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Title	Retention of water in subducted slabs under core-mantle boundary conditions
Author(s)	Tsutsumi, Yutaro; Sakamoto, Naoya; Hirose, Kei et al.
Citation	Nature Geoscience, 17(7), 697-704 https://doi.org/10.1038/s41561-024-01464-8
Issue Date	2024-07
Doc URL	https://hdl.handle.net/2115/94015
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Type	journal article
File Information	41819_1_merged_1715273080.pdf



Retention of water in subducted slabs under core-mantle boundary conditions

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The hydrated SiO₂ phase is a main carrier of water in subducting slabs in the lower mantle. Assuming its dehydration under high temperatures above the core-mantle boundary, it has been speculated that seismic anomalies observed in this enigmatic region and the uppermost core might be attributable to water released from slabs. Here we report melting experiments on a hydrous basalt up to conditions of the core-mantle boundary region at 25–144 gigapascals and 2,900–4,100 kelvin. Secondary ion mass spectrometry measurements with high-resolution imaging techniques reveal that the SiO₂ phase and SiO₂-AlOOH solid solution contain 0.5–3.6 wt% and ~3.5 wt% H₂O, respectively, coexisting with melts holding 0.9–2.6 wt% H₂O. The

large solubility into SiO₂ and high SiO₂/melt partition coefficient of water under high temperatures of the core-mantle boundary region suggest that practically water does not escape from subducted slabs at the base of the mantle. Even if the core-mantle boundary temperature is high enough to melt subducted crustal materials, most of the H₂O would remain in the solid residue rather than entering a partial melt. Previously proposed consequences of slab dehydration are therefore unlikely to be responsible for chemical heterogeneities in the lowermost mantle and the topmost core.

The carriers of water in cold subducting slabs (former oceanic lithosphere) have been explored by experiments and theory at high pressures^{1,2}. While the slab's crustal layer may almost completely dehydrate at shallower than ~300 km depth³, the hydrated slab mantle^{4,5} can transport H₂O by hydrous phases down to the uppermost lower mantle (~700 km depth) or at most to the mid-lower mantle (~1,500 km depth)⁶⁻⁹ depending on slab temperatures (see red stars in [Fig. 1](#))^{1,2}. Beyond these depths, part of the released water will dissolve into surrounding, nominally anhydrous minerals of bridgmanite and ferropericlasite in the depleted peridotite layer of the slab, whose water capacity has been reported to be <0.01 wt%^{10,11} to 0.1 wt% level^{12,13}. It will also migrate into an overlying crustal layer of the slab and hydrate the dense SiO₂ phase^{2,14-18} in particular, which is a

major constituent of subducting crustal materials^{19–22}. Recent experiments performed along the cold to normal lower-mantle geotherm (Fig. 1) demonstrated that SiO₂ can hold up to ~3.5^{17,18} or ~10 wt% H₂O^{15,16}. It suggests that the hydrated SiO₂ phase, rather than AlOOH-Mg_{0.5}Si_{0.5}OOH-FeOOH solid solution (ss.)^{1,23–25}, is likely to be a primary host for water in the slab at greater depths (>700–1,500 km) to the bottom of the mantle^{2,14–17}.

Regardless of the high capacity of water in dense SiO₂, it has been speculated^{1,2} and recently demonstrated by experiments on the SiO₂-H₂O system¹⁷ that the subducted slabs will dehydrate under high temperatures in the core-mantle boundary (CMB) region. The temperature at the CMB has been estimated to be 3,400 K^{26,27} to ~4,000 K^{28,29} considering the solidus temperature of a pyrolitic lowermost mantle or 3,470–3,880 K based on seismic shear velocity and attenuation³⁰. The released water will give rise to chemical heterogeneities indicated by seismological observations. Water can induce melting and melt migration, and hydration of lowermost mantle materials which might create the large low shear velocity provinces (LLSVPs)^{1,31}. Furthermore, the reaction of water with core metal may form superoxidized FeOOH_x (x<1)^{32–34} that could account for the ultralow velocity zones (ULVZs)³⁵. Also, it might develop a low-velocity layer in the topmost core^{36,37}.

