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ARTICLE

Host and guest joining forces: a holistic approach for metalorganic frameworks in nonlinear optics⁺

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Metal-organic frameworks (MOFs) are an interesting candidate for nonlinear optics (NLO) application. However, current design strategies for MOFs in NLO are typically limited to either engineering the MOF itself or using the MOF to align NLO-active molecules within its pores. But more design factors can be considered when engineering MOFs and choosing guest molecules. The NLO emission of host and guests can be combined instead of using only either one of the two. The interaction between host and guest can be source for further improvement by changing the symmetry, dipole moment, bond lengths, charge distribution, etc. Additionally, unstable NLO molecules can potentially be stabilized by accommodation in the MOF. Here, we demonstrate a new strategy for MOFs by combining MOF-177 as a host and Li@C₆₀ as guest for NLO meaning both the MOF itself as well as guest molecules are emitting NLO signal, as well as further increasing the emission using their interaction elongating bonds within the MOF, thus fully using the MOFs potential for second harmonic generation. Using this approach, the overall emission can be boosted by 40% compared to MOF-177 alone. Furthermore, the accommodation into MOF-177 stabilizes Li@C₆₀ that is normally unstable under ambient conditions without a counterion.

Introduction

Metal-organic frameworks (MOFs) have attracted great attention in applications such as gas storage and gas separation, catalysis or sensors due to their large surface area.¹⁻⁷ Recently, MOFs are further considered in an area of applications even larger than these primary fields. This is because their structure, consisting of metal ions and organic linkers, allows facile modification for various purposes by changing or modifying the linkers and/or metal nodes. In this way, many properties of MOFs can be tailor-made to suit exactly the intended application. Accommodation of metallofullerenes is an interesting way of further controlling the properties of MOFs and their accommodated guests.⁸⁻¹⁰

A more recent field for MOFs is nonlinear optics (NLO).¹¹⁻¹³ NLO has played a central role in modern photonics for a variety of applications ranging from laser frequency conversion, ultrafast information processing, sensors, switches, laser amplifiers etc.¹⁴⁻¹⁸ Current materials in NLO are often based on inorganic materials like salts and semiconductors.¹⁹ However, these materials have disadvantages like low optical damage threshold, being difficult and expensive to manufacture.²⁰ For this reason, there has been a push to use organic materials as they allow flexible design, good NLO properties and high optical damage threshold.²⁰ MOFs as a class of inorganic-organic hybrids can combine the best of both worlds. A different strategy uses the ability of MOFs to accommodate NLO active molecules into their pores. Such a strategy can be viable for molecules that have strong NLO properties but poor crystallinity.

However, the combined emission of MOF and guest molecules has not been scrutinized so far. This means that either the pores of the MOF or the MOF itself are not efficiently used. Overall NLO performance could be enhanced if both NLO active MOF and guest are combined to obtain the desired emission. Furthermore, interaction between host and guest can further improve emission, *e.g.* by increasing the dipole moment or causing structural changes that increase non-centrosymmetry. Finally, accommodation in the MOF can potentially stabilize NLO-active molecules that would otherwise be unstable under ambient conditions or are difficult to crystalize with high quality. Considering all of these factors means fully taking advantage of the design flexibility of MOFs and organic compounds in general.

Here, we demonstrate how choosing suitable MOF and guest pair can significantly improve NLO emission and fully use the MOF volume. The guest molecule chosen in this study is lithium encapsulating C₆₀ (Li@C₆₀). It has been shown previously that Li@C₆₀ has superior NLO properties compared to ordinary C₆₀.²¹⁻²³ As Li@C₆₀ is a relatively large molecule, a MOF with large pores is necessary to accommodate it.²⁴⁻²⁶ For this reason, MOF-177 was chosen for its capability to accommodate C₆₀.^{8,9,27} Second-harmonic generation (SHG) can be increased by 40% by accommodating Li@C₆₀ into MOF-177. By additionally comparing the performance with C₆₀ accommodating MOF-177, we discuss the improvement of NLO emission by considering NLO of the guest molecule itself and the interaction between Li@C₆₀ and MOF-177.

