



Title	Isotope Microscopic Observation of Osteogenesis Process Forming Robust Bonding of Double Network Hydrogel to Bone
Author(s)	Nonoyama, Takayuki; Wang, Lei; Tsuda, Masumi et al.
Citation	Advanced Healthcare Materials, 10(3), 2001731 https://doi.org/10.1002/adhm.202001731
Issue Date	2021-02-03
Doc URL	https://hdl.handle.net/2115/84005
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Type	journal article
File Information	Re_manuscript.pdf



1 **Isotope Microscopic Observation of Osteogenesis Process Forming Robust Bonding of**
2 **Double Network Hydrogel to Bone**

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7
8 ((Optional Dedication))

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44 Keywords: Tough Hydrogel, Hydroxyapatite, Osteogenesis, Bone Absorbable Materials,
45 Isotope Microscopy

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1 Abstract

2 Tough double network (DN) hydrogels are promising substitutes of soft supporting
3 tissues such as cartilage and ligaments. For such applications, it is indispensable to robustly fix
4 the hydrogels to bones with medically feasible methods. Recently, we succeeded in robustly
5 bonding the DN hydrogels to defected bones of rabbits *in vivo*. The osteo-conductance was
6 achieved by mineralizing in the surface layer of the DN hydrogels with calcium-phosphate-
7 hydroxide salt hydroxyapatite (HAp) nano crystals of low crystallinity. Spontaneous
8 osteogenesis penetrating into the semi-permeable hydrogels occurs to form a gel/bone
9 composite layer of 40 μm -thick within four weeks after the implantation. In this work, we
10 implanted the ^{44}Ca isotope-doped HAp/DN hydrogel in a defect of rabbit femoral bone,
11 and analyzed the dynamic osteogenesis process at the gel/bone interface by tracing the calcium
12 isotope ratio $^{44}\text{Ca}/^{40}\text{Ca}$ using isotope microscopy. The synthetic HAp hybridized on the surface
13 layer of DN gel dissolves rapidly in the first two weeks by inflammation, and then the immature
14 bone with a gradient structure starts to form in the gel region, reutilizing the dissolved Ca ions.
15 These results reveal, for the first time, that HAp hybridized in the DN hydrogel is reutilized for
16 the osteogenesis to form the bone/gel composite layer at the gel/bone interface. These facts also
17 help to understand the lifetime of bone absorbable materials and to elucidate the mechanism of
18 spontaneous, non-toxic, but strong fixation of hydrogels to bones.

19

20 Main text

21 Hydrogels, a class of soft materials storing a large amount of water in three-dimensional
22 polymeric network, possess flexibility, low friction and high permeability of small molecules
23 and ions. Owing to the recent development of several strategies to toughen the soft materials,
24 many hydrogels have acquired sufficiently high strength and toughness as load-bearing
25 structural biomaterials [1-4]. Double network (DN) hydrogels, a representative in the class of
26 tough hydrogels, possess modulus, strength, and toughness that are comparable to those of

1 cartilages while containing ~90 wt% water [1,2]. The invention of tough and strong hydrogels
2 provides novel strategies to treat injured component tissues of human joint, such as articular
3 cartilage, meniscus, and ligament, using various artificial tissues and implants to induce tissue
4 regeneration [5-7]. For example, localized replacement of only the injured articular cartilage
5 with the artificial cartilage would provide a potential surgical option, which is less invasive than
6 the currently available total joint replacement using hard metallic or sintered ceramics
7 components [8]. To apply hydrogel materials for artificial cartilage applications, a robust
8 fixation to bone tissue is indispensable. However, it is extremely challenging to establish a non-
9 toxic and medically feasible method because no glues work well on such a watery surface of
10 hydrogels. Calcium phosphate salts such as low-crystalline hydroxyapatite (HAp) and β -
11 tricalcium phosphate (β -TCP) have been applied to promote bone regeneration as bone
12 absorbable materials because of their excellent bio-absorbable and osteoconductive properties
13 [9,10]. Recently, we developed a technique to make a robust fixation of DN gel to bone tissues
14 *in vivo*. The technique is by mineralizing low-crystalline HAp nanoparticles with ~600 nm
15 diameter to the surface layer of the semi-permeable DN hydrogel [11-15]. When HAp-
16 hybridized DN gel is implanted in a defect created in bones, the HAp induces osteogenesis
17 penetrating into the gel matrix, which forms the strong bonding of the DN gel to bone within 4
18 weeks. On the other hand, the pristine DN gel without HAp-hybridization did not show any
19 bonding to bone at all even after 12 weeks implantation. This technique provides a general,
20 non-toxic, and medically feasible approach to form robust fixation of the semi-permeable DN
21 hydrogels to bones by the formation of gel/bone composite layer.

22 In this work, we intend to clarify the dynamic process of HAp-induced osteogenesis
23 penetrating into the DN gel matrix, which results in the robust fixation of gel to bone tissue.
24 We focus on the reutilization of the synthetic HAp during the osteogenesis process. Generally,
25 at the early stage of the bone reconstruction process, chondrocytes construct extracellular
26 matrix (ECM) composed of collagen and proteoglycan at the defect, and then chondrocytes

1 secrete alkaline phosphatase to mineralize HAp from body fluid in the ECM. After that,
2 osteoblasts penetrate into the mineralized ECM with the construction of blood vessel and
3 produce immature woven bone [16,17]. This bone is subsequently deconstructed by osteoblast-
4 induced osteoclast, and then osteoblast remodels functionalized mature bone. In the presence
5 of the bone absorbable materials, it has been generally accepted that such material is dissolved
6 to calcium and phosphate ions by osteoclast and then reutilized to construct bone tissue by
7 osteoblast in the osteogenesis process [18,19]. However, direct evidence to prove the
8 reutilization has not been obtained because both synthetic calcium phosphate implants and bone
9 tissues are made of the same inorganic compound and cannot be distinguished at all. Currently,
10 the only evidence of the “resorbable” is that the material disappears from the implantation
11 location over time [20].

12 To reveal the HAp reutilization, isotope analysis is a strong solution. Yurimoto et al.
13 have developed isotope microscopy, based on the secondary ion mass spectrometry, to evaluate
14 not only geological minerals and meteorites in cosmic science [21-23] but also biominerals [24,
15 25]. In this work, we apply this unique and powerful technique to analyze the HAp reutilization
16 in osteogenesis process at the bone-gel interface. When a HAp/DN hydrogel with isotope-
17 anomalous calcium phosphate salt is implanted in the defect of bone that possesses natural
18 isotope ratio, the implant and tissue can be distinguished based on isotope ratio, and then the
19 reutilization of implant HAp in the osteogenesis should be evaluated by tracing the isotope ratio
20 of regenerated bone. In this study, we implanted the DN hydrogels containing isotope
21 anomalous HAp nanoparticles in the defected femoral trochlea of the patellofemoral joint of
22 rabbits, and the time and spatial evolutions of the isotope anomaly were observed. For the first
23 time, we identified the reutilization of synthetic HAp implant during the osteogenesis. This
24 work strongly supports our previously reported explanation that the dissolution and diffusion
25 of a small amount of HAp nanocrystals stimulate the cells on the bone region to perform
26 osteogenesis forming the gel/bone layer. This work also gives important insight into

1 understanding the bone regeneration process induced by other implanted materials, which is
2 beneficial for the design and development of novel bone-resorbable materials.

