



HOKKAIDO UNIVERSITY

Title	High pressure densification and dielectric properties of perovskite-type oxynitride SrTaO ₂ N
Author(s)	Masubuchi, Yuji; Kawamura, Fumio; Taniguchi, Takashi et al.
Citation	Journal of the European Ceramic Society, 35(4), 1191-1197 https://doi.org/10.1016/j.jeurceramsoc.2014.10.028
Issue Date	2015-04
Doc URL	https://hdl.handle.net/2115/58051
Type	journal article
File Information	YMasubuchi-JEurCeramSoc-2015-35-1191-1197-1.pdf



High pressure densification and dielectric properties of perovskite-type oxynitride SrTaO₂N

Yuji Masubuchi^{*1}, Fumio Kawamura², Takashi Taniguchi², and Shinichi Kikkawa¹

Affiliations

¹Faculty of Engineering, Hokkaido University, N13 W8, Kita-ku, Sapporo,
060-8628, Japan

²National Institute for Materials Science, 1-1 Namiki, Tsukuba, 305-0044, Japan

*Corresponding author: Y. Masubuchi; Address: Faculty of Engineering,
Hokkaido University, N13 W8, Kita-ku, Sapporo 060-8628, Japan; Tel/Fax:
+81-(0)11-706-6742/6740; E-mail: yuji-mas@eng.hokudai.ac.jp

Abstract

Perovskite-type SrTaO_2N was densified using a belt-type high pressure apparatus at 2.5-7.7 GPa with and without sample heating. The relative density of the samples treated at 2.5 GPa was 64% and was increased to 75% in those treated at 7.7 GPa without sample heating. The color of the compact changed from orange to brown with densification and the crystals were fractured to nanometer size grains. The density was increased up to 86% with heating above 800 °C at 7.7 GPa and the sample color changed to black, accompanied by electrically conductive nature. The permittivity of compacts that were densified to a relative density of around 75% with heating at 600 °C and without sample heating were 200 and 90 at 100 Hz, respectively.

Keywords: SrTaO_2N , Perovskite oxynitride, High pressure densification, Dielectric property

1. Introduction

Perovskite-type oxynitrides, SrTaO_2N and BaTaO_2N have attracted much attention since the report of their superior dielectric properties by Kim et al. in 2004 [1]. The bulk ceramics had a low relative density value around 55% even after a powder compact is cold isostatically pressed (CIP) at 430 MPa and sintered at 1020 °C in NH_3 flow [1]. Their permittivity of several hundred was reported in the temperature range between 180 and 300 K, with a relatively large dielectric loss > 1 [1]. However there have been no other reports to support the interesting information. Two independent reports in 2011 indicate the presence of *cis*-type TaO_4N_2 in SrTaO_2N from neutron diffraction data [2,3]. The *cis*-configuration of TaO_4N_2 was maintained up to 1100 °C from high temperature neutron diffraction measurements [4].

To clarify the intrinsic dielectric properties of the oxynitride, highly densified SrTaO_2N ceramics were prepared by sintering at 1400 °C in 0.2 MPa of N_2 and subsequent annealing at 1000 °C in NH_3 flow [5,6]. SrCO_3 (2.5 wt%) was necessary as a sintering additive to compensate for the partial loss of SrO during high temperature sintering. The sample color changed from orange to black with sintering due to a slight loss of nitrogen, although the sintered ceramics was

single phase of SrTaO_2N . Less dense SrTaO_2N ceramics with relative density of $< 85\%$ recovered both the orange color, stoichiometric nitrogen content and electrical insulating behavior after the annealing in NH_3 . The nitrogen loss was not recovered in fully dense SrTaO_2N ceramics. The dielectric constant of SrTaO_2N changed in the range between 450 and 9000 with the relative density. The recovery in the post annealing was more difficult in the denser ceramics. Sintering with hot isostatic pressing (HIPping) at 1400°C under 196 MPa has also been applied to prepare highly densified SrTaO_2N ceramics [7]. The color of the SrTaO_2N ceramics also changed from brown to dark gray by HIPping and the original brown color was recovered by post-annealing in NH_3 . Neither high temperature sintering in nitrogen atmosphere nor HIPping suppressed the nitrogen loss from SrTaO_2N . Thus, the sintered bodies must be post-annealed in NH_3 to compensate for the nitrogen loss.

