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Magnetic anisotropy in the van der Waals ferromagnet VI_3

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A comprehensive study of magnetocrystalline anisotropy of a layered van der Waals ferromagnet VI_3 was performed. We measured angular dependences of the torque and magnetization with respect to the direction of the applied magnetic field within the basal ab plane and a general orthogonal plane to ab , respectively. A twofold butterflylike signal was detected by magnetization in the orthogonal plane. This signal symmetry remains conserved throughout all magnetic regimes as well as through the known structural transition down to the lowest temperatures. The maximum of the magnetization signal and the resulting magnetization easy axis is significantly tilted from the normal to the basal ab plane by $\sim 40^\circ$. The close relation of the magnetocrystalline anisotropy to the crystal structure was documented. In contrast, a two-fold-like angular signal was detected in the paramagnetic region within the ab plane in the monoclinic phase, which transforms into a six-fold-like signal below the Curie temperature T_C . With further cooling, another six-fold-like signal with an angular shift of $\sim 30^\circ$ grows approaching T_{FM} . Below T_{FM} , in the triclinic phase, the original six-fold-like signal vanishes, being replaced by a secondary six-fold-like signal with an angular shift of $\sim 30^\circ$.

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I. INTRODUCTION

Magnetic van der Waals (vdW) materials have recently become hot subjects of interest because of their potential use in atomically thin devices for spintronic and optoelectronic functionalities [1–5]. Exploring new chemistry paths to tune their magnetic and optical properties enables significant progress in fabricating heterostructures and ultracompact devices [6–9] by mechanical exfoliation [10] or well-controlled element deposition [11]. Despite belonging to a well-studied family of transition metal trihalides, VI_3 has received significant attention only recently [10,12,13]. Nevertheless, the first studies pointed out that the magnetism of VI_3 is very complicated. This is probably related to the incomplete high spin state t_{2g} shell in V^{3+} ions here, which allows for a Jahn-Teller distortion. The detailed results of ^{51}V and ^{127}I NMR spectroscopy complemented by temperature dependences of specific-heat and magnetization [14] revealed two distinct ferromagnetic (FM) phases, the ground-state one having the critical temperature $T_{\text{FM}} = 26$ K and another existing between T_{FM} and $T_C = 49.5$ K [14]. VI_3 behaves at low temperatures as a hard ferromagnet with a high coercive field $\mu_0 H_c \sim 1$ T at 2 K for the magnetic field applied parallel to the c axis ($H \parallel c$) [10,12], while H_c value is considerably lower in the perpendicular direction ($H \parallel ab$). In fields above 2 T, the mag-

netization M in $H \parallel c$ is proportional to $\sim 1.2 M$ in $H \parallel ab$. This anisotropy of magnetization persists at least up to 9 T [10,12].

A crucial property of the vdW materials with considerable application potential is the strong magnetocrystalline anisotropy [2]. In layered materials, two limits of the easy axis are in-plane (XY model) and out-of-plane (Ising model). The anisotropy features of VI_3 were investigated by magnetization measurements. These have shown a different value of saturated magnetization (higher along the c axis) and a significantly larger low-temperature magnetization loop hysteresis along the c axis [12,15,16]. The Raman scattering study has also identified an acoustic magnon mode (the spin-wave gap) and a two-magnon mode in the low-temperature FM state. With the help of linear spin-wave theory calculations, the magnetic anisotropy and intralayer exchange strength are estimated as 1.16 and 2.75 meV. The sizeable magnetic anisotropy implies the stability of the FM order in the 2D limit with a high critical field [15].

First-principles calculations have been used to investigate the magnetism and magnetic anisotropy of VI_3 theoretically, however, mostly in the single-layer or bilayer limit [17–20], with only a few works considering the bulk form [10,21]. Overall, magnetocrystalline anisotropy with an easy axis along c has been predicted for both bulk and single layer in

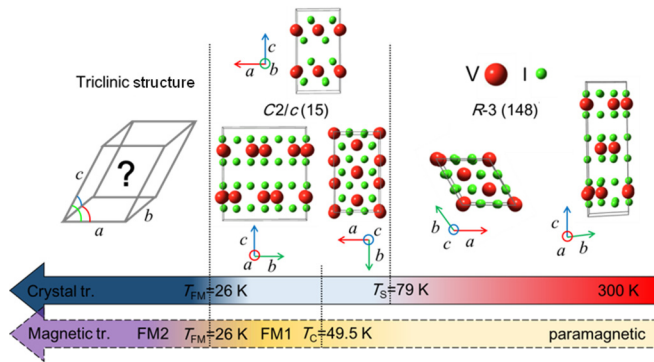


FIG. 1. The scheme shows the evolution of crystal and magnetic transitions in VI_3 below room temperature. The trigonal structure corresponding to the 79–300 K temperature range was taken from Ref. [25], the monoclinic structure, which is stable between $T_S = 79$ K and $T_{FM} = 26$ K was taken from Ref. [12], and the so far unknown triclinic structure, proposed in Ref. [25], is finally established below $T_{FM} = 26$ K. For the transition temperatures, we have used the notation and values from Refs. [14,26].

these works. A high orbital momentum on V was found in a calculation based on the linear augmented plane wave method, which led to a robust magnetic anisotropy dominated by a single-ion contribution of 15.9 meV per formula unit. Based on these findings, it was concluded that VI_3 behaves as an Ising-like ferromagnet, similar to CrI_3 [22]. In a different calculation based on the projector augmented-wave framework, the energy difference between the in-plane and out-of-plane magnetization orientation was found to be strongly dependent on both Hubbard U as well as strain. Furthermore, this energy difference changes sign at $U \sim 3.5$ eV, close to the expected value of U [23].