3 **Figure 1** shows the schematic illustration of the experimental procedures to prepare the
4 $^{44}\text{HAp}/\text{DN}$ gel (**Figure 1a**), to implant and evaluate the reutilization of the implanted HAp in
5 the osteogenesis *in vivo* (**Figure 1b**). The cylindrical DN gel plug with ^{44}HAp nanoparticles
6 hybridized in the surface layer of its sidewall was implanted in the osteochondral defect created
7 in the rabbit femoral groove of the bilateral patellofemoral joints. The $^{44}\text{HAp}/\text{DN}$ gel implant
8 with surrounding cancellous bone tissue was harvested at 2, 4, 6 and 12 weeks after the surgery,
9 and the 14 μm -thick sections in transverse plane processed by using cryostat under frozen
10 temperature were placed on the silicon wafer. The defrosted and dried slice was coated by gold
11 evaporation. The boundary between $^{44}\text{HAp}/\text{DN}$ gel and cancellous bone of rabbit was observed
12 by the isotope microscope.

13 The optical microscopic images of the slice specimen showed the center circle region
14 of $^{44}\text{HAp}/\text{DN}$ gel plug and surrounding cancellous bone (**Figure 2**). An obvious gap between
15 gel and bone, indicated by red arrow in the figure, was observed at 2 weeks (**Figure 2a-i**). A
16 minor connection between gel and bone, indicated by red arrow in the figure, was observed at
17 4 weeks (**Figure 2b-i**). The gap was hardly observed and most of the interface was connected
18 at 6 weeks (**Figure 2c-i**). The interface was connected completely at 12 weeks (**Figure 2d-i**).
19 No large gap was observed at the sample harvest even for 2 and 4-week samples because the
20 gel plug with 4.5-mm diameter was press-fitted in the created defect (4.3 mm of inner diameter).
21 The gap appeared in the drying process of slice due to less adhesion of $^{44}\text{HAp}/\text{DN}$ gel plug to
22 bone tissue at the boundary. The ^{44}HAp coating region became unclear with increasing
23 implantation period, indicating dissolution and diffusion of HAp over time.

24 The key parameter of this study is the ratio of target isotope ^{44}Ca to major ^{40}Ca
25 ($^{44}\text{Ca}/^{40}\text{Ca}$). The atomic fractions of ^{40}Ca and ^{44}Ca in nature are 96.94% and 2.086%,
26 respectively [26], which gives the natural isotope ratio $^{44}\text{Ca}/^{40}\text{Ca}$ of 0.022. On the other hand,

1 $^{44}\text{Ca}/^{40}\text{Ca}$ of the prepared ^{44}HAp is calculated as 0.132 considering the mixing ratio of $^{44}\text{CaCl}_2$
2 and $^{\text{nat}}\text{CaCl}_2$ (1 : 9 = $^{44}\text{CaCl}_2$: $^{\text{nat}}\text{CaCl}_2$) and purity of isotope reagent (^{40}Ca 2.89%, ^{44}Ca 97.00%)
3 **(Supporting Experimental Section).**

4 First, we measured the values of $^{44}\text{Ca}/^{40}\text{Ca}$ of natural cancellous bone tissue and pristine
5 ^{44}HAp coated DN gel plug. The measurement was performed on the sample implanted at time
6 zero. The isotope images of ^{40}Ca , ^{44}Ca , and $^{44}\text{Ca}/^{40}\text{Ca}$ for the natural cancellous bone were taken
7 at a position far from the implanted position **(Supporting Figure S1a)**. The sponge morphology
8 can be observed from the individual isotope images. The $^{44}\text{Ca}/^{40}\text{Ca}$ of natural bone tissue
9 calculated from the data points in the black dotted square of the image was 0.020 ± 0.001 .
10 Moreover, $^{44}\text{Ca}/^{40}\text{Ca}$ of ^{44}HAp near the surface of the implanted DN gel was 0.132 ± 0.001 ,
11 measured by a spot analysis **(Supporting Figure S1b)**. The measured $^{44}\text{Ca}/^{40}\text{Ca}$ values of
12 natural bone (0.020 ± 0.001) and $^{44}\text{HAp}/\text{DN}$ gel (0.132 ± 0.001) were well-consistent with the
13 theoretical values.

14 Next, the isotope images around the boundary between the cancellous bone and the
15 $^{44}\text{HAp}/\text{DN}$ gel at 2, 4, 6 and 12 weeks after the implantation are shown in **Figure 2**. The images
16 of all periods were acquired at the connected regions between gel and bone observed in the
17 optical microscopic images, showing by the red squares. The $^{44}\text{Ca}/^{40}\text{Ca}$ image at 2 weeks
18 **(Figure 2a-iv)** showed a clear border between the gel and bone; Moreover, the $^{44}\text{HAp}/\text{DN}$ gel
19 (0.090) was lower than the pristine $^{44}\text{HAp}/\text{DN}$ gel (0.132), indicating the dissolution and
20 diffusion of ^{44}HAp . On the other hand, the $^{44}\text{Ca}/^{40}\text{Ca}$ in the bone region remained the same
21 value (0.020), meaning that the bone facing to the $^{44}\text{HAp}/\text{DN}$ gel was the pre-existed tissue, not
22 the regenerated one. Besides, it indicates that the dissolved Ca^{2+} ions from the mineralized HAp
23 did not diffuse into the bone tissue because of the bony solidity. From the relatively
24 homogeneous image, the decrement of $^{44}\text{Ca}/^{40}\text{Ca}$ due to dissolution and diffusion of Ca^{2+} ions
25 should uniformly occur in the whole gel region **(Figure 2a-iv)**. These results indicate that the
26 osteogenesis hardly penetrated the gel region in a short period of 2 weeks. This is consistent

1 with our previous observation that the adhesion of HAp-coated DN gel to bone at 2 weeks was
2 weak [11,12]. A clear $^{44}\text{Ca}/^{40}\text{Ca}$ border was still observed at 4 weeks (**Figure 2b-iv**). However,
3 the $^{44}\text{Ca}/^{40}\text{Ca}$ value of gel near the boundary with $\sim 30\ \mu\text{m}$ thick (white arrowed, 0.073) was
4 slightly lower than that of inner bulk (0.080) in the 4-week image. The slight decrement of gel
5 near the boundary could be associated to the ingrowth of immature bone, as shown by TEM
6 observation in our previous report [11,12]. At 6 weeks after the implantation, the boundary
7 became very diffusive, and the gradient distribution of $^{44}\text{Ca}/^{40}\text{Ca}$ was observed (**Figure 2c-iv**).
8 This continuous morphology from gel to bone implied that the regenerated bone, which was
9 more mature than that of 4 weeks, penetrated into the gel region. The gradient structure at the
10 boundary became wider in 12 weeks (**Figure 2d-iv**). These results clearly indicate that during
11 the implantation, the ^{44}HAp is gradually dissolved and diffused, and then the osteogenesis
12 propagates into the gel implant to form the robust bonding of gel to bone tissue.

