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1 **Proposing a three-phase model for predicting the mechanical properties of**  
2 **mortar and concrete**

3  
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27 **Abstract**

28 The interfacial transition zone (ITZ) around aggregates is known as the weakest zone in mortar and  
29 concrete, which highly influences the mechanical properties. In this study, the influence of ITZ on  
30 mechanical properties such as Young's modulus and Poisson's ratio of cementitious materials is  
31 evaluated using a multi-scale model (developed in five hierarchical levels: nano-scale cement  
32 hydrates~ scale of concrete). In the proposed model, the microstructure of mortar/ concrete is  
33 considered as a three-phase material: fine/ coarse aggregates, ITZ of aggregates and bulk paste/  
34 mortar. The primary input is the microstructure of cement paste and ITZ, which are predicted as the  
35 function of curing time using coupled cement hydration-thermodynamic model. The ITZ volume  
36 fraction is analytically computed based on aggregate particle size distribution. Multi-level  
37 homogenization methods (based on three-phase sphere model for two-phase composite material) are  
38 finally implemented to predict the effective properties of mortar and concrete. Here, the equivalent  
39 matrix consisted of fine/ coarse aggregate, bulk paste/ mortar and ITZ are obtained with the first and  
40 second levels homogenization procedures for mortar/ concrete. To validate the predictability of the  
41 models for mortar and concrete, predicted values are compared with independent experimental data  
42 sets. The results show a reasonable agreement with experimental results of Poisson's ratio and  
43 Youngs modulus (in the error range of 5 GPa). The influences of ITZ on properties of mortar and  
44 concrete are also discussed based on the outcomes.

45

46 **Keywords:** Mechanical properties; Mortar; Concrete; Multi-scale model; interfacial transition zone  
47 (ITZ); Homogenization approach

48

49 **1. Introduction**

50 Mortar and concrete are the most widely used building materials in the global construction industry.  
51 Owing to the diverse components such as cement hydrates, aggregates, and cement-aggregate  
52 interfacial zones, they highly reveal a complicated microstructure. Number of previous studies

53 disclosed that their behaviours on the application of loads are determined by intrinsic properties,  
54 which, at the micro-scale, are influenced by the water/cement ratio (w/c), clinker compositions and  
55 degree of hydration of cement, while aggregate properties play a significant role at the macroscale  
56 [1,2]. The properties of bulk paste existing in the matrix of mortar and concrete are time dependent.  
57 As the crystalline and semi-crystalline hydrates continue to evolve with hydration period, the  
58 materials become more complicated to understand [3,4]. Therefore, optical determination of the  
59 mechanical properties of mortar and concrete from their microstructure is of a great interest in  
60 industrial applications these days.

61

62 The occurrence of cement-aggregate interface (also referred to as interfacial transition zone, ITZ) is  
63 related to the wall-effect originated by the aggregate surface that disturbs the normal packing of  
64 cement particles [5–7]. During casting, the spatial arrangement of anhydrous grains becomes looser  
65 in the vicinity of aggregate particles. In addition, micro-bleeding of fresh mix leads to the  
66 accumulation of water near the surface of aggregate particles during the vibration of mixture and  
67 before setting. As the result, clinker content exhibits an increasing distribution with the distance from  
68 the aggregate surface, and a higher local w/c is attained in the ITZ of fresh mortar and concrete [5].  
69 In fact, the ITZ is not a definite zone, but a region of transition, and a thickness ranging between 10–  
70 40  $\mu\text{m}$  is documented for typical mortars and concrete [5,6,8,9]. Because of its high porosity  
71 compared to bulk paste matrix, ITZ is typically considered to reveal adverse effects on the  
72 performance of mortars and concretes, particularly on transport and mechanical properties to a great  
73 extent [5,6,8,9].

74

75 The thickness of the ITZ is found to be influenced by various casting factors such as curing age, w/c  
76 ratio, aggregates properties (type, roughness, shape and size), aggregate content and supplementary  
77 cementitious materials [6,10–13]. Recently, Huang et al. [13] investigated the ITZ properties of both  
78 normal aggregates and light weight aggregates. Their outcomes indicated that the ITZ between light

79 weight aggregate and cement paste is stronger and bonded without visible cracks compared to normal  
80 weight aggregates. High surface roughness of the lightweight aggregate can be the reason, which  
81 allows to have an effective mechanical interlocking, bringing advantages related to the strength of  
82 the aggregate-cement system. It should also be noted that w/c ratio at the interface would be low due  
83 to high water absorption of the lightweight aggregates, which leads to increased interfacial bonding  
84 strength in ITZ. Due to the tightened environmental policy on limiting aggregate exploitation and  
85 ever-increasing price, finding substitutes for natural aggregate has drawn a great deal of attention  
86 [14–16]. Involving artificial aggregates can be one potential alternative; following are some of such  
87 widely used materials: coal fly ash, paper ash, cement kiln dust, municipal solid waste incineration  
88 fly ash and bottom ash [17–19]. Jiang et al. [15] studied the performance of artificial aggregates on  
89 strength, ITZ and drying shrinkage, concluded that the artificial aggregates using waste concrete  
90 powder exhibits an enhancement on ITZ thickness, i.e., lower ITZ thickness (30  $\mu\text{m}$ ) was obtained  
91 for artificial aggregates compared to that of natural aggregate (60  $\mu\text{m}$ ) while improving mechanical  
92 properties of ITZ. This was attributed to the further reaction in aggregates located at the outer layer  
93 made by waste concrete powder with cement matrix, thus resulting in low porosity.

94  
95 The concrete is characterized as a multi-phase material with several different representative scales.  
96 At macroscopic scale, concrete is regarded as a homogeneous material, while at mesoscopic scale, it  
97 is pondered to be consisting of coarse aggregates and mortar matrix [8]. Further subdivisions of the  
98 mortar matrix produce fine aggregates and hardened cement paste with pores embedded inside. It is  
99 well known that it is unsuitable to model the concrete at a single scale, as its intrinsic features range  
100 from nanometer-sized pores to millimeter-sized aggregates. Therefore, a multi-scale model can be  
101 more appropriate to link different length features of concrete, hence to evaluate macro-mechanical  
102 properties from micro-composition and micro-mechanical properties [20,21]. During the analysis, the  
103 properties computed at one scale, micrometers for instance, are inputted to the subsequent scale of  
104 the model, such as millimeters. For evaluating the effective properties of multi-inclusion composites

105 (such as mortar and concrete), homogenization stepping scheme was repeatedly shown to be an  
106 effective pathway [20–24]. In this homogenization method, the properties are assessed by averaging  
107 the stress and strain fields of material representative volume element (RVE) featuring a “mesoscopic”  
108 length scale which is much larger than the characteristic length scale of particles (inhomogeneities),  
109 but smaller than the characteristic length scale of a macroscopic specimen [20].

110  
111 As yet, many research were undertaken to determine the Young’s modulus and Poisson’s ratio of  
112 concrete by both analytical and numerical methods. Earlier days, researchers attempted to model the  
113 concrete simply as a two-phase material consisting coarse aggregates and mortar matrix [25–27].  
114 Although the two-phase model could provide approximate estimations, insightful results were not  
115 achieved. The variation between real values and estimated values (in two-phase models) was later  
116 found to be due to the ignorance of ITZ in concrete [25]. Thenceforth, the three-phase concept was  
117 incorporated by researchers in their models proposed to concrete by considering the ITZ as a single  
118 layer around aggregates [25,26]. It should be noted, however, modelling the ITZ realistically into the  
119 concrete composite was continuously a challenging process. For instance, during the prediction of  
120 Young’s modulus of concrete, Li et al. [28] failed to capture the overlapping between ITZs, which  
121 inevitably affected the accuracy of the prediction. In fact, the unavailability of experimental data on  
122 ITZ is the major hurdle that limits the developments of reliable and insightful simulations of concrete,  
123 which consequently leads the researchers to deal with lot of assumptions on properties of ITZ [20,29–  
124 31]. Particularly, in most of the existing models, the ITZ layer was assumed to be uniform with  
125 constant mechanical properties [5,30,32]. But, in real case, the clinker content in ITZ reveals a  
126 gradient increase with the distance from aggregate surface [6]. Moreover, by ignoring time-dependent  
127 effects, the mechanical properties of ITZ were assumed to be a specific ratio/ function of cement paste  
128 properties [29,31]. Therefore, it is still needed to develop reliable models by considering both gradient  
129 and time-dependent effects in ITZ for precise computations of Young’s modulus and Poisson’s ratio  
130 of concrete.

131

132 The model proposed in this research work systematically integrates (i) cement hydration model, (ii)  
133 thermodynamic model, (iii) multi-scale model and (iv) homogenization scheme to predict the  
134 mechanical properties of mortar and concrete in a realistic manner. By representing mortar/ concrete  
135 as a three-phase composite material and based on previously proposed model for w/c distribution in  
136 ITZ by Nedeau [6], the mechanical properties of ITZ and bulk paste/ mortar is computed with  
137 hydration period separately. Unlike most of the previously proposed models (wherein constant  
138 properties of ITZ and bulk paste were used), the computation of mechanical properties of ITZ and  
139 bulk paste is based on the detailed microstructure of hydrated cement paste including two types of C-  
140 S-H, capillary porosity, chemical shrinkage and other hydration products. Since the microstructure of  
141 cement matrix varies depending on curing time, chemical composition of clinker, w/c and curing  
142 temperature, the realistic computations of ITZ and bulk paste are emphasized in this proposed  
143 framework. Finally, the homogenization method is applied to predict the mechanical properties of the  
144 materials. The validity of the proposed model is verified with independent sets of experimental data,  
145 and the effect of ITZ on the Young's modulus and Poisson's ratio is investigated through sensitivity  
146 analysis for mortar and concrete.

147

## 148 **2. Modelling strategy**

149 The steps followed in the proposed model for mortar and concrete are clearly depicted in **Figure. 1**.  
150 The fundamental parameters such as clinker compositions, properties of clinker and aggregates,  
151 mixture recipe and boundary conditions are the necessary input parameters for computations. As  
152 conceptually illustrated in **Figure. 2**, both the mortar and concrete are considered as three-phase  
153 materials: (i) mortar: fine aggregates, ITZ and bulk paste, and (ii) concrete: coarse aggregates, ITZ  
154 and mortar. It should be noted, as the ITZ consists of low amount of clinker and high amount of water,  
155 it can be reasonably considered as a cement paste with higher w/c (compared to the initial w/c of fresh  
156 concrete mix/ mortar mix). The required mechanical properties of ITZ and bulk paste are separately

157 computed from the cement paste model developed in our previous work [33]. Finally, homogenization  
158 method was used to predict the mechanical properties of mortar and concrete. The elaboration of  
159 modelling approach can be found in subsequent sections.