Nevertheless, the solubility and partitioning of water into Al-bearing hydrous SiO₂ included in subducted former crustal rocks, especially mid-oceanic ridge basalt (MORB), have not been examined at high pressures and temperatures (*P-T*) of the CMB region, and therefore the dehydration of subducted slabs and their consequences have not been verified yet.

Water in Al-bearing SiO₂ at CMB

We have conducted 19 separate melting experiments on a hydrous MORB up to the lowermost mantle pressure range and 4,100 K in a laser-heated diamond-anvil cell (Fig. 1 and Table 1, see Methods). Melting textures and chemical compositions including the H₂O contents in melt and coexisting solids were examined on recovered samples (Fig. 2). They exhibit a concentric texture; a round pocket of quenched partial melt at the center, surrounded by the SiO₂ phase, CaSiO₃ perovskite (davemaoite), SiO₂-AlOOH ss.⁷ and bridgmanite at 25–59 GPa (Fig. 2a). At 100 GPa and above, SiO₂ was still present next to a melt pool, and bridgmanite/post-perovskite appeared closer to melt than at lower pressures (Fig. 2b). Such zonation represents the sequence of crystallization with decreasing temperature^{38,39} and is different from that of an anhydrous MORB material^{40,41} because of the addition of water and the lesser amount of CaO in the present sample. Synchrotron X-ray diffraction (XRD) patterns of these samples indicate that CaCl₂-type

SiO₂ appeared down to at least 25 GPa at 2,900 K (22 GPa at 300 K), consistent with earlier experimental⁴² and theoretical⁴³ findings that CaCl₂-type SiO₂ is stabilized with respect to stishovite in the presence of Al₂O₃ and H₂O, while it forms above 65–75 GPa at ~1,500–2,000 K in Al-free, dry SiO₂⁴⁴ and SiO₂-H₂O systems¹⁷. They also show a change in the crystal structure of SiO₂ from CaCl₂-type to α -PbO₂-type (seifertite)¹⁹ above 128 GPa in the lowermost mantle (Fig. 1). The coexistence of SiO₂ and SiO₂-AlOOH ss. phases (Fig. 2a and Extended Data Fig. 1) suggests a miscibility gap in the SiO₂–AlOOH system as predicted by earlier theoretical calculations⁷.

Water concentrations in the SiO₂ phase, SiO₂-AlOOH ss. and partial melts were determined by secondary ion mass spectrometry (SIMS) coupled with high-resolution imaging techniques (Fig. 2 and Extended Data Fig. 2) (see Methods), while in earlier studies the H₂O content in SiO₂ was estimated from Fourier-transform infrared absorption spectra^{14,18}, or volume expansion based on their relationship that was observed at 1 bar^{15,16} or theoretically calculated at high pressures¹⁷. The present SIMS analyses show the hydrogen (water) abundances with ± 1 to 10% fractional uncertainty; i) 0.5–3.6 wt% and 1.7–2.2 wt% H₂O in hydrated SiO₂ with the CaCl₂- and α -PbO₂-type structures, respectively, ii) 3.4–3.5 wt% H₂O in SiO₂-AlOOH ss. and iii) 0.9–2.6 wt% H₂O in partial melts (Table 1).

The water content in Al-bearing CaCl₂-type SiO₂ substantially increased with increasing pressure to 3.6 wt% H₂O at 121 GPa under 3,450 K (Figs. 3a, b). It is much higher than the recent H₂O solubility estimate of only 0.4 wt% at this *P-T* condition based on experiments on the Al-free SiO₂-H₂O system¹⁷. Such contradiction suggests that the charge-coupled substitution $\text{Si}^{4+} = \text{Al}^{3+} + \text{H}^{+}$ is a key mechanism for water incorporation into Al-bearing CaCl₂-¹⁸ and α -PbO₂-type SiO₂ (Fig. 3c), in contrast to interstitial incorporation of molecular H₂O into Al-free SiO₂¹⁷. Our XRD study finds that the magnitude of orthorhombic distortion was enhanced with increasing pressure (Fig. 3d) and such crystallographic distortion correlates strongly with its water content (Fig. 3e). The enhanced distortion shortens the distance between specific pairs of oxygen atoms, making hydrogen bonds stronger and stabilizing hydrogen in SiO₂^{43,45}. Moreover, previous molecular dynamics simulations⁴³ of hydrated SiO₂ showed superionic proton motion at 60 GPa and >1,300 K, hopping around possible sites with diffusion along the *c*-axis. Such superionic proton motion could lead to the entropic stabilization of the hydrated SiO₂ phase to high temperatures. The overall increase in water concentration in CaCl₂-type SiO₂ with increasing pressure results in an increase in the partition coefficient of water, $D_{\text{H}_2\text{O}}$ (CaCl₂-SiO₂/melt) from ~0.2 at 25 GPa to 1.4 under deep lower-mantle conditions above 120 GPa (Fig. 3f).