13 To reveal the reutilization of implant HAp in the osteogenesis, the line profile of
14 $^{44}\text{Ca}/^{40}\text{Ca}$ around the boundary was further studied. Since the ^{44}HAp was gradually dissolved
15 and diffused, it is difficult to identify the initial boundary between gel and bone from the
16 $^{44}\text{Ca}/^{40}\text{Ca}$ line profile. Alternatively, we used the sulfur element contained in the PAMPS 1st
17 network of the DN gel to determine the initial boundary. It was confirmed that the DN gel does
18 not degrade in such a short implantation period [27], so the line profile of sulfur should not
19 change with time. **Figure 3a** shows the line profiles of $^{44}\text{Ca}/^{40}\text{Ca}$ ratio measured by isotope
20 microscopy, and the intensities of calcium and sulfur measured by energy dispersed X-ray
21 spectroscopy (EDS) at the boundary region. The measurement range of line analysis was wider
22 than the imaging analysis of **Figure 2**. Some $^{44}\text{Ca}/^{40}\text{Ca}$ data of the line profiles at the boundary
23 was failed to collect because of the poor ionization efficiency in the isotope microscopic
24 analysis. The dash lines in the $^{44}\text{Ca}/^{40}\text{Ca}$ curves were drawn based on the reference to the line
25 profiles generated from mapping image (**Supporting Figure S2**). An abrupt drop in the
26 normalized sulfur intensity (red curves) was observed for all the samples harvested at different

1 implanted times, which identifies the initial boundary position between the gel and the bone,
2 and was set as zero position in **Figure 3a**. The gel region and the bone region were highlighted
3 in blue and orange colors, respectively. Although the $^{44}\text{Ca}/^{40}\text{Ca}$ profile (black curve) around the
4 boundary at 2 weeks was not collected (**Figure 3a-i**), we speculate a relatively sharp line profile
5 from the corresponding imaging data (**Figure 2**). The high normalized Ca intensity (blue curve)
6 in the bone region (1.0 to 1.2) than that in the gel region (~ 0.2) far from the boundary is
7 consistent with the fact that the HAp density of bone was much higher than that of ^{44}HAp in
8 the gel. The $^{44}\text{Ca}/^{40}\text{Ca}$ ratio was decreased to 0.09 from the initial value (0.132) after 2 weeks.
9 With increasing in the implantation period, the $^{44}\text{Ca}/^{40}\text{Ca}$ ratio decreased from gel to bone
10 regions, and the gradient width was expanded (**Figure 3a-ii-iv**). Specifically, $^{44}\text{Ca}/^{40}\text{Ca}$ was
11 decreased inside the gel and saturated at lower values far from the boundary, while $^{44}\text{Ca}/^{40}\text{Ca}$
12 out of the gel was increased with the time. These results further confirm the dissolution of
13 ^{44}HAp in the gel and diffusion to outside of the gel. Furthermore, the gradient width of Ca
14 intensity also became expanded with time, but in the opposite direction: the Ca intensity in the
15 gel region increased while that in the bone region decreased slightly at the boundary. The region
16 that both Ca intensity and $^{44}\text{Ca}/^{40}\text{Ca}$ ratio changed indicates the newly regenerated bone by
17 osteogenesis, which is shown by hatched lines in **Figure 3a**. That is, the right region with 0.020
18 of the $^{44}\text{Ca}/^{40}\text{Ca}$ ratio and ~ 1.2 of Ca intensity was pre-existed bone. In contrast, the hatched
19 region, where both $^{44}\text{Ca}/^{40}\text{Ca}$ ratio and the Ca intensity were in gradient structure, is where the
20 bone was newly regenerated. After 12 weeks of implantation, the newly formed bone region
21 was about $\sim 450\ \mu\text{m}$ -thick ($\sim 200\ \mu\text{m}$ inside the gel and $\sim 250\ \mu\text{m}$ outside of the gel), consistent
22 with our previous observation by TEM [11]. From the fact that the $^{44}\text{Ca}/^{40}\text{Ca}$ ratio located
23 outside of gel matrix is obviously higher than the natural ratio 0.020, the regenerated bone
24 reutilized the HAp, because the isotope anomaly could not come from ingested feeds and
25 reutilization of pre-existed bone. Therefore, this work gives the first direct evidence that the
26 implanted HAp is reutilized to reconstruct bone tissues.

1 From the time evolution of $^{44}\text{Ca}/^{40}\text{Ca}$, we can estimate the characteristic time of
2 implanted HAp *in vivo* in the relatively early stage of osteogenesis. **Figure 3b** shows the
3 $^{44}\text{Ca}/^{40}\text{Ca}$ at a position = -60 μm of line profile as a function of the implantation period. The
4 plots were well-fitted by a simple attenuation function, and the characteristic time of ~ 16 days
5 was estimated. This characteristic time is well consistent with the period of ~ 14 days for the
6 formation of first immature (woven) bone.

7 Based on the above results, we present a schematic illustration of the primary bone
8 remodeling with $^{44}\text{HAp}/\text{DN}$ gel implant, as shown in **Figure 3c**. After the implantation of
9 HAp/DN gel, the inflammation occurs, which causes HAp dissolution because the
10 environmental pH becomes acidic by inflammation. So, the Ca ions diffuse to surrounding
11 regions at a relatively fast rate in this earliest stage. Over ~ 14 days, the inflammation subsides,
12 and the HAp dissolution is suppressed. Alternately, the immature bone is formed, which also
13 acts as a physical barrier to prevent further Ca diffusion. Therefore, the decreasing rate of Ca
14 becomes slower after 14 days and then reaches equilibrium over 40 days. After this early period,
15 the reconstruction to the mature bone is dominantly controlled by cellular activities of osteoclast
16 and osteoblast. As a result, newly formed bones penetrated in the gel form a strong bonding
17 between the gel and the bone.

18 The unique isotope microscopy firstly revealed the reutilization of synthetic HAp
19 implant to osteogenesis *in vivo*. When we implant HAp-coated tough gel in the defect created
20 at the femoral groove, the HAp is chemically dissolved and diffused due to acidic conditions
21 by inflammation at the earliest stage of surgery. After the subside of the inflammatory, the
22 immature woven bone is formed by using ions dissolved from the implant. This woven bone
23 formation penetrates into gel matrix to achieve robust fixation. The dissolution and diffusion of
24 HAp get slow down over 2 weeks because of the physical barrier of woven bone and less
25 inflammation. After this early stage, the reconstruction from woven bone to functional mature
26 bone is performed by the biological reaction of osteoblast and osteoclast. The isotopic analysis

1 can trace the lifetime and reutilization of HAp implants *in vivo*, so it is a strong tool to evaluate
2 the molecular process of bone-related materials.

