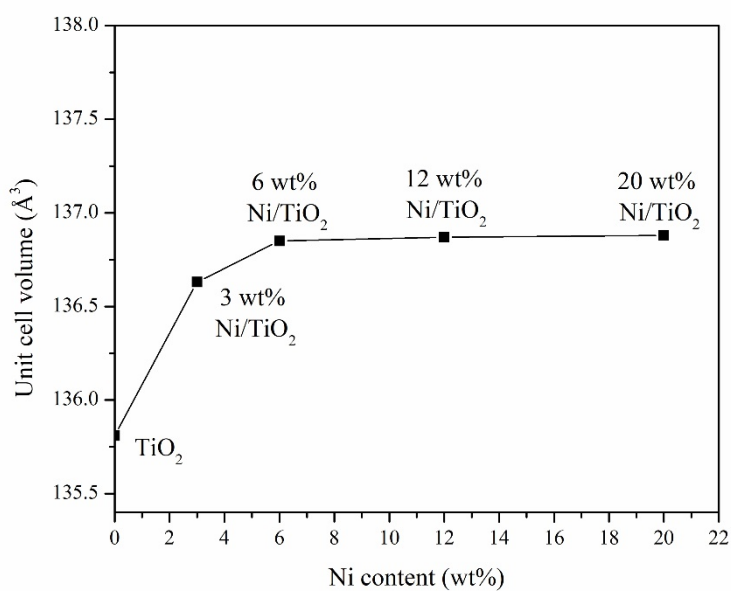
362 Besides the TiO_2 anatase diffraction peaks, another additional important peak was also
363 observed, i.e., NiO diffraction peak at 43.5° . This peak was attributed to a face centered-cubic
364 crystalline structure of NiO. It was noteworthy that this peak was not observed in the low Ni
365 content sample (3wt% Ni/ TiO_2). This was probably due to completely incorporate of adding
366 Ni ion into TiO_2 and the solid solution was formed. In contrast, the NiO diffraction peak was
367 obviously appeared for Ni loading $\geq 6\text{wt}\%$. This result can imply that addition of Ni was
368 existed in two forms, i.e., incorporation of Ni ion into TiO_2 lattice and dispersion of NiO onto
369 TiO_2 surface. Therefore, there was a limitation or saturated level of incorporation of Ni ion into
370 TiO_2 lattice and then the excess of Ni added amount can disperse onto TiO_2 surface. From this
371 behavior, both shifting of XRD peaks and existing of NiO peak were observed. In order to
372 estimate the saturated value of Ni concentration, the relationship between unit cell volume and
373 Ni content was constructed as illustrated in Figure 7. It can be seen that the unit cell volume
374 was increased upon addition of Ni and the expansion of lattice was almost constant at higher
375 Ni loading more than 6wt%. Form this plot, the saturated Ni incorporated amount was in the
376 range between 3-6wt%. Therefore, from our results, we assumed that the saturated Ni
377 incorporated amount was 3wt% which mean that all Ni addition was incorporated into TiO_2
378 lattice at this added level. Increasing of Ni content at higher amount than 6wt%, the excess
379 amount of adding Ni was dispersed onto TiO_2 surface and more intense of NiO diffraction peak
380 was observed which indicated to higher crystallinity and crystalline size of NiO phase. To
381 evidence the substitution of Ni ion into TiO_2 lattice and dispersion of NiO phase onto catalyst
382 surface, X-ray Absorption Spectroscopy (XAS) results were used and discussed in the next
383 part.

384

385



386 **Figure 6** (a) XRD diffraction patterns of all catalysts.
 387 (b) magnification of (101) crystallographic plane for all catalysts to compare
 388 the shifting of the diffraction peak.



389 **Figure 7** Relationship between unit cell parameters of all catalysts against Ni content.
 390

391 **Table 1** Structural and physical properties of all synthesized samples including lattice
 392 parameter, cell volume, crystal size, lattice strain and Ni content.

Samples	Lattice parameter ^a (Å)		Cell volume ^a (Å ³)	Crystal size ^b (nm)		Lattice strain ^c	Ni contents ^d (wt%)
	a = b	c		Anatase	Ni		
TiO ₂	3.778	9.516	135.81	14.92	-	3.65 × 10 ⁻⁴	-
3wt% Ni/TiO ₂	3.791	9.506	136.63	11.41	3.99	7.35 × 10 ⁻⁴	2.92
6wt% Ni/TiO ₂	3.793	9.501	136.85	10.63	4.55	1.66 × 10 ⁻³	6.07
12wt% Ni/TiO ₂	3.801	9.475	136.87	8.37	3.98	4.03 × 10 ⁻³	13.5
20wt% Ni/TiO ₂	3.780	9.483	136.88	7.84	5.31	7.83 × 10 ⁻³	21.5

^aCalculated from the plane (101), (004), (100) by Bragg's equation.

^bCalculated from the plane (101) for anatase and (111) for Ni using Scherer's equation.

^cCalculated from Williamson-Hall (W-H).

^dCalculated from ICP-OES results.

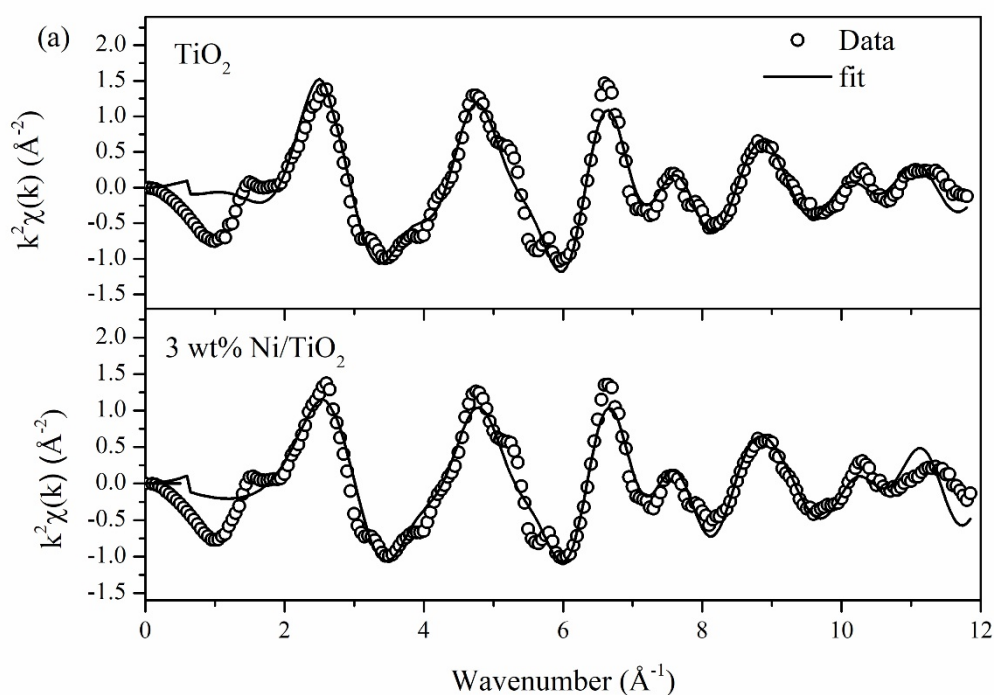
393 **3.2.2 Structural study of Ni-loaded TiO₂ catalysts by X-ray Absorption**
 394 **Spectroscopy (XAS)**

395 As discussed above that there were two forms of Ni addition; Ni substitution
 396 and NiO dispersion, thus in this part, X-ray Absorption Spectroscopy (XAS) results were used
 397 to evidence the existing of these two forms. The first form, substituted of Ni ion into TiO₂,
 398 Extended X-ray Absorption Fine Structure (EXAFS) of 3wt% Ni/TiO₂ was analyzed and
 399 discussed to confirm the substitution of Ni ion into TiO₂ lattice. Another form, dispersed NiO,
 400 linear combination between substitution Ni and NiO was performed and then the percentage
 401 composition of substitution Ni and NiO were determined.

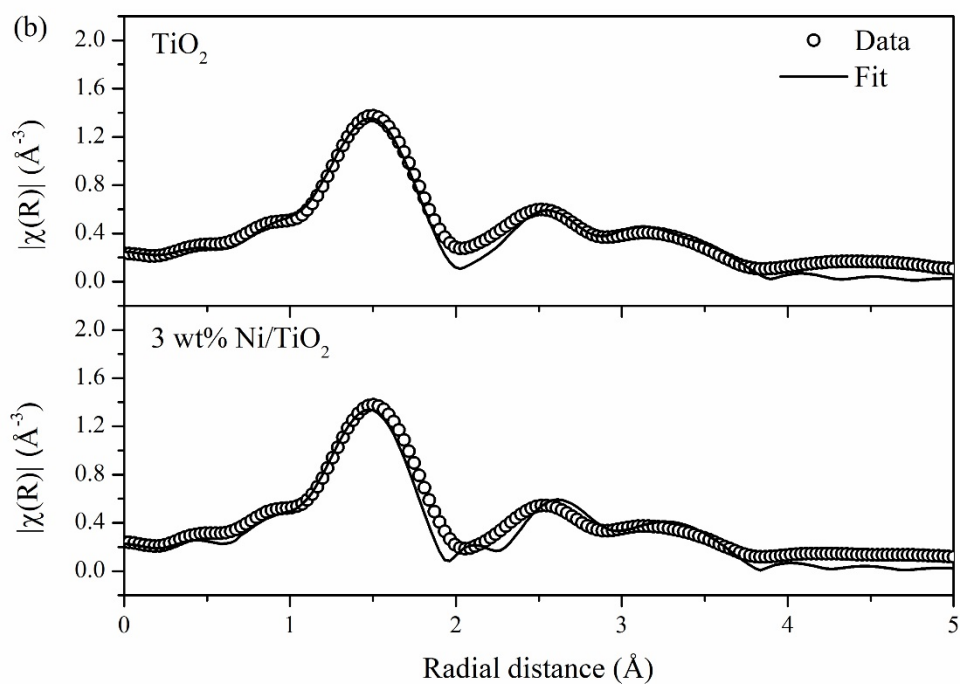
402 For Ni substitution form, analysis of EXAFS result of 3wt% Ni/TiO₂ was
 403 used since we assumed the complete substitution of all added Ni amount for this sample. The
 404 Ti K-edge EXAFS oscillation spectra and Fourier transformed functions for TiO₂ and 3wt%
 405 Ni/TiO₂ by using anatase TiO₂ (space group I4₁/amd) as a model for fitting are shown in Figure
 406 8. For Ni-loaded TiO₂ sample, the FT-EXAFS spectra were obtained by fitting with anatase
 407 TiO₂ and replacing one atom of the nearest Ti atom with Ni. The k²-weighted Fourier transform
 408 was used (Figure 8(a)). The weight EXAFS data in k space was in the range from 3-10 Å⁻¹ and
 409 was transformed to R space without phase collection (Figure 8(b)). The structural parameters
 410 of TiO₂ and 3wt% Ni/TiO₂ for Ti K-edge are summarized in Table 2. For Figure 8(b), the sharp
 411 peak of TiO₂ at ~1.5 Å was referred to the oscillation of Ti-O in the first shell. Upon

412 incorporation of Ni in second shell of TiO₂ lattice, the Ti–O bond distance were distorted (as
413 shown in Table 2), while the Ti–Ni bond length was elongated due to substitution by larger
414 cation and then resulted in shortening of Ti–Ti bond distance.