Currently, there is a controversy concerning the VI_3 crystal structure and its conjunction with magnetism. Kong *et al.* have reported a trigonal ($R-3$) structure at room temperature with a structural transition below $T_S = 78$ K [13]. A contradictory result has been reported by Son *et al.* [12] showing a structural phase transition at $T_S = 79$ K where the crystal symmetry lowers with cooling from the trigonal $P-31c$ structure to a monoclinic $C2/c$ in which the system becomes ferromagnetic at temperature $T_C = 50$ K. Vice versa, Tian *et al.* [10] claim that the VI_3 structural phase transition is similar to the structural transition of CrI_3 [24], i.e., they observed a lowering of the symmetry from low-temperature trigonal $R-3$ to high-temperature monoclinic $C2/m$ with heating through T_S . A detailed crystal structure study by Doležal *et al.* has confirmed the trigonal ($R-3$) crystal structure at room temperature in agreement with Kong [13] and found a subsequent symmetry lowering to a triclinic structure at $T_{FM2} = 32$ K when the ferromagnetic phase FM I transforms to a different ferromagnetic phase FM II. The details of the triclinic structure, however, remain unresolved. Structural and magnetic transitions are schematically summarized in Fig. 1 as well as the used notation.

The connection of the structure with the magnetic phase transition in VI_3 suggests a considerable role of magnetoelectric interactions in this compound. It is also supported by magnetostriction-induced changes of the monoclinic-structure

parameters at T_C . Surprisingly, the temperature of structure transition is decreased by magnetic fields applied along the trigonal c axis and is intact by a magnetic field applied within the basal plane. These results indicate that the magnetic field applied within the threefold axis of the VI_6 octahedrons stabilizes the symmetry of the V honeycombs in the basal plane [25].

The direct observation of the VI_3 magnetocrystalline anisotropy was revealed by the angular dependence of magnetization at temperature 2 K in various magnetic fields H rotated within the ab plane and out-of-plane (orthogonal plane). The out-of-plane anisotropy exhibits a “butterflylike” pattern [16], different from the other 2D FM semiconductor $\text{Cr}_2\text{Ge}_2\text{Te}_6$ [27]. On the other hand, the angular signal of magnetization within the basal ab plane shows a six-fold-like symmetry. However, anisotropy studies so far performed were mostly focused on the lowest temperature magnetic regime below T_{FM} . Further, detailed measurements throughout the entire temperature interval covering all detected magnetic regimes and structural phases are missing and highly desirable to understand complex magnetic interactions and anisotropy in the VI_3 vdW ferromagnet. The VI_3 magnetic structure, exchange interactions, and magnetocrystalline anisotropy are much more complicated than the earlier studied Cr tri-halides [28–32].

This paper focuses on studying magnetocrystalline anisotropy features of VI_3 in a wide temperature interval by angular-dependent magnetic torque and magnetization measurements for the magnetic field applied out-of-plane marked as “ ac ” (a general perpendicular plane to the ab basal plane) and within the ab basal plane with respect to room temperature trigonal structure.

II. EXPERIMENTAL METHODS

The single crystals were prepared by the chemical vapor transport method, as described elsewhere [12], and come from the identical batch of samples used in the work of Doležal *et al.* [25]. The angular dependence of the magnetization was measured by MPMS (Quantum Design) devices using thin single crystals with a sample area of $1.4 \times 1.8 \text{ mm}^2$ within the principal trigonal basal ab plane about angle (ϕ) and a general orthogonal plane to basal plane about angle (θ). The sample was placed on a handmade rotator with a rotation axis orthogonal to the external magnetic field. Magnetic torque was measured using a commercial Quantum Design Torque magnetometer chip attached on a 370° rotator installed inside a 9T PPMS apparatus. To measure the angular dependence of torque within the principal ab plane with respect to magnetic field orientation, an L-shaped copper element had to be attached to the torque chip (Fig. 2). The sample was fixed with Apiezon grease. The torque magnetometer chip can typically be calibrated. However, because of the installed copper L-element, calibration was not possible to finalize, and so we do not focus on the absolute value of the torque. The scale of the data for each temperature was normalized to have the same amplitude. The L-element mass contribution was tested by the rotation of the chip in zero magnetic field at room temperature and the observed signal was negligible compared to the signal from the sample in the magnetic field below T_C .