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Fig. 1 a) SEM image of $Li@C_{60}$ CMOF-177. b) EDX spectrum obtained. Only the small amounts of PF₆ are detected. All is due to the sample holder, Ag is due to metal sputtering.

Results and discussion

Characterization of Li@C₆₀⊂MOF-177

 $Li@C_{60}$ was accommodated into MOF-177 (Figure S1) by dissolving $Li^+@C_{60}(PF_6^-)$ powder in a 1:1 (v:v) mixture of odichlorobenzene (oDCB) and acetonitrile (ACN) (see experimental section for details). In order to explore the accommodation of Li@C₆₀ into MOF-177 more, we performed scanning electron microscopy (SEM) with energy dispersive Xray (EDX) spectroscopy (see Figure 1). As can be seen from the EDX spectrum (Figure 1 b), there are only small amounts of phosphorus (P) and fluorine (F) present. In the powder of Li⁺@C₆₀(PF₆⁻), PF₆⁻ is used as a counterion in order to balance the positive charge of Li⁺. It is known that Li⁺@C₆₀ is not stable under ambient conditions unless paired with a counterion.28 Therefore, the small amount of PF₆ indicates that another molecule, i.e. solvent molecules, or the MOF itself is serving as a counterion. It should be noted that chlorine (Cl) is present in the samples, meaning oDCB is co-accommodated. However, no nitrogen (N) that would point to ACN is detected.

We carried out attenuated total reflection measurement of Fourier-transform infrared spectroscopy (FT-IR) to further investigate the solvent molecules being accommodated. The results can be seen in Figure S2. FTIR shows the presence of both PF₆ and oDCB in Li@C₆₀⊂MOF-177, however no ACN is accommodated. Similarly, C₆₀⊂MOF-177 does show some amount of toluene accommodated.

Because Li is not detectable in EDX, inductively coupled plasma atomic emission spectroscopy (ICP-OES) was performed (Figure S3) showing a content of Li. The quantitative analyses of Li by ICP-OES is difficult because of the ionization interreference of Li with Na (added in the solution to be pH7).²⁹ The chemical formula of $\text{Li}@C_{60} \subset \text{MOF-177}$ was estimated to be $\{\text{Zn}_4O(\text{btb})_2 \bullet 0.3 \quad [(\text{Li}@C_{60})(\text{PF}_6)] \bullet (\text{oDCB})\}_n \quad \text{by}$

thermogravimetry (Figure S4, btb: benzene-1,3,5-tribenzoate). A single unit cell of MOF-177 ($[Zn_{32}O_8(btb)_{16}]_n$) can accommodate 16 C₆₀ molecules according to literature³⁰, meaning that 15 % of the available space is filled with $Li@C_{60}$. To further investigate the interaction of MOF-177 and Li@C₆₀, we have performed Raman spectroscopy (Figure 2). An interesting feature is the disappearance of the $A_{\alpha}(2)$ mode, normally positioned at 1470 cm⁻¹ and shifted slightly lower by 3 cm⁻¹ in case of Li@C₆₀[PF_6^-] powder²¹, after the accommodation into MOF-177 (Figure 2a, red highlighted and inset). This disappearance of the $A_q(2)$ mode cannot be seen after dissolving Li@C₆₀ in a mixture of ACN/oDCB (see Figure S5). This suggests that Li@C₆₀ is not interacting with either oDCB or ACN, despite it being co-accommodated, but actually with the MOF itself, suggesting the MOF could serve as a counterion. A similar change in the Raman spectrum cannot be seen when accommodating ordinary C₆₀ into MOF-177 (see Figure S6).

A similar vanishing of the $A_g(2)$ mode has been reported by Denisov *et al.* when studying superconducting films of K_3C_{60} and Rb_3C_{60} .³¹ When Raman was measured with 1064 nm laser excitation, no $A_g(2)$ mode could be observed, which has been attributed to a distortion of C_{60} due to the Jahn-Teller effect.