The thermal stability of perovskite-type oxynitrides has been investigated using a combination of thermogravimetry (TG) and mass spectroscopy (MS) in different atmospheres [8]. The oxidation reaction of SrTaO_2N begins at 480°C in 10-20% O_2 diluted with Ar to produce $\text{SrTaO}_{3.5}$. TG-MS measurement in Ar indicated SrTaO_2N released N_2 gas above 1120°C

and the color was changed from orange to dark orange with black particles, although the perovskite phase was retained, as confirmed by X-ray diffraction (XRD) measurement. Thermal annealing of SrTaO_2N in an evacuated silica tube also partially released nitrogen and a secondary phase of rock-salt type $\text{TaO}_{0.9}$ was formed together with the perovskite phase at 1100 °C [6]. Thin film formation is another way of material shaping different from ceramic sintering. SrTaO_2N thin films were grown epitaxially on a SrTiO_3 substrate by nitrogen plasma-assisted pulsed laser deposition method [9]. The compressively strained SrTaO_2N thin film was composed of small domains that exhibited ferroelectric response and a surrounding matrix that exhibited relaxor-type behavior.

Densification below the decomposition temperature is required to obtain highly densified SrTaO_2N ceramics without compositional change. The synthesis and sintering of metal nitrides have been examined on optical, magnetic, electronic and refractory materials, such as GaN, FeN_x , BN, and AlN [10-15]. However, little research has been conducted on synthesis of oxynitrides, e.g. $\text{Ga}_3\text{O}_3\text{N}$, and REZrO_2N (RE = Pr, Nd, and Sm) suppressing a release of nitrogen under high pressure in the order of gigapascals [16,17]. Sintering of TaON has been investigated under high pressures of 3 and 5.5 GPa at temperatures above

900 °C using a belt-type high pressure apparatus, while the green compacts were not densified but were oxidized to Ta₂O₅ during HIPping in Ar [18,19].

In this study, densified SrTaO₂N were prepared by using a belt-type high pressure apparatus to avoid the thermal decomposition under pressures of several gigapascals, with and without sample heating. The electrical properties of the densified ceramics were then investigated with respect to their phase decomposition.

2. Experimental

2.1. High pressure and high temperature experiments

SrTaO₂N was prepared by the ammonolysis reaction of SrTaO_{3.5}, as reported in our previous paper [6]. The SrTaO₂N precursors were encapsulated in Pt or Au tubes (6 mm diameter, 2-3 mm height) under a dried Ar atmosphere. The encapsulated samples were set in the belt-type high pressure apparatus with a NaCl pressure transmitting medium and a graphite heater [20,21]. High pressure was applied in the range of 2.5-7.7 GPa at room temperature and the sample was then heated to temperatures in the range of 600-1000 °C and held for 30 or 60 min, as summarized in Table 1. After sintering under pressure, the

sample was quenched to room temperature within a few minutes by turning off the heating power under the pressure. After release the pressure, the SrTaO₂N compacts were removed from the Pt (Au) wrapping.

2.2. Characterization

The crystalline phases were characterized using powder XRD (Ultima IV, Rigaku) with monochromatized Cu K α radiation. XRD patterns were measured for the polished surface of the high pressure compacts. The relative densities of the compacts were determined using bulk densities measured from their dimensions and weight. Scanning electron microscopy (FE-SEM; JSM-6500F, Jeol) was used to observe the microstructure of the fractured surface. Transmission electron microscopy (TEM; JEM-2010, Jeol) observations were performed using thin slices prepared by focused ion beam milling (FIB; JIB-4600F, Jeol). The dielectric properties were measured using an LCR meter (4274A, Hewlett-Packard) in the frequency range from 10² to 10⁶ Hz at room temperature with both surfaces of the compacts coated with silver paste as electrodes. Diffuse reflectance of the SrTaO₂N precursor and the high pressure compacts were measured with a UV-Vis spectrometer (V-550, Jasco).

3. Results and discussion

3.1. High pressure compacts obtained at ambient temperature

SrTaO₂N compacts obtained at high pressures were single phase of perovskite as shown in Fig. 1. The diffraction lines broadened with the increase of pressure, while their lattice parameters were almost constant at room temperature, as summarized in Table 1. The relative density of the compact formed at 2.5 GPa was 64 %, which reached 75% at pressures above 5.5 GPa without sample heating. The compact formed at 2.5 GPa was bright orange colored, similar to the SrTaO₂N precursor powder. However, the color of the compact formed at 7.7 GPa had changed to dark brown, as shown in Fig. 2. Optical microscopic images of the fracture surfaces of the samples are shown in Fig. S1 of the Supporting Information (SI). As the pressure was increased, brown or black particles appeared together with the original orange particles. UV-Vis spectra for the powdered samples had a band gap energy of approximately 2.05 eV which was comparable with the reported value of 2.1 eV [1]. The compacts became dark in color, but without impurity phases, although the crystallinity was lowered, which could be related to a microstructural change. A high density of

defects on the particle surface was proposed as a reason for the optical change in perovskite-type tantalum oxynitrides [22].