415 From the results of XRD, it can be seen that both Ni incorporated and
416 dispersed phase of NiO were observed, especially for high Ni content than 6wt%. In order to
417 determine the compositions of these two forms of Ni species, linear combination of XANES
418 spectra were employed. For this approach, we assumed that all Ni loading at 3wt% was
419 completely substituted into TiO₂ lattice, thus the excess amount of added Ni (higher than 3wt%)
420 was existed in NiO dispersed form. Therefore, the linear combination between substituted Ni
421 and dispersed NiO was constructed and was used as a standard to estimate the composition
422 between two forms of Ni species. For this work, Ni K-edge XANES spectrum of 3wt% Ni/TiO₂
423 was used as a standard for Ni substitution while that of NiO was used as a standard for NiO
424 dispersion phase. Figure 9 displays the percentage of two forms of Ni species obtained from
425 linear combination fitting. It was seen that NiO phase increased with Ni content which was
426 corresponding to XRD results.



427 **Figure 8** (a) Ti K-edge EXAFS oscillation data with $k^2\chi(k)$ for TiO₂ and 3wt% Ni/TiO₂.



428 **Figure 8** (b) Fourier transform data of Ti K-edge EXAFS for TiO₂ and 3wt% Ni/TiO₂.

429 **Table 2** Structural parameter for Ti K-edge EXAFS fitting analysis.

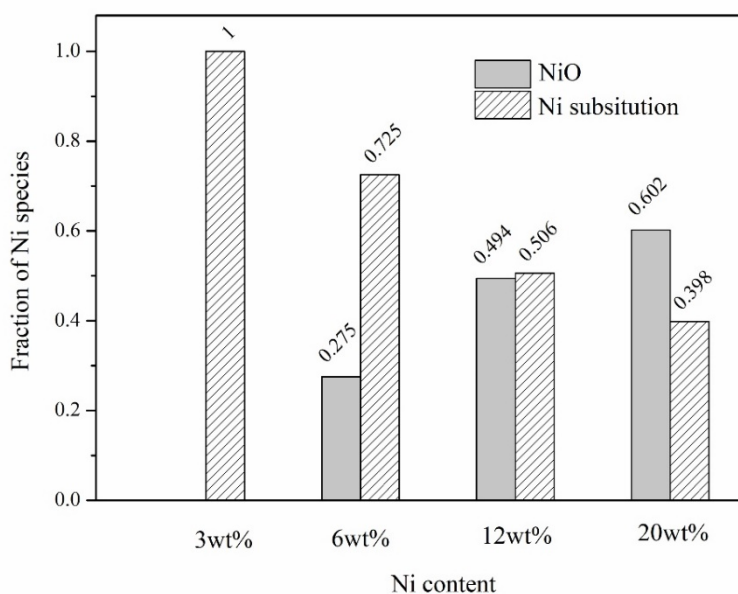
Samples	Shells	Best fit EXAFS parameters			
		N	σ^2	ΔE (eV)	R (Å)
TiO ₂	Ti-O	2.3	0.003	1.439	1.908
	Ti-O	1.2	0.001		1.996
	Ti-Ti	2.3	0.007		3.088
3wt% Ni/TiO ₂	Ti-O	1.9	0.001	1.474	1.927
	Ti-O	0.9	0.003		1.947
	Ti-Ti	1.4	0.002		3.051
	Ti-Ni	0.5	0.003		3.291

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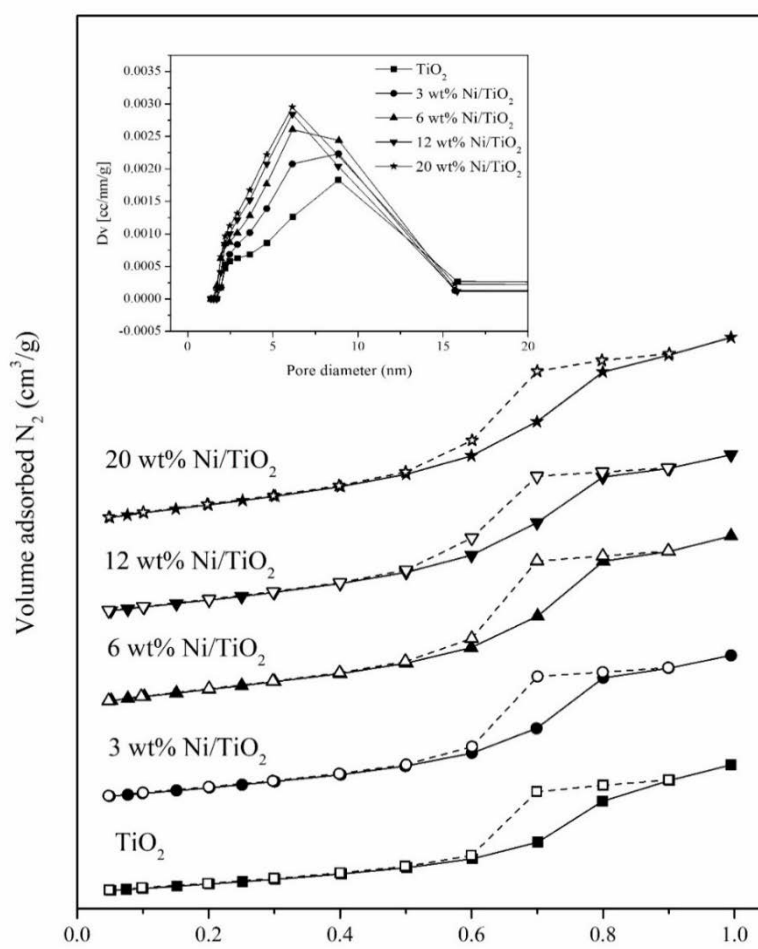
434 **Figure 9** The comparison between Ni substitution amount and NiO phase of all Ni-
 435 loaded catalysts which obtained from comparing to a linear combination of XANES data of
 436 Ni K-edge between 3wt% Ni/TiO₂ and NiO.

437 3.2.3 Surface and physical properties of catalysts

438 The surface properties of TiO₂ and modified-TiO₂ catalysts were estimated
 439 by N₂ adsorption-desorption technique. The N₂ adsorption-desorption isotherms are displayed
 440 in Figure 10. All isotherms exhibited an adsorption-desorption feature corresponded to
 441 mesoporous materials. The surface area, pore size and pore volume of samples are summarized
 442 in Table 3. The BET surface area of pure TiO₂ was 72 m²/g with pore volume of 0.15 cm³/g.
 443 The Ni modified TiO₂ samples exhibited higher BET surface area and pore volume than that
 444 of original TiO₂ upon increasing of Ni addition amount. The surface area of 3, 6, 12 and 20wt%
 445 Ni/TiO₂ were ca. 92, 95, 100 and 108 m²/g, respectively. These results were in agreement with
 446 XRD analysis which mentioned the prevention of anatase crystal growth by Ni addition and
 447 led to lowering of TiO₂ crystalline sizes. Higher surface area was important to improve the
 448 active surface sites to adsorb gas substrates for CO₂ methanation [29]. The pore size
 449 distributions of the synthesized samples were in the range of 2-16 nm (inset). The pore diameter
 450 distributions of TiO₂ and 3wt% Ni/TiO₂ gained the maximum value at 9 nm whereas higher Ni
 451 contents samples (6, 12 and 20wt% Ni/TiO₂) provided the smaller of diameter at a maximum
 452 peak around 6 nm. Pore volumes of Ni-added samples were higher than that of pure TiO₂.

453 The percentage of added Ni content in all Ni-loaded catalysts were
 454 determined by inductively couple plasma-optical emission spectroscopy and the results are
 455 summarized in Table 1. The obtained results of Ni content from ICP-OES in all Ni-loaded
 456 catalysts exhibited a similar value to nominated added amount (2.92, 6.07, 13.5 and 21.5wt%
 457 Ni for 3, 6, 12 and 20wt% Ni/TiO₂, respectively).

458 The morphology of fresh TiO₂ and Ni supported TiO₂ are showed in Figure
 459 11(a-e). The shape of particles for all samples were uniformly spherical and there was no
 460 change of the anatase morphology for all Ni catalysts. The means of anatase particle sizes were
 461 approximately in the range of 8-15 nm and the particle sizes were decreased upon increasing
 462 of Ni content which was in agreement with XRD results. NiO was characterized using high-
 463 resolution TEM for 20wt% Ni/TiO₂ resulting in Figure 11(f). The detection of lattice spacing
 464 0.210 nm can be indexed as (200) plane of FCC Ni which were related to XRD patterns. This
 465 result was confirmed the existence and dispersion of NiO.



484 **Figure 10** N₂ adsorption-desorption isotherms^o and pore size distribution (inset) of all
 485 catalysts; TiO₂, 3, 6, 12 and 20wt% Ni-loaded TiO₂. Solid symbol is adsorption
 486 branch and open symbol is desorption branch.

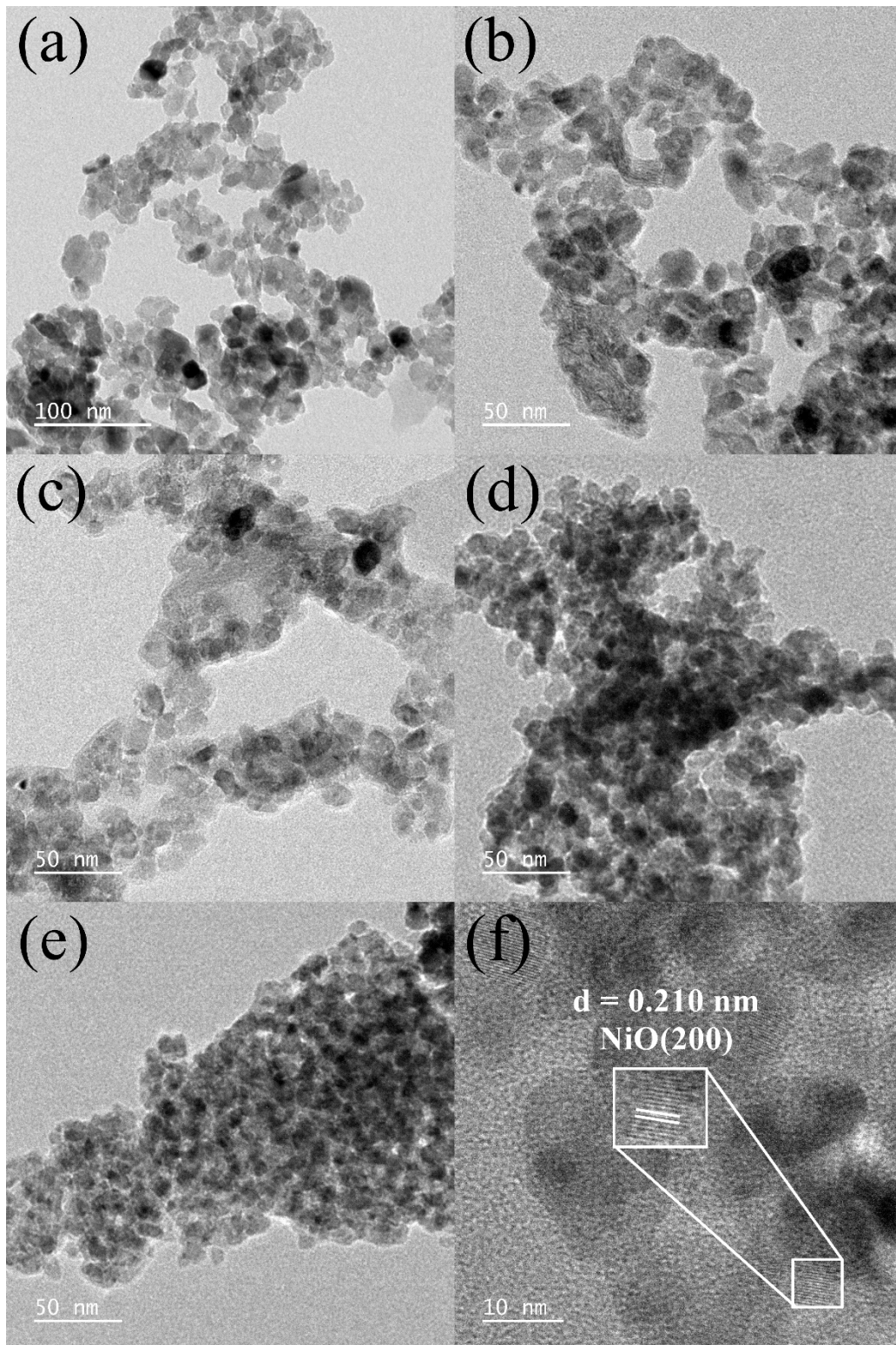
487 **Table 3** Surface properties of samples (TiO₂, 3, 6, 12 and 20wt% Ni/TiO₂).