For a better understanding of the interaction between MOF-177 and Li@C₆₀, a closer look at the Raman spectra, in particular at 415 cm⁻¹ is useful. This peak originates from MOF-177 and can be attributed to the Zn-O bond vibration.^{32,33} Figure 2b shows the Raman peak for MOF-177 after soaking in the oDCB/ACN solution (black) and those with accommodated C_{60} (blue) and $Li@C_{60}$ (red), respectively. When $Li@C_{60}$ is accommodated, a peak shift to 412 cm⁻¹ is observed, while such a shift was not observed for the other two, indicating that accommodation of Li@C₆₀ is the origin of this shift. Raman peaks shifting to lower frequencies is typically associated with strain in the system and in particular with tensile stress.³⁴ Similar shifts have been reported for phosphorus-oxide bonds as well as cobalt-doped Zn-O nanorods.^{35,36} This would indicate a lengthening of the Zn-O bonds in MOF-177. Likely, the positive charge of Li⁺ is attracting oxygen and repelling zinc causing the bond length to



Fig. 2 a) Raman spectra of Li@C₆₀[PF₆⁻] (blue), Li@C₆₀ \subset MOF-177 (red) and MOF-177 (black) as well as oDCB (green) and ACN (yellow). The A₈(2) mode peak of C₆₀ is marked with a red square. It disappears after accommodation into MOF-177. Excitation wavelength was set to 785 nm. **b)** Close-up of the MOF-177 Raman peak resulting from the Zn-O bond. Shown are the spectra obtained from MOF-177 after storage in a 1:1 (v:v) oDCB/ACN solution (black), as well as with accommodated C₆₀ (blue) and Li@C₆₀ (red). A peak shift is observed only when Li@C₆₀ is accommodated. Excitation wavelength was set to 633 nm. The spectra have been background-corrected, see **Figure S7** for details.

increase. Similar interactions are causing the C_{60} cage in $Li^+@C_{60}(PF_{6}^-)$ crystals to distort, as the nonpolar C_{60} repulses the Li^+ and/or PF_{6}^- . Similar forces in $Li@C_{60}$ accommodating MOF-177 ($Li@C_{60}$ ⊂MOF-177) could explain the above observed disappearance of the $A_g(2)$ mode of C_{60} as it can feel repulsive forces resulting from interaction with Li and O. Furthermore, the electrostatic interaction between $Li@C_{60}$ and MOF-177 explain how $Li@C_{60}$ can be stable without PF_{6}^- as a counterion. To investigate the homogeneity of C_{60} and $Li@C_{60}$ in MOF-177, Raman scattering intensity mapping was carried out. C_{60} and $Li@C_{60}$ were distributed in MOF-177 over several microns (Figures S8a-b). Additionally, C_{60} and $Li@C_{60}$ were detected in the inner part of MOF-177 (Figures S8c-e). These results indicate that C_{60} and $Li@C_{60}$ were encapsulated in the whole volume of MOF-177.

As SHG is a 2nd order NLO effect, it requires breaking of centrosymmetry in order to be observed. Due to this condition, it should be noted that MOF-177 originally has a centrosymmetric structure³⁷ and thus the emission of 2nd order NLO effects is not obvious. To check if this signal was not produced by the solvent molecules, we exposed MOF-177 to a vacuum for 30 minutes and performed NLO measurements. Indeed, SHG was clearly observed (see Figure S10). We can safely conclude that the signal is generated in the MOF itself (no signal could be observed from the coverslip). MOFs are known to have a soft structure due to their organic linkers meaning the linkers can e.g. rotate, which could cause the breakdown of

Nonlinear optical properties

In order to assess the NLO properties of the Li@C₆₀⊂MOF-177 composite, we have performed SHG measurements under microscopy conditions. Figure 3 shows the average intensities of SHG obtained from MOF-177 (soaked in either toluene or a 1:1 (v:v) mixture of oDCB/ACN), C₆₀ accommodating MOF-177 (C₆₀ \subset MOF-177) and Li@C₆₀ \subset MOF-177. As can be seen, the average emission after accommodation of Li@C₆₀ has increased by roughly 40% compared to empty MOF-177 (soaked in toluene). This effect is not observed when C₆₀ is accommodated. In fact, C_{60} shows only a marginal improvement that does not exceed standard deviation. MOF-177 soaked in oDCB/ACN also shows a substantial increase in emission, meaning the enhancement in Li@C₆₀⊂MOF-177 is partially due to Li@C₆₀ and partially due to oDCB. It should be mentioned that some part of the emitted SHG is likely re-absorbed by Li@C60 lowering the efficiency slightly (see Figure S9).