3
4 **Supporting Information**

5 Supporting Information is available from the Wiley Online Library or from the author.
6

7 **Acknowledgements**

8 This research was financially supported by JSPS KAKENHI (Grant Number 17H06144,
9 JP17H06376). We thank the Open Facility, Global Facility Center of Hokkaido University, for
10 isotope microscopic measurements. The authors thank Prof. Kuniharu Ijro (Hokkaido
11 University) for precious suggestion on the work. The authors also thank Dr. Masanori Kikuchi
12 (National Institute for Materials Science), Prof. Dr. Chikara Ohtsuki (Nagoya University) and
13 Prof. Dr. Kunio Ishikawa (Kyusyu University) for discussions on the HAp re-utilization.

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15 Received: ((will be filled in by the editorial staff))

16 Revised: ((will be filled in by the editorial staff))

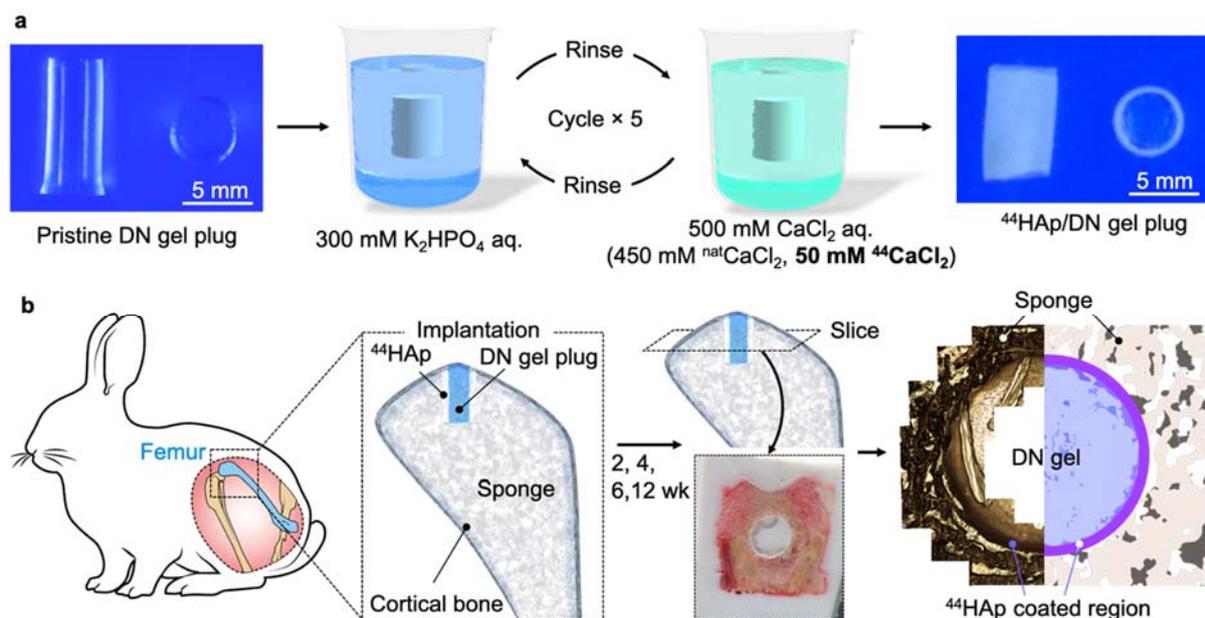
17 Published online: ((will be filled in by the editorial staff))
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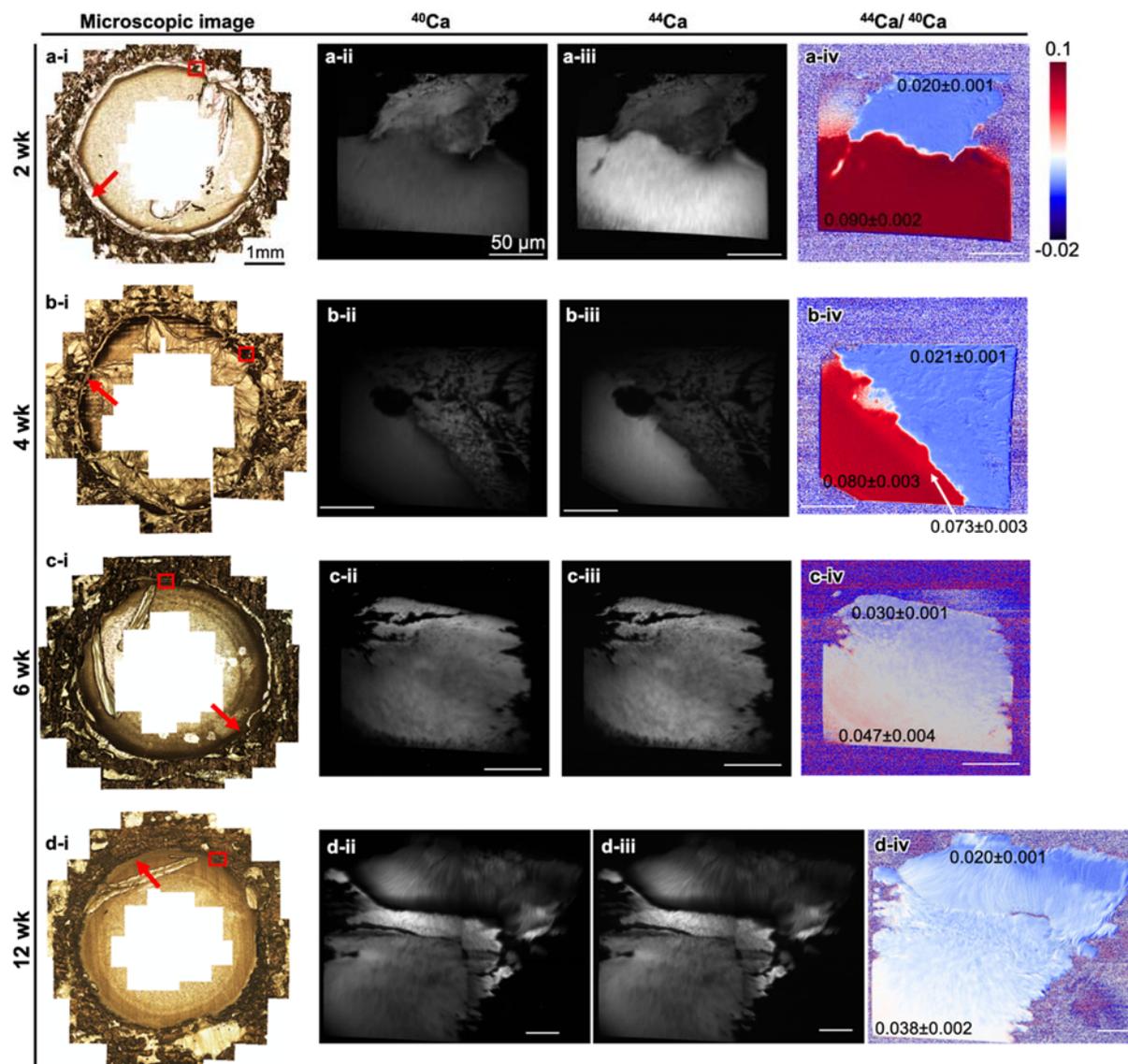
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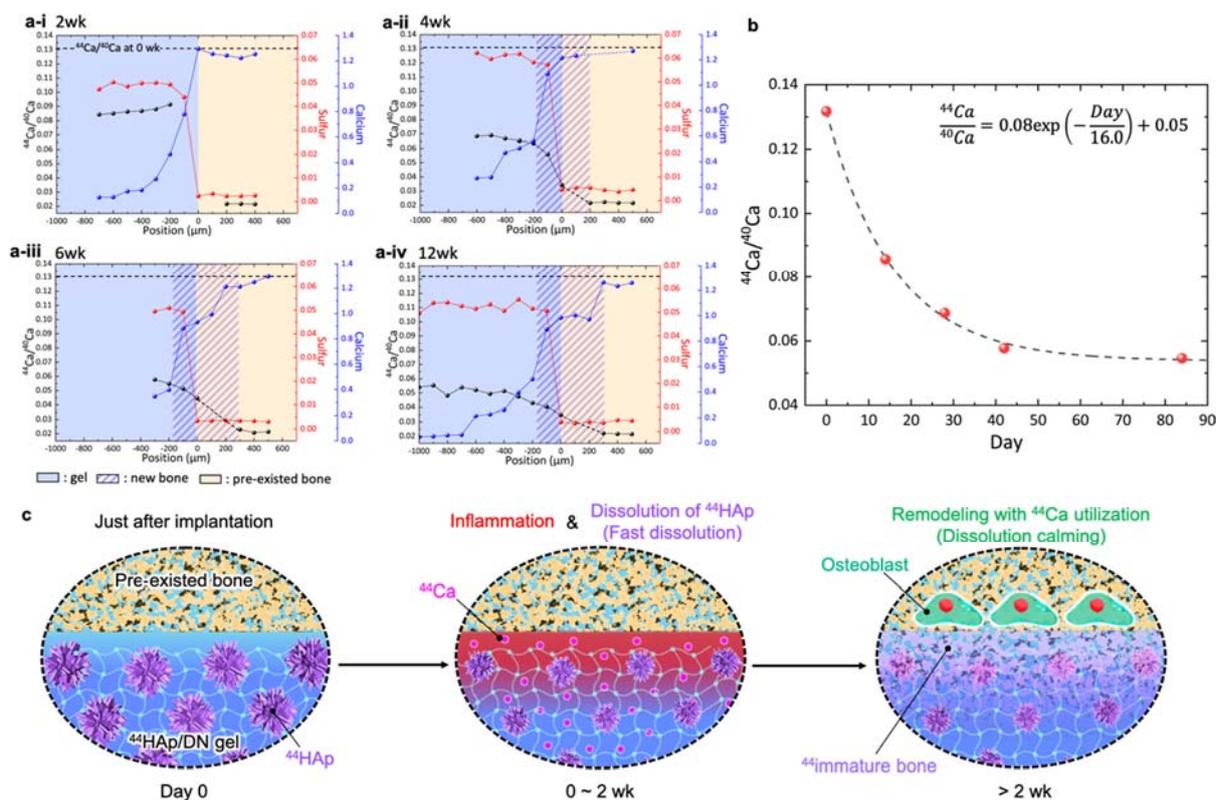
11 **Figure 1.** Experimental procedures to prepare the implanted specimens for isotope microscopic
 12 observation. (a) The ^{44}HAp coating to the subsurface of DN gel plug by the alternative soaking
 13 of K_2HPO_4 aq. and CaCl_2 aq. containing 10% $^{44}\text{CaCl}_2$. (b) The implantation of ^{44}HAp -coated
 14 DN gel into the defected femoral trochlea of the patellofemoral joint of rabbits and the
 15 schematic image and optical microscopic image of the sliced specimen in the transverse plane.