The starting SrTaO_2N powder showed a coagulation of plate-like crystals and a particle size distribution from several micrometers to several hundreds of nanometers, as shown in Fig. 3(a). Then fracture surfaces of the high pressure compacts obtained at 2.5 and 7.7 GPa without heating revealed dense microstructures where the morphology of the secondary particles was maintained. The primary particle size seems to be smaller than that of the original powder (Figs. 3(b) and 3(c)). Residual pores were evident in the TEM image of the compact obtained at 7.7 GPa, as depicted in Fig. 3(d), because no grain growth or sintering between SrTaO_2N grains occurred under the non-heating condition. The magnified image shown in Fig. 3(e) indicates that the grain surface was fractured into small particles of several tens of nanometers. The nanosized surface structure may be formed by inter-grain stress that arises during high pressure compaction. Similar diffraction line broadening has been reported for oxides, nitrides and carbides, where the grain size was reduced to the nanometer scale under microstructural strain [13,23,24]. The electron diffraction shown in Fig. 3(f) suggests that the fractured grains in Fig. 3(e)

retained the perovskite-type crystal lattice. No amorphous halo pattern was observed or extra diffraction spots that would indicate the presence of a secondary phase. These observations suggest that the grain was fractured into nanosized particles during the room temperature compaction, as indicated by the XRD line broadening in Fig. 1, while the crystalline perovskite structure was still maintained.

3.2. High pressure compacts obtained at high temperature

High pressure compacts obtained at ambient temperature and pressures above 5.5 GPa had a maximum relative density of 75%. No secondary phases were observed in their XRD patterns; however, the diffraction lines were broadened due to the formation of nanosized particles that were responsible for the color change from orange to brown. To improve the density and crystallinity, high pressure sintering at high temperature was performed at a pressure of 7.7 GPa. Figure 4 shows that the relative density increased with temperatures above 800 °C and reached 83% at 1000 °C. The product pressed at 600 °C was colored brown, similar to that obtained at ambient temperature. However, the color changed to black when sintered above 800 °C. The sintered pellets were

also single phase of perovskite structure, as shown in Fig. S2 in SI. The lattice parameters were almost comparable with those of the powder precursor but became slightly smaller along the *a*-axis with the increased sintering temperature under pressure of 7.7 GPa (Table 1). Fracture surface of the pellet obtained at 600 °C was almost similar to that of the high pressure compact obtained without heating as shown in Fig. 5(a). However, sintering at 1000 °C slightly disrupted the particles to make a denser morphology in Fig. 5(b). The diffuse reflectance of the black pellets sintered above 800 °C was reduced in the near infrared region due to the appearance of electrical conductive nature in the sintered pellets as shown in Fig. S3 in SI. Dense phase is stabilized under high pressure as in the case of graphite to diamond phase transition. The observed lattice shrinkage was also induced by the application of high pressure, and this shrinkage may be compensated by the partial loss of nitrogen when the high pressure was released.

3.3. Dielectric property of the high pressure compacts

Figure 6 shows complex impedance plots for the high pressure compacts. Insulative characteristics are suggested by observation of only a large

partial semi-circle for the compacts obtained under pressure of 2.5 and 7.7 GPa at room temperature and that formed at 600 °C under 7.7 GPa. The compacts had slightly lower relative density below 75%, and the sample color was orange or brown. At higher heating temperature above 800 °C and at 7.7 GPa, the sintered pellets were black and the complex impedance plots were semi-circles. The resistivities were estimated from the intersection with the real axis and were 25 MΩcm and 12 kΩcm for the pellets sintered at 800 °C and 1000 °C, respectively. The conductivity may be related to a partial loss of nitrogen and the formation of rock-salt type TaO_{0.9} impurity, as reported in our previous paper [6]. Black colored conductive pellets were previously obtained when SrTaO₂N was sintered with a SrCO₃ additive at high temperatures above 1300 °C under a 0.2 MPa N₂ atmosphere or when HIPped above 1200 °C at 196 MPa [5-7]. The TaO_{0.9} impurity phase has also appeared in powder compacts annealed at 1100 °C in a silica tube under vacuum [6]. Nitrogen release from SrTaO₂N and thermal decomposition of the oxynitride have also been confirmed in a temperature range between 1120 °C and 1200 °C by thermal analysis with mass spectroscopic analysis [8]. Both decomposed products contained black particles due to the presence of the reduced Ta⁵⁺ species, and which also indicates the