Samples	Surface area ^a (m ² /g)	Pore volume ^a (cm ³ /g)	Pore size ^a (nm)	H ₂ consumption ^b (mmol/g)	H ₂ chemisorbed ^c (μmol/g)
TiO ₂	71.48	0.1490	8.340	-	-
3wt% Ni/TiO ₂	91.80	0.1743	7.596	13.9	5.8
6wt% Ni/TiO ₂	95.41	0.1776	7.745	31.1	13.8
12wt% Ni/TiO ₂	100.3	0.1749	6.947	62.5	25.5
20wt% Ni/TiO ₂	107.6	0.1934	7.191	76.2	43.4

^aEstimated from N₂ adsorption-desorption.

^bEstimated from H₂-TPR.

^cEstimated from H₂ chemisorption.



488 **Figure 11** TEM images of (a) TiO₂, (b) 3, (c) 6, (d) 12, (e) 20wt % Ni-loaded TiO₂ and
489 (f) HRTEM for 20wt % Ni-loaded TiO₂.

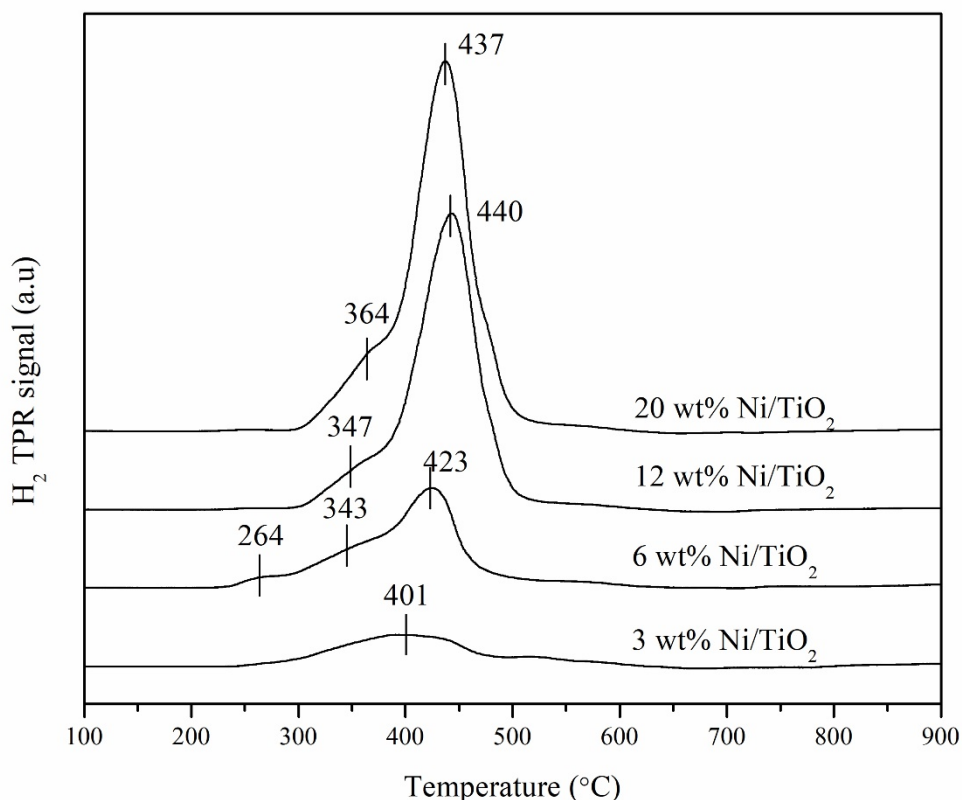
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3.3.3 H₂ Temperature Program Reduction (H₂-TPR) and H₂ chemisorption

H₂ Temperature Program Reduction (H₂-TPR) is attributed to the interaction of metal active site and support. Figure 12 demonstrates the TPR profiles of 3, 6, 12 and 20wt% Ni-loaded TiO₂ catalysts. The reduction peaks were observed in the range of 250-450 °C which were assigned as conversion of Ni²⁺ to Ni⁰ on surface which interacted with TiO₂ support [30,31]. The different reduction temperature of NiO was attributed to different degree of interaction between NiO and TiO₂ support. This behavior depended on the preparation method and calcination temperature since these factors caused a different NiO dispersion state, NiO sizes and degree of interaction with TiO₂ and thus resulted in different reducibility of NiO particles. The shifting of NiO reduction peaks to higher temperature were observed upon increasing of Ni content which indicated to high dispersion of NiO species with stronger interaction with TiO₂ support [32]. Moreover, H₂ consumption considerably enhanced with the increase of Ni additions as follows: 3 < 6 < 12 < 20wt% Ni/TiO₂ as shown in Table 3. The signal of H₂-TPR is referred to the amount of H₂ consumption, i.e., more intense of reduction peak implies to high H₂ consumption to reduce the metal in a catalyst. It was clearly seen that H₂ consumption was related to amount of NiO phase reduction to Ni metal; 20wt% Ni-modified TiO₂ provided 76.2 mmol/g of H₂ which was 5.5 times higher than H₂ consumption of 3wt% Ni-modified TiO₂.

In order to validate the exposed NiO and/or Ni active species on catalyst surface, H₂-chemisorption experiment was conducted. Due to H₂ favor to adsorb on metal active site, thus higher amount of H₂ chemisorbed can indicate to higher amount of metal active site. Table 3 illustrates the amount of H₂ chemisorbed of all catalysts. It is seen that the H₂ chemisorbed amount was increased with increasing of Ni loading on catalyst surface which corresponded to higher area of Ni adsorption sites for H₂. Since H₂ is one of the reactants for CO₂ methanation, thus the results of H₂ chemisorption was related to enhance of catalytic activities of Ni-loaded catalysts which will be discussed in the latter part.



518 **Figure 12** H₂ TPR profiles of TiO₂, 3, 6, 12 and 20wt% Ni-loaded TiO₂.

519 3.3 X-ray Absorption Spectroscopy analysis

520 From catalytic activities results, it was found that addition of Ni into TiO₂ lattice
 521 led to structural, physical and redox properties changing which also evidenced the altering of
 522 catalytic activities performance. In this part, X-ray Absorption Spectroscopy (XAS) was used
 523 to investigate the electronic state changing of Ni and Ti in freshly prepared catalysts and
 524 monitor the changing of the catalysts during CO₂ methanation. The changing of catalysts
 525 properties was studied by using data analysis of two regions; X-ray Absorption Near Edge
 526 Structure (XANES) and Extended X-ray Absorption Fine Structure (EXAFS). All of data
 527 reduction from XAS can be used to understand and describe the effect of Ni addition into TiO₂
 528 in improving CO₂ methanation.

529 3.3.1 X-ray Absorption Near Edge Structure (XANES) for monitoring Ni 530 and Ti oxidation states

531 This part is divided into two parts; 1) ex-situ XANES analysis for freshly
 532 synthesized catalyst and 2) in-situ XANES analysis for monitoring Ni and Ti oxidation state
 533 during CO₂ methanation.