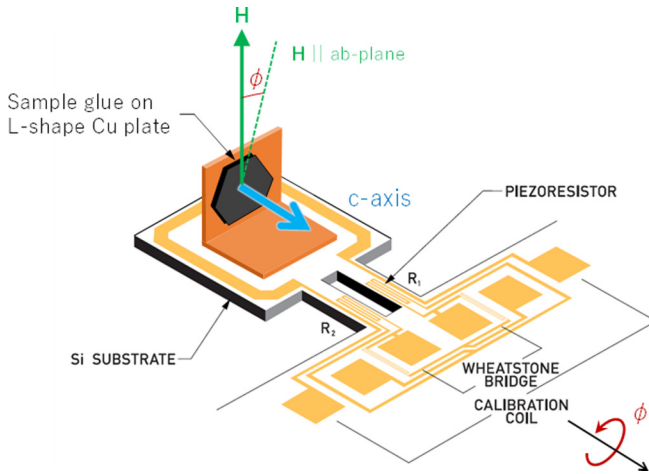


FIG. 2. The scheme of the magnetic torque chip with the installed sample attached to the copper L element. The scheme was taken from the Quantum Design PPMS manual and modified. The red arrow marks the direction of the chip rotation (ϕ) within the principal ab plane of the sample. The green arrow marks the direction of the external magnetic field.

The magnetic field is rotating within the ab plane during the torque measurement. Since the magnetic field was applied within the plane of the thin sample, the shape of samples and the demagnetization factor are expected not to have a significant effect on the results.

All samples used for measurements were stored under the inert Ar (6N purity) atmosphere in a glovebox. The contact with air was minimized only at the time of the installation of the rotators into the MPMS and PPMS instruments.

III. RESULTS

Within our study, we have performed a series of angular-dependent magnetization scans within the orthogonal plane, perpendicular to the ab plane (Fig. 3). Our low-temperature (2 K) and high-magnetic-field (5 T) angular magnetization results in the form of a butterflylike signal in polar projection agreeing with the previous result in Ref. [16]. Both the symmetry and positions of the maxima of the angular magnetization signal remain conserved within this plane throughout all magnetic regimes and the structural transition to the triclinic phase [25] at T_{FM} . The maximum magnetization signal tilted approximately 40° out of the normal to the basal ab plane.

The angular-dependent magnetic torque measurements within the basal ab plane in identical magnetic field 5 T at various temperatures are shown in Fig. 4.

Angular-dependent magnetic torque provides information mainly about the magnetocrystalline anisotropy strength. In contrast, the angular-dependent magnetization measurements show the size of the magnetic moment at the given orientation of the sample with respect to the direction of the external magnetic field. The isothermal scans of the torque have clearly revealed significant evolution of the magnetocrystalline anisotropy (Fig. 4). A double peaklike signal in the paramagnetic region in the monoclinic phase transforms to a six-fold-like signal below T_C . The six-fold-like symmetry is

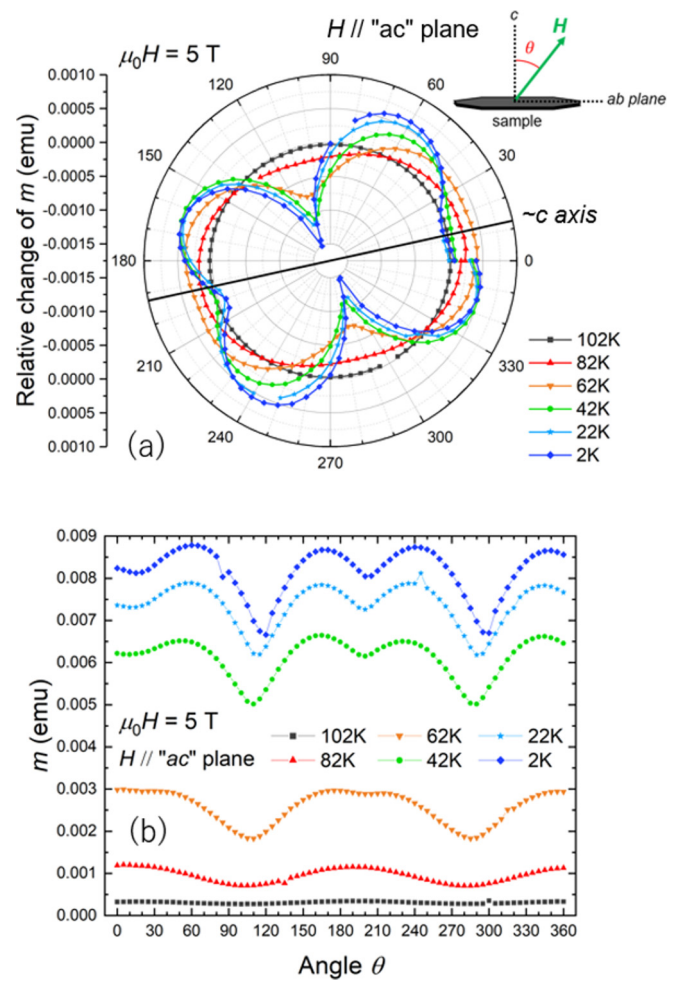


FIG. 3. Angular-dependent magnetization isotherms in magnetic field 5 T rotated within the orthogonal “ ac ” plane. The polar plot (a) and classical plot (b) shows selected curves at specific magnetic regimes; above T_C (102, 82, 62, and 52 K), below T_C and above T_{FM} (42 K), and below T_{FM} (22 and 2 K). The black solid line around 12° in Fig. 3(a) shows the approximate position of the c axis, which is determined from the maximum value of the paramagnetic signal.

present down to the lowest temperatures. However, a detailed analysis of the curves between T_C and T_{FM} shows signatures of the gradual splitting of each maximum when approaching T_{FM} . At a temperature of 30 K in the vicinity of T_{FM} the signal is split into two sets of six-fold-like signals of similar intensity mutually rotated by 30° [see Fig. 4(b), green line]. The original six-fold-like signal, which has emerged at T_C , becomes gradually reduced with decreasing temperature in favor of the intensity of the second set. The original set vanishes below T_{FM} whereas the second set remains down to the lowest temperature at the fixed angular position.