The H₂O contents in Al-bearing α -PbO₂-type SiO₂ formed at 3,650–4,100 K and 128–144 GPa, corresponding to conditions of the Earth’s CMB region, were almost constant at 2 wt% (Fig. 3a). They are substantially higher than recently reported in the SiO₂-H₂O system (0.3–1.0 wt% H₂O)¹⁷ at 2,820–4,100 K and 142–152 GPa, likely because of the Si⁴⁺ = Al³⁺ + H⁺ charge-coupled substitution in α -PbO₂-type SiO₂ as well (Fig. 3c). $D_{\text{H}_2\text{O}}$ (α -PbO₂-SiO₂/melt) ranged from 1.1 to 2.1 (Fig. 3f), indicating that water is partitioned into hydrated SiO₂ more than into coexisting partial melts under CMB conditions. The nearly constant ~2 wt% H₂O in α -PbO₂-type SiO₂ found in all of six separate runs, while water concentrations in coexisting melts varied by a factor of two, might suggest that ~2 wt% is the maximum solubility of water and $D_{\text{H}_2\text{O}}$ (α -PbO₂-SiO₂/melt) could be higher when the SiO₂ phase is not saturated with water. While it was not found in our melting experiments, hydrous Al-rich hexagonal niccolite (hNT)-type SiO₂¹⁶ was previously reported above 84 GPa⁴⁶. It is similar in composition to CaCl₂-structured SiO₂-AlOOH ss. observed in this study and might appear in a hydrated MORB material coexisting with α -PbO₂-type SiO₂ under high P - T conditions of the CMB region.

Escape of Water from Subducted Slabs at CMB?

It has been considered¹ that the high-temperature stability limit for hydrous phases in the slab at lower-mantle depths is set by the dehydration curve of δ -AlOOH⁴⁷, which is close to the normal mantle geotherm (Fig. 1). While FeOOH_x was reported³³ to be stable to 3,300 K at 125 GPa, it should be a minor phase in the slab^{48,49}. Recent experiments^{17,50} showed that CaCl₂-type SiO₂, instead of the α -PbO₂-type phase, is stable at CMB conditions in the Al-free SiO₂-H₂O system and contains no water above $\sim$ 3,600 K. Contrary to these studies suggesting nearly complete dehydration of the slab at the base of the mantle, our experiments demonstrate that Al-bearing SiO₂ in subducted MORB adopts the α -PbO₂-type structure in the lowermost mantle and can possess $\sim$ 2 wt% H₂O even at 4,100 K (Figs. 3a, b), which corresponds to the temperature for the core side of the CMB (Fig. 1). The pressure of the phase transition between the CaCl₂- and α -PbO₂-type structures is found to be similar between dry¹⁹ and hydrous MORB materials.

As argued above, the hydrated SiO₂ phase is a main host for water transport by subducting slabs in the shallow to deep lower mantle^{2,14,17}, unless the crustal layer is delaminated at the 660 km discontinuity⁵¹. It constitutes 15–30% of MORB crust^{19,20} and 20–75% of continental crust materials^{21,22} and changes crystal structure to α -PbO₂-type in the lowermost mantle (Fig. 1). Once slabs reach the bottom of the mantle, they are heated to 3,400–4,000 K^{26–30}. The present experiments demonstrate that Al-bearing α -

PbO₂-type SiO₂ in the slab crust does not release water even at such high temperatures as long as it carries H₂O less than ~2 wt% (Figs. 3a, b). Indeed, the amount of H₂O in SiO₂ during subduction in the lower mantle is most likely smaller than 2 wt% considering its water storage capacity of ~3.5 to ~1.0 wt% in the shallow to deep lower mantle as recently estimated by Lin & Mao⁵⁰. Even when it is more than 2 wt%, hNT-type SiO₂-AlOOH ss.⁴⁶ might form above the CMB and admit water that is not stored in SiO₂. Additional Al₂O₃ necessary to form SiO₂-AlOOH ss. could derive from coexisting bridgmanite/post-perovskite in MORB¹⁸. Furthermore, an Al- and/or Fe-rich part of the slab crust may include AlOOH-Mg_{0.5}Si_{0.5}OOH-FeOOH ss.^{1,23-25} as an additional water-bearing phase. It will decompose¹ at least at temperatures along the normal lower-mantle geotherm before arriving at the CMB, and instead SiO₂-AlOOH ss. might be produced.