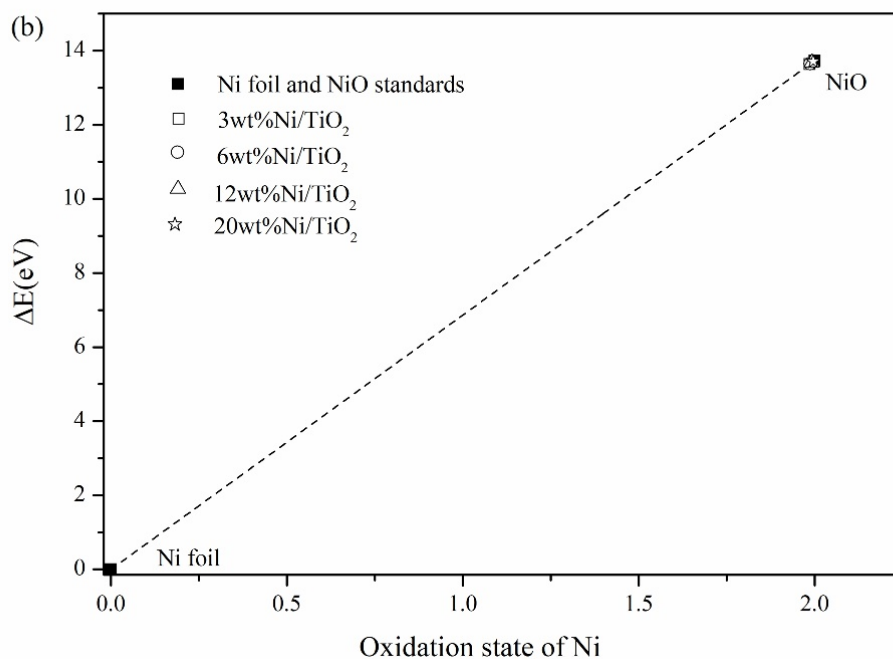
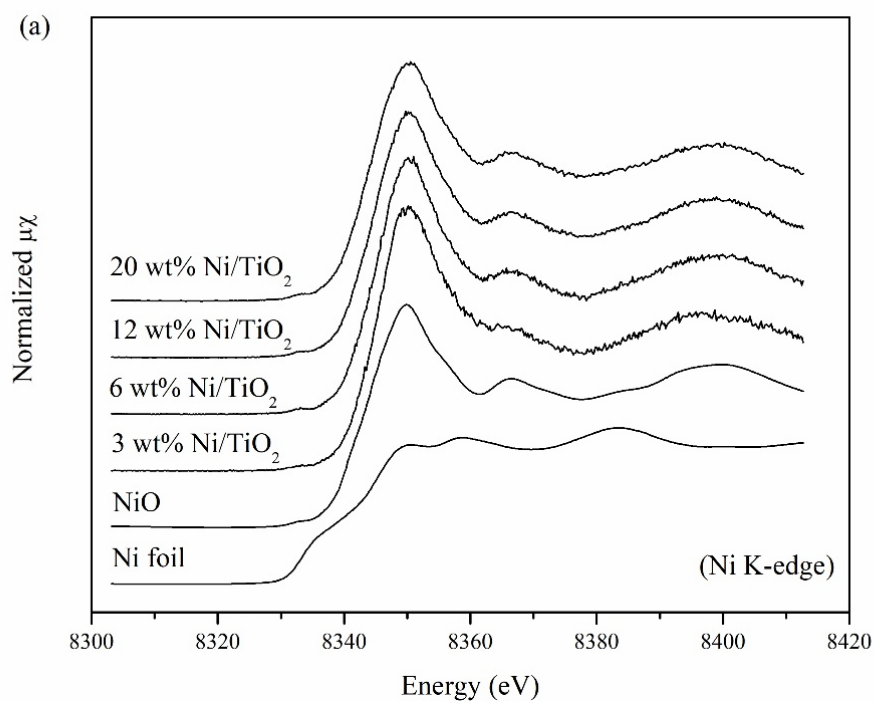
534 Figure 13(a) illustrates Ni K-edge XANES spectra of all freshly prepared
535 Ni-loaded TiO₂ catalysts compared with Ni foil and NiO standards. The Ni foil spectrum
536 exhibits an edge energy at around 8333 eV which is attributed to the electron transition from
537 1s to 3d orbital [33]. For NiO standard, the white line peak at 8347 eV was observed which
538 corresponded to multiple scattering process in octahedral NiO [33,34]. The oxidation state of
539 probe atom can be initially estimated by comparing with the standard XANES spectra with
540 certain oxidation state. From Figure 13(a), the Ni XANES spectra of Ni-loaded samples were
541 almost identical to NiO standard spectrum which indicated to oxidation state of Ni-loaded TiO₂
542 in all samples were 2+. Moreover, in order to elucidate the oxidation state of Ni, the relationship
543 between ΔE (different of edge energy relative with Ni foil) and oxidation state of Ni was
544 constructed. The edge energy of all samples can be obtained by taking the first derivative of
545 those spectra and the turning point was assigned as an edge energy. Figure 13(b) displays the
546 relationship between ΔE and oxidation states of Ni for all Ni-loaded samples compared with
547 two Ni standards. It can be seen that the apparent oxidation states of Ni for all Ni-loaded
548 samples were 2+. Normalized Ti K-edge XANES spectra of all fresh as-synthesized samples
549 display in Figure 14. Characteristic peaks of TiO₂ support and Ni-loaded samples were marked
550 as pre-edge peaks A₁–A₃ and B, a shoulder peak as C and a white line peak as D as shown in
551 Figure 14(a) [35]. The first pre-edge peak (A₁) was corresponding to a quadrupole transition to
552 3d (*t_{2g}*) state of the TiO₆ octahedron while that of A₃ and B peaks were assigned as dipole
553 transition from 1s to hybridized p-d state of *t_{2g}* and *e_g*, respectively. The A₂ was not noticed in
554 the larger size of anatase and rutile TiO₂, but this peak appeared in the case of the small sized
555 nanostructures [36]. The shoulder C was noted as transition shakedown process while that of
556 D was assigned as transition of 1s to out-of-plane 4p_z orbital [37]. The structures of all fresh
557 samples were detected as anatase TiO₂ structure with consisted of Ti⁴⁺ surrounding with six-
558 fold coordinate which were related to XRD results. It is known that formation of oxygen
559 vacancies within TiO₂ lattice lead to change of Ti⁴⁺ oxidation state to Ti³⁺ and distortion in the
560 local TiO₆ octahedral coordination [38]. In the first change, the changing of Ti⁴⁺ oxidation state
561 to Ti³⁺ was not observed from our results which might be due to small amount of metal addition
562 when comparing with bulk TiO₂ concentration, thus, the changing of Ti³⁺ was too small. On
563 the other hand, the distortion in TiO₂ octahedral is caused from elongation or shortening of Ti–
564 O bond length. Changing of Ti–O bond length lead to alteration of Ti 3d and O 2p density of
565 state (DOS) and the changing of Ti–O orbital overlap results in changing the total DOS and its
566 corresponding to electronic transition which reflected to the pre-edge feature [39]. The

567 changing of pre-edge feature is usually used to indicate the influence of metal substitution into
568 host lattice. The pre-edge intensities were related to the symmetry of the crystal lattice, thus, a
569 distortion and oxygen vacancies can cause an increase of pre-edge peaks [38-42]. Figure 14(b)
570 shows Ti K-edge in the pre-edge region for TiO₂ and 3wt% Ni/TiO₂ and 20wt% Ni/TiO₂. The
571 pre-edge intensities increased with increasing of Ni content which was attributed to higher
572 distortion and oxygen vacancies formation within TiO₂ lattice. This result can also confirm the
573 substitution of Ni²⁺ ions into Ti⁴⁺ sites which led to oxygen vacancies formation and distortion
574 of TiO₂ lattice. This phenomenon was in agreement with XRD and Ti EXAFS analysis for
575 3wt% Ni/TiO₂ as discussed in the first part.

576 Second part is in-situ XANES analysis for monitoring Ni and Ti oxidation
577 states during CO₂ methanation. In-situ Time-Resolve X-ray Absorption Spectroscopy (in-situ
578 TR-XAS) was used to obtain the XANES results. 20%wt Ni/TiO₂ synthesized sample
579 performed the highest CO₂ methanation activity, thus, this sample was selected to study by in-
580 situ TR-XAS.

581 The original Ni oxidation state of freshly prepared Ni catalysts was +2
582 which were confirmed by XANES results in previous part. In in-situ experiment, catalyst
583 surface was cleaned up and reduced by using H₂ during heating from 50 to 450 °C as shown in
584 Figure 15(a). During H₂ pretreatment, the characteristic peak of Ni species did not change in
585 the temperature range from 50 to 340 °C. After that, the white line intensity of Ni²⁺ (8347 eV)
586 obviously decreased and then the feature of spectra was slightly changed with increase of
587 pretreatment temperature and completely converted at 450 °C. After the temperature reached
588 to 450 °C, the pretreatment temperature was hold at this point for 90 min to maintain metallic
589 Ni active species before starting the reaction [43,44]. It indicated that the H₂ pretreatment
590 process resulted in reduction of NiO to Ni⁰. Figure 15(b) shows in-situ TR-XAS results of
591 20wt% Ni/TiO₂ during CO₂ methanation with H₂/CO₂ ratio = 4 at reaction temperature range
592 of 150 to 550 °C. The oxidation state of Ni remained unchanged during CO₂ methanation (Ni⁰)
593 within the whole range of reaction temperature.

594 Figure 16(a) shows in-situ TR-XAS results of Ti K-edge for 20wt% Ni/TiO₂ over H₂
595 pretreatment condition at 50 to 450 °C. It can be seen that the spectra of 20wt% Ni/TiO₂ under
596 pretreatment condition remained unchanged (Ti⁴⁺) and after conducting the CO₂ methanation,
597 the oxidation state of Ti was still unchanged which can be observed by the same spectra feature
598 of Ti K-edge for both pretreatment and reduction process (Figure 16(b)).

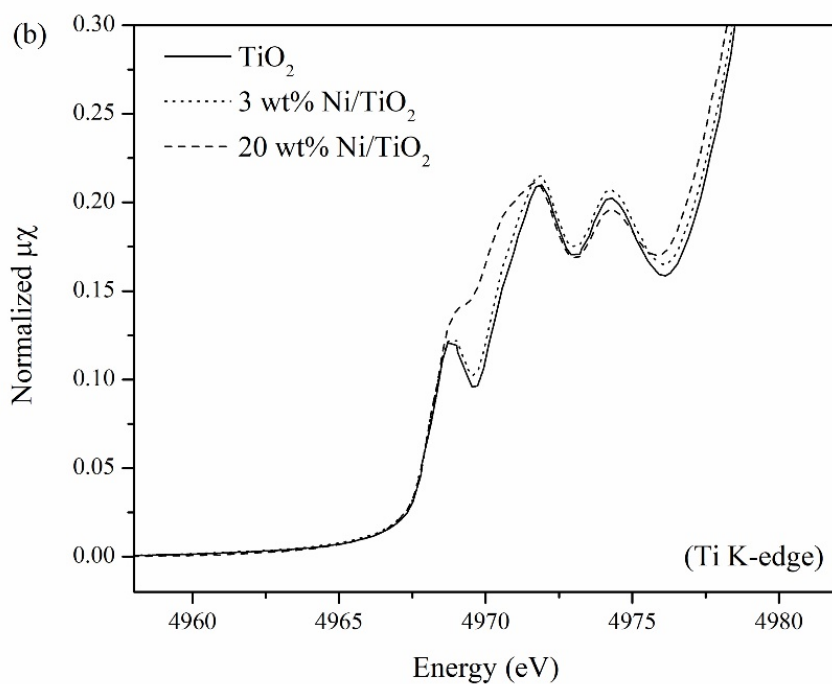
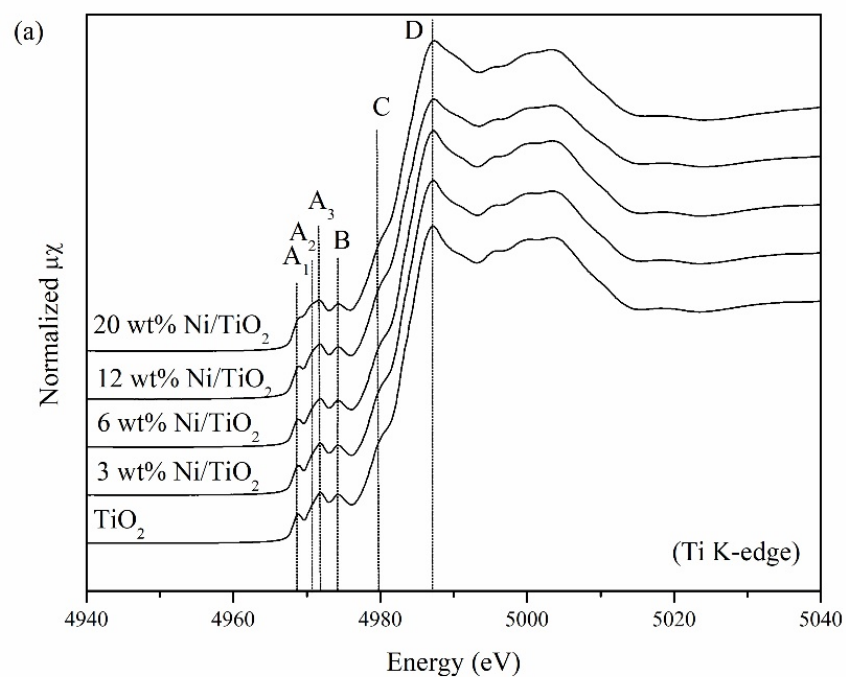


599 **Figure 13** (a) Normalized Ni K-edge XANES spectra of fresh samples (pure TiO₂, 3, 6, 12
 600 and 20wt% Ni/TiO₂) and references (NiO, Ni foil).