The angular-dependent scans of magnetization isotherms with magnetic field 5 T applied within the basal ab plane are shown in Fig. 5.

The angular-dependent magnetization measurements have shown complementary results with the magnetic-torque measurements. One six-fold-like signal was detected below T_C . With further cooling, a clear signature of a second six-fold-like signal rotated by $\sim 30^\circ$ is observed [see green line in

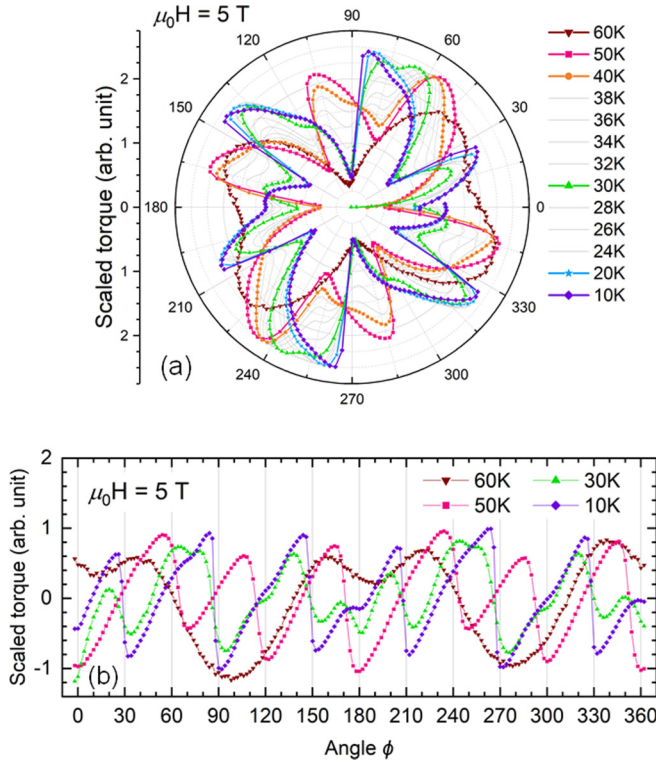


FIG. 4. Angular-dependent magnetic torque scans in magnetic field 5 T rotated within the *ab* plane. The polar plot (a) shows all recorded temperatures. The classical plot (b) shows selected curves at specific magnetic regimes; above T_C (60 K), in the lower vicinity of T_C (50 K), in the upper vicinity of T_{FM} (30 K), and well below T_{FM} (10 K). The intensity of the torque signal is on a relative scale.

Fig. 5(b). The second signal gradually grows at the expense of the original one's intensity. Below T_{FM} , the original six-fold-like signal vanishes, and only the second one is detected. The signal in the paramagnetic regime at temperatures below T_s is relatively weak, with two-fold-like broad maxima around $\sim 15^\circ$ and $\sim 195^\circ$ [Fig. 5(b)] which can be distinguished analogically to the symmetry of magnetic torque data. The averaged angular value of the magnetic moment at given temperatures is plotted in Fig. 5(c) and saturates around value $1.7 \mu_B/V$, which is lower than expected for the V^{3+} ion.

IV. DISCUSSION

Our detailed experimental study of magnetocrystalline anisotropy by angular-dependent magnetization has unambiguously proved that the magnetic moment is canted from the *ab* plane normal direction by $\sim 40^\circ$. A somewhat surprising result is the robustness of the anisotropy throughout the entire temperature interval covering both ferromagnetic phases and particularly the structural transition from the monoclinic to likely triclinic phase accompanying the magnetic order-to-order transition at T_{FM} .

A more complex angular response of the magnetization and torque was detected in the basal *ab* plane. In the following discussion, we will place the *ab* plane in the plane of the VI_3 layer and take the direction of the twofold axis as the *b* axis. From now, we will simply call the normal to the *ab* plane the