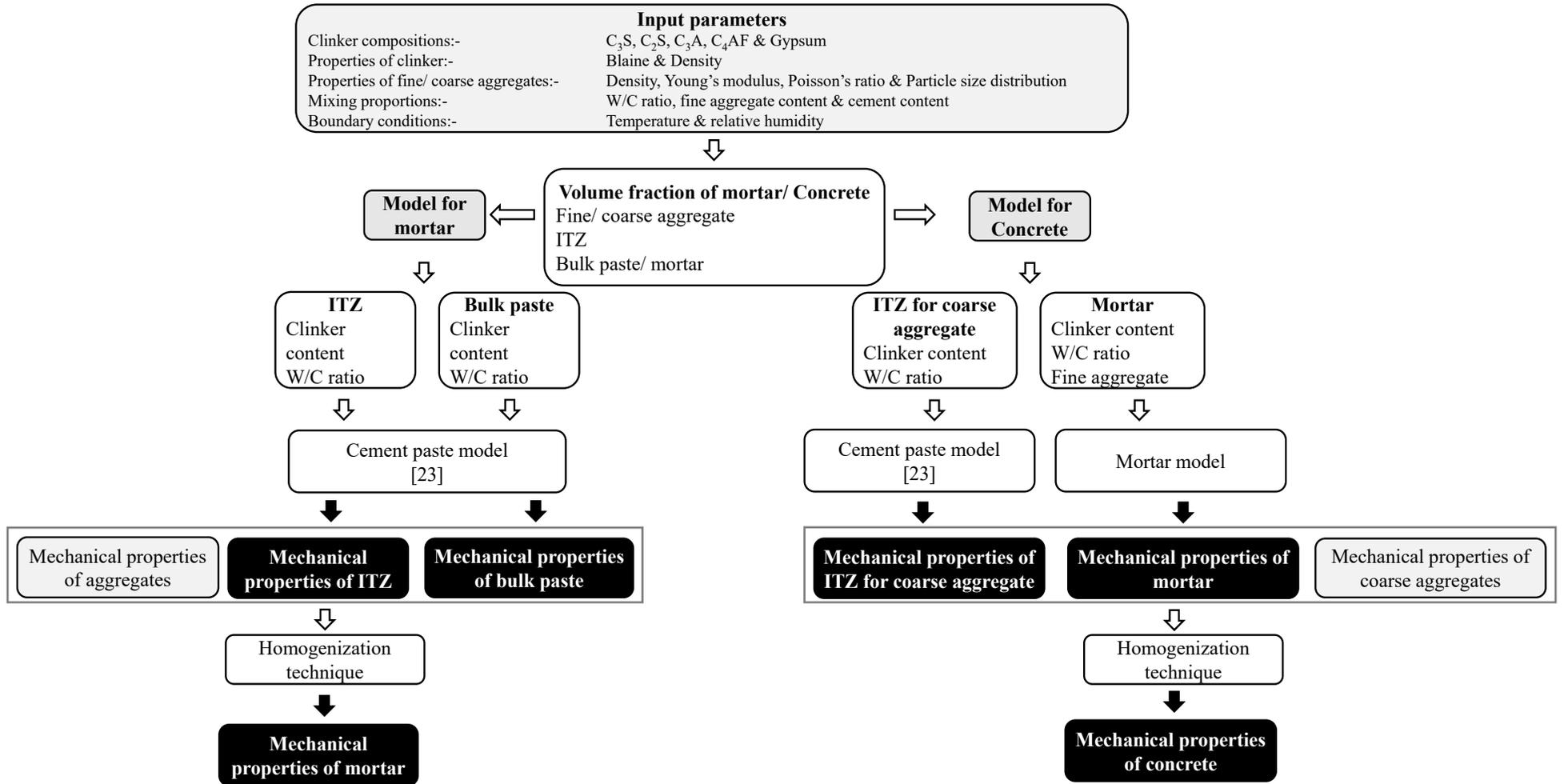
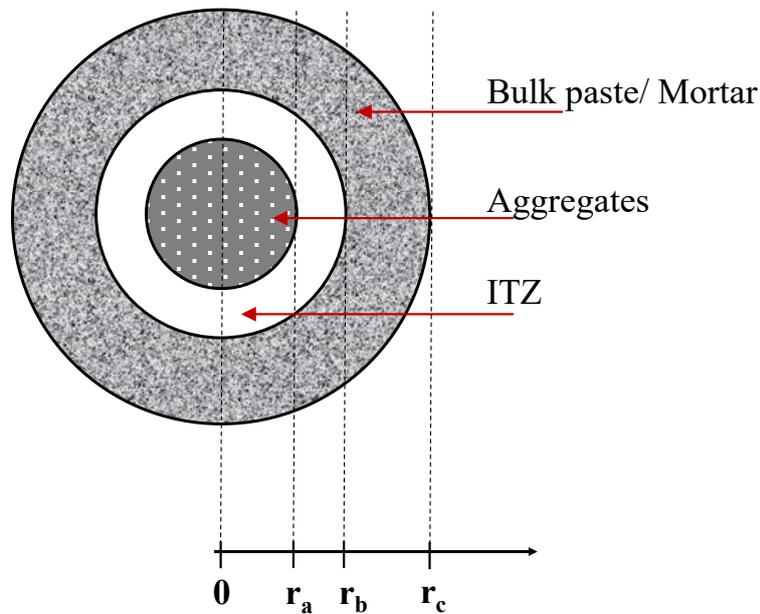


Figure. 1. Flow diagram of the model proposed for mortar and concrete



163

164

## 165 2.1 Properties of ITZ

166 The development of ITZ imposes adverse effect on the mechanical properties; therefore, the  
 167 properties of ITZ such as volume fraction and mechanical properties are required to be predicted in a  
 168 realistic manner. As discussed earlier, the aggregate properties are one of the parameters that  
 169 determine the thickness of the ITZ [34,35], and the mechanical properties of ITZ vary with the  
 170 hydration period. Therefore, for a specific degree of hydration (i.e., for a specific curing period), the  
 171 ITZ can be considered as a single shell with uniform mechanical properties. Moreover, based on the  
 172 previous studies, the thickness is presumed to be 15  $\mu\text{m}$  for fine aggregate in mortar and 40  $\mu\text{m}$  for  
 173 coarse aggregate in concrete [36–38].

174

175 The volume fraction of aggregate in typical concrete is more than 60%, thereby the spacing between  
 176 adjacent aggregates can only be a few times the typical ITZ thickness [39,40]. Similar to that  
 177 implemented in previous studies, the volume fraction of ITZ is computed herein by considering the  
 178 overlapping of ITZ shells using the ‘void exclusion probability’ [41]. Principally, the void exclusion  
 179 probability is the volume fraction of the space not occupied by all the spheres and ITZ shells, i.e.,

180 fraction of the bulk paste. Hence, the ITZ volume fraction  $f_{ITZ}$  for an ITZ thickness of  $h$  can be  
 181 expressed as,

182

$$183 \quad f_{ITZ} = (1 - f_{agg})(1 - \exp(-\pi\rho(\alpha_1 h + \alpha_2 h^2 + \alpha_3 h^3))) \quad (1)$$

184 where,

$$185 \quad \alpha_1 = \frac{4\langle R^2 \rangle}{1 - f_{agg}} \quad (2)$$

$$186 \quad \alpha_2 = \frac{4\langle R \rangle}{1 - f_{agg}} + \frac{12\epsilon_2 \langle R^2 \rangle}{(1 - f_{agg})^2} \quad (3)$$

$$187 \quad \alpha_3 = \frac{4}{3(1 - f_{agg})} + \frac{8\epsilon_2 \langle R \rangle}{(1 - f_{agg})^2} + \frac{16A\epsilon_2^2 \langle R^2 \rangle}{3(1 - f_{agg})^3} \quad (4)$$

$$188 \quad \epsilon_2 = \frac{2\pi \langle R^2 \rangle}{3} \quad (5)$$

189

190 with  $A$  being equal to 2, 3, and 0 for the Carnahan-Starling, scaled- particle, and Percus-Yevick  
 191 approximation, respectively [41]. However, by comparing with numerical exact model data, Garboczi  
 192 and Bentz [42] suggested that  $A = 0$  is always the best choice for the simulation, thus,  $A$  is taken as  
 193 zero in this work.  $\rho$  is the total number of aggregates per unit volume, and  $\alpha_1$ ,  $\alpha_2$  and  $\alpha_3$  are functions  
 194 of mean aggregate radius  $\langle R \rangle$  and mean square aggregate radius  $\langle R^2 \rangle$  according to the aggregate size  
 195 distribution. According to previous studies [42,43], with the assumption of uniform distribution by  
 196 volume of aggregates, the remaining parameters are calculated from aggregate size distribution as,

197

$$198 \quad \rho = \sum \frac{9f_{agg}f_i}{4\pi(r_{i+1}^3 - r_i^3)} \ln \frac{r_{i+1}}{r_i} \quad (6)$$

$$199 \quad \langle R^n \rangle = \sum \frac{9f_{agg}f_i}{4\pi\rho(r_{i+1}^3 - r_i^3)} \int_{r_i}^{r_{i+1}} r^{n-1} dr \quad (7)$$

200

201  $f_{agg}$  is the volume fraction of aggregate.  $f_i$  is the fraction of the total volume of aggregate that has a  
 202 radius between  $r_i$  and  $r_{i+1}$ ,  $r_i < r_{i+1}$ . A typical sieve analysis is expressed in terms of the mass  
 203 fraction passing or retained by a certain sieve size, which can easily be converted to the form given

204 here. If aggregates of different size have all the same density, then mass fractions are the same as  
205 volume fractions. Once the ITZ volume fraction is known, the volume fraction of bulk paste ( $f_{bulk}$ )  
206 is obtained by simple subtraction.

207

208 To represent the concrete or mortar of practical use, the proportion of each phase constituent in the  
209 composite sphere should satisfy the following conditions (refer to **Figure. 2**) [35].

210

$$211 \quad f_{agg} = \frac{r_a^3}{r_c^3} \quad (8)$$

$$212 \quad f_{agg} + f_{itz} = \frac{r_b^3}{r_c^3} \quad (9)$$

213

214 The local volume fraction of anhydrous cement in bulk paste ( $\alpha_{c,bulk}$ ) and ITZ ( $\alpha_{c,ITZ}$ ) are computed  
215 from Nadeau [6]. Moreover, the effective w/c of bulk paste and ITZ is calculated based on Eq 10.