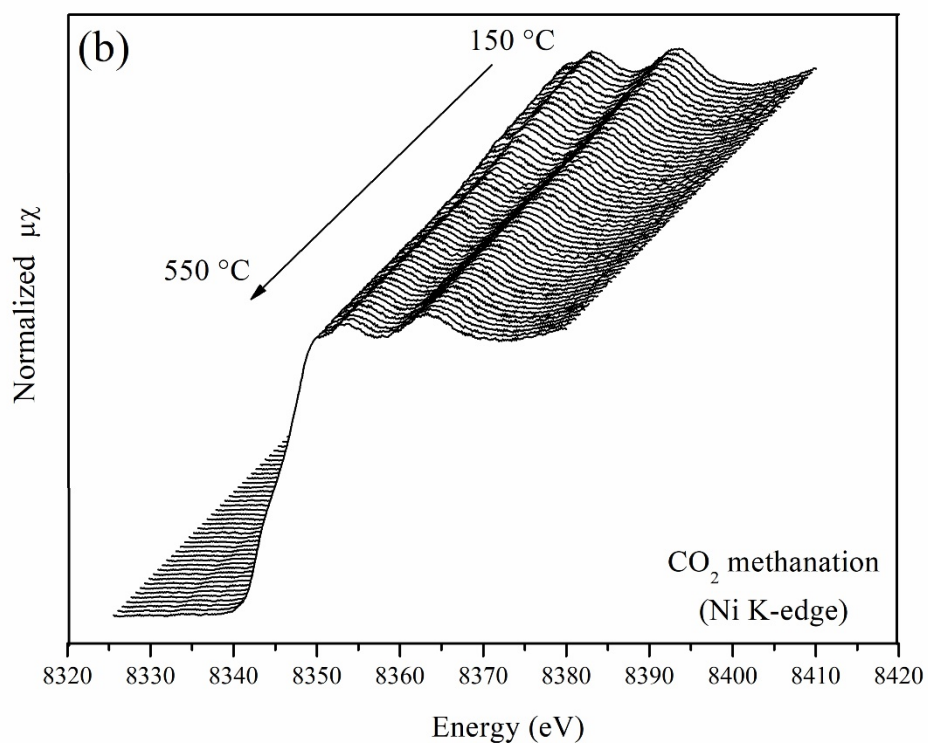
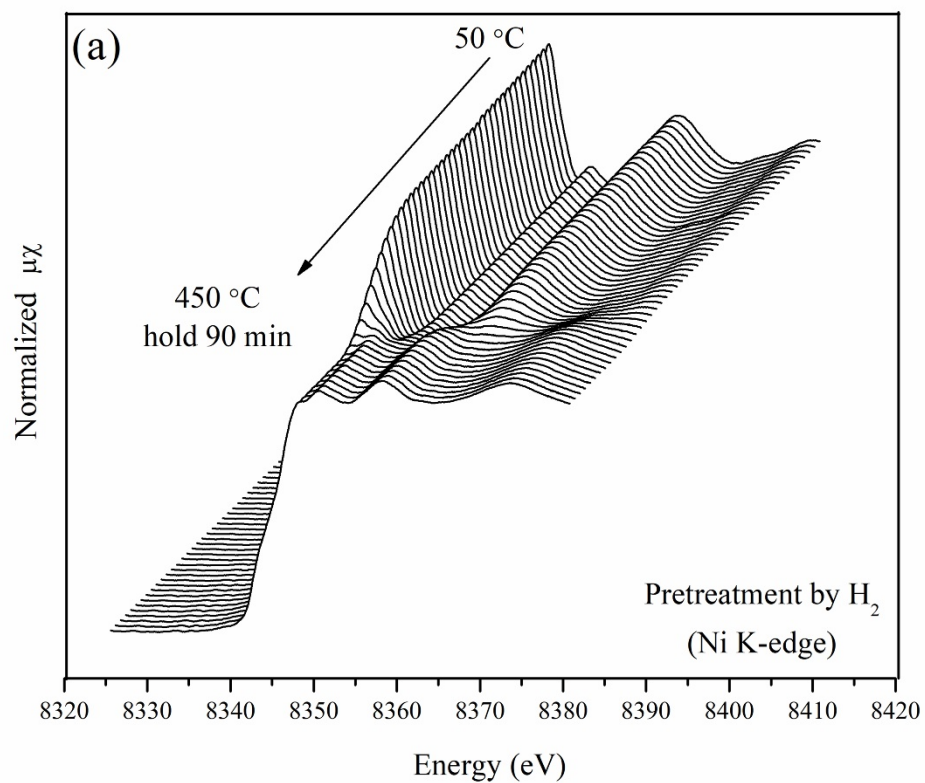
601 (b) Relationship between ΔE and oxidation state of Ni for determining the
 602 oxidation state of Ni in all synthesized catalysts.

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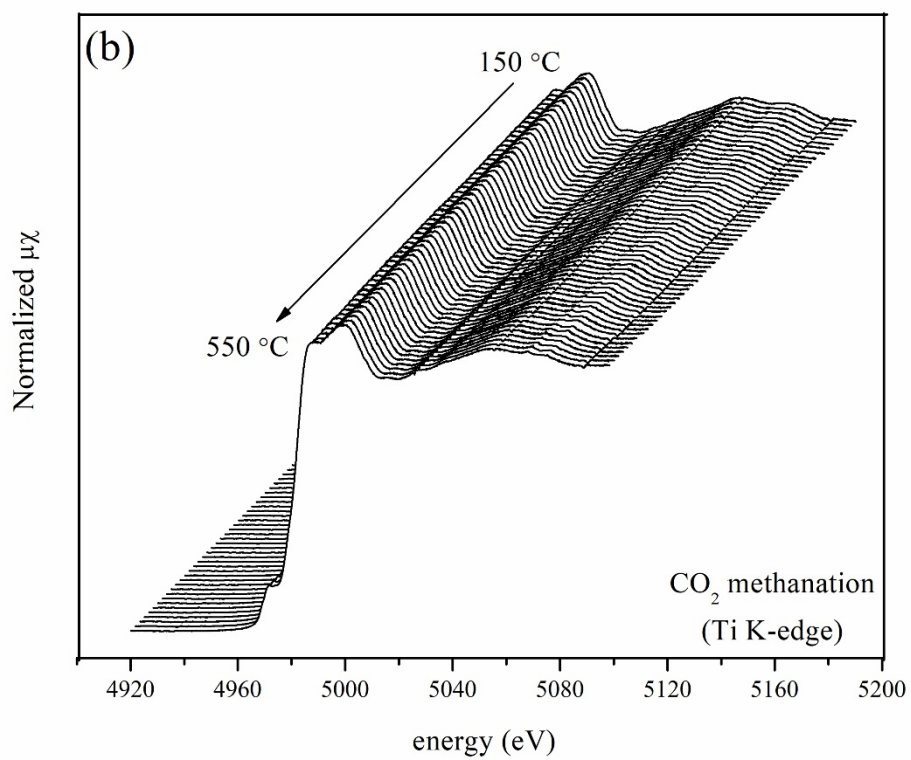
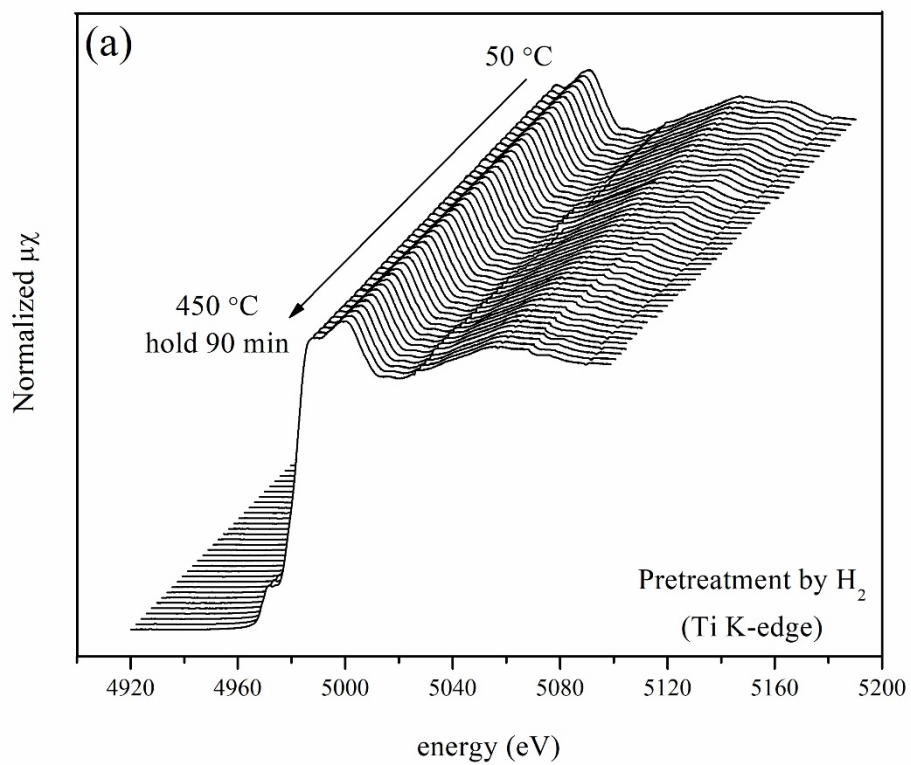
604



605 **Figure 14** (a) Normalized Ti K-edge XANES spectra of fresh samples (pure TiO₂, 3, 6,
 606 12 and 20wt% Ni/TiO₂).
 607 (b) Ti K-edge XANES spectra in pre-edge region of TiO₂, 3wt% Ni/TiO₂ and
 608 20wt% Ni/TiO₂.



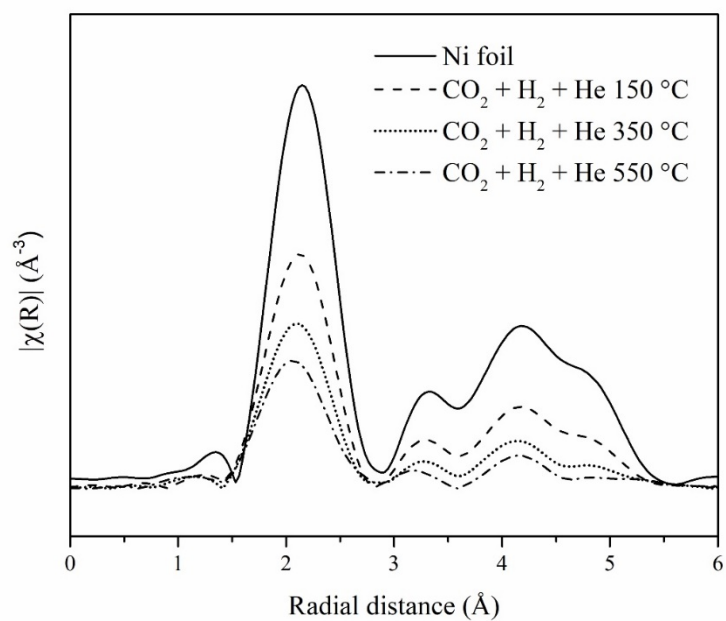
609 **Figure 15** in-situ Ni K-edge XANES spectra obtained from TR-XAS experiment of
 610 20% wt Ni/TiO₂: (a) pretreatment condition and (b) CO₂ methanation.