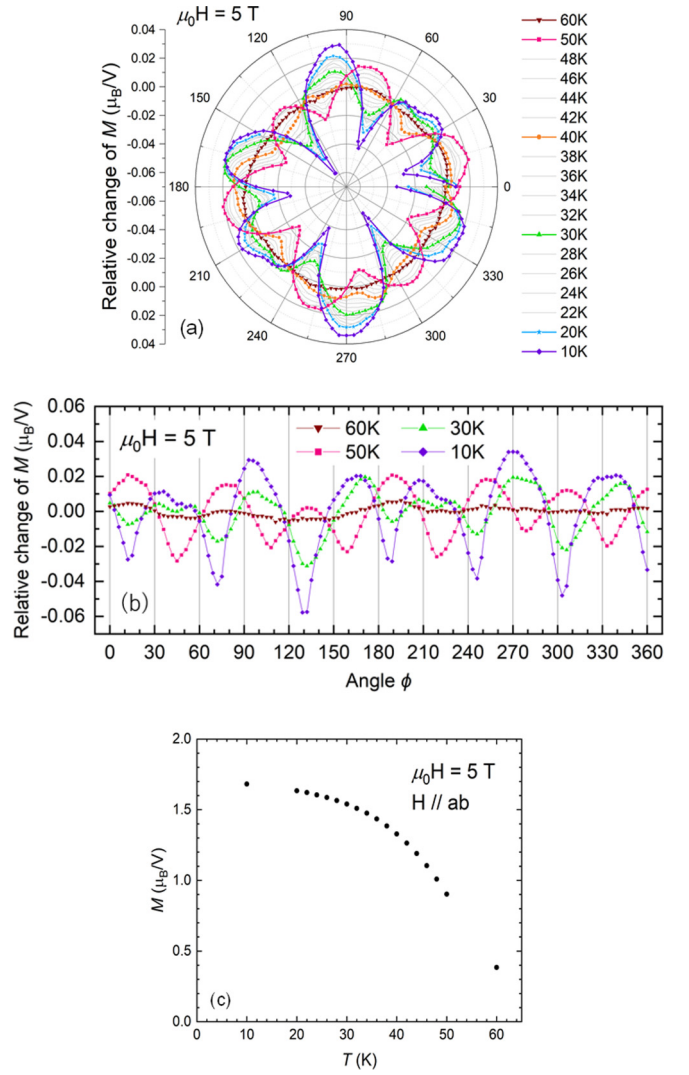


FIG. 5. Angular-dependent magnetization isotherms in THE magnetic field of 5 T rotated within the *ab* plane. The polar plot (a) shows all recorded temperatures. The classical plot (b) shows selected curves at specific magnetic regimes; above T_C (60 K), below T_C (50 K), in the upper vicinity of T_{FM} (30 K), and well below T_{FM} (10 K). Panel (c) shows the temperature evolution of the magnetization averaged over the entire angular scan. Note that the initial value of the angle ϕ here does not necessarily need to coincide with Fig. 4.

“*c* axis”, although the crystallographic *c* axis depends on the respective space group.

When the system reveals three-fold-rotational symmetries around the *c* axis, which is the case of the trigonal $R\bar{3}$ space group for VI_3 above T_s , only the $\sin 6\phi$ term is present in the tensor τ_z of magnetization torque [Eq. (1), see Appendix].

$$\tau_z = M_x H_y - M_y H_x = \chi_{330} H^6 \sin 6\phi. \quad (1)$$

Focusing on the paramagnetic region at 60 K below T_s (Fig. 6), the best agreement for experimental torque data was found using the following formula:

$$\tau(\phi) = k_2 \sin 2\phi + k_4 \sin 4\phi + k_6 \sin 6\phi, \quad (2)$$

where twofold and fourfold signals have been observed. This is a direct indication that the symmetry of the crystal below

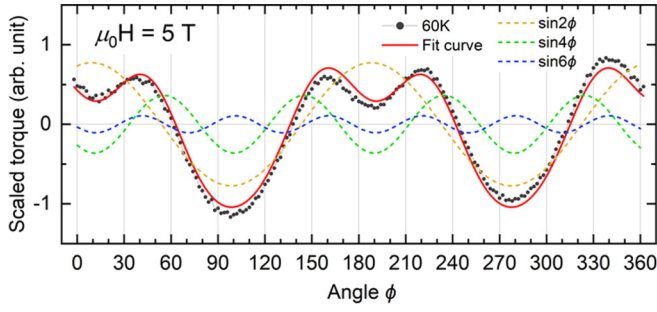


FIG. 6. Angular dependence of torque at 60 K and fitting of the result by combining trigonometric functions.

T_s is lowered at least to a monoclinic one at this temperature because the $\sin 2\phi$ and $\sin 4\phi$ contributions are forbidden for threefold rotational symmetries in the tensor τ_z [Eq. (1)] for a trigonal structure.

Since the highest point group of a single layer in VI_3 is D_{3d} , there are three possible orientations of twofold axes when symmetry is reduced to the monoclinic structure (Fig. 7). These form three domains associated with each other on a threefold axis perpendicular to the ab plane. On the other hand, when a magnetic field is applied to the ab plane of a system with twofold rotations around the b (or a) axis, the torque in the paramagnetic state can be expressed only by the sum of sine functions [Eq. (2)].

If the volume ratio of the three domains is $A : B : C$ ($A + B + C = 1$), the torque applied to the entire system is

$$\begin{aligned}\tau_{\text{tot}}(\phi) &= A\tau(\phi) + B\tau(\phi - \pi/3) + C\tau(\phi - 2\pi/3) \\ &= D\{k_2 \sin 2(\phi + \alpha/2) + k_4 \sin 4(\phi - \alpha/4) \\ &\quad + k_6 \sin 6\phi\},\end{aligned}\quad (3)$$

where D and α are constants determined by the volume ratio of the domains. As shown in Fig. 6, the phase of the sine function of the second and fourth oscillation terms is out of phase, which indicates that this is the sum of multiple domains. Since the phase shift due to the domain ratio does not occur in the sixfold oscillation term, the angle at which the torque becomes zero corresponds directly to the main axis of the crystal.