216

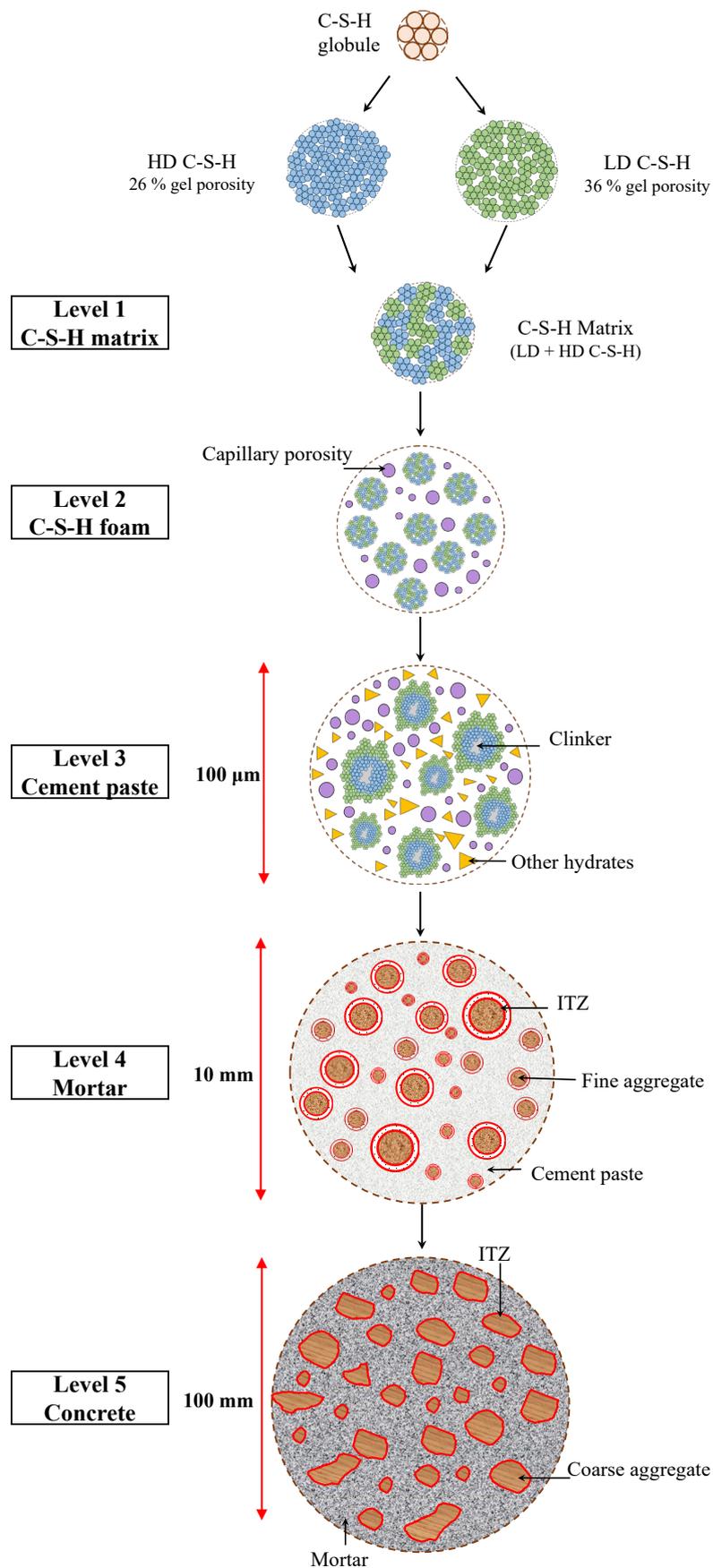
$$217 \quad w/c_{bulk \text{ or } ITZ} = \frac{1 - \alpha_{c,bulk \text{ or } \alpha_{c,ITZ}}}{\rho_c \alpha_{c,bulk \text{ or } \alpha_{c,ITZ}}} \quad (10)$$

218

## 219 **2.2 Multi-scale model**

220 In computational material science, concrete is characterized as a multi-phase material with different  
221 representative scales [8]. At mesoscopic scale, it is focused to be consisting of coarse aggregates and  
222 mortar matrix; the mortar matrix can be subdivided as fine aggregates and hardened cement paste;  
223 the hardened cement paste consists of several hydration products, unreacted clinker and capillary  
224 porosity. Therefore, a multi-scale model can be more appropriate for realistically predicting the  
225 behaviour of concrete and evaluating the impact of compositions on the mechanical properties.

226 **Figure. 3** presents the organization of the multi-scale model proposed in this study, describing the  
227 levels and constitutions for modelling the concrete.



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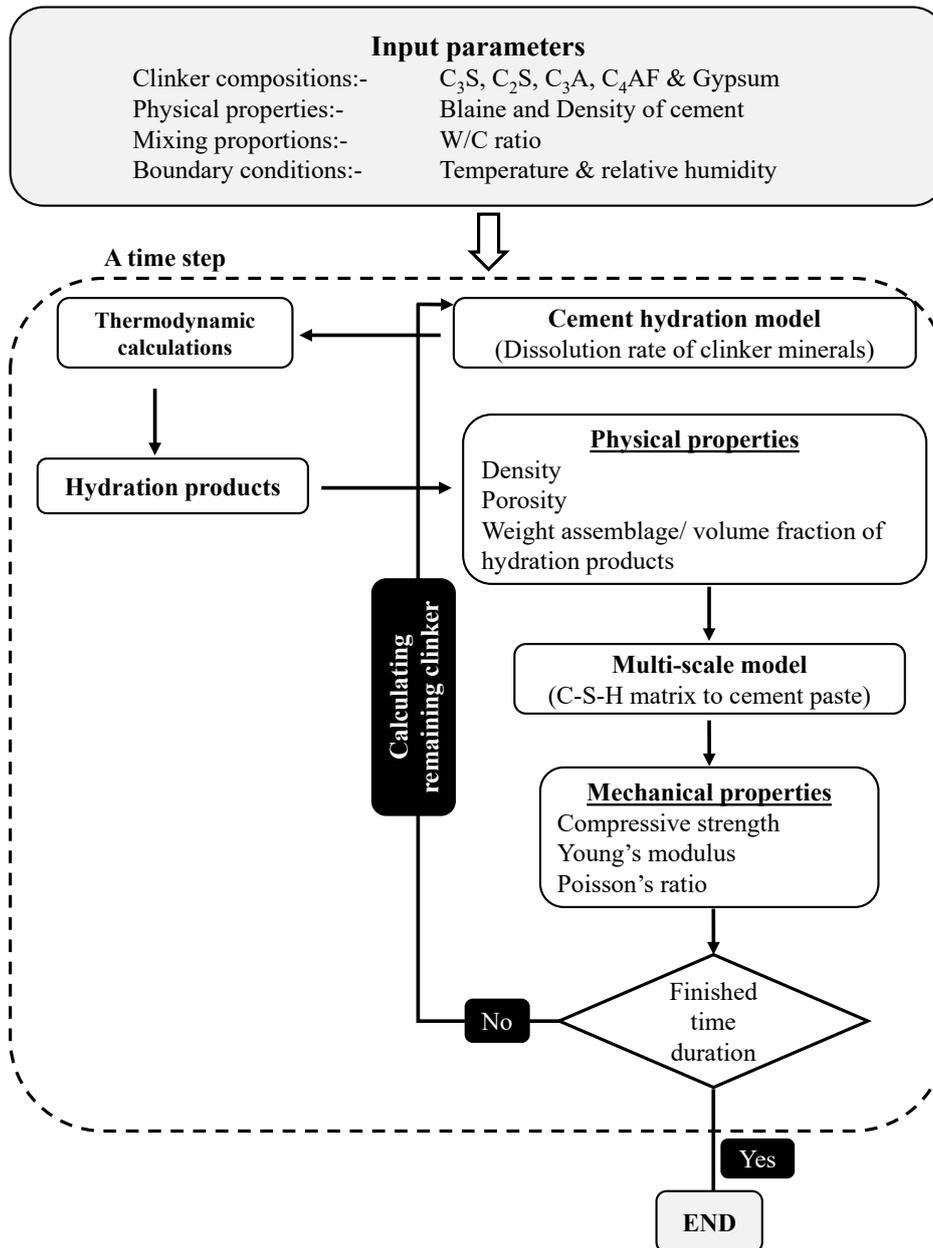
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230

**Figure. 3.** Representation of hierarchical model (multi-scale) proposed for concrete

231 **2.2.1 Model description for cement paste**

232 The development of cement paste model (Level 1-3 of the multi-scale model) is described briefly in  
233 this section. It should be noted that further detail of cement paste model can be found in our previous  
234 work [33]. Here, as the first step, the popular hydration model proposed by Parrot and Killoh [44]  
235 was implemented to predict the hydration products based on the considered input parameters such as  
236 clinker composition, boundary condition and mixing conditions. Thermodynamic calculations were  
237 performed using the open source geochemical (IPHREEQC) based on the estimated dissolution rate  
238 of clinker minerals to predict the hydration products. Using the predicted volume fraction of hydration  
239 products such as two types of C-S-H (Low density C-S-H and High density C-S-H), portlandite,  
240 ettringite, monosulfate, hydrotalcite, Fe-siliceous hydrogarnet and calculated chemical shrinkage  
241 based on the reaction rate of each clinker minerals, the capillary porosity with hydration period was  
242 predicted. The multi-scale model initiating from C-S-H matrix (Level 1) and ending up with cement  
243 paste (Level 3) was then used to predict the mechanical properties of cement paste. **Figure. 4** depicts  
244 the overall procedure adopted in the cement paste model.

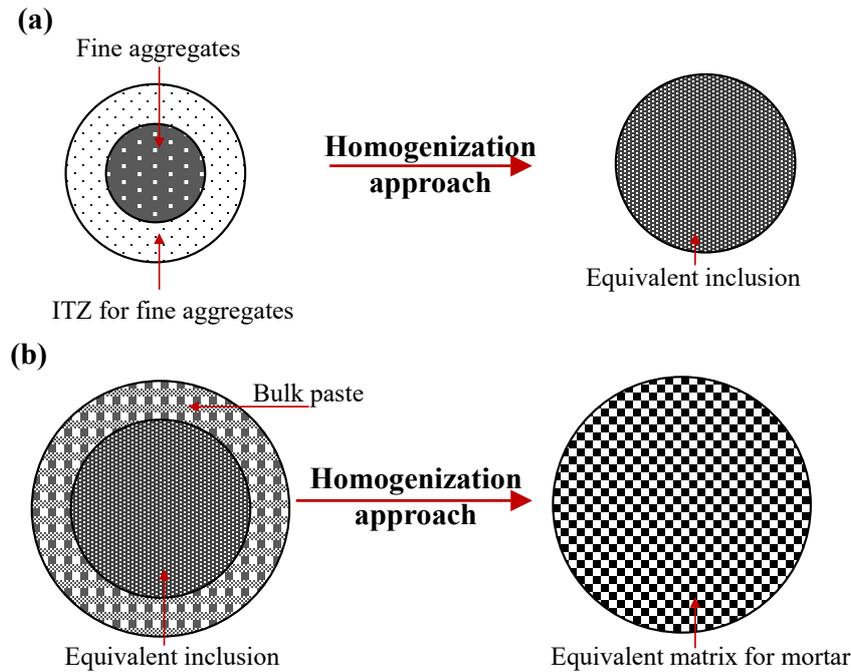


246 **Figure. 4.** Flow diagram illustrating the procedure used to develop the cement paste model  
 247

248 **2.2.2 Model description for mortar**

249 In the model proposed for mortar, the aggregate is assumed to be inert with the cement in order to  
 250 eliminate the complexity of the model. The properties of ITZ and bulk paste vary with w/c ratio,  
 251 degree of reaction, aggregate content, and clinker properties. As illustrated in **Figure. 1**, to predict  
 252 the mechanical properties of ITZ and bulk paste, following parameters are required: (i) clinker content,  
 253 (ii) w/c ratio and (iii) volume fraction of ITZ and bulk paste. Clinker content in ITZ and bulk paste is  
 254 obtained from Nadeau [6], while the w/c ratio and volume fraction are computed using Eq. 10 and  
 255 Eq. 1, respectively. The above-mentioned computations are then inputted to cement paste model

256 (which is the integration of cement hydration model, thermodynamic model, and multi-scale model)  
 257 to predict the mechanical properties. The predicted mechanical properties (of ITZ and bulk paste),  
 258 volume fraction of all three phases and considered mechanical properties of fine aggregate are finally  
 259 used to predict the mechanical properties of mortar such as Young's modulus and Poisson's ratio  
 260 using the homogenization method (as illustrated in **Figure. 5**).