Fig. 3 Average maximum intensity obtained from different types of MOF-177 with standard deviations. Averages were obtained from 40 spectra for each type. SHG was excited using 820 nm irradiation with a pulse duration of 120 fs and a repetition rate of 80 MHz.



Fig. 4 Polarized SHG analysis performed on MOF-177. Test 1: a linear polarized pulsed beam is rotated 360° and all emitted SHG collected. Test 2: the beam is rotated 360° and the emitted SHG measured with an arbitrary polarization using an analyzer. Test 3: the beam polarization and analyzer are rotated 360° in parallel. Test 4: the beam polarization and analyzer are rotated 360° but perpendicular to each other. Data has been corrected to account for polarization dependence of the microscope and spectrograph. See section 11 in the ESI for details.

centrosymmetry. In addition to this, deterioration of the structure of MOF-177 when exposed to air has been reported previously.³⁸

In order to explore how SHG can be emitted from a centrosymmetric crystal, we have performed polarized SHG analysis under ambient conditions as reported by van der Veen *et al.*³⁹ By conducting a series of tests using a polarizer and an analyzer, the point group of a crystalline sample can be determined. The results can be seen in Figure 4. All tests showed zero symmetry axes and the SHG never vanished. Following the flow chart described by van der Veen *et al.*, it can be concluded that the point group of MOF-177 in this case is P1 pointing to a non-centrosymmetric triclinic crystal structure. A triclinic structure has been theoretically predicted previously after accommodation of C_{60} .²⁸ No other point groups could be detected for several crystals which is why we assume P1 is the dominant point group present in the sample. .

Our measurements suggest that removal from a solvent and exposure to air could already cause changes to the structure of MOF-177.

There are several possibilities that could explain the increase in SHG intensity in case of $Li@C_{60} \subset MOF-177$: 1) the presence of a guest molecule itself, 2) structural changes (meaning change of point/space group) of the host and/or guest due to accommodation, 3) increased dipole moment due to interaction between host and guest.

Despite its centrosymmetry, bulk SHG emission from C_{60} films has been reported and was attributed to magnetic dipole and electric quadrupole moments.^{40,41} This can explain the slight increase in emission for C_{60} ⊂MOF-177.

Seeing as C_{60} is causing a slight increase in emission, it is likely that part of the larger increase in emission strength of $Li@C_{60} \subset MOF-177$ is simply due to the presence of $Li@C_{60}$ as well as oDCB. It has been shown by Campbell *et al.* as well as our group that $Li@C_{60}$ has stronger NLO properties compared to ordinary C_{60} and thus stronger emission from $Li@C_{60} \subset MOF-177$

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compared to C_{60} \subset MOF-177 is expected.²¹⁻²³ However, there are more factors contributing to the overall increased emission.

Although change of the crystal structure of MOF-177 after accommodation of C_{60} has been predicted²⁸, in this case it does not seem to be associated with any meaningful increase in SHG intensity as it should cause measurable increases in both $Li@C_{60} \subset$ MOF-177 and $C_{60} \subset$ MOF-177 and the crystal structure is the same for empty MOF-177 in our study. Small additional structural changes induced by $C_{60}/Li@C_{60}$ cannot be excluded by this study, however they do not seem to cause a noticeable change in NLO emission.

Furthermore, the interaction between Li⁺ and the Zn-O bond of MOF-177 is causing tensile strain on the Zn-O bond. Increased length of the bond means an increased dipole moment if the charges remain the same. Such an increased dipole moment can further improve overall NLO emission.⁴²

Conclusions

We have shown how the NLO emission of MOF-177 can be significantly enhanced by accommodating $\text{Li}@C_{60}$ into the pores and how the interaction between the two is likely further increasing the emission properties. We have further shown that MOF-177 slightly deviates from the reported crystal structure when exposed to air as well as the stabilization of $\text{Li}@C_{60}$ by accommodation into the MOF.