At 50 K near the ferromagnetic transition at T_C , a sixfold saw-tooth-like signal was observed (Fig. 4). This is due to a rapid change in sign due to switching of the ferromagnetic domains, indicating that the magnetization is not fully polarized in the direction of the external magnetic field. This implies that the magnetocrystalline energy due to the VI_6 octahedron

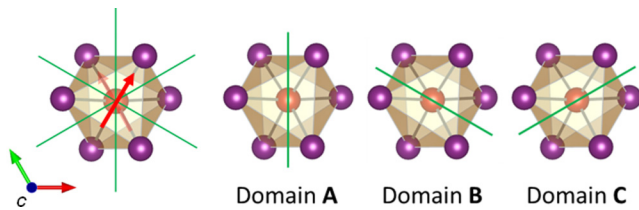


FIG. 7. Schematic view of three twofold axes and three possible domains when VI_3 transforms from the trigonal to the monoclinic structure.

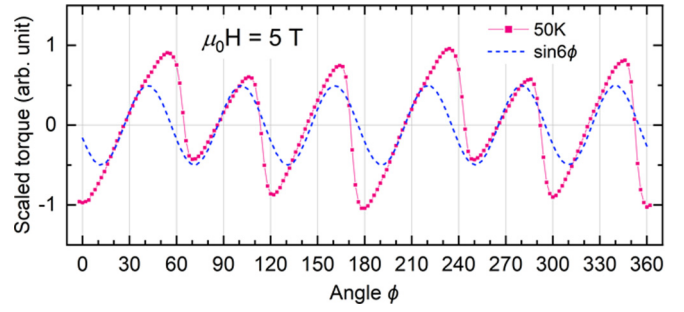


FIG. 8. The angular dependence of torque at 50 K. The dashed blue line is a scaled graph of the sixfold oscillation term from the 60 K result (Fig. 6).

is sufficiently large with respect to the Zeeman energy. The result can be understood qualitatively by considering the VI_6 octahedron to be a regular octahedron as a first approximation. The magnetocrystalline anisotropy energy $E_a^c(\theta, \phi)$, represented by the symmetric representation of the point group O_h , can be written as follows:

$$\begin{aligned}E_a^c(\theta, \phi) &= K_1 \left\{ \frac{(7\cos^4\theta - 6\cos^2\theta + 3)}{12} \right. \\ &\quad \left. - \frac{\sqrt{2}}{3} \sin^3\theta \cos\theta \cos 3\phi \right\}.\end{aligned}\quad (4)$$

This equation is obtained from the commonly used magnetic anisotropy equation $K_1(\alpha_x^2\alpha_y^2 + \alpha_y^2\alpha_z^2 + \alpha_z^2\alpha_x^2)$ where α_i represents the direction cosine. We obtain this equation by making the $[111]$ axis into a new c axis, which corresponds to the threefold rotation axis of the VI_6 octahedron by an orthogonal transformation. This formula confirms that the easy axis direction of spontaneous magnetization has six values, $\phi = 0, \pi/3, 2\pi/3, \pi, 4\pi/3, 5\pi/3$, independent of the sign of K_1 . Therefore, when the magnetic field is rotated within the ab plane, the moment switches to the most stable direction among these easy axes that exist every 60° , as shown in Fig. 8.

When the magnetic field is in the middle of the two easy axes, an abrupt change in the sign of the torque takes place. Therefore, the main axis of the crystal is also considered to be in this position. In Fig. 8, the six oscillations in the fitting results for the paramagnetic state are shown together, and the two results correspond to each other.

As the temperature was lowered further within the ferromagnetic region, magnetic torque measurements indicate there was a significant change in the easy magnetization axis (Fig. 4). Although this change does not have a clear boundary temperature, another component with an easy axis tilted about $\sim 30^\circ$ gradually grows and finally replaces the easy axis direction at the lowest temperature. This change cannot be explained, for example, by a change in the sign of the magnetic anisotropy energy $E_a^c(\theta, \phi)$. Previous reports have suggested the presence of a first-order structural phase transition from monoclinic to triclinic at around 30 K [25], and we expect to see a qualitative change reflecting this observation. Detailed crystal structure analysis at low temperatures below T_{FM} is highly desirable to understand the mechanism of the transition. The surprising result is that the orthogonal plane signal is

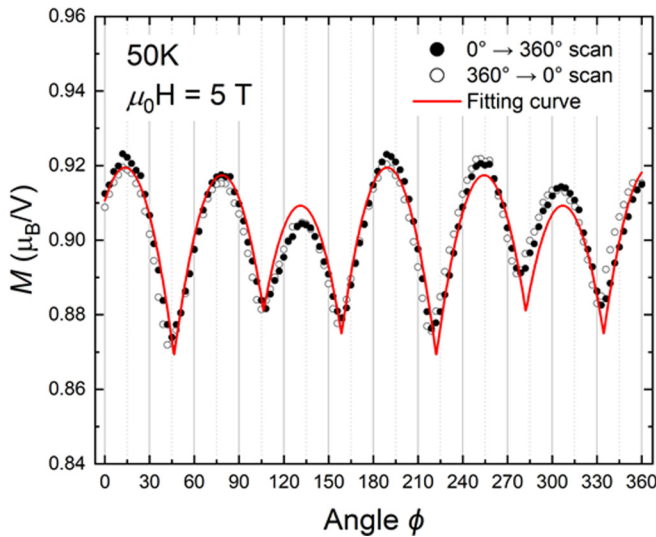


FIG. 9. Angular-dependent magnetization isotherm at 50 K and in magnetic field 5 T rotated within the ab plane and the fit at 50 K given by Eq. (5).

insensitive to signal symmetry change within the ab plane. We can only speculate that the FM I and FM II magnetic structures are very similar, only mutually rotated by 30° .