261  
 262 **Figure. 5.** Two step homogenization procedure of mortar: (a) the first step homogenization of fine  
 263 aggregates and ITZ and (b) the second step homogenization of equivalent inclusion and bulk paste  
 264

265 The two-step homogenization method employed herein was developed by Christensen and Lo [45].  
 266 The method is based on three-phase sphere model for two-phase composite material, which was  
 267 developed to compute the effective bulk and shear modulus by assuming that each individual  
 268 aggregate has homogeneous particle size with equivalent elastic properties and radius. The origin of  
 269 this model was from the study by Eshelby [45]. Following the evaluations, Christensen [45]  
 270 concluded that the homogenization model can be more reliable than other generally used models such  
 271 as differential scheme and Mori-Tanaka model [46], because the stress-strain field interactions  
 272 between different inclusions were considered in this model [47]. This homogenization model was

273 also implemented in several previous studies to predict the effective properties of mortar and concrete  
 274 [20,47]. As per the first step, the effective properties of two-phase composite made up of fine  
 275 aggregates and ITZ (shown in **Figure. 5(a)**) are computed based on Eq. 11- Eq. 19 [45].

276

$$277 \quad K_{equ} = K_{ITZ} + \frac{\phi_{agg}(K_{agg}-K_{ITZ})(3K_{ITZ}+4G_{ITZ})}{3K_{ITZ}+4G_{ITZ}+3(1-\phi_{agg})(K_{agg}-K_{ITZ})} \quad (11)$$

$$278 \quad B\left(\frac{G_{equ}}{G_{ITZ}}\right)^2 + C\left(\frac{G_{equ}}{G_{ITZ}}\right) + D = 0 \quad (12)$$

$$279 \quad B = 8\left(\frac{G_{agg}}{G_{ITZ}} - 1\right)(4 - 5v_{ITZ})\eta_1\phi_{agg}^{\frac{10}{3}} - 2\left(63\left(\frac{G_{agg}}{G_{ITZ}} - 1\right)\eta_2 + 2\eta_1\eta_3\right)\phi_{agg}^{\frac{7}{3}} +$$

$$280 \quad 252\left(\frac{G_{agg}}{G_{ITZ}} - 1\right)\eta_2\phi_{agg}^{\frac{5}{3}} - 50\left(\frac{G_{agg}}{G_{ITZ}} - 1\right)(7 - 12v_{ITZ} + 8v_{ITZ}^2)\eta_2\phi_{agg} + 4(7 - 10v_{ITZ})\eta_2\eta_3$$

$$281 \quad (13)$$

282

$$283 \quad C = -4\left(\frac{G_{agg}}{G_{ITZ}} - 1\right)(1 - 5v_{ITZ})\eta_1\phi_{agg}^{\frac{10}{3}} + 4\left(63\left(\frac{G_{agg}}{G_{ITZ}} - 1\right)\eta_2 + 2\eta_1\eta_3\right)\phi_{agg}^{\frac{7}{3}} -$$

$$284 \quad 504\left(\frac{G_{agg}}{G_{ITZ}} - 1\right)\eta_2\phi_{agg}^{\frac{5}{3}} + 150\left(\frac{G_{agg}}{G_{ITZ}} - 1\right)(3 - v_{ITZ})v_{ITZ}\eta_2\phi_{agg} + 3(15v_{ITZ} - 7)\eta_2\eta_3$$

$$285 \quad (14)$$

$$286 \quad D = 4\left(\frac{G_{agg}}{G_{ITZ}} - 1\right)(5v_{ITZ} - 7)\eta_1\phi_{agg}^{\frac{10}{3}} - 2\left(63\left(\frac{G_{agg}}{G_{ITZ}} - 1\right)\eta_2 + 2\eta_1\eta_3\right)\phi_{agg}^{\frac{7}{3}} +$$

$$287 \quad 252\left(\frac{G_{agg}}{G_{ITZ}} - 1\right)\eta_2\phi_{agg}^{\frac{5}{3}} + 25\left(\frac{G_{agg}}{G_{ITZ}} - 1\right)(v_{ITZ}^2 - 7)\eta_2\phi_{agg} - 3(5v_{ITZ} + 7)\eta_2\eta_3$$

$$288 \quad (15)$$

$$289 \quad \eta_1 = \left(\frac{G_{agg}}{G_{ITZ}} - 1\right)(49 - 50v_{agg}v_{ITZ}) + 35\left(\frac{G_{agg}}{G_{ITZ}}\right)(v_{agg} - 2v_{ITZ}) + 35(2v_{agg} - v_{ITZ})$$

$$290 \quad (16)$$

$$291 \quad \eta_2 = 5v_{agg}\left(\frac{G_{agg}}{G_{ITZ}} - 8\right) + 7\left(\frac{G_{agg}}{G_{ITZ}} + 4\right) \quad (17)$$

$$292 \quad \eta_3 = \frac{G_{agg}}{G_{ITZ}}(8 - 10v_{ITZ}) + (7 - 5v_{ITZ}) \quad (18)$$

$$293 \quad \Phi_{agg} = \frac{f_{agg}}{f_{agg} + f_{ITZ}} \quad (19)$$

294

295 where  $K_{equ}$  and  $G_{equ}$  are the effective bulk modulus and shear modulus of the equivalent inclusions  
 296 after the first step of homogenization.  $K_{agg}$ ,  $G_{agg}$  and  $\nu_{agg}$  are the bulk modulus, shear modulus and  
 297 Poisson's ratio of fine aggregates.  $K_{ITZ}$ ,  $G_{ITZ}$  and  $\nu_{ITZ}$  are the bulk modulus, shear modulus and  
 298 Poisson's ratio of ITZ, and which are obtained from the previous work [33].

299

300 Since the mortar consisting of bulk paste and equivalent inclusion (as elucidated in **Figure. 5(b)**), the  
 301 effective mechanical properties of mortar can be similarly computed by homogenization approach.  
 302 The Eq. 11 - Eq. 19 are amended by replacing the effective properties of equivalent inclusion ( $\Phi_f$ ,  
 303  $G_{equ}$ ,  $K_{equ}$  and  $\nu_{equ}$ ) and bulk paste properties ( $G_{bulk}$ ,  $K_{bulk}$  and  $\nu_{bulk}$ ) in the place of aggregate  
 304 properties ( $\Phi_{agg}$ ,  $G_{agg}$ ,  $K_{agg}$  and  $\nu_{agg}$ ) and ITZ properties ( $G_{ITZ}$ ,  $K_{ITZ}$  and  $\nu_{ITZ}$ ) to predict effective  
 305 bulk modulus and shear modulus of mortar ( $K_{mor}$ ,  $G_{mor}$ ). The formula for volume fraction of  
 306 equivalent inclusion in mortar is provided in Eq. 20. The bulk paste properties ( $G_{bulk}$ ,  $K_{bulk}$  and  
 307  $\nu_{bulk}$ ) are obtained from the previous work [33].

308

$$309 \quad \Phi_f = \frac{f_{agg} + f_{ITZ}}{f_{agg} + f_{ITZ} + f_{bulk}} \quad (20)$$

$$310 \quad E_{mor} = \frac{9K_{mor}G_{mor}}{3K_{mor} + G_{mor}} \quad (21)$$

$$311 \quad \nu_{mor} = \frac{3K_{mor} - 2G_{mor}}{2(3K_{mor} + G_{mor})} \quad (22)$$

312

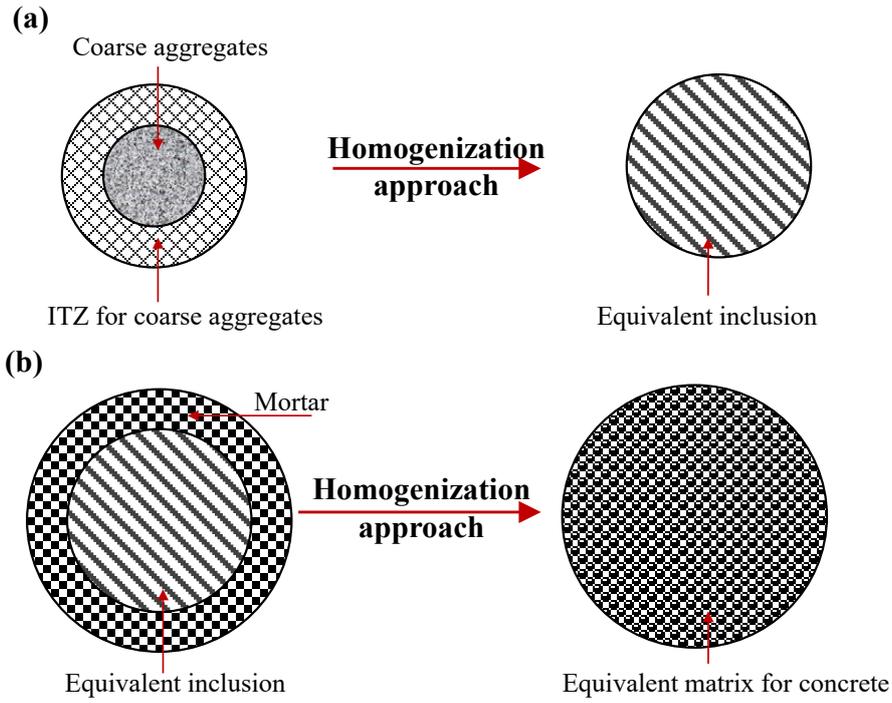
313 Theoretically, two sets of moduli are required for the computation of Young's modulus and Poisson's  
 314 ratio of a material; therefore, bulk modulus of mortar ( $K_{mor}$ ) together with shear modulus of mortar  
 315 ( $G_{mor}$ ) were used herein to predict the Young's modulus ( $E_{mor}$ ) and Poisson's ratio ( $\nu_{mor}$ ) from the

316 fundamental relationship presented in Eq.21 and Eq. 22. Finally, the predicted results were validated  
317 using experimental results.

318

### 319 **2.2.3 Model description for concrete**

320 In the proposed model (refer **Figure. 1**), concrete is reliably assumed as three phase material  
321 consisting of coarse aggregate, ITZ and mortar. In order to eliminate the complexity of the simulation,  
322 coarse aggregate is considered as chemically inactive with cement paste similar to that considered for  
323 fine aggregate. The volume fraction of ITZ for coarse aggregate ( $f_{c.ITZ}$ ) is computed with aid of Eq.  
324 1 using the particle size distribution of coarse aggregate and ITZ thickness. Similar procedure that  
325 has been described for mortar is used to estimate the clinker content and w/c in ITZ for coarse  
326 aggregate and mortar (existing in concrete). Based on the computed w/c and clinker content, the  
327 mechanical properties of ITZ paste is predicted using the cement paste model [33]. Besides, the  
328 mechanical properties of mortar are predicted using the computed w/c, clinker content and fine  
329 aggregate content as described in section 2.2.1. The predicted mechanical properties (of ITZ and  
330 mortar), volume fraction of all three phases and considered mechanical properties of coarse aggregate  
331 are finally used to predict the mechanical properties of concrete using the homogenization method  
332 (illustrated in **Figure. 6**).



333

334 **Figure. 6.** Two step homogenization procedure for concrete: **(a)** the first step homogenization of  
 335 coarse aggregates and ITZ and **(b)** the second step homogenization of equivalent inclusion and  
 336 mortar matrix

338 As elucidated in **Figure. 6(a)**, the coarse aggregates and ITZ are considered in the first step. The Eq.  
 339 11 – Eq. 19 has been modified by replacing coarse aggregates properties ( $\phi_{C.agg}$ ,  $G_{C.agg}$ ,  $K_{C.agg}$  and  
 340  $\nu_{C.agg}$ ) and ITZ properties of coarse aggregates ( $G_{C.ITZ}$ ,  $K_{C.ITZ}$  and  $\nu_{C.ITZ}$ ) in the place of fine  
 341 aggregates properties ( $\phi_{agg}$ ,  $G_{agg}$ ,  $K_{agg}$  and  $\nu_{agg}$ ) and ITZ properties of fine aggregates ( $G_{ITZ}$ ,  $K_{ITZ}$   
 342 and  $\nu_{ITZ}$ ) to predict the bulk ( $K_{equ, con}$ ) and shear modulus ( $G_{equ, con}$ ) of equivalent inclusion for  
 343 concrete. The formula for volume fraction of coarse aggregate in equivalent inclusion is given in Eq.  
 344 23.