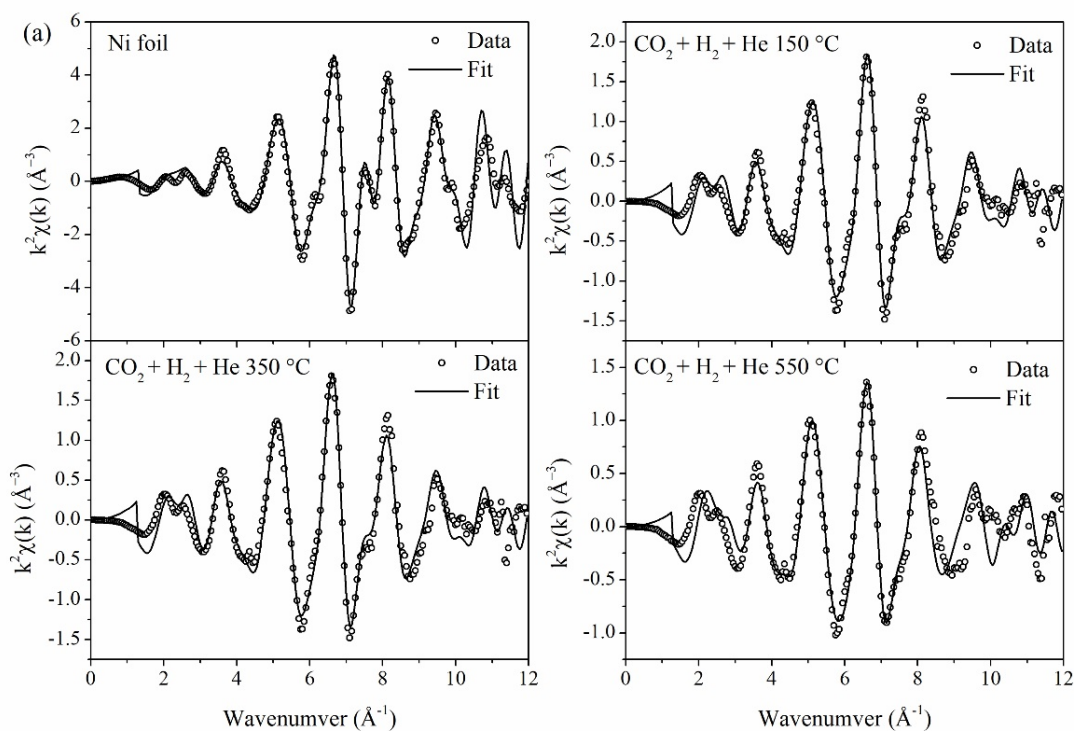
611 **Figure 16** in-situ Ti K-edge TR-XAS experiment of 20% wt Ni/TiO₂: (a) pretreatment
 612 condition and (b) CO₂ methanation.
 613

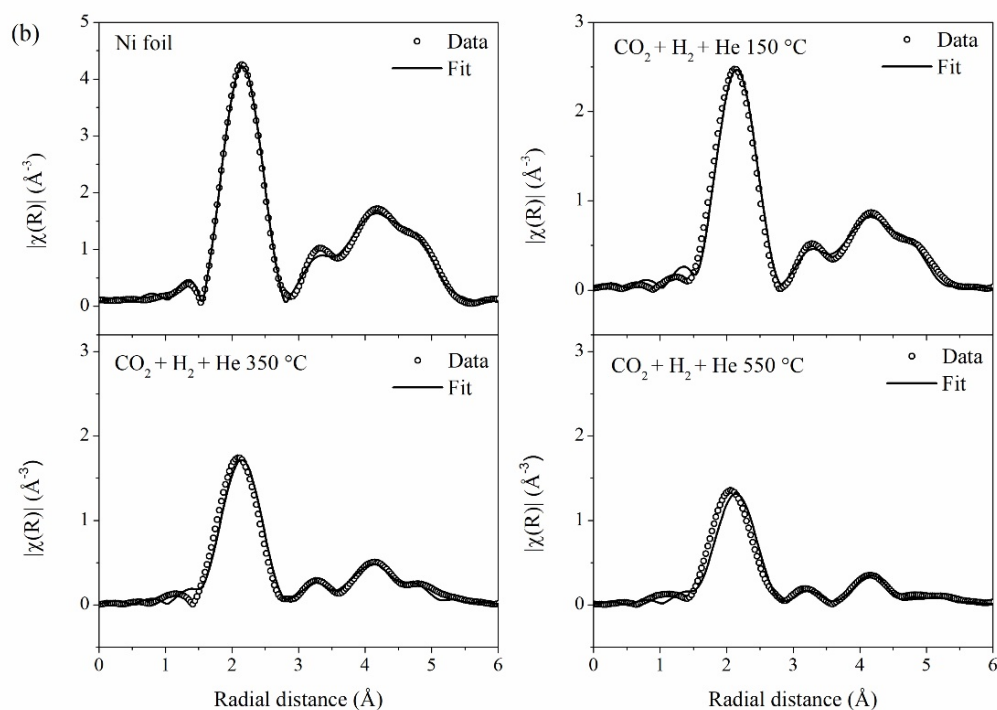
3.3.2 Extended X-ray absorption Fine Structure (EXAFS) study for monitoring Ni oxidation state during CO₂ methanation

Several works reported that Ni species on the catalyst surface is an important active site for enhancing CO₂ methanation rate since H₂ favors to adsorb on metal active site. Therefore, in this part, EXAFS analysis data was conducted to understand the state of Ni, structural changing of Ni during CO₂ methanation. The Fourier transformed function without phase correction of in-situ Ni K-edge EXAFS oscillation for 20wt% Ni/TiO₂ at the reaction temperature of 150, 350 and 550 °C is shown in Figure 17. The feature of all XANES spectra were similar to that of Ni foil. However, the amplitude of the spectra decreased with increased reaction temperature. Since, the amplitude is related to the coordination number around Ni probe atom, thus, EXAFS analysis can provide the information on the changing of Ni species during CO₂ methanation. The experimental data was normalized and the spectra were amplified by k² weight with R space window in the range of 1-5 Å and k range of 3-10 Å⁻¹. Figure 18(a) and 18(b) displays the k² weight and the radial function of Ni K-edge EXAFS spectra for 20wt% Ni/TiO₂, respectively. The calculated EXAFS of all catalysts were obtained by fitting with the cubic closed-packed (CCP) Ni structure. The best fitting analysis parameters are summarized in Table 4. The fitting results showed a first strongest peak at around 2.0 Å (2.48 Å from fitting) which assigned to a single scattering path of absorbing Ni atom with the nearest neighboring Ni atom (Ni–Ni). The calculated coordination number for Ni foil was 9.3 which indicated that Ni probe atom was surrounded by approximately 9 atoms. From Table 4, the calculated coordination number of 20wt% Ni/TiO₂ at 150, 350 and 550 °C were reduced to 7.6, 7.1 and 6.0, respectively. The lowering of coordination number of 20wt% Ni/TiO₂ with increasing of reaction temperature implied to higher level of unsaturated Ni–Ni coordination. High unsaturated of Ni–Ni coordination number can promote the dissociation of hydrogen on catalyst surface which can lead to high catalytic activity [45,46]. Therefore, high unsaturated Ni–Ni coordination number at high temperature (indicated by lowering of C.N.) provided high possibility of H₂ adsorption. However, the adsorption ability of H₂ on an active site was one of the factor to describe the catalytic performance. In order to describe the catalytic activities enhancement, understanding on the adsorption behavior of two reactants should be studied.



645 **Figure 17** Fourier Transform functions of *in-situ* EXAFS oscillation (without phase
 646 collection) of Ni foil and 20wt% Ni/TiO₂ at 150, 350 and 550 °C under CO₂
 647 methanation.





648 **Figure 18** (a) The k^2 weight EXAFS spectra of 20wt% Ni/TiO₂ at 150, 350 and 550 °C
 649 under CO₂ methanation.
 650 (b) The radial function of Ni K-edge EXAFS of 20wt% Ni/TiO₂ at 150, 350 and
 651 550 °C under CO₂ methanation.

652
 653 **Table 4** The best fitting parameters of 20wt% Ni/TiO₂.

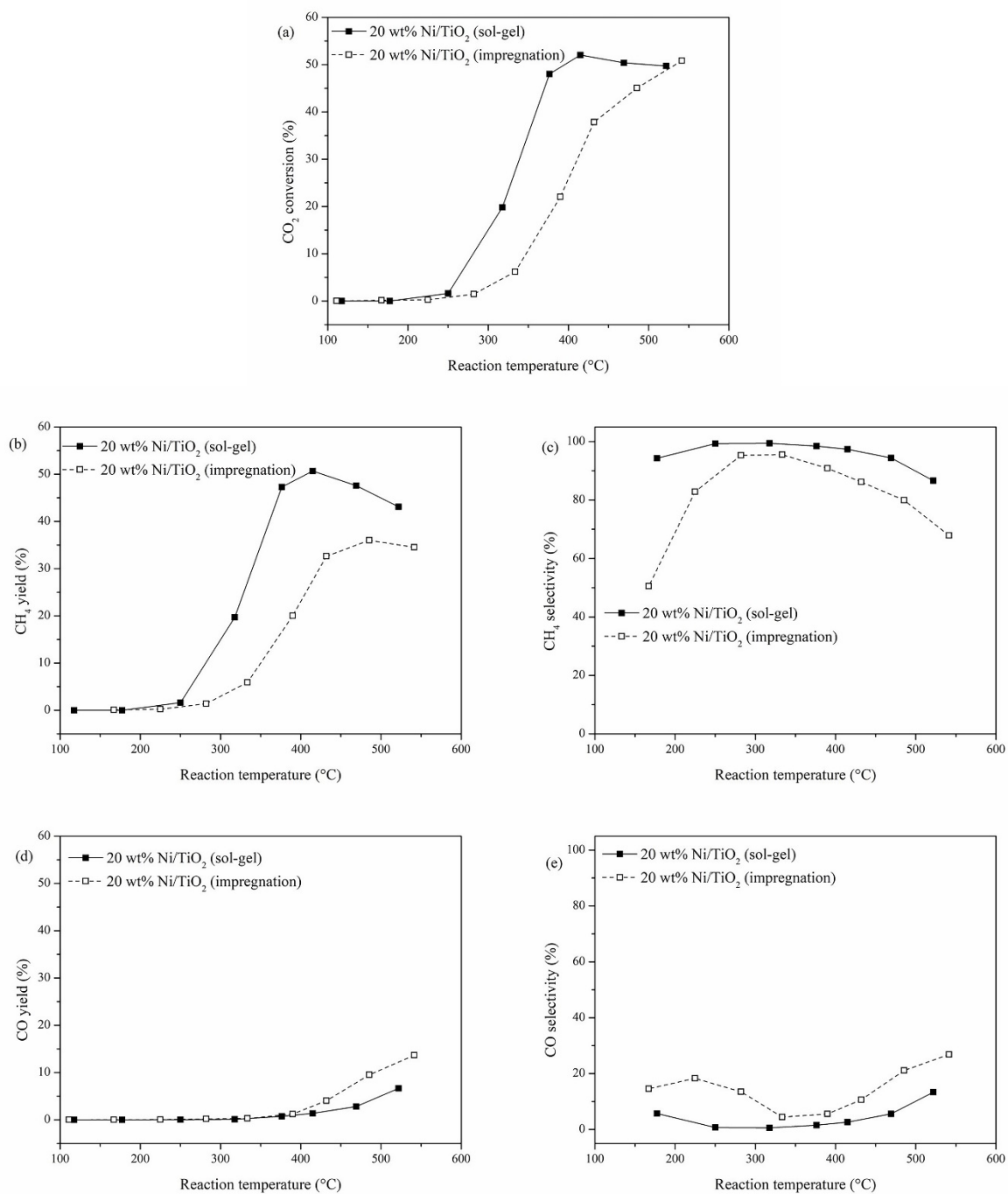
Samples	Shells	Best fit EXAFS parameters			
		N	σ^2	ΔE (eV)	R (Å)
Ni foil	Ni–Ni	9.3	0.0056	6.776	2.48
CO ₂ + H ₂ + He 150 °C	Ni–Ni	7.6	0.0088	6.417	2.48
CO ₂ + H ₂ + He 350 °C	Ni–Ni	7.1	0.0120	6.223	2.48
CO ₂ + H ₂ + He 550 °C	Ni–Ni	6.0	0.0130	6.240	2.48

654
 655 **3.4 The role of added Ni on enhancing CO₂ methanation**
 656 It has been reported that a different preparation method had a significant effect on
 657 physical and chemical properties of the catalysts which can be altered to the catalytic activities.
 658 In this work, sol-gel method was used to synthesize the catalyst for using in CO₂ methanation.