conductive nature of the product. The color of pellets sintered under a high pressure of 7.7 GPa was changed to black above 800 °C, which is approximately 200 °C lower than the decomposition temperature of SrTaO₂N in Ar or vacuum. The slight shrinkage of the unit cell volume could be related to partial decomposition and the release of nitrogen. Pressure-induced phase decomposition has been reported in many oxides with various crystal structures [25,26]. The FeTiO₃ perovskite-type oxide (as a high pressure polymorph) was also decomposed into a mixture of rock-salt type “FeO” and TiO₂ at high pressure above 67 GPa [27,28]. The high pressure-induced decomposition can be understood by the formation of a denser assemblage of the decomposed oxides that is favorable at high pressure rather than the parent ternary oxide. Pressure-induced phase decomposition has also been suggested in ternary nitride based on density functional calculations [29]. SrTaO₂N has a theoretical density of 7.95 g/cm³ [2], which is less dense than those of simple tantalum oxides and nitrides such as 9.93 g/cm³ for Ta₃N₅ [30], 8.31 g/cm³ for Ta₂O₅ [31] and around 15 g/cm³ for rock-salt type TaO and TaN [32,33]. Rock-salt type TaO_{0.9} has been reported to appear as a high temperature impurity phase in SrTaO₂N [6]. High pressure inducing microstructural stress on the SrTaO₂N

grains decreases its thermal stability because higher density phases such as TaO or TaN with theoretical densities around 15 g/cm³ may stabilize under the high pressure rather than SrTaO₂N with a lower density of 7.9 g/cm³.

Permittivity and dielectric loss ($\tan\delta$) were measured for the orange and brown insulative pellets. The relative dielectric constants for the high pressure compacts obtained at 2.5 and 7.7 GPa without sample heating were 60 and 90, respectively as shown in Fig. 7. Heating at 600 °C under a pressure at 7.7 GPa significantly increased the dielectric constant to 200 at 100 Hz. The dielectric loss for the sintered pellet were slightly increased above $\tan\delta > 0.1$. Their permittivity had a small frequency dependence. The permittivity and low dielectric loss values measured for the high pressure compacts were comparable to those of previously measured for dense SrTaO₂N ceramics sintered at 1450 °C under 0.2 MPa N₂ atmosphere with subsequent post-annealing at 1000 °C in NH₃ flow [6]. High pressure compaction at low heating temperatures and at ambient temperature densified the thermally unstable SrTaO₂N up to a relative density of 75%, while conductive behavior appeared in the high pressure compacts sintered above 800 °C, which is 200 °C lower than the decomposition temperature of SrTaO₂N under vacuum or Ar flow.

Conclusion

High pressure densification was investigated for perovskite-type SrTaO_2N with and without sample heating. The products were phase pure perovskites and the relative density was increased for samples pressed at ambient temperature from 64 at 2.5 GPa to 75% at 7.7 GPa. In addition to the densification, the sample color changed from orange to brown and the crystallinity was decreased due to the formation of nanoparticles, while the perovskite structure was retained. Heating above 800 °C at 7.7 GPa promoted densification up to 85% accompanied by a slight contraction of the crystal lattice. The sintered ceramics were black and exhibited electrical conductivity due to a partial loss of nitrogen. The decomposition reaction may occur under high pressure sintering above 800 °C, which is lower than the temperature of thermal decomposition at ambient pressure or in a vacuum. The presence of crystalline phases that are denser than SrTaO_2N may depress the stability of the perovskite phase at high pressure. High pressure compacts with a relative density of 75% prepared at 600 °C and 7.7 GPa were insulative and had a permittivity of 200 (100 Hz) at room temperature.

Acknowledgement

This work was partly supported by a Grant-in Aid for Scientific Research (A) (No. 24245039) from the Japan Society for the Promotion of Science (JSPS). Some TEM observations was conducted at Hokkaido University, supported by the Nano-technology Platform Program of the Ministry of Education, Culture, Sports, Science and Technology (MEXT), Japan. Y. M. acknowledges financial support from the Murata Science Foundation.