345

$$346 \quad \phi_{C.agg} = \frac{f_{C.agg}}{f_{C.agg} + f_{C.ITZ}} \quad (23)$$

347 where,  $f_{C.agg}$  is the volume fraction of coarse aggregate in concrete.

348

349 In the second step, the equivalent inclusion and mortar matrix are considered for the homogenization  
 350 method as shown in **Figure. 6(b)**. The effective mechanical properties of concrete can be similarly  
 351 computed by homogenization approach. The Eq. 11 - Eq. 19 were altered by replacing the effective  
 352 properties of equivalent inclusion ( $\emptyset_{C.f}$ ,  $G_{equ,con}$ ,  $K_{equ,con}$  and  $\nu_{equ, con}$ ) and mortar properties  
 353 ( $G_{mor}$ ,  $K_{mor}$  and  $\nu_{mor}$ ) in the place of coarse aggregate properties ( $\emptyset_{C.agg}$ ,  $G_{C.agg}$ ,  $K_{C.agg}$  and  
 354  $\nu_{C.agg}$ ) and ITZ properties of coarse aggregates ( $G_{C.ITZ}$ ,  $K_{C.ITZ}$  and  $\nu_{C.ITZ}$ ) to predict effective bulk  
 355 modulus and shear modulus of concrete ( $K_{Con}$ ,  $G_{Con}$ ). The volume fraction of equivalent inclusion in  
 356 concrete is given in Eq. 24.

357

$$358 \quad \emptyset_{C.f} = \frac{f_{C.agg} + f_{C.ITZ}}{f_{C.agg} + f_{C.ITZ} + f_{mor}} \quad (24)$$

359

360 where,  $f_{mor}$  is volume fraction of mortar in concrete. Once the volume fraction of ITZ for coarse  
 361 aggregate is known, the volume fraction of mortar ( $f_{mor}$ ) is obtained by simple subtraction ( $f_{mor} =$   
 362  $1 - f_{C.agg} - f_{C.ITZ}$ ).

363

364 The Young's modulus ( $E_{Con}$ ) and the Poisson's ratio ( $\nu_{Con}$ ) of concrete are predicted from the  
 365 fundamental relationship presented in Eq. 21 and Eq. 22 by replacing the concrete moduli in the place  
 366 of mortar properties. Finally, the predictability of the model is validated using the experimental results  
 367 available in the literature.

368

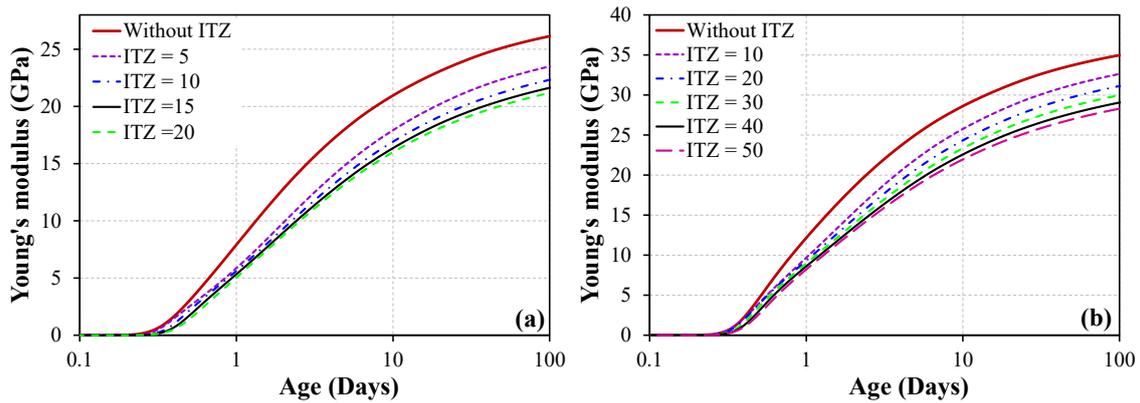
### 369 **3. Results and discussion**

#### 370 **3.1 Sensitivity analysis of ITZ**

371 **Figure. 7** presents the Young's modulus of mortar and concrete predicted as a function of hydration  
 372 period for different ITZ thickness. It should be mentioned that the prediction corresponds to the w/c  
 373 of 0.6. The considered volume fraction of fine aggregate for mortar is 0.6, and cement: fine aggregate:

374 coarse aggregate weight ratio for concrete is 1: 1.5: 3.2. The results clearly demonstrate that  
 375 increasing the thickness of the ITZ results in a reduction in predicted Young's modulus. It is well  
 376 known that the ITZ is the weakest zone in mortar and concrete [48]. The increase in ITZ thickness  
 377 directly upsurges the volume fraction of ITZ, leading to the decrease in effective elastic modulus of  
 378 the mortar and concrete. Similar behavior was observed by Simeonov and Ahmad [26] and Li et al.  
 379 [28]. However, when the ITZ thickness exceeds 15  $\mu\text{m}$  for mortar and 40  $\mu\text{m}$  for concrete, the  
 380 predicted results do not show significant variation (refer **Figure. 7**). It is apparent that the volume  
 381 fraction of ITZ increases with the increasing thickness, yet the volume fraction of bulk paste decreases,  
 382 ensuing a negative effect on effective Young's modulus. Meanwhile, the local w/c in ITZ and bulk  
 383 paste increases with ITZ thickness based on the model proposed by Nadeau [6], which leads to a  
 384 positive effect on modulus. Since the above effects compromise each other, further increase in ITZ  
 385 thickness does not show significant reduction in final outcomes. Relatively a similar ITZ thickness  
 386 (in a range between 10-50  $\mu\text{m}$ ) was also reported for sand and coarse aggregates in many previous  
 387 works [5,6,8,9]. On the above basis, therefore, the ITZ thicknesses of 15 and 40  $\mu\text{m}$  are reliably  
 388 chosen for the computations in the proposed model.

389



390

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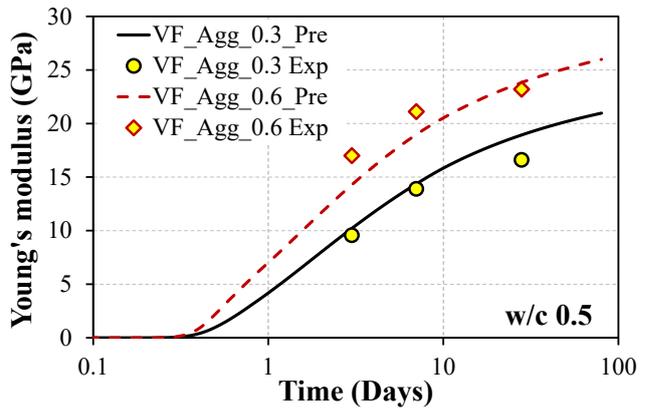
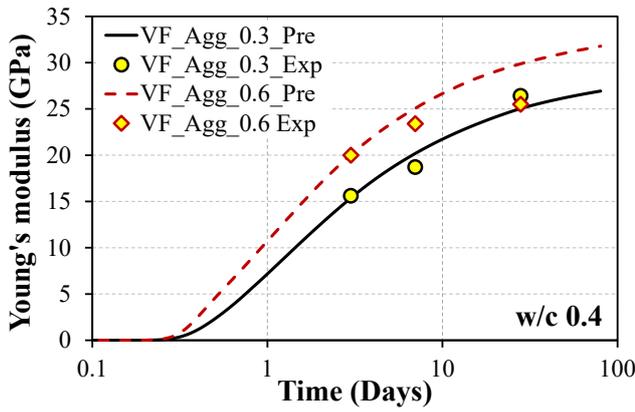
394

### 395 3.2 Young's modulus of mortar

396 To reveal the predictability of the proposed model, the predicted Young's moduli of mortar are  
397 compared with two independent sets of experimental results (refer **Figure. 8** and **Figure. 9**). In  
398 **Figure. 8**, the raw experimental data were employed for the comparison. The experimental conditions  
399 and procedures corresponding to the set of data compared in **Figure. 8** are detailed in **Appendix A**.  
400 The data used in **Figure. 9** for validating the predicted results were obtained from previous report  
401 [49]. The comparisons indicate that the predicted results exhibit a good agreement with the both  
402 experimental sets for range of w/c and aggregate volume fractions. Slight variations are however  
403 witnessed in certain readings. For instance, 5 GPa variation could be seen between experimental and  
404 predicted results for the mortar with w/c of 0.4, aggregate volume fraction of 0.6 and curing period  
405 of 28 days (in **Figure 8**). It should be noted that in computations, the Young's modulus and Poisson's  
406 ratio of fine aggregate were considered to be 37.7 GPa and 0.2 respectively [50].

407  
408 Regardless of w/c and aggregate volume fractions, the Young's modulus increases with hydration  
409 period. Besides, relatively higher values are predicted for the mortar with lower w/c, which is in  
410 consistent with that of hydrated cement paste. This could mainly be attributed to the increased  
411 formation of prime binding agent (i.e., C-S-H) at low w/c during the hydration. In due course, the  
412 decrease in the porosity could lead to the development of denser microstructure [33], resulting an  
413 increase in Young's modulus of mortar matrix with hydration period. Moreover, for a specific w/c  
414 ratio, high moduli are computed for high content of fine aggregate (refer to **Figure. 8** and **Figure. 9**).  
415 This is because of the increased stiff inclusion in the paste and decreased effective water content,  
416 which lead to the decrease in porosity in bulk paste [10,51]. The tendencies predicted herein for  
417 Young's modulus of different cases are in consistent with those reported in previous studies  
418 [29,49,52–54].