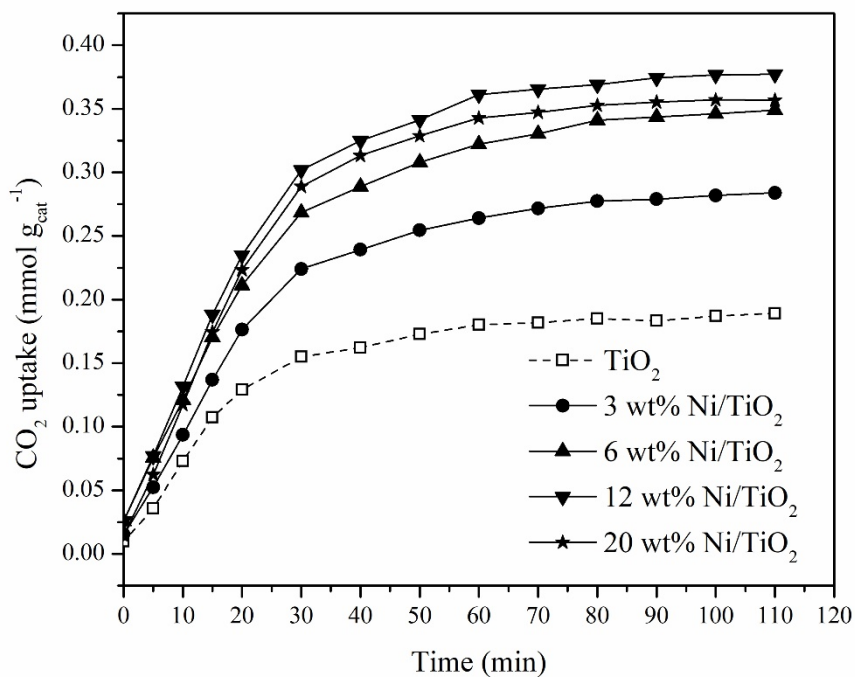
659 Since, this method is one of the reducible methods for preparing the catalyst, thus this method
660 was employed. In the catalytic reaction, adsorption of the reactants on the catalyst surface is
661 one of the most important elementary steps to control and dominate the catalytic reaction rate.
662 Therefore, to speed up the reaction rate, enhancing the reactants adsorption ability on catalyst
663 surface must be done. For CO₂ methanation, CO₂ and H₂ adsorption on catalyst surface can be
664 enhanced by different approach. For CO₂ molecules, providing the basicity sites on the catalyst
665 surface can increase the CO₂ adsorption capacity by acid-base interaction or existing of oxygen
666 vacancy can also improve the adsorption capacity [47-49]. While H₂ molecules, it is known
667 that H₂ favors to adsorb on metallic metal sites, especially on transition metals group, in
668 dissociative form, thus dispersing of metal active sites on the catalyst surface can increase the
669 H₂ adsorption capacity. Consequently, modification of the catalyst by above approaches would
670 lead to increase of adsorption rate of reactants and also speed up the reaction rate. Preparation
671 and modification by sol-gel method was used to alter the surface and catalyst properties which
672 can provide both oxygen vacancies and metal active sites for CO₂ and H₂ adsorption,
673 respectively. For CO₂ methanation, nickel was usually used to modify on support surface to
674 increase the activity of the catalyst. In this work, added Ni on TiO₂ support acted as a bi-
675 functional role in enhancing CO₂ methanation rate. First, addition of Ni during TiO₂ sol-gel
676 process led to incorporation of Ni²⁺ into TiO₂ lattice and resulted in strain and unbalanced
677 charge and then oxygen vacancy was formed (this result can be evidenced by EXAFS data.).
678 The oxygen vacancy was an active site for CO₂ adsorption; i.e., CO₂ interacted with oxygen
679 vacancies by strong interaction and weakened the C=O bond [50,51]. Second, the excess
680 amount of added Ni was existed in NiO form which might be located on the catalyst surface
681 and this NiO phase was an active site for H₂ adsorption (after reduction to Ni⁰ by pretreatment
682 process before starting the reaction). Moreover, increasing of NiO crystallinity upon increase
683 of Ni content was observed (increase of NiO peak intensity in XRD pattern) which resulted in
684 lowering or inhibiting of TiO₂ crystal growth [52]. From this effect, smaller crystalline size of
685 TiO₂ was occurred and higher surface area was observed. Figure 19(a)-19(c) illustrates the
686 comparison of CO₂ methanation catalytic activities between 20wt% Ni/TiO₂ obtained by sol-
687 gel and impregnation method. It is seen that addition of Ni into TiO₂ support by sol-gel method
688 exhibited higher catalytic activity than that by impregnation method. Although, impregnation
689 method is one of the simplest preparation method but the obtained catalyst properties were
690 lowered, especially surface area and porosity which was an important properties that can
691 dominate to the reduction rate. From Figure 19(d)-19(e), beside the higher catalytic activity of

692 catalyst obtained from sol-gel method, this catalyst was more selectively toward to CH₄ than
693 CO while the impregnated catalyst exhibited an opposite result.

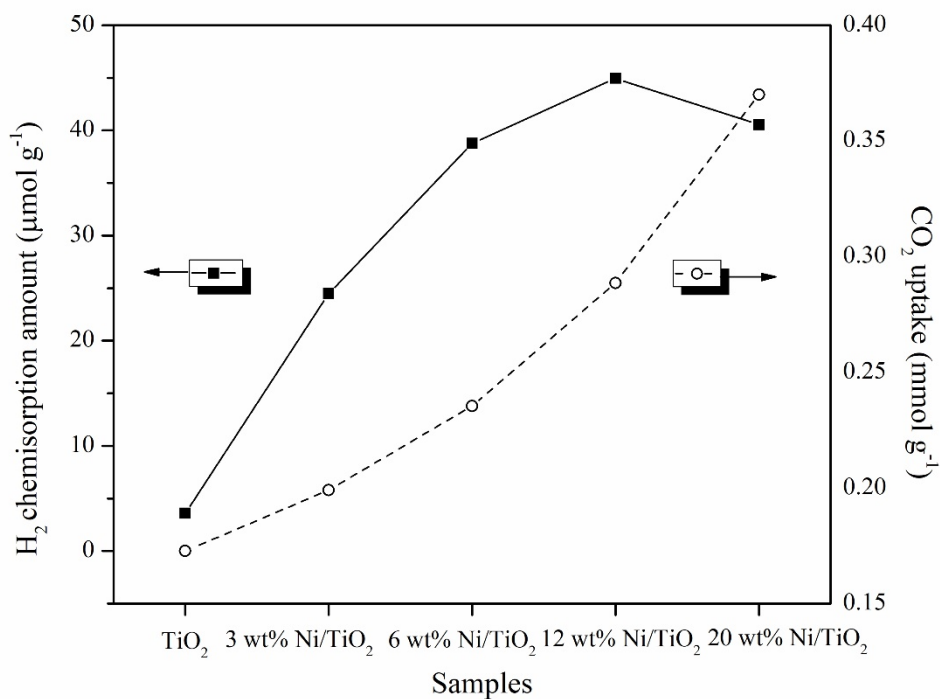
694 In order to evidence the role of added Ni in enhancing the reaction rate, CO₂ and
695 H₂ adsorption studies were conducted. Since, in catalyst cycle, the adsorption of reactants on
696 catalyst surface is usually assigned as a rate determining step, therefore, investigation of
697 adsorption process can provide the information that can be used to understand and describe the
698 role of catalyst in enhancing the reaction rate. For CO₂ methanation, increasing of CO₂ and H₂
699 adsorption can enhance the reaction rate, since, higher amount of adsorbed CO₂ and H₂ on
700 catalyst surface would lead to more possibility to react and convert to products. In the case of
701 H₂ adsorption, the results are illustrated in H₂-chemisorption part and are summarized in Table
702 3. It is assumed that only H₂ was adsorbed on Ni active site, thus higher amount of H₂
703 chemisorbed would be implied to higher Ni active area on catalyst surface. The results from
704 Table 3 showed that H₂ chemisorption amount was increased with increasing of Ni content
705 (5.80, 13.8, 25.5 and 43.4 μmol g⁻¹ for 3, 6, 12 and 20wt%, respectively). For CO₂ adsorption
706 study, thermogravimetric analysis (TGA) was used. The CO₂ uptake experiment was conducted
707 at 40 °C (the lowest temperature for the apparatus) and ambient pressure for 110 min. Figure
708 20 illustrates the data reduction from TGA experiment which expressed in the term of CO₂
709 uptake at interval time. The results showed that the CO₂ uptake increased with time and almost
710 constant within 60 min. The amount of CO₂ uptake was increased with Ni content (0.189, 0.248,
711 0.349 and 0.377 mmol g⁻¹ for 0, 3, 6 and 12wt% Ni, respectively). The results from XRD and
712 EXAFS indicated that added Ni²⁺ was substituted into TiO₂ lattice and led to formation of
713 oxygen vacancies which was reactive for CO₂ adsorption. However, increased Ni content up to
714 20wt%, the CO₂ uptake was slightly dropped. This result can be used to relate with the highest
715 catalytic activities of 20wt% Ni/TiO₂. Since, the adsorption sites for CO₂ and H₂ were co-exist
716 on the catalyst surface then if higher amount of adsorbed CO₂, inhibition and suppression of
717 H₂ adsorption on metal active site was probably occurred and then led to decrease of surface
718 reaction between adsorbed CO₂ and H₂ and finally lower the catalytic activity. Therefore, an
719 appropriate amount of adsorbed CO₂ and H₂ on catalyst surface would lead to high catalytic
720 activities. Figure 21 shows the relationship between H₂ chemisorption and CO₂ uptake as a
721 function of Ni content. It was found that 20wt% Ni/TiO₂ can maximize the CO₂ methanation
722 rate. This result caused from the improvement of catalyst properties by addition of Ni which
723 led to existence of active sites with appropriate performance to promote the CO₂ and H₂
724 adsorption.



725 **Figure 19** (a) CO₂ conversion, (b) CH₄ yield, (c) CH₄ selectivity, (d) CO yield and (e) CO
 726 selectivity of 20wt% Ni-loaded TiO₂ by preparing with different preparation
 727 methods; sol-gel and impregnation. The H₂/CO₂/He ratio was 24:6:10, total flow
 728 rate was 40 mL min⁻¹ and WHSV was 48,000 mL g⁻¹ h⁻¹ at reaction temperature
 729 100-550 °C.



730 **Figure 20** CO₂ uptake from TGA experiment for 0, 3, 6, 12 and 20wt% Ni/TiO₂.



731 **Figure 21** The relationship between H₂ chemisorption and CO₂ uptake as a function of Ni
 732 content.