References

- [1] Kim YI, Woodward PM, Baba-Kishi KZ, Tai CW. Characterization of the structural, optical, and dielectric properties of oxynitride perovskites AMO_2N (A = Ba, Sr, Ca; M = Ta, Nb). *Chem Mater* 2004;16:1267-76.
- [2] Zhang YR, Motohashi T, Masubuchi Y, Kikkawa S. Local anionic ordering and anisotropic displacement in dielectric perovskite SrTaO_2N . *J Ceram Soc Jpn* 2011;119:581-6.
- [3] Yang M, Oró-Solé J, Rodgers JA, Jorge AB, Fuertes A, Attfield JP. Anion order in perovskite oxynitrides. *Nat Chem* 2011;3:47-52.
- [4] Clark L, Oró-Solé J, Knight KS, Fuertes A, Attfield JP. Thermally robust anion-chain order in oxynitride perovskites. *Chem Mater* 2013;25:5004-11.
- [5] Zhang YR, Motohashi T, Masubuchi Y, Kikkawa S. Sintering and dielectric properties of perovskite SrTaO_2N ceramics. *J Eur Ceram Soc* 2012;32:1269-74.
- [6] Sun SK, Zhang YR, Masubuchi Y, Motohashi T, Kikkawa S. Additive sintering, postannealing, and dielectric properties of SrTaO_2N . *J Am Ceram Soc* 2014;97:1023-7.

- [7] Zhang YR, Masubuchi Y, Motohashi T, Kikkawa S, Hirota K. Hot isostatic press sintering and dielectric properties of SrTaO_2N ceramics. *Ceram Int* 2013;39:3377-80.
- [8] Aguiar R, Logvinovich D, Weidenkaff A, Reller A, Ebbinghaus SG. Thermal oxidation of oxynitride perovskites in different atmospheres. *Thermochim Acta* 2008;471:55-60.
- [9] Oka D, Hirose Y, Kamisaka H, Fukumura T, Sasa K, Ishii S, Matsuzaki H, Sato Y, Ikuhara Y, Hasegawa T. Possible ferroelectricity in perovskite oxynitride SrTaO_2N epitaxial thin films. *Sci Rep* 2014;4:4987/1-6.
- [10] Kawamura F, Watanabe K, Takeda T, Taniguchi T. Synthesis of high-crystallinity cubic-GaN nanoparticles using the Na flux method - A proposed new usage for a belt-type high-pressure apparatus. *J Cryst Growth* 2011;321:100-5.
- [11] Saitoh H, Utsumi W, Kaneko H, Aoki K, Synthesis of $\text{Al}_x\text{Ga}_{1-x}\text{N}$ alloy by solid-phase reaction under high pressure. *Jpn J Appl Phys* 2004;43:L981-3.
- [12] Dzivenko DA, Zerr A, Miehe G, Riedel R. Synthesis and properties of oxygen-bearing c- Zr_3N_4 and c- Hf_3N_4 . *J Alloys Compd* 2009;480:46-9.

- [13] Takagi K, Akada M, Ozaki K, Kobayashi N, Ogawa T, Ogata Y, Takahashi M. High-pressure sintering behavior of α'' -Fe₁₆N₂ nanopowder. J Appl Phys 2014;115:103905/1-5.
- [14] Li XL, Ma HA, Zheng YJ, Liu Y, Zuo GH, Liu WQ, Li JG, Jia X. AlN ceramics prepared by high-pressure sintering with La₂O₃ as a sintering aid. J Alloys Compd 2008;463:412-6.
- [15] Taniguchi T, Akaishi M, Yamaoka S. Sintering of cubic boron nitride without additives at 7.7 GPa and above 2000 °C. J Mater Res 1999;14:162-9.
- [16] Huppertz H, Hering SA, Zvoriste CE, Lauterbach S, Oeckler O, Riedel R, Kinski I. High-pressure synthesis, electron energy-loss spectroscopy investigation, and single crystal structure determination of a spinel-type gallium oxonitride Ga_{2.79}□_{0.21}(O_{3.05}N_{0.76}□_{0.19}). Chem Mater 2009;21:2101-7.
- [17] Yang M, Rodgers JA, Middler LC, Oró-Solé J, Jorge AB, Fuertes A, Attfield JP. Direct solid-state synthesis at high pressures of new mixed-metal oxynitrides: RZrO₂N (R= Pr, Nd, and Sm). Inorg Chem 2009;48:11498-500.
- [18] Matizamhuka WR, Herrmann M, Sigalas I, Sempf K. Preparation, hardness, and fracture toughness of tantalum oxynitride ceramics. J Am Ceram Soc 2007;90:2280-2.