The topmost 2 to 5 km^{4,5} in average of the slab mantle is hydrated at the onset of subduction and transports water into the deep mantle. When hydrous phases no longer exist at shallow to middle lower mantle depths¹, bridgmanite and ferropericlase in the uppermost slab mantle become water-saturated² (davemaoite will not appear in a depleted peridotite layer). The solubility of water in these nominally anhydrous minerals has been highly controversial but could be as much as ~0.1 wt%^{12,13}. When slabs reach the lowermost mantle under high temperatures, the H₂O storage capacity of bridgmanite and

ferropericlasite will diminish, leading to the exsolution of water from the topmost slab mantle, which turns into hydrous silicate melts. Even so, as soon as such hydrous melts migrate into the neighboring slab crust that is originally 6–7 km thick, all of the H₂O in hydrous melts would redissolve into the SiO₂ phase, fully consuming the silicate melts until SiO₂ becomes saturated with water. Note that both the slab crust and mantle layers have the post-perovskite-dominant lithology in the lowermost mantle^{52,53} and are thus easily deformed^{54,55}. The stretched and thinned slab mantle layer may be intercalated between the slab crust layers, and the hydrous melts can meet the SiO₂ phase by migrating either upward or downward.

Limited Role of Water in the Lowermost Mantle

The dehydration of subducted slabs at the base of the mantle has been stressed, and its consequences have been extensively discussed^{1,2,34,37,50}. However, our finding of the high capacity of H₂O (~2 wt% in α -PbO₂-type SiO₂ and ~0.3–0.6 wt% in subducted MORB crust) suggests that practically water does not escape from slabs under high temperatures of the CMB region. Therefore, hydration around subducted slabs¹ will not occur and thus not contribute to the formation of low velocity anomalies observed in the LLSVPs^{1,31}. The ULVZs have often been attributed to the presence of silicate melt above the CMB³⁵,

but the origin of such molten silicate is not likely caused by water. Indeed, the addition of water to MORB does not induce partial melting until α -PbO₂-type SiO₂ becomes H₂O-saturated. It is possible that the CMB temperature is high enough to cause partial melting of subducted crustal materials⁴⁰. Even in this case, water will be largely retained in a solid residue rather than being incorporated into a partial melt; employing $D_{\text{H}_2\text{O}}(\alpha\text{-PbO}_2\text{-SiO}_2/\text{melt}) = 2.1$ obtained at 128 GPa and 3,850 K close to CMB conditions and the phase proportions found in run #12 (Table 1), we found 73% of H₂O remains in the residue when the degree of partial melting is 31 wt%.

The accumulation of water-bearing slabs may have large impacts on viscosity, electrical conductivity, and seismic velocities and attenuation in the lowermost mantle. The incorporation of water largely diminishes the hardness of crystals such as SiO₂ quartz⁵⁶. The water-bearing, less viscous slabs might have weaker interactions with LLSVPs, which helps the stability of their structures⁵⁷. The hydrated subducted slabs may cause high electrical conductivity anomalies in the deep lower mantle. The anomalies will be significant when superionic proton motion occurs in Al-bearing hydrous SiO₂, which was found in earlier theoretical simulations⁴³ under mid-lower mantle P - T conditions. Seismic velocities of dense dry SiO₂ phases are high compared to those of typical lower-mantle minerals⁵⁸. The incorporation of Al + H in SiO₂ usually decreases the shear and

compressional velocities but increases in the lowermost mantle because of the effect of ferroelastic-type phase transition between stishovite and CaCl_2 -type phase⁵⁹, enhancing high velocity anomalies caused by subducted slabs above the CMB⁵⁸.