16



1
 2 **Figure 2.** Optical microscopic images and isotope microscopic images. Optical microscopic
 3 images (i), raw images of ^{40}Ca (ii) and ^{44}Ca (iii) mappings, and color mapping of $^{44}\text{Ca}/^{40}\text{Ca}$
 4 ratio (iv) at 2 (a), 4 (b), 6 (c) and 12 (d) weeks. Red arrows indicate the boundary between bone
 5 and gel. Isotope images (ii to iv) were acquired at the red squares. The global optical
 6 microscopic images of the implanted $^{44}\text{HAp/DN}$ gel with sponge were obtained by the
 7 reconstruction of many micro-images.

8



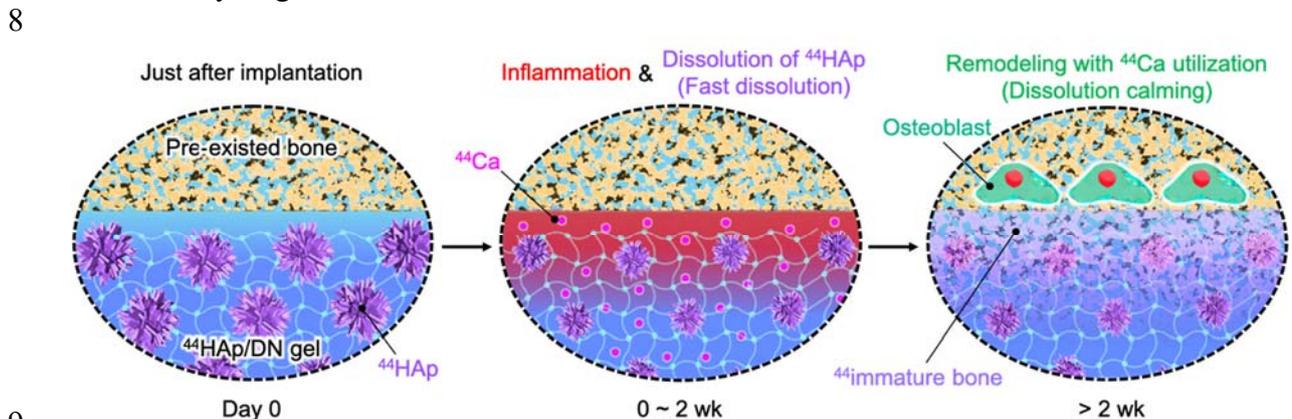
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 2 **Figure 3.** (a) Line profiles of $^{44}\text{Ca}/^{40}\text{Ca}$ ratio measured by isotope microscopy, normalized
 3 sulfur and calcium intensities measured by EDS. The origin of position is defined as the position
 4 that sulfur intensity is dramatically dropped to zero. Blue, orange and hatched purple are gel,
 5 pre-existing bone and new bone regions, respectively. The dash line parts in the curves of
 6 $^{44}\text{Ca}/^{40}\text{Ca}$ are drawn based on the reference to the line profiles generated from mapping image
 7 (Fig. S2). (b) Chronological change of $^{44}\text{Ca}/^{40}\text{Ca}$ ratio at the subsurface of ^{44}HAp -coated DN
 8 gel. The characteristic decay time of $^{44}\text{Ca}/^{40}\text{Ca}$ is 16.0 days that are close to the period of woven
 9 bone formation ($\sim 2\text{wk}$). R^2 of the fitting curve was 0.999. (c) Schematic illustrations of the
 10 primary bone remodeling with $^{44}\text{HAp/DN}$ gel implant according to the results of Figure 3(a)
 11 and 3(b). Because the implanted environment just after surgery is acidic due to inflammation,
 12 the ^{44}HAp is rapidly dissolved to calcium ions and phosphate ions, and these ions diffuse,
 13 partially into bone regions, as revealed in Figure 3(b). After two weeks, the inflammation calms
 14 and the immature bone is generated by osteoblasts, utilizing the dissolved ^{44}Ca ions, as revealed