- [19] Matizamhuka WR, Sigalas I, Herrmann M. Synthesis, sintering and characterisation of TaON materials. *Ceram Int* 2008;34:1481-6.
- [20] Fukunaga O, Takeuchi S, Taniguchi T. High-pressure synthesis of cubic BN using Fe-Mo-Al and Co-Mo-Al alloy solvents. *Diamond Relat Mater* 2011;20:752-5.
- [21] Taniguchi T, Yamaoka S. Spontaneous nucleation of cubic boron nitride single crystal by temperature gradient method under high pressure. *J Cryst Growth* 2001;222:549-57.
- [22] Kim YI. Effects of KCl flux on the morphology, anion composition, and chromaticity of perovskite oxynitrides, CaTaO_2N , SrTaO_2N , and LaTaON_2 . *Ceram Int* 2014;40:5275-81.
- [23] Nadeau JS. Very high pressure hot pressing of silicon carbide. *Am Ceram Soc Bull* 1973;52:170-4.
- [24] Machon D, Sinitsyn VV, Dmitriev VP, Bdikin IK, Dubrovinsky LS, Kuleshov IV, Ponyatovsky EG, Weber HP. Structural transition in Cu_2O at pressures up to 11 GPa. *J Phys Condens Matter* 2003;15:7227-35.
- [25] Marezio M, Remeika JP, Jayaraman. A. High-pressure decomposition of synthetic garnet. *J Chem Phys* 1966;45:1821-4.

- [26] Ravindran TR, Arora AK, Mary TA, High-pressure Raman spectroscopic study of zirconium tungstate. *J Phys Condens Matter* 2001;13:11573-88.
- [27] Leinenweber K, Utsumi W, Tsuchida Y, Yagi T, and Kurita K. Unquenchable high-pressure perovskite polymorphs of MnSnO_3 and FeTiO_3 . *Phys Chem Mater* 1991;18:244-50.
- [28] Nishio-Hamane D, Yagi T, Ohshiro M, Niwa K, Seto Y. Decomposition of perovskite FeTiO_3 into wüstite $\text{Fe}_{1-x}\text{Ti}_{0.5x}\text{O}$ and orthorhombic FeTi_3O_7 at high pressure. *Phys Rev B*, 2010;82:092103/1-4.
- [29] Römer SR, Braun C, Oeckler O, Schmidt PJ, Kroll P, Schnick W. $\text{HP-Ca}_2\text{Si}_5\text{N}_8$ – A new high-pressure nitridosilicate: synthesis, structure luminescence, and DFT calculations. *Chem Eur J* 2008;14:7892-902.
- [30] Strähle J. Die kristallstruktur des tantal(V)-nitrides Ta_3N_5 . *Z Anorg Allg Chem* 1973;402:47-57.
- [31] Stephenson NC, Roth RS. Structural systematics in the binary system Ta_2O_5 – WO_3 . V. The structure of the low-temperature form of tantalum oxide $L\text{-Ta}_2\text{O}_5$. *Acta Crystallogr* 1971;B27:1037-44.
- [32] Schönberg N. An x-ray investigation of the tantalum-oxygen system. *Acta Chem Scand* 1954;8:240-5.

[33] Brauer G, Kiliani W. Darstellung und kristallstruktur von niob-tantal-nitridphasen (Nb,Ta)N₋₁. Z Anorg Allgem Chem 1979;452:17-26.

Table 1 Lattice parameters and relative densities of SrTaO₂N compacts

Pressure / GPa	Temp. / °C (duration / min.)	Lattice parameters		Relative density, ρ / %
		a / nm	c / nm	
SrTaO ₂ N powder		0.56972(1)	0.80786(2)	-
2.5	-	0.56973(4)	0.8082(1)	64(1)
5.5	-	0.56983(6)	0.8082(2)	76(1)
7.7	-	0.56985(6)	0.8083(2)	75(2)
7.7	600 (60)	0.56974(9)	0.8085(2)	71(2)
7.7	800 (60)	0.56963(1)	0.8083(1)	82(1)
7.7	1000 (30)	0.56952(3)	0.8084(1)	83(2)

Figure captions

Figure 1. XRD patterns for the SrTaO₂N (a) powder and high pressure compacts obtained at (b) 2.5, (c) 5.5, and (d) 7.7 GPa and at ambient temperature.

Figure 2. Relative density of high pressure compacts obtained at ambient temperature. Inset photographs indicate the color of the compacts.

Figure 3. SEM images of the SrTaO₂N starting powder (a) and fracture surfaces of the high pressure compacts obtained at (b) 2.5 and (c) 7.7 GPa. TEM image of the compact obtained at (d) 7.7 GPa and (e) a magnified image of the area indicated by a red square in (d), and (f) an electron diffraction pattern.

Figure 4. Relative density of the high pressure compacts as a function of the heating temperature. Inset photographs indicate the color of the compacts.