By considering both host and guest for NLO emission, the overall signal can be greatly improved and the whole volume of the MOF, both the framework as well as the pores, are used efficiently for NLO. Additionally, due to the interaction between host and guest, the emission can be further strengthened. In the future, by carefully considering all factors highly efficient NLO active MOFs can be created: 1) choosing an NLO active host, 2) choosing an NLO active guest and 3) using interaction between host and guest to further enhance NLO properties. Using this holistic approach to MOF, their performance can be greatly improved. Furthermore, the approach can be potentially expanded to other organic porous materials as design flexibility is a common feature of organic materials.

Experimental

 $Li^+ @C_{60}(PF_6^-)$ powder that was purified through recrystallization was purchased from Idea International Co. Ltd.

Microcrystals of MOF-177 were synthesized according to the literature (J. Mater. Chem., 2007, 17, 3197–3204).

Introduction of C_{60} into MOFs: As-synthesized crystals were immersed into a saturated toluene solution of fullerene C_{60} (10 mL) at room temperature. The suspension was heated to 60° C in an oven. After 1 day, the supernatant was removed by decantation, another 10 mL of saturated fullerene solution was added to the residue, and the resulting suspension was again allowed to stand at 60 °C. After 1 week with six solutionreplacement cycles, the inclusion complex was obtained (Nat. Chem., 2010, 2, 780–783). Introduction of $Li@C_{60}$: As-synthesized crystals were immersed into oDCB/ ACN solution of $Li^+@C_{60}(PF_6^-)$ (1 mg/mL) at room temperature. After 1 day, the supernatant was removed by decantation, another solution of $Li^+@C_{60}(PF_6^-)$ was added to the residue, and the resulting suspension was again allowed to stand at room temperature.

After soaking MOF-177 crystals in the solutions of C_{60} or [Li⁺@ C_{60}], the residual C_{60} and [Li⁺@ C_{60}] were repeatedly washed out with their solvents until the washouts become colorless. C_{60} ⊂MOF-177 and Li@ C_{60} ⊂MOF-177 are brown and dark brown, respectively (Figure S12).

ICP-OES: Li@C₆₀ \subset MOF-177 was decomposed by HClaq and neutralized by NaOHaq. The resultant solution was mixed with acetonitrile to dissolve organic components. The final solution was analyzed by ICPE-9000, SHIMADZU CORPORATION.

Thermogravimetry of $[Li^+@C_{60}](PF_6^-)$, MOF-177 and $Li@C_{60} \subset$ MOF-177 were carried out with Thermo plus EVO II TG8120 HUM-1F, Rigaku.

Raman spectroscopy: Raman spectra of microcrystals were obtained by inVia Reflex (Renishaw) with 785 nm laser. Measurements of peak position at 415 cm⁻¹ were obtained with the same setup as describes under *"NLO measurements"* but with 633 nm excitation from a He-Ne laser and a 645 long-pass filter to block excitation light in front of the spectrograph.

NLO measurements have been performed on an inverted optical microscope (Ti-U, Nikon). The samples were illuminated by a femtosecond (fs) pulse at 820 nm (Mai Tai HP, Spectra-Physics, 120 fs, 80 MHz). The excitation beam was focused by an objective lens (10x Plan Fluor, NA 0.30, Nikon). The backscattered NLO signal was collected through the same objective. Spectra were recorded using a charge-coupled device (CCD) camera (DU920P, Andor) with a spectrograph (iHR320, Horiba). Out of focus signal was removed using a pinhole. Excitation light was blocked by a 750 short-pass filter (ET750sp, Chroma). Spectra were obtained from the central regions of MOFs. Integration time for every spectrum was 5 s. For each MOF (MOF-177 (toluene), MOF-177 type of (0dichlorobenzene/acetonitrile) $C_{60} \subset MOF-177$, Li@ $C_{60} \subset MOF-$ 177) 40 spectra were recorded (20 for MOF-177 in odichlorobenzene/acetonitrile) and the average intensities calculated from these. Measurements were carried out under ambient conditions.

UV-vis measurement was carried out on a Lambda 950 (PerkinElmer) spectrophotometer using blank correction.

Conflicts of interest

There are no conflicts to declare.

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