- 1 by Figure 3(a). The formation of ^{44}Ca immature bone act as a physical barrier to ion diffusion,
- 2 which prevents decrement of $^{44}\text{Ca}/^{40}\text{Ca}$ at later time as revealed in Figure 3(b).

1 The table of contents

2
3 T. Nonoyama*, L. Wang, M. Tsuda, Y. Suzuki, R. Kiyama, K. Yasuda, S. Tanaka, K. Nagata,
4 R. Fujita, N. Sakamoto, N. Kawasaki, H. Yurimoto, J. P. Gong*

5
6 Isotope Microscopic Observation of Osteogenesis Process Forming Robust Bonding of Double
7 Network Hydrogel to Bone



10 The ^{44}Ca calcium isotope-doped HAp/DN hydrogel was implanted in a defect of rabbit femoral
11 bone, and the dynamic osteogenesis process at the gel/bone interface was traced by the calcium
12 isotope ratio $^{44}\text{Ca}/^{40}\text{Ca}$ using an isotope microscopy. It is revealed that the osteogenesis
13 penetrates into the DN hydrogel re-utilizing dissolved HAp to form robust bonding.

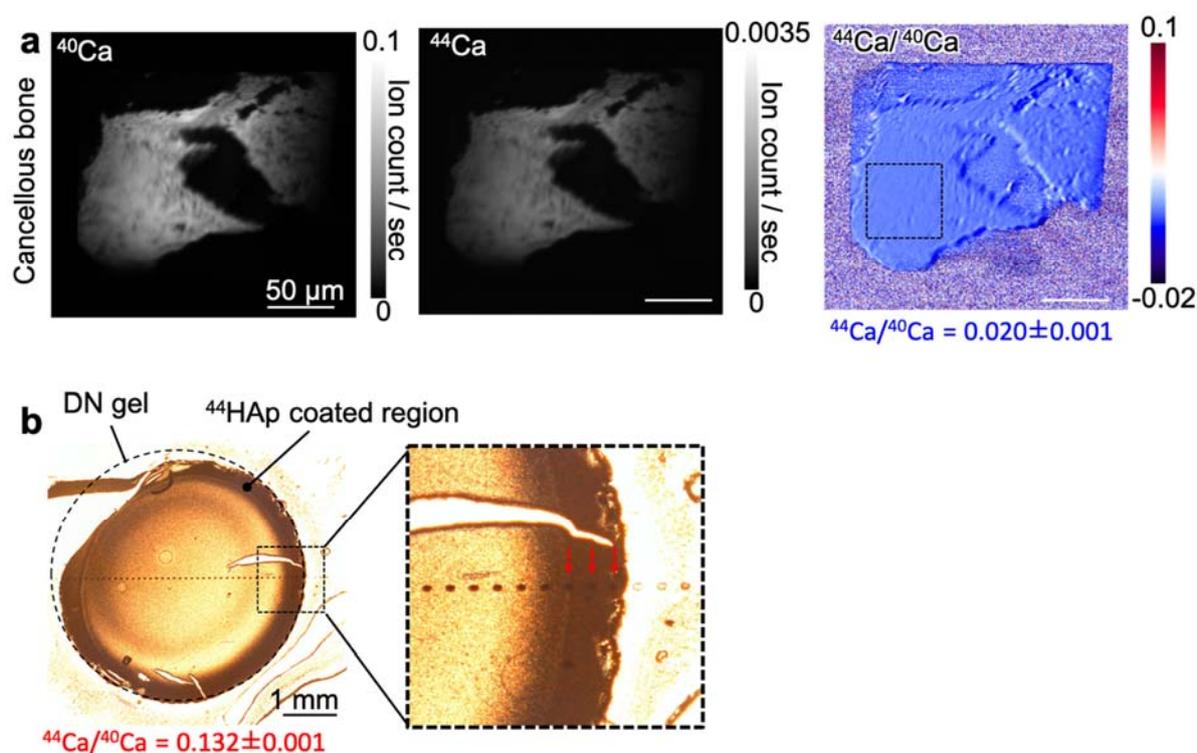
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3 Copyright WILEY-VCH Verlag GmbH & Co. KGaA, 69469 Weinheim, Germany, 2018.