Figure 5. SEM images of the high pressure compacts sintered at 600 °C (a) and 1000 °C (b) under a pressure of 7.7 GPa.

Figure 6. Complex impedance plots for the high pressure compacts. The inset shows an expanded low frequency plot for the compacts.

Figure 7. Frequency dependence of the permittivity and dielectric loss for the high pressure compacts.

Figure 1, Y. Masubuchi, et al.,

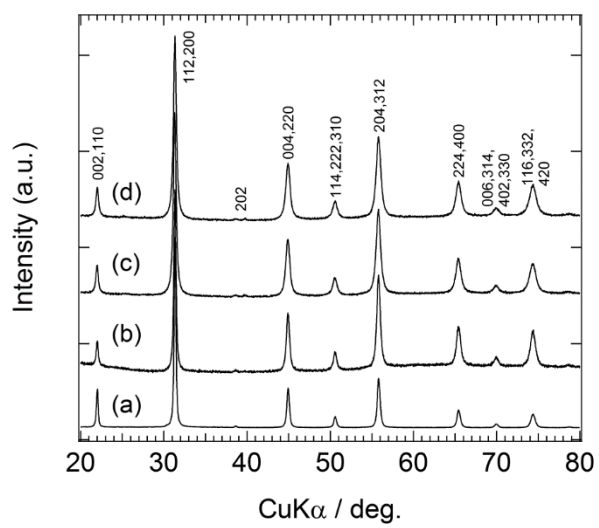