The anomaly of the phase transition at T_{FM} is very small in most of the macroscopic measurements, so even if the triclinic term component in the magnetic anisotropy energy becomes nonzero below T_{FM} , its magnitude may be small. Furthermore, the x-ray data indicate that the triclinic phase differs only slightly from the monoclinic one; it is rather a distortion. Therefore, we speculate that this 30° change may not directly reflect a triclinic term, but rather a significant reorientation of the other highly symmetric terms.

The results of the angular dependence of magnetization in Fig. 5 are in qualitative agreement with the torque result;

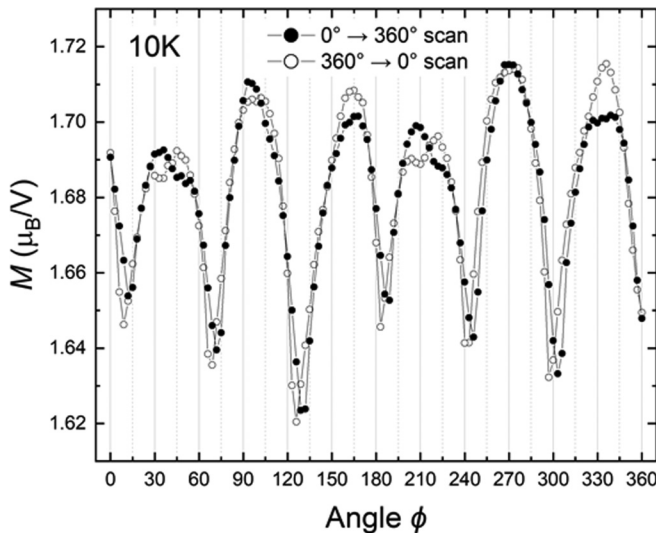


FIG. 10. Angular dependence of the magnetization at 10 K rotated in a clockwise and anticlockwise direction in an identical magnetic field.

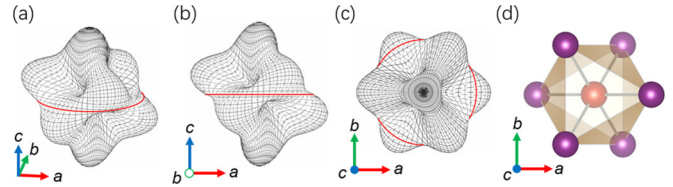


FIG. 11. (a)–(c) The plot shows angular dependence of the calculated magnitude of magnetocrystalline anisotropy energy $E_a^c(\theta, \phi)$ in the simplified O_h representation for $K_1 > 0$ [Eq. (4)] as a function of radius. The principal axis (θ) is taken in the $[111]$ direction of the cube which corresponds to the trigonal c axis of the VI_3 crystal. The magnetic moment is then stable in the $\langle 100 \rangle$ direction. The red solid line shows $\theta = 90^\circ$ (d) VI_6 octahedron with orientation corresponding to (c).

we obtained six-fold-like signals. The folding back when the magnetization is reduced corresponds to ferromagnetic domain switching. The original easy axis changes with decreasing temperature about $\sim 30^\circ$ below T_{FM} , being conserved down to the lowest temperature of 10 K. For simplicity, to analyze the data, we have decomposed the model into spontaneous magnetization M_s , which is independent of the magnitude of the magnetic field, and polarized magnetization M_p , which follows the magnetic field. In this model, the result is fitted with a function of $|\cos\phi| + \text{const}$ in Eq. (5), as shown in Fig. 9:

$$M_s \left\{ A \left| \cos \left(\frac{\pi}{w} (\phi - \phi_0) \right) \right| + B \left| \cos \left(\frac{\pi}{w} (\phi - \phi_0 - 60) \right) \right| + C \left| \cos \left(\frac{\pi}{w} (\phi - \phi_0 - 120) \right) \right| \right\} + M_p. \quad (5)$$

Fitting result $M_s = 0.44(1) \mu_B/\text{V}$ and $M_p = 0.62(1) \mu_B/\text{V}$ is roughly consistent with magnetization results of previous experiments at 50 K.

Figure 10 shows the angular dependence of the magnetization at 10 K. The hysteresis is not noticeable at 50 K, but some non-negligible hysteresis is observed at 10 K. Hysteresis, such as the intersection of clockwise and anticlockwise data that exists around 30° and 210° , is typically expected to occur at the minima of magnetization where the domain switches. This suggests that the two components, one developed at around 50 K and the other developed at a lower temperature, coexist as independent signals.

The observed anisotropy within the ab plane can be understood using projections of the magnetocrystalline anisotropy energy $E_a^c(\theta, \phi)$ in a simplified O_h representation for the $[111]$ direction as the principal axis of the VI_6 octahedron for magnetic moment stable in the $\langle 100 \rangle$ direction of the cube, i.e., the direction of Iodine that leads to the six-fold-like signal [see Figs. 11(c) and 11(d)]. On the other hand, the projection of the easy axis in the “ ac ” plane [Fig. 11(b)] leads to a two-fold-like signal, in which the expected moment flipping position is not every 90° .