733

734 4. Conclusion

735 In this work, the role of added Ni into TiO₂ support in enhancing CO₂ methanation rate
736 was studied. The Ni supported TiO₂ with different amount of Ni loading as 3, 6, 12 and 20wt%
737 Ni were synthesized by sol-gel method. The preparation method by addition of Ni during TiO₂
738 sol-gel process led to structural and surface properties changing which could alter to their
739 catalytic activities. The results show that added Ni modified the catalyst properties in 2 ways;
740 i) Ni²⁺ was substituted into TiO₂ lattice which led to strain and unbalanced charge and then
741 creating the oxygen vacancy. This effect can be evidenced by the results from distortion of Ti–
742 O, Ti–Ti bond length (EXAFS analysis) and the enlargement of TiO₂ unit cell (XRD results).
743 ii) upon increasing of Ni content, the excess amount of Ni was presented in NiO formed and
744 probably located on the catalyst surface. This can be seen from the diffraction peak of NiO in
745 XRD pattern. Moreover, the crystallinity of NiO increased with increasing Ni content which
746 could inhibit to TiO₂ crystal growth and resulted in smaller TiO₂ crystalline sizes and then
747 higher surface area was occurred. The existence of oxygen vacancy and NiO phase can be
748 evidenced by XANES linear combination analysis. The results showed that NiO phase was
749 increased with increasing of Ni loading. These two species that produced from addition of Ni
750 during TiO₂ sol-gel process has a significant role in enhancing the CO₂ methanation rate. Both
751 oxygen vacancy and NiO phase were an active site for promoting the adsorption of CO₂ and
752 H₂ on catalyst surface, high possibility of surface reaction between these two adsorbed species
753 were occurred and finally the reaction rate was enhanced. However, excess amount of adsorbed
754 CO₂ on surface can inhibit the H₂ adsorption and lead to lowering of CO₂ rate.

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761 Reference

- 762 [1] J. Gao, Y. Wang, Y. Ping, D. Hu, G. Xu, F. Gu, F. Su, RSC Adv. 2, (2012), 2358-2368.
763 [2] W. Li, H. Wang, X. Jiang, J. Zhu, Z. Liu, X. Gao, C. Song, RSC Adv. 8, (2018), 7651-
764 7669.
765 [3] J. Gao, Q. Liu, F. Gu, B. Liu, Z. Zhong, F. Su, RSC Adv. 5, (2015), 22759-22776.

- 766 [4] M.S. Duyar, A. Ramachandran, C. Wang, R.J. Farrauto, J. CO₂ Util. 12, (2015), 27-33.
- 767 [5] D.C. Upham, A.R. Derk, S. Sharma, H. Metiu, E.W. McFarland, Catal. Sci. Technol. 5,
768 (2015), 1783-1791.
- 769 [6] M. Schubert, S. Pokhrel, A. Thomé, V. Zielasek, T.M. Gesing, F. Roessner, L. Mädler,
770 M. Bäumer, Catal. Sci. Technol. 6, (2016), 7449-7460.
- 771 [7] P. Panagiotopoulou, Appl., Catal. A. 542, (2017), 63-70.
- 772 [8] J. Gao, Q. Liu, F. Gu, B. Liu, Z. Zhong, F. Su, RSC Adv. 5, (2015), 22759-22776.
- 773 [9] H.C. Wu, Y.C. Chang, J.H. Wu, J.H. Lin, I.K. Lin, C.S. Chen, Catal. Sci. Technol. 5,
774 (2015), 4154-4163.
- 775 [10] R. Zhou, N. Rui, Z. Fan, C.J. Liu, Int. J. Hydrogen Energy. 41, (2016), 22017-22025.
- 776 [11] J. Ashok, M.L. Ang, S. Kawi, Catal. Today. 281, (2017), 304-311.
- 777 [12] L.A.A. Varilla, N. Seriani, J.A. Montoya, J. Mol. Model. 25, (2019), 1-8.
- 778 [13] Y.X. Pan, C.J. Kia, D. Mei, Q. Ge, Langmuir. 26, (2010), 5551-5558.
- 779 [14] X. Li, Z.J. Zhao, L. Zeng, J. Zhao, H. Tian, S. Chen, K. Li, S. Sang, J. Gong. Chem.
780 Sci., 9, (2018), 3426-3437.
- 781 [15] B. Li, H. Metiu, J. Phys. Chem. 114, (2010), 12234-12244.
- 782 [16] X. Yan, Y. Liu, B. Zhao, Z. Wang, Y. Wang, C.J. Lia, Int. J. Hydrogen Energy. 38,
783 (2013), 2283-2291.
- 784 [17] A. Niltrarach, S. Kityakarn, A. Worayingyong, J.T. Thienprasert, W. Klysubun, P.
785 Songsiriritthigul, S. Limpijumnong. Phys B Condens Matter. 407, (2012), 2915-2918.
- 786 [18] X. Su, J. Xu, B. Liang, H. Duan, B. Hou, J. Energy. Chem. 25, (2016), 553-565.
- 787 [19] Q. Liu, J. Gao, F. Gu, X. Lu, Y. Liu, H. Li, Z. Zhong, B. Liu, G. Xu, F. Su, J. Catal. 326,
788 (2015), 127-138.
- 789 [20] P. Kidkhunthod, Adv. Nat. Sci. Nanosci. Nanotechnol. 8 (2017).
- 790 [21] W. Klysubun, P. Kidkhunthod, P. Tarawarakarn, P. Sombunchoo, C. Kongmark, S.
791 Limpijumnong, S. Rujirawat, R. Yimnirun, G. Tumcharern, K. Faungnawakij, J.
792 Synchrotron Radiat. 24, (2017), 707-716.

- 793 [22] X. Gao, A. Traitangwong, M. Hu, C. Zuo, V. Meeyoo, Z. Peng, C. Li, *Energy Fuels*. 32,
794 (2018), 3681-3689.
- 795 [23] V.R. Akshay, B. Arun, G. Mandal, M. Vasandhara, *Phys. Chem. Chem. Phys.* 21,
796 (2019), 2519-2532.
- 797 [24] M. Manzoor, A. Rafiq, M. Ikram, M. Nafees, S. Ali, *Int. Nano Lett*, 8, (2018), 1-8.
- 798 [25] I. Ganesh, A.K. Kumar, P.P. Kumar, P.S.C. Sekhar, K. Radha, G. Padmanabham, G.
799 Sundararajan, *Sci World J.* 2012, 2012, 13-20.
- 800 [26] P.M. Shirage, A.K. Rana, Y. Kumar, S. Sen, S.G. Leonardi, G. Neri, *RCS Adv.* 6,
801 (2016), 82733-82742.
- 802 [27] K.C. Chanapatttharapol, S. Krachumram, Y. Poo-arporn, *Solid State Sci.* 99, (2020),
803 106066.
- 804 [28] S. Tiwari, G. Bajpai, T. Srivastava, S. Viswakarma, P. Shirage, S. Sen, S. Biring, *Scr.*
805 *Mater.* 129, (2017), 84-87.
- 806 [29] S.V. Moghaddam, M. Rezaei, F. Meshkani, R. Daroughegi, *Int. J. Hydrogen Energy.* 43,
807 (2018), 16522-16533.
- 808 [30] J. van de Loosdrecht, A.M. van de Kraan, A.J. van Dillen, J.W. Geus, *J. Catal.* 170,
809 (1997), 217-226.
- 810 [31] S.W. Ho, C.Y. Chu, S.G. Chen, *J. Catal.* 178, (1998), 34-48.
- 811 [32] J. Liu, C. Li, F. Weng, S. He, H. Chen, Y. Zhao, M. Wei, D.G. Evans, X. Duan, *Catal.*
812 *Sci. Technol.* 3, (2013), 2627-2633.
- 813 [33] X. Meng, C. Wan, Y. Wang, X. Ju, *J. Alloys Comp.* 735, (2018), 1637-1647.
- 814 [34] K.I. Pandya, R.W. Hoffman, J. McBeen, W.E. O'Grady, *J. Electrochem. Soc.* 137, 1990,
815 383-388.
- 816 [35] F. Farges, G.E. Brown Jr, J.J. Rehr, *Phys. Rev. B.* 56, (1997), 1809-1819.
- 817 [36] A. Sharma, M. Varshney, H.J. Shin, B.H. Lee, K.H. Chae, S. O. Won, *Mater. Chem.*
818 *Phys.* 191, (2017), 129-144.
- 819 [37] J.H. Yang, H. Piao, A. Vinu, A.A. Elzatahty, S.M. Paek, J.H. Choy, *RCS, Adv.* 5,
820 (2015), 8210-8215.

- 821 [38] B. Bharti, S. Kumar, H.N. Lee, R. Kumar, *Sci Rep*, 6, (2016), 1-12.
- 822 [39] M. Sahoo, A.K. Yadav, S. Ghosh, S.N. Jha, D. Bhattacharyya, T. Mathews, *Phys.*
823 *Chem. Chem. Chem. Phys.* 21, (2019), 6198-6206.
- 824 [40] B.J. Hsieh, M.C. Tsai, C.J. Pan, W.N. Su, J. Rick, J.F. Lee, Y.W. Yang, B.J. Hwang,
825 *NPG Asia Mat*, 9, (2017), e403.
- 826 [41] K. Huang, K. Sasaki, R.R. Adzic, Y. Xing, *J. Mater. Chem.* 22, (2012), 16824-16832.
- 827 [42] H. Song, T.G. Jeong, Y.H. Moon, H.W. Chun, K.Y. Chung, H.S. Kim, B.W. Cho, Y.T.
828 Kim, *Sci. Rep.* 4, (2015), 1-8.
- 829 [43] L. Huang, L. Yan, M. Tang, G. Wang, Z. Qin, H. Ge, *ACS Omega*, 3, (2018), 18967-
830 18975)
- 831 [44] J.A. Rodriguez, J.C. Hanson, A.I. Frenkel, J.Y. Kim, M. Pe´rez, *J. Am. Chem. Soc.*,
832 124, (2002), 346-354.
- 833 [45] D. Hu, J. Gao, Y. Ping, L. Jia, P. Gunawan, Z. Zhong, G. Xu, F. Gu, F. Su, *Ind. Eng.*
834 *Chem. Res.* 51, (2012), 4875-4886.
- 835 [46] P. van Helden, J.A. van den Berg, I.M. Ciobîˆcaˆ, *Catal. Sci. Technol.* 2, (2012) 491-
836 494.
- 837 [47] W. Göpel, G. Rocker, R. Feierabend, *Phys. Rev. B.* 28, (1983), 3427-3438.
- 838 [48] J. Xu, X. Su, H. Duan, B. Hou, Q. Lin, X. Liu, X. Pan, G. Pei, H. Geng, Y. Huang, T.
839 Zhang, *J. Catal.* 333, (2016), 227-237.
- 840 [49] X. Jia, X. Zhang, N. Rui, X. Hu, C. Liu, *Appl. Catal. B Environ.* 244, (2019), 159-169.
- 841 [50] F. Wang, S. He, H. Chen, B. Wang, L. Zheng, M. Wei, D.G. Evans, X. Duan, *J. Am.*
842 *Chem. Soc.* 138, (2016), 6298-6305.
- 843 [51] M.Y.S. Hamid, M.L. Firmansyah, S. Triwahyono, A.A. Jalil, R.R. Mukti, E.
844 Febriyanti, V. Suendo, H.D. Setiabudi, M. Mohamed, W. Nabgan, *Appl. Catal. A.* 532,
845 (2017), 86-94.
- 846 [52] M. Pudukudy, Z. Yaakob, A. Kadier, M.S. Takriff, N.S.M. Hassan, *Int. J. Hydrogen*
847 *Energy.* 42, (2017), 16495-16513.