Figure 2, Y. Masubuchi, et al.,

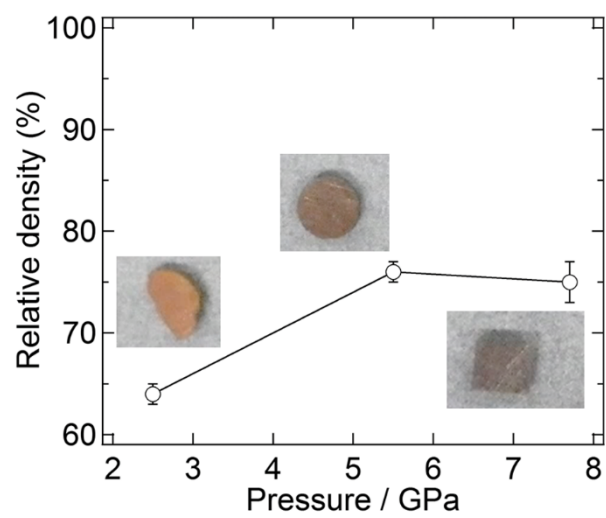


Figure 3, Y. Masubuchi, et al.,

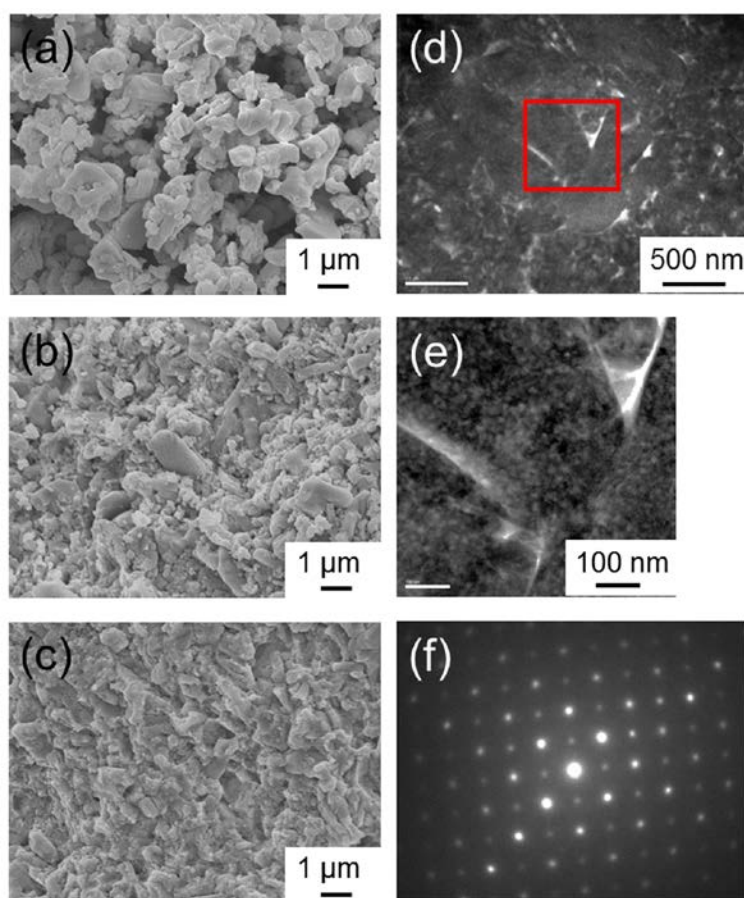


Figure 4, Y. Masubuchi, et al.,

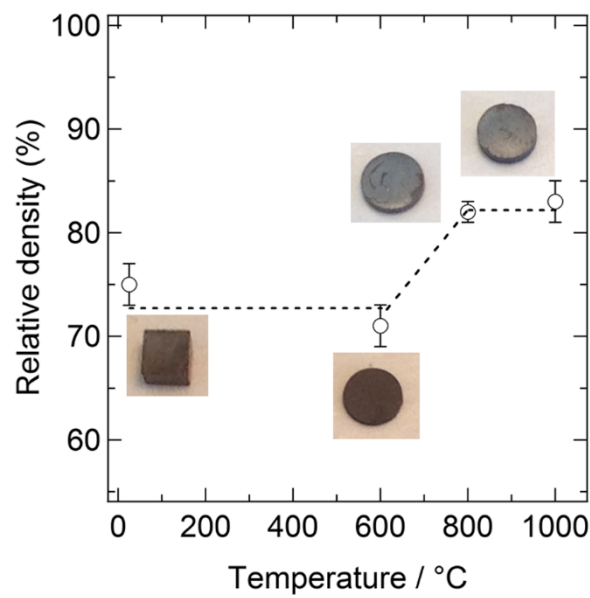


Figure 5, Y. Masubuchi

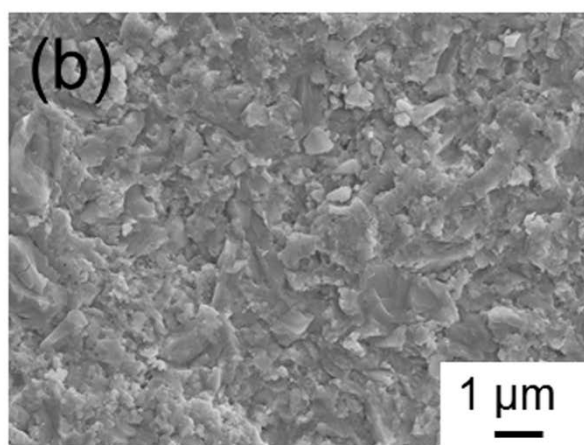
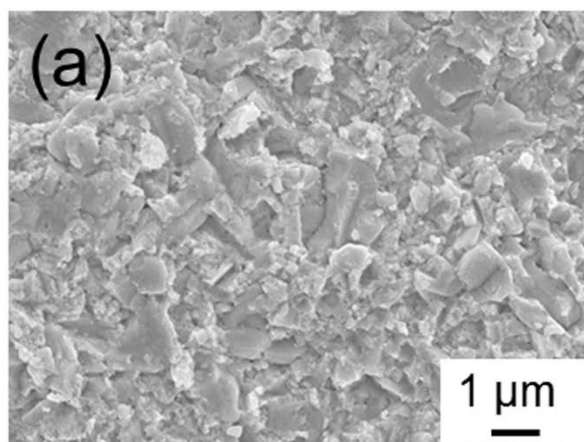


Figure 6, Y. Masubuchi, et al.,

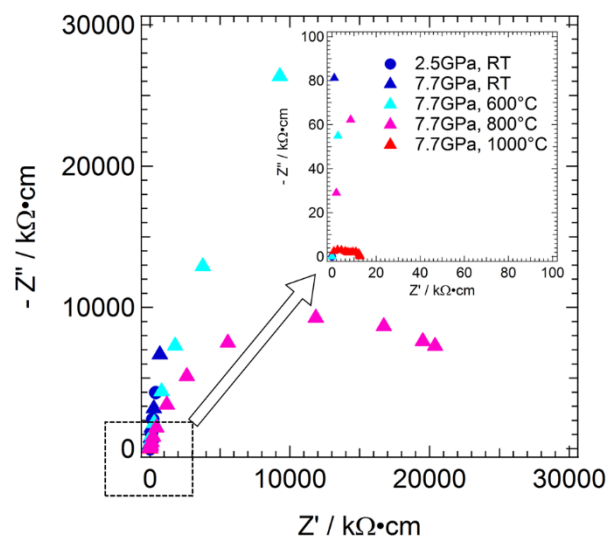


Figure 7, Y. Masubuchi, et al.,