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5 Supporting Information

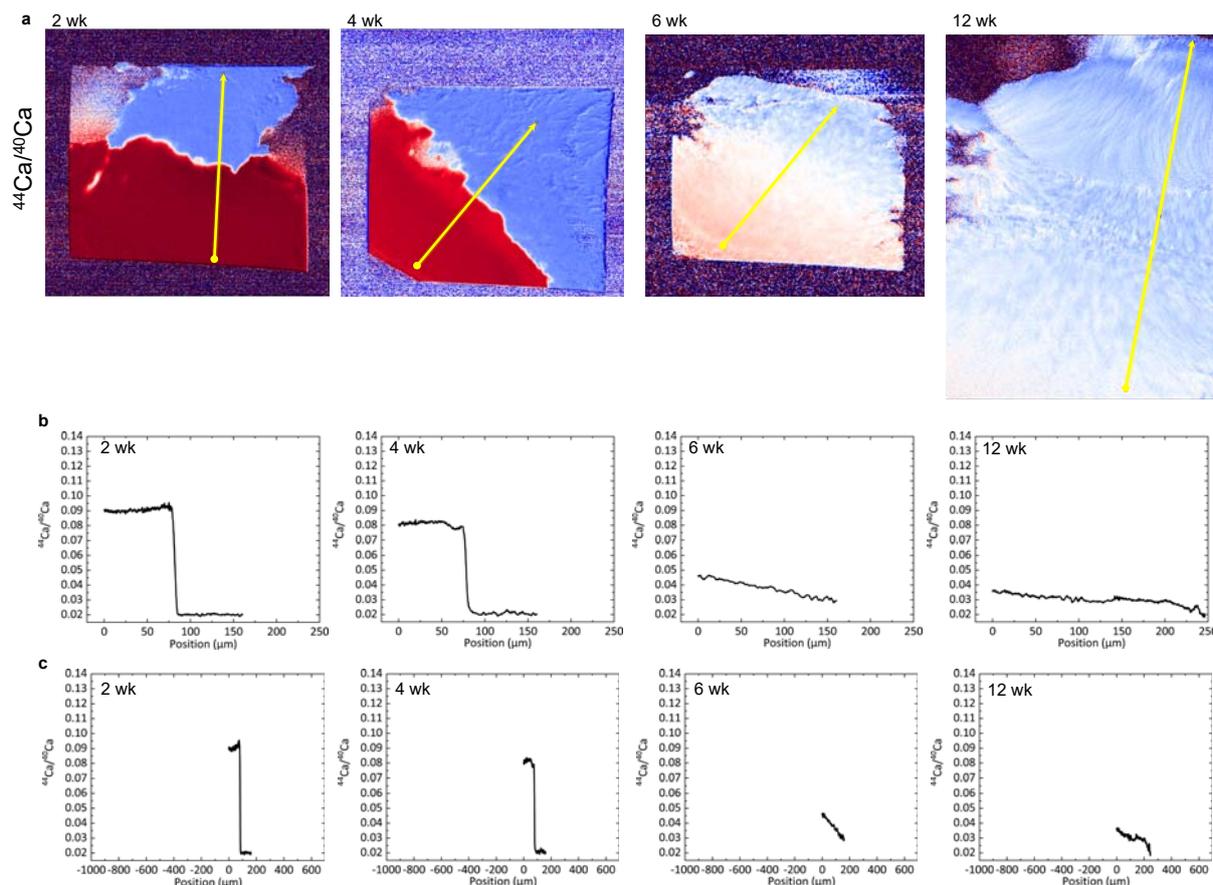
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8 **Isotope Microscopic Observation of Osteogenesis Process Forming Robust Bonding of**
9 **Double Network Hydrogel to Bone**

10
11 *Takayuki Nonoyama*^{1,2*}, *Lei Wang*^{2,3}, *Masumi Tsuda*^{2,3,5}, *Yuki Suzuki*⁴, *Ryuji Kiyama*⁴,
12 *Kazunori Yasuda*^{1,2}, *Shinya Tanaka*^{2,3,5}, *Kousuke Nagata*⁶, *Ryosuke Fujita*^{7,8}, *Naoya Sakamoto*⁷,
13 *Noriyuki Kawasaki*⁶, *Hisayoshi Yurimoto*^{6,7}, *Jian Ping Gong*^{1,2,5*}



15
16 **Figure S1. (a)** Isotope microscopic images of ^{40}Ca , ^{44}Ca and $^{44}\text{Ca}/^{40}\text{Ca}$ of the cancellous bone
17 taken at a position far from the implanted HAp/DN gel. The black dotted area in the $^{44}\text{Ca}/^{40}\text{Ca}$
18 image was used to estimate the $^{44}\text{Ca}/^{40}\text{Ca}$ ratio. **(b)** Optical microscopic image of spot
19 measurement of ^{44}HAp /DN gel slice. The horizontal dots were the history of spot measurements.
20 The three dots at the HAp/DN gel hybridized layer shown by red arrows were used to calculate
21 the isotope ratio.

22



1
2 **Figure S2.** Color mapping of $^{44}\text{Ca}/^{40}\text{Ca}$ ratio (a) and the corresponding line profiles (b). (c) The
3 line profiles of (b) plotted in the same scale with that in **Figure 3a** for comparison, and the
4 position in the horizontal axis is arbitrary. The tendencies of $^{44}\text{Ca}/^{40}\text{Ca}$ line profiles generated
5 from imaging data (**Figure S2c**) were almost consistent with those of spot analysis (**Figure 3a**).
6 The 4wk profile was slightly different from those of **Figure 3a** due to different measurement
7 positions.

8 **Materials & Methods**

9 **1. Materials**

10 2-Acrylamido-2-methyl propanesulphonic acid (AMPS) was provided by Toagosei Co. Ltd.
11 (Tokyo, Japan). Dimethylacrylamide (DMAAm), *N,N'*-methylenebisacrylamide (MBAA), 2-
12 oxoglutaric acid (α -keto), regular calcium chloride ($^{\text{nat}}\text{CaCl}_2$), dipotassium phosphate (K_2HPO_4),
13 and tris(hydroxymethyl)aminomethane (Tris) were purchased from Fujifilm Wako Pure
14 Chemical Industries, Ltd. (Osaka, Japan). ^{44}Ca calcium chloride (isotopic composition: ^{40}Ca
15 **2.89%**, ^{42}Ca 0.06%, ^{43}Ca 0.03%, ^{44}Ca **97.00 \pm 0.2%**, ^{46}Ca <0.002%, ^{48}Ca 0.02%) was purchased
16 from Trace Sciences International Corp. (Ontario, Canada). Optimal cutting temperature
17 (O.C.T.) compound and plastic mold for freezing specimen was purchased from Sakura Finetek
18 Japan Co., Ltd. (Tokyo, Japan). Cryo-support film was purchased from Leica Biosystems
19 GmbH (Wetzlar, Germany). A silicon wafer for a substrate of the specimen was purchased from
20 The Nilaco Corporation (Tokyo, Japan).

21 **2. Synthesis of DN hydrogel**

22 The PAMPS/PDMAAm DN gel was synthesized using previously reported methods [1,11].
23 PAMPS gel was obtained by radical polymerization from an aqueous solution containing 1 M
24 AMPS as a monomer, 4 mol% (to monomer) MBAA as a cross-linker, and 0.1 mol% (to
25 monomer) α -keto as an initiator. The solution was injected into a cell consisting of a pair of
26 glass plates separated by a silicone rubber. The PAMPS gel was synthesized by irradiating the
27
28

1 cell with a UV lamp (wavelength 365 nm) for about 8 h. The obtained PAMPS hydrogel (first
 2 network) was then immersed in an aqueous solution containing 2 M DMAAm, 0.1 mol% (to
 3 monomer) MBAA, and 0.1 mol% (to monomer) 2-oxoglutaric acid for 1 day until reaching
 4 equilibrium. The sample was sandwiched between two glass plates, and the second network
 5 (PDMAAm) was subsequently polymerized in the presence of the PAMPS hydrogel by
 6 irradiating with UV light for 6 h. After polymerization, the PAMPS/PDMAAm DN gel was
 7 immersed in pure water for 1 week, and the water was changed twice daily to remove any
 8 unreacted chemicals.