V. CONCLUSIONS

We have performed a detailed study of the magnetocrystalline anisotropy of VI_3 by angular-dependent torque

magnetometry and magnetization within the ab plane and the orthogonal plane throughout the entire 10–60 K temperature interval covering both the two ferromagnetic phases and the paramagnetic phase. Irrespective of the temperature change, the symmetry of the twofold butterflylike angular signal within the orthogonal plane to basal ab plane remains conserved throughout both FM phases. On the other hand, substantial development of the signal was detected within the ab plane. The double-peak-like signal in the paramagnetic region in the monoclinic phase transforms to the six-fold-like signal below T_C as a product of three possible monoclinic domains. The original six-fold-like signal appearing at T_C disappears below T_{FM} and the second set of six-fold-like signal rotated by about 30° survives down to the lowest temperatures. The gradual emergence of the secondary six-fold-like signal growth between T_C and T_{FM} and the angular shift of magnetization maxima about $\sim 30^\circ$ below T_{FM} is a subject of further research and it conflicts with the predicted scenario of the first-order transition at T_{FM} . Since the torque contains a periodic function of 2ϕ and 4ϕ , we corroborate that the threefold symmetry present in the trigonal lattice is broken in the studied temperature range.

Our findings are in agreement with the presence of low symmetry monoclinic and triclinic phases at low temperatures. We have unambiguously shown that the easy axis is not perpendicular to the layers, but canted by around 40° from the normal to the ab plane in both FM phases which fundamentally differentiate the complex magnetism and anisotropy of VI_3 from the Ising ferromagnetism of CrI_3 . For further progress in this field, it is essential to have detailed knowledge about the predicted triclinic crystal structure below T_{FM} . Thus, a neutron diffraction study to reveal the VI_3 magnetic structure is highly desirable.

ACKNOWLEDGMENTS

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APPENDIX: TORQUE EXPRESSION IN THE PARAMAGNETIC STATE

Considering the situation where time-reversal symmetry is preserved, we expand the magnetization as a function of the magnetic field as follows:

$$M_i = \chi_{ij}^{(2)} H_j + \chi_{ijkl}^{(4)} H_j H_k H_l + \chi_{ijklmn}^{(6)} H_j H_k H_l H_m H_n + \dots (i, j, k = x, y \text{ or } z).$$

Each $\chi^{(k)}$ in this formula represents the magnetic susceptibility tensors of the k th rank, and here, the general sum reduction rule is used. Since these tensors are symmetrical with respect to the interchange of indices, the distinction between independent components is determined only by the number of the subscripts x , y , and z . Therefore, for simplicity, we have decided to denote these tensors by the number of its xyz , for example, as follows:

$$\chi_{yz}^{(2)} \equiv \chi_{011}, \quad \chi_{xyyz}^{(4)} \equiv \chi_{121}, \quad \chi_{xxxxxx}^{(6)} \equiv \chi_{600}.$$

We consider the situation where a magnetic field is applied in the $ab(xy)$ plane, so $(H_x, H_y, H_z) = (H \cos \phi, H \sin \phi, 0)$. Then the general equation for the torque up to the 6th order can be written as follows:

$$\begin{aligned} \tau_z = & \left\{ \frac{1}{2}(\chi_{200} - \chi_{020})H^2 + \frac{1}{4}(\chi_{400} - \chi_{040})H^4 + \frac{5}{32}(\chi_{600} + \chi_{420} - \chi_{240} - \chi_{060})H^6 \right\} \sin 2\phi \\ & - \left\{ \chi_{110}H^2 + \frac{1}{2}(\chi_{310} + \chi_{130})H^4 + \frac{5}{16}(\chi_{510} + 2\chi_{330} + \chi_{150})H^6 \right\} \cos 2\phi \\ & + \left\{ \frac{1}{8}(\chi_{400} - 6\chi_{220} + \chi_{040})H^4 + \frac{1}{8}(\chi_{600} - 5\chi_{420} - 5\chi_{240} + \chi_{060})H^6 \right\} \sin 4\phi \\ & - \left\{ \frac{1}{2}(\chi_{310} - \chi_{130})H^4 + \frac{1}{2}(\chi_{510} - \chi_{150})H^6 \right\} \cos 4\phi + \left\{ \frac{1}{32}(\chi_{600} - 15\chi_{420} + 15\chi_{240} - \chi_{060})H^6 \right\} \sin 6\phi \\ & - \left\{ \frac{1}{16}(3\chi_{510} - 10\chi_{330} + 3\chi_{150})H^6 \right\} \cos 6\phi. \end{aligned}$$

Generally, the tensor is limited in its components by the point group symmetry of the crystal. Equation (1) in the discussion considers a situation in which there are threefold rotational symmetries around the $c(z)$ axis. In this case, each component of susceptibility has the following relationship to each other:

$$\begin{aligned} \chi_{200} = \chi_{020}, \quad \chi_{110} = 0, \quad \chi_{400} = 3\chi_{220} = \chi_{040}, \quad \chi_{310} = \chi_{130} = 0, \\ \chi_{600} = 5\chi_{420} = 5\chi_{240} = \chi_{060}, \quad \chi_{510} = -\chi_{330} = \chi_{150}. \end{aligned}$$

Taking this into account, the equation of torque is expressed as follows:

$$\tau_z = \chi_{330}H^6 \cos 6\phi.$$

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