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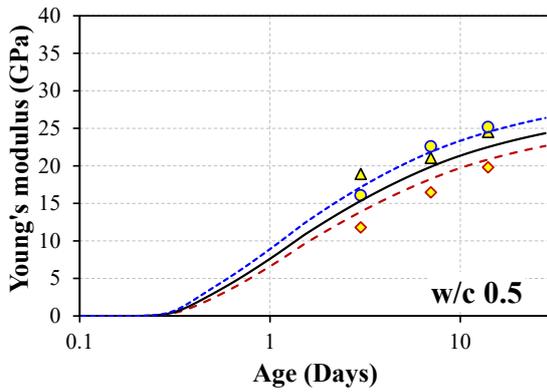
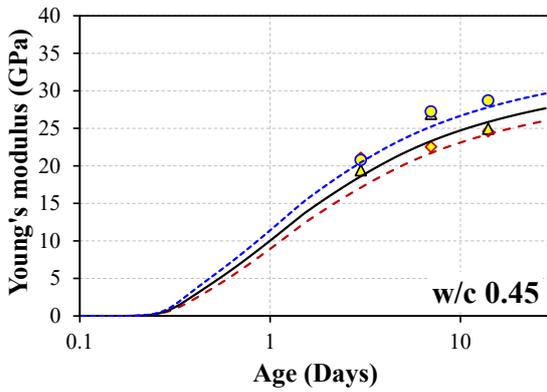
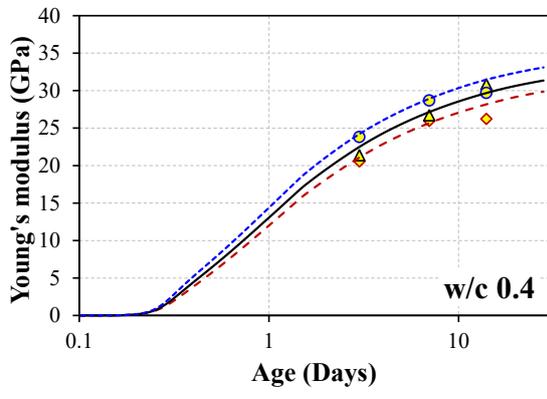
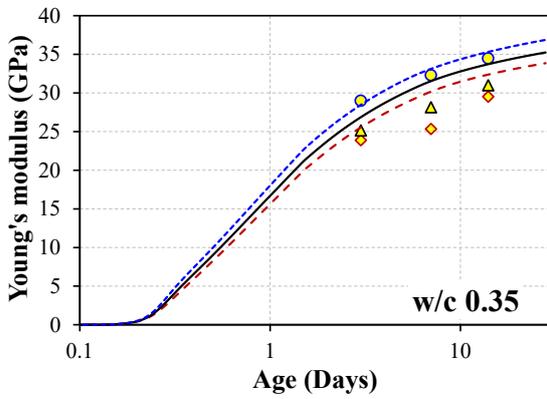


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(VF\_Agg stands for volume fraction of aggregate)



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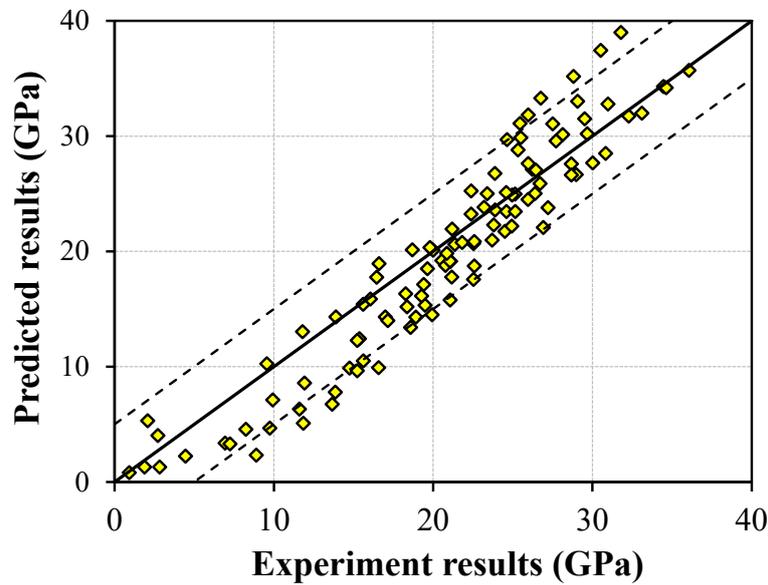
aggregate 0.35-0.55 (the experiment results were obtained from ref. [49]) (VF\_Agg stands for

426

volume fraction of aggregate)

427

428 **Figure. 10** presents the compilation of predicted Young's modulus plotted against the experimental  
429 results of mortars corresponding to w/c ratios ranging from 0.3 to 0.55 and aggregate volume fraction  
430 from 0.2 to 0.6. The experimental data were obtained from both measurements and literature [49,53].  
431 As illustrated in **Figure. 10**, the predicted results of Young's modulus fall within the error range of 5  
432 GPa for all three sets of experiment results. Based on the sets of comparisons (**Figure. 8**, **Figure. 9**,  
433 and **Figure. 10**), it is perceived that the proposed model can be reliable for predicting the Young's  
434 modulus of mortar with different w/c ratios and aggregate volume fractions.



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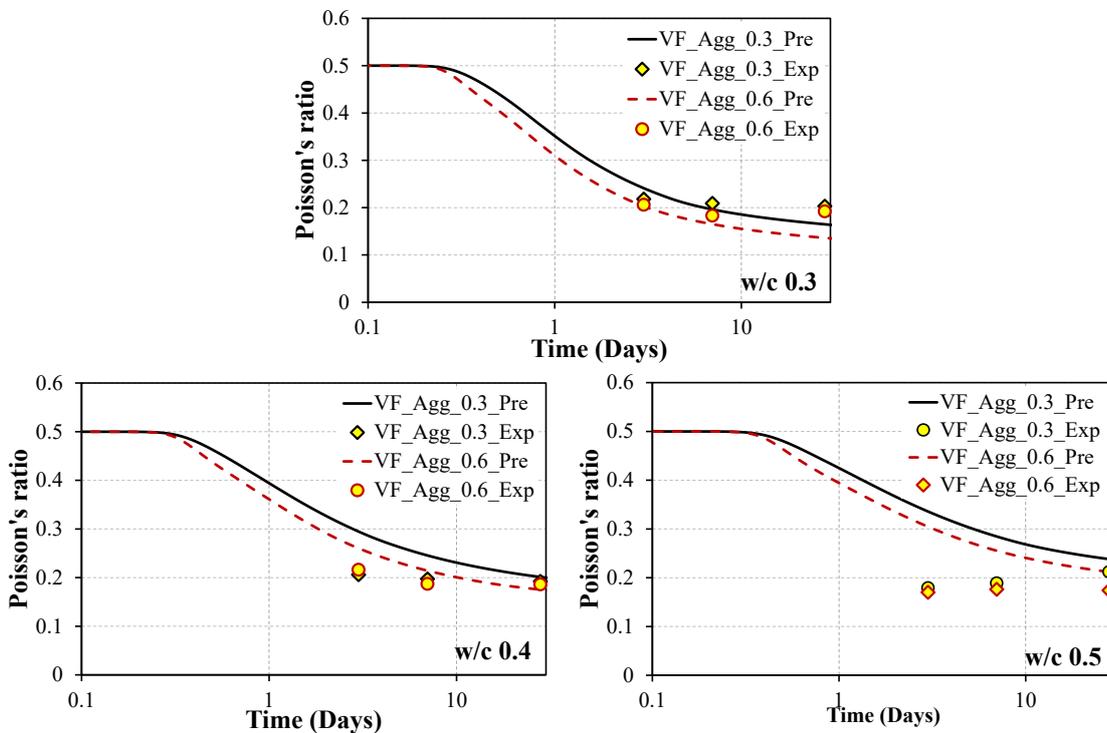
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439

### 440 3.3 Poisson's ratio of mortar

441 **Figure. 11** presents the variation of Poisson's ratio of mortar with hydration period for w/c of 0.3,  
442 0.4 and 0.5 and fine aggregates volume fraction of 0.3 and 0.6 (The experimental conditions and  
443 procedures corresponding to the experiment data presented in **Figure. 11** are detailed in **Appendix**  
444 **A.**). It can be observed that the predicted results appear to slightly overestimate the Poisson's ratio  
445 for w/c ratio of 0.5, while those show a better agreement with experimental results for the w/c ratios  
446 of 0.3 and 0.4. The tendency observed for w/c ratio of 0.5 increases with increasing hydration period,

447 which is in contrast with those observed for other w/c ratios. In fact, during the early stage, the mortar  
 448 exists in suspension form within the water, thus the Poisson's ratio of fresh mix is equal to Poisson's  
 449 ratio of water (0.5) [55,56], and which is captured by the proposed model (**Figure. 11**). When the  
 450 hydration is in progress, the Poisson's ratio of mortar decreases due to the formation of denser  
 451 microstructure by replacing the solids in the place of water. Thus, the increasing tendency observed  
 452 for w/c of 0.5 might be due to experimental error. It can also be noticed that the Poisson's ratio  
 453 increases with the increase in w/c ratio, which is similar to the tendency observed for cement paste  
 454 [33,57]. For a specific water cement ratio, the Poisson's ratio decreases with the increase in volume  
 455 fraction of fine aggregates (refer to **Figure. 11**). It is clear that sand is considerably stiffer than the  
 456 cement paste; therefore, the increasing stiffer inclusions in paste tends to restrain the lateral expansion  
 457 of the softer matrix. In fact, both Young's modulus and Poisson's ratio of the aggregate influence the  
 458 elastic properties such as Poisson's ratio and elastic modulus of mortar [58].  
 459

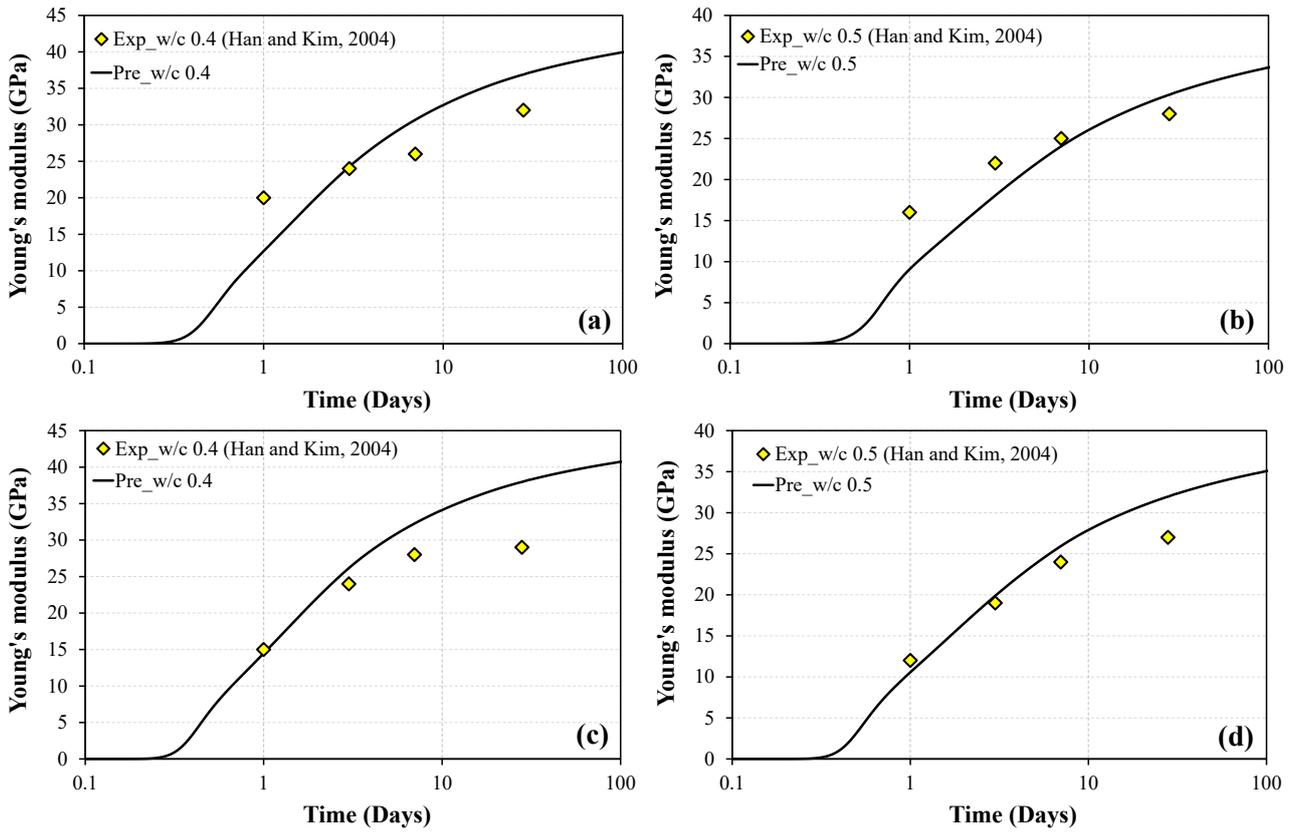


460  
 461 **Figure. 11.** Validation of Poisson's ratio of the mortar for w/c 0.3-0.5 with volume fraction of  
 462 aggregate 0.3 and 0.6 (VF\_Agg stands for volume fraction of aggregate)  
 463