10 3. ⁴⁴Ca doped HAp (⁴⁴HAp) nano-particles mineralization in DN hydrogel

11 To form the ⁴⁴Ca doped HAp (⁴⁴HAp) /DN layer, the DN gel was first immersed in 1 M Tris
 12 buffer (pH 9.0) to adjust the pH within the hydrogel because HAp is the most stable in an
 13 aqueous solution at pH 9. The gel was then dipped in 300 mM K₂HPO₄ aq for 110 sec, distilled
 14 water for 10 sec, 500 mM CaCl₂ aq consisting of 450 mM ^{nat}CaCl₂ and 50 mM ⁴⁴CaCl₂ for 110
 15 sec, and distilled water for 10 sec. This alternative dipping process was repeated for 5 cycles to
 16 mineralize ⁴⁴HAp. After the alternative dipping, the hydrogels were dipped in 500 mM CaCl₂
 17 aqueous solution containing 10 mol% ⁴⁴CaCl₂ (pH 11.0) at body temperature for 24 h to ripen
 18 the mineralized amorphous calcium phosphate (ACP) and metastable calcium phosphate
 19 crystals to low crystalline HAp. Subsequently, a hybridized HAp/DN layer was formed on the
 20 surface of the DN gel. The mineralized layer was approximately 500-μm thick. Considering the
 21 natural isotope ratio of calcium (⁴⁰Ca: 96.94%, ⁴⁴Ca: 2.086%) and purity of ⁴⁴Ca isotope reagent,
 22 the ⁴⁴Ca/⁴⁰Ca of the ⁴⁴HAp on the surface of DN gel is simply estimated as follows,

$$23 \quad {}^{44}\text{Ca}/{}^{40}\text{Ca} = \frac{2.086\% \times \frac{9}{10} + 97.00\% \times \frac{1}{10}}{96.94\% \times \frac{9}{10} + 2.89\% \times \frac{1}{10}} = 0.132$$

24 4. Surgery

25 4.1. Animals

26 Animal experiments were carried out at the Institute of Animal Experimentation at Hokkaido
 27 University according to the Rules and Regulations of the University's Animal Care and Use
 28 Committee. Total of 4 mature female Japanese White rabbits, weighing 4.0 ± 0.5 kg, were used
 29 in this study. The operation for each animal was performed under intravenous anesthesia
 30 (pentobarbital, 25 mg/kg) and sterile conditions. Due to the high costs of ⁴⁴CaCl₂ isotope
 31 chemical and isotope microscopy, one rabbit per each implantation period was used. The
 32 evidence of robust bonding of HAp/DN gels to rabbit bone tissues was well-confirmed in our
 33 previous reports [11-13].

34 4.2. Implantation of ⁴⁴HAp/DN in osteochondral defect

35 To prepare a specimen of isotope microscopy, we prepared cylindrical DN gel rods (4.5 mm in
 36 diameter and 10 mm in height). The surface layer of the rods was mineralized with ⁴⁴HAp. Then,
 37 a 4-mm-high cylindrical plug without HAp coating at its top and bottom was created by cutting
 38 a section from the cylindrical rod.

39 An osteochondral defect (4.3 mm in diameter and 4 mm depth) was created in the femoral
 40 groove of the bilateral patellofemoral joints. A 4-mm-high cylindrical plug having plain HAp
 41 coating was implanted in the opposite knee as a control. All the rabbits were sacrificed by
 42 pentobarbital injection at 2, 4, 6 and 12 weeks after the surgery.

43 5. Isotope imaging

44 5.1. Specimen preparation

1 The harvested rabbit femur with $^{44}\text{HAp/DN}$ implant was cut into 1.5-cm cube. The
2 sample was embedded in the O.C.T. cryo-compound in plastic mold, and then gently frozen in
3 liquid nitrogen pool. The slices (14 μm -thick) of middle part of $^{44}\text{HAp/DN}$ plug with rabbit
4 bone were prepared using the cryostat (Leica CM 3050S, Leica Biosystems GmbH, Wetzlar,
5 Germany), and transferred on cryo-support film. After the slices were completely dried, the
6 films were fixed on a flat silicon wafer by the double-sided carbon tape. Finally, the gold
7 coating treatment on the surface of specimen was applied by physical vapor deposition (Quick
8 Cool Coater SC-701MC, Sanyu electron Co., Ltd., Tokyo, Japan). The low magnification
9 optical images of specimens were acquired by optical microscopy (Nikon Eclipse E600 POL,
10 Nikon Instech Co., Ltd., Tokyo, Japan), and the acquired raw images were merged to make a
11 panoramic image by the photo-merge processing of Photoshop CC 2014 (Adobe, Systems
12 Incorporated, California, US). The slices from the middle position of the cylindrical gel plug
13 were used for analysis.

14 15 **5.2. Isotope microscopy**

16 The Hokkaido University isotope microscope system, a Cameca ims-1270 secondary
17 ion mass spectrometer (SIMS) equipped with a stacked CMOS-type active pixel sensor
18 (SCAPS), was applied to visualize $^{44}\text{Ca}/^{40}\text{Ca}$ distributions on the sample surface [15]. For two-
19 dimensional isotope imaging, an $^{16}\text{O}^-$ primary beam of 23 keV was irradiated on the sample
20 surface of approximately $200 \times 200 \mu\text{m}^2$ with beam currents ranging from c. 20 to 40 nA. The
21 positive secondary ion images of $^{40}\text{Ca}^+$ and $^{44}\text{Ca}^+$ on the sample surface were collected by the
22 SCAPS detector, with exposure times ranging between 15 and 100 seconds after pre-sputtering
23 for cleaning of the sample surface. The spatial resolution of the Ca isotope images was about 1
24 μm . For line and spot analyses, an $^{16}\text{O}^-$ primary beam of 23 keV and 50 nA was focused on the
25 sample surface of $\sim 30 \mu\text{m}$ diameter. The positive secondary ions of $^{40}\text{Ca}^+$ and $^{44}\text{Ca}^+$ were
26 simultaneously collected for 30 s in each spot by two Faraday cups in the multi-collection mode.

27 28 **6. Energy dispersed X-ray spectroscopy**

29 The quantitative analysis of sulfur and calcium elements of the same samples was
30 performed by spot scanning of energy dispersed X-ray spectroscopy on a scanning electron
31 microscope (JEOL JSM-6010LA, Japan) with energy dispersed X-ray spectrometer (EDS, EX-
32 94400T4L11). The condition of imaging was 20 kV of acceleration voltage, 60 nm of spot size,
33 10 mm of working distance, 100 sec of duration, and 70 of magnification. The intensities of
34 sulfur and calcium were normalized by that of zinc oxide as a standard sample. ZAF correction
35 (Z: atomic number correction, A: absorbance correction, F: fluorescence correction) was
36 applied to quantitate the obtained data on the software.

37 38 **7. Statistical analysis**

39 The mean and standard deviation of isotope ratio $^{44}\text{Ca}/^{40}\text{Ca}$ in selected area were
40 calculated by using imageJ ver. 1.52q. For Figure 2, the area and number of data point were
41 10×10 pixels and 100, respectively. In Figure S1a, area and number of data point were 140×140
42 pixels and 19,600, respectively. Due to the spot analysis, in the calculation of the isotope ratio
43 in Figure S1b, three data points in the right edge of HAp coated region were used.