### 464 3.4 Young's modulus of concrete

465 **Figure. 12** and **Figure. 13** illustrate the predictability of the model for the Young's modulus of  
466 concrete with two independent sets of experimental results obtained from previous studies. **Figure.**  
467 **12** shows the comparison of Young's modulus for different w/c ratio and different types of cement  
468 (Type I and Type V). The experiment results used herein for the comparison belong to Han and Kim  
469 [59]. The second set of experimental data compared in **Figure. 13** were obtained from Corinaldesi  
470 and Moriconi [60] for w/c, volume fraction of fine aggregate and volume fraction of coarse aggregates  
471 of 0.56, 0.12 and 0.5, respectively. During the computation, the Young's modulus and Poisson's ratio  
472 of coarse aggregates were considered to be 50 GPa and 0.18, respectively [61], and for the fine  
473 aggregates, they were respectively 37.7 GPa and 0.20. Based on the considered aggregates properties,  
474 the proposed model shows a good capability to predict the Young's modulus of concrete for different  
475 w/c and wide range of aggregates content. There are however slight variations observed for some  
476 points (refer **Figure. 12**), which could possibly be due to the assumptions made during prediction.  
477 Since the required parameters such as particle size distribution and properties of aggregates (Young's  
478 modulus and Poisson's ratio) were not available in the literature, they were assumed based on the  
479 instructions given in literature. Additionally, the aggregates were assumed to be inert and ITZ  
480 properties to be constant with distance from aggregate for a specific time. Nevertheless, the  
481 development of Young's modulus of concrete reveals almost a similar tendency of the development  
482 of Young's modulus of mortar.

483



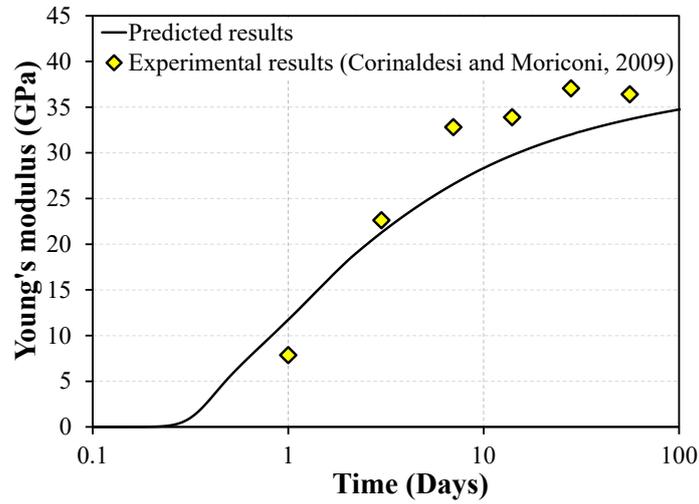
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485

**Figure. 12.** Validation of Young's modulus of concrete (experiment results were obtained from ref. [59] of (a) and (b) for Type I cement and (c) and (d) for Type V cement.

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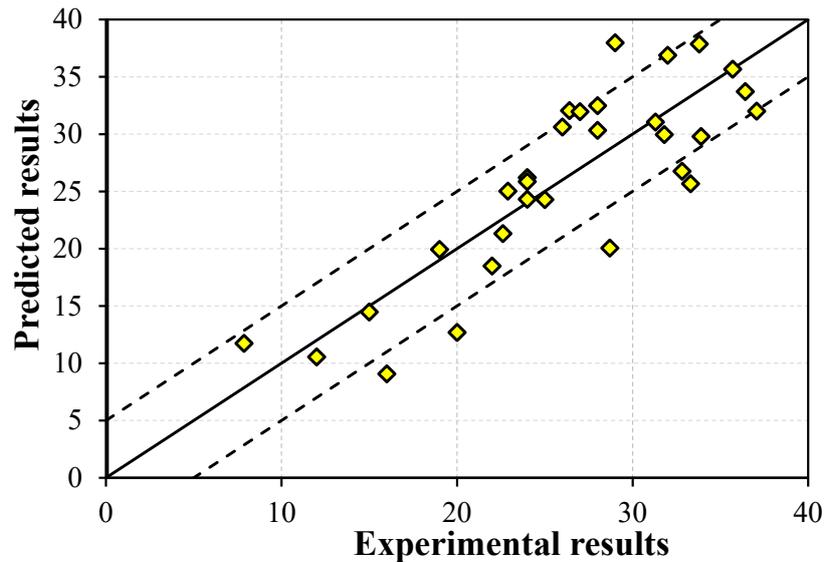
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**Figure. 13.** Validation of Young's modulus of concrete (experiment results were obtained from ref. [60])

490

491

492 **Figure. 14** shows the overall validation for 30 number of measured Young's modulus of concrete  
 493 with the w/c of 0.4 to 0.6. All the experiment results were taken from previous studies [59,60,62].  
 494 The predicted results of Young's modulus fall within the error range of 5 GPa for all three sets of  
 495 experiment results as depicted in **Figure. 14**. Based on the sets of validations (**Figure. 12**, **Figure. 13**  
 496 and **Figure. 14**), it is verified that the proposed model can be reliable for predicting the Young's  
 497 modulus of concrete with different w/c ratios and aggregate volume fractions.



499 **Figure. 14.** Comparison of predicted Young's modulus plotted against the experiment results

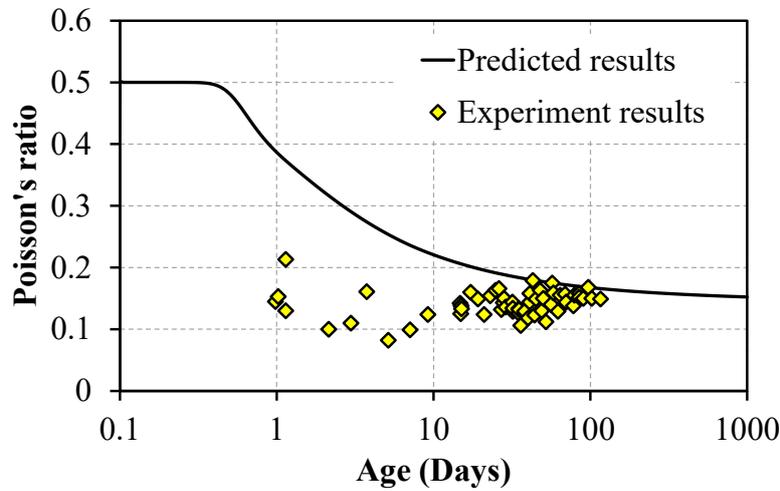
500

### 501 3.5 Poisson's ratio of concrete

502 The validation of Poisson's ratio with hydration period of concrete for w/c of 0.5, sand to cement  
 503 ratio of 1.5 and coarse aggregate to cement ratio of 3 is presented in **Figure. 15**. The experiment  
 504 results used for the comparison were taken from Allos and Martin [63]. The predicted results show  
 505 relatively good agreement with experiment results (after 10 days); however, the model appears to  
 506 overpredicts the Poisson's ratio at the early stage. According to the prediction, the Poisson's ratio of  
 507 concrete decreases with the increase in hydration period. This is understandable that the formation of  
 508 denser microstructure during the hydration replaces the solids in the place of water. Particularly, the  
 509 greatest decrease in Poisson's ratio occurs until 10 days, and after 10 days, the decrease becomes very  
 510 mild. A similar behaviour was also observed by Narayan Swamy [58]. Therefore, to understand and

511 overcome the discrepancy in early-stage prediction, more validations with independent sets of  
512 experiments are required and left for the future work.

513



515

516 **Figure. 15.** Validation of Poisson's ratio of concrete with hydration period (the experimental  
517 data were obtained from ref. [63])

518

### 519 3.6 Effect of ITZ on Young's modulus and Poisson's ratio of mortar and concrete.

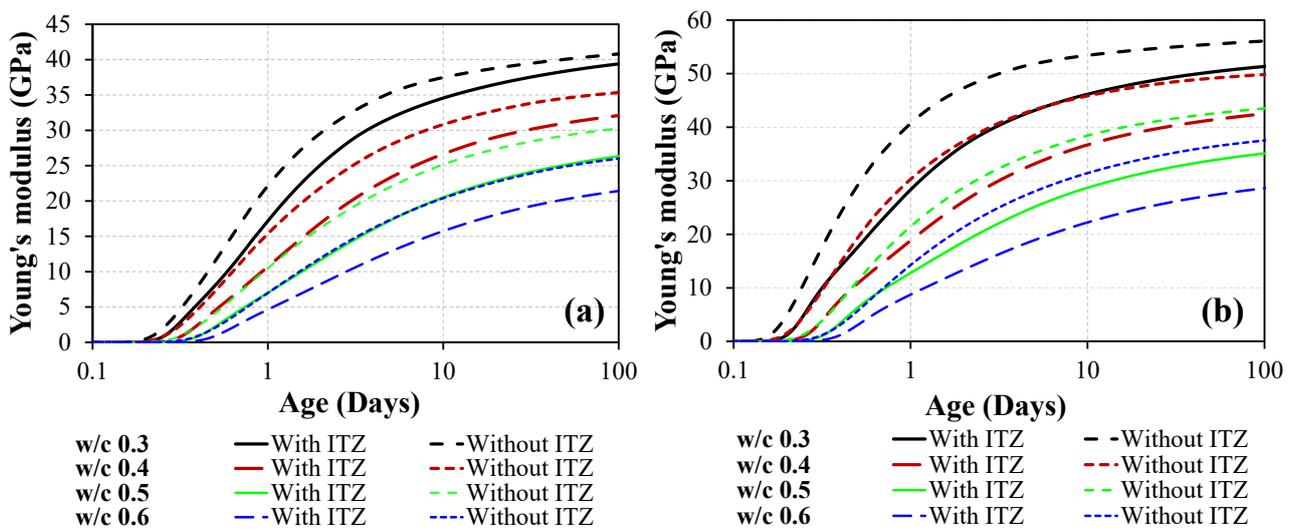
520 The Young's moduli of mortar and concrete computed with and without considering the ITZ are  
521 compared in **Figure. 16** for different w/c ratios. The considered fine aggregate volume fraction for  
522 mortar is 0.6 and cement: fine aggregate: coarse aggregate weight ratio for concrete is 1: 1.5: 3.2. It  
523 is evident that the model overestimates the modulus when the effect of ITZ is neglected for both  
524 mortar and concrete (**Figure. 16(a)** and **Figure. 16(b)**, respectively). Besides, the variation of  
525 Young's modulus between the cases of with and without ITZ decreases with increasing hydration  
526 period. As mentioned earlier, ITZ can also be considered as a distinct cement paste with low amount  
527 of clinker and high amount of water [6,35,64,65]. At the early age, the ITZ remains the weakest region  
528 with very high amount of porosity. However, during the hydration process, the ITZ becomes hardened  
529 cement paste with increased mechanical properties. Owing to the above reason, the variation in later  
530 age appeared to be decreasing. Relatively a similar tendency can be observed for all the w/c ratios

531 (Figure. 16). However, the deviation observed for concrete (Figure. 16(b)) is higher compared to the  
 532 mortar results (Figure. 16(a)), which is mainly due to the higher ITZ effect in concrete (attributing  
 533 to fine aggregate and coarse aggregate). For instance, after one day of w/c for 0.4 mortar, the modulus  
 534 of without-ITZ case is approximately 40 % higher than that of with-ITZ case, whereas the variation  
 535 after 28 days is reduced to around 10 % (see Figure. 16(a)). For the w/c 0.4 of concrete, after one  
 536 day, the modulus of without-ITZ case is approximately 60 % higher than that of with-ITZ case, while  
 537 the variation after 28 days is reduced to around 20 % (in Figure. 16(b)).

538

539 The difference between with ITZ and without ITZ cases is observed to increase with w/c ratio for  
 540 both the mortar and concrete. This could be due to the high migration of free water to the surface of  
 541 aggregates for high w/c matrix. In fact, the rate of migration is higher for the paste with higher w/c  
 542 ratios; as the result, there would be a higher w/c ratio in ITZ, while the remaining paste would have  
 543 a lower ratio. Since the elastic properties of the matrix and the ITZ are directly related to the water  
 544 content and their porous structure, greater migration process of the water (at higher w/c ratio) led to  
 545 more negative effect on Young's modulus. Similar observation was reported by Simeonov and  
 546 Ahmad [26].

547



548

549 **Figure. 16.** Comparison of predicted Young's modulus of (a) mortar and (b) concrete for the cases

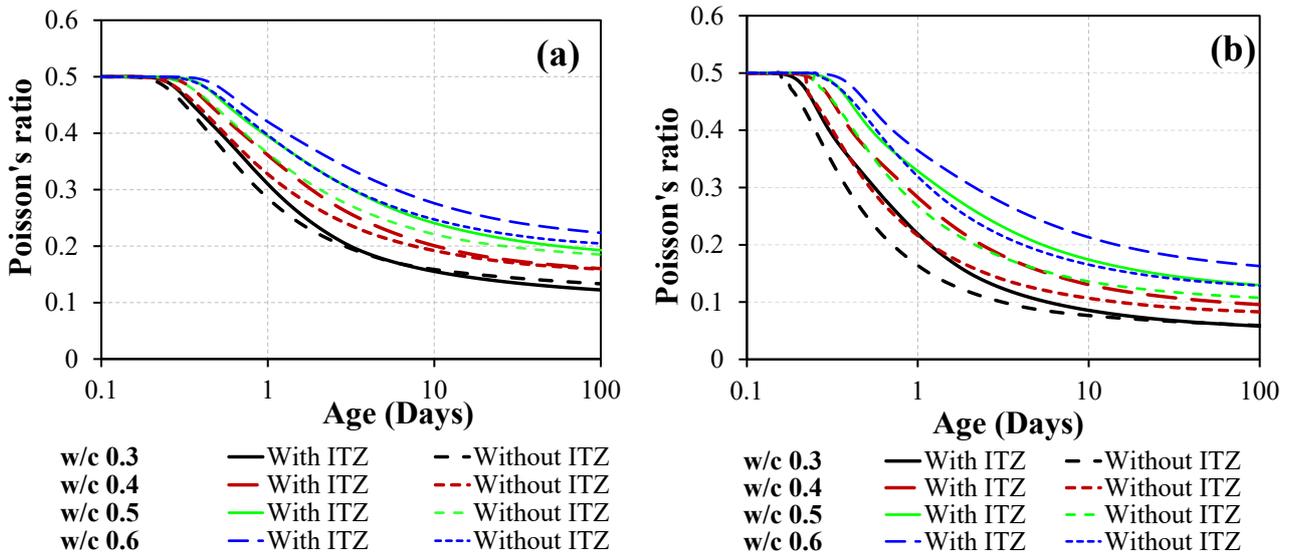
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of with and without ITZ for w/c 0.3-0.6

551

552 The comparison of predicted Poisson's ratio of mortar and concrete for the cases with and without  
553 ITZ is depicted in **Figure. 17**. The results presented herein are corresponding to the w/c ratio of 0.3-  
554 0.6 and to the fine aggregate volume fraction of 0.6 for mortar whereas cement: fine aggregate: coarse  
555 aggregate weight ratio for concrete is 1: 1.5: 3.2. It can be seen that in the predicted results of both  
556 cases, the Poisson's ratio increases with increase in w/c ratio, which is in consistent with the tendency  
557 reported by Stefan et al. [55]. If the mortar and concrete is considered as two-phase matrix by  
558 neglecting ITZ, the effective Poisson's ratio is stiffer than that predicted for with-ITZ case for all w/c  
559 ratios as reflected in **Figure. 17**. Moreover, the variation between the predicted results from both  
560 cases reduce with hydration period, which is probably because the mortar and concrete matrix become  
561 hardened solid including ITZ due to the hydration process of clinker. Nevertheless, the deviation  
562 observed for concrete (**Figure. 17(b)**) is higher compared to the mortar results (**Figure. 17(a)**), which  
563 is mainly due to the higher ITZ effect in concrete (attributing to fine aggregate and coarse aggregate).  
564 For instance, after one day of w/c for 0.4 mortar, the Poisson's ratio of without-ITZ case is  
565 approximately 10 % lower than that of with-ITZ case, whereas the variation after 28 days is reduced  
566 to around 2 % (see **Figure. 17(a)**). For the w/c 0.4 of concrete, after one day, the Poisson's ratio of  
567 without-ITZ case is approximately 20 % lower than that of with-ITZ case, while the variation after  
568 28 days is reduced to around 15 % (in **Figure. 17(b)**). Moreover, similar to that observed in Young's  
569 modulus (refer to **Figure. 16**), the variation between with and without ITZ cases increase with w/c  
570 ratio for both mortar and concrete.

571



573 **Figure. 17.** Comparison of predicted Poisson's ratio of (a) mortar and (b) concrete for the cases of  
 574 with and without ITZ for w/c 0.3-0.6

575

#### 576 4. Conclusions

577 In this research work, the mechanical properties such as Young's modulus and Poisson's ratio of  
 578 mortar and concrete were reliably predicted using the developed multi-scale model and  
 579 homogenization method. The mortars and concretes were assumed as three-phase composites, and  
 580 the ITZ for both coarse and fine aggregate were reliably considered as cement paste with high water  
 581 content. The mechanical properties of ITZ and bulk paste were based on the detailed microstructure  
 582 of hydrates during hydration reaction. The proposed models for mortar and concrete were validated  
 583 with experiment results for wide range of aggregate contents and w/c ratios.

584

585 The predicted Young's modulus excellently captured the realistic behaviour for the w/c ratios ranging  
 586 from 0.3 to 0.6 and aggregate content from 0.2 to 0.6. The computation of Poisson's ratio showed  
 587 relatively good agreement with the experimental data for mortar and concrete. However, the model  
 588 slightly overpredicted the Poisson's ratio at early stage (before 10 days) for concrete; thus, further  
 589 validations are recommended with more independent experimental results to generalize the Poisson's  
 590 ratio model of concrete. Moreover, the effect of ITZ on Young's modulus and Poisson's ratio were

591 demonstrated using the proposed model for mortar and concrete. At the same time, the results  
592 predicted via typical way, i.e., two-phase model, was shown to reveal a stiffer material compared to  
593 that of both experiment results and three-phase model for both mortar and concrete. The effect of ITZ  
594 was higher for concrete compared to mortar due to the high amount of aggregates content.

595

596 Overall, the proposed novel three-phase models for mortar and concrete can be used to accurately  
597 predict the mechanical properties. One of the marked merits of this model is, if the chemical  
598 composition is known, the Young's modulus and Poisson's ratio can be computed, which would  
599 reduce the waste of time, cost, material and manpower compared to typical approaches.

600

#### 601 **Declaration of Competing Interest**

602 The authors declare that they have no known competing financial interests or personal relationships  
603 that could have appeared to influence that work reported in this paper.

604

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759

760 **Appendix A: Material and experimental methodology**

761

762 Material

763 The cement used for experimental works of mortar was ordinary Portland cement (OPC). The  
764 chemical composition and physical properties of the cement are detailed in **Table A1**. Fine aggregates  
765 used were natural mountain sand with specific gravity of 2.57 g/cm<sup>3</sup>, obtained from Kakegawa city  
766 in Shizuoka prefecture of Japan.

767 **Table A1.** Chemical composition and physical properties of cement

	SiO <sub>2</sub>	20.89
	Al <sub>2</sub> O <sub>3</sub>	5.44
	Fe <sub>2</sub> O <sub>3</sub>	2.94
	CaO	65.11
Chemical composition (%)	MgO	1.54
	SO <sub>3</sub>	2.08
	Na <sub>2</sub> O	0.27
	K <sub>2</sub> O	0.53
	TiO <sub>2</sub>	0.26

	P <sub>2</sub> O <sub>5</sub>	0.14
	MnO	0.05
	LOI	0.71
Physical properties	Blaine (cm <sup>2</sup> /g)	3450
	Density (g/cm <sup>3</sup> )	3.16

768

769 Mix design

770 The experiments were carried out for mortar at three different w/c: 0.3, 0.4 and 0.5 and two different  
771 sand contents (in total volume fraction): 0.3 and 0.6. The mix proportions for each of these mixes are  
772 stated in **Table A2**.

773

774 **Table A2.** Mixing proportion of mortar

Water (g)	Cement (g)	Sand (g)	w/c	Volume fraction of sand
1023	3408	2322	0.3	0.3
585	1947	4644		0.6
1173	2931	2322	0.4	0.3
669	1674	4644		0.6
1287	2571	2322	0.5	0.3
735	1470	4644		0.6

775

776 Sample preparation

777 According to mix proportion of mortar,  $\phi$ 50 x 100 mm cylinder samples were prepared. After  
778 demolding, the samples were sealed and cured in an environment of 20 °C for 3, 7 and 28 days.

779 Young's modulus test

780 The Young's modulus was tested in accordance with JIS A 1149. The Young's modulus was the  
781 average value of test results of three samples.

782 Poisson's ratio test

783 The Poisson's ratio experiment was carried out in accordance with JHS 307, and the Poisson's ratio  
784 value was the average value of three samples.

785 Experiment results

786 The experiment results of Young's modulus and Poisson's ratio are shown in **Table A3** for three sets  
 787 of w/c and two sets of sand content.

788

789

790

**Table A3.** Mechanical properties of mortar

W/C	Sand (volume %)	Time (Days)	Young's modulus (kN/mm <sup>2</sup> )	Poisson's ratio
0.3	30	3	-	0.218
		7	-	0.209
		28	-	0.203
	60	3	-	0.206
		7	-	0.183
		28	-	0.192
0.4	30	3	15.6	0.206
		7	18.7	0.197
		28	26.4	0.192
	60	3	20.0	0.216
		7	23.4	0.187
		28	25.5	0.186
0.5	30	3	9.56	0.179
		7	13.9	0.189
		28	16.6	0.212
	60	3	17.0	0.170
		7	21.1	0.176
		28	23.2	0.174