It has also been proposed³⁴ that the reaction of water with core metal generates oxygen-rich patches (ORP) at the CMB. They consist mainly of FeOOH_x ($x < 1$)³² formed by superoxidation of iron by water, which exhibits low shear and compressional wave velocities³³ that could account for the ULVZs and exceedingly high electrical conductivity^{48,49}. In addition, recent experiments³⁷ argued the formation of a hydrogen-rich, silicon-deficient layer at the top of the core as a consequence of the reaction between water and Fe-Si core liquid, which can cause low velocity anomalies³⁶. However, the water-metal reaction at the CMB should be minimal because practically water does not escape from subducted slabs. The reaction is restricted only at a direct contact between water-bearing SiO_2 crystals and core metal. The hydrated slabs are eventually recycled by being involved in plumes without leaving water in the lowermost mantle. During upwelling, Al-bearing SiO_2 will release water in the shallow lower mantle and the mantle transition zone and finally almost fully dehydrate around 20 GPa where the orthorhombic distortion of the CaCl_2 -type structure disappears (Figs. 3d, e).

Acknowledgements The authors acknowledge K. Yonemitsu for assisting FIB and EPMA analyses and G. Helffrich for discussion. Comments by W. Panero on an earlier version of the manuscript were helpful. Synchrotron XRD measurements were carried out at BL10XU, SPring-8 (proposals no. 2020A0072, 2020A0066, 2021A0072 and 2021A1481). This work was supported by JSPS Kakenhi to K.H. and H.Y. and by the “Imaging Platform” program by MEXT.

Author contributions The project was designed by K.H. and led by Y.T. SIMS analyses were performed by N.S., H.Y., S.T. and Y.T. XRD data were collected by Y.T. and Y.O. Y.T., K.H. and K.U. wrote the manuscript, and all authors commented on it.

Competing interests The authors declare no competing interests.

Additional information

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Table 1 | Chemical compositions and water concentrations in hydrated SiO₂, SiO₂-AlOOH ss. and melt

	#1		#2		#3		#4		#5	
Pressure (GPa)	25 (3)		29 (3)		31 (3)		41 (4)		42 (4)	
Temperature (K)	2900 (150)		3050 (150)		3300 (170)		3000 (150)		2950 (160)	
Duration (sec)	10		4		10		10		6	
Phase	CaCl ₂ -SiO ₂	melt	CaCl ₂ -SiO ₂	melt	CaCl ₂ -SiO ₂	melt	CaCl ₂ -SiO ₂	melt	CaCl ₂ -SiO ₂	melt
FE-EPMA (wt%)										
SiO ₂	87.49 (26)	38.22 (52)	86.74 (47)	41.79 (80)	86.29 (137)	37.95 (67)	87.06 (137)	41.54 (136)	84.26 (278)	38.18 (92)
TiO ₂	0.06 (3)	2.43 (10)	0.03 (2)	1.15 (4)	0.06 (2)	1.53 (7)	0.10 (0)	2.21 (28)	0.00 (0)	1.36 (4)
Al ₂ O ₃	8.72 (80)	9.88 (97)	7.37 (55)	7.78 (35)	7.08 (58)	10.19 (8)	8.80 (52)	8.20 (55)	6.82 (94)	8.08 (1)
FeO	0.22 (5)	15.91 (61)	0.16 (10)	15.71 (47)	0.15 (2)	16.98 (51)	0.13 (4)	7.59 (271)	0.25 (0)	22.32 (135)
MgO	0.20 (1)	10.57 (111)	0.17 (8)	7.81 (52)	0.04 (1)	9.85 (4)	0.07 (0)	8.74 (54)	0.10 (7)	7.39 (6)
CaO	0.07 (3)	7.18 (63)	0.14 (9)	6.16 (38)	0.05 (0)	5.33 (18)	0.07 (0)	6.81 (61)	0.13 (9)	5.15 (7)
Na ₂ O	0.07 (9)	0.40 (7)	0.35 (22)	1.25 (11)	0.11 (9)	0.58 (11)	0.04 (4)	1.16 (51)	0.02 (1)	1.05 (1)
K ₂ O	0.01 (1)	0.13 (2)	0.01 (1)	0.25 (3)	0.01 (1)	0.12 (1)	0.02 (0)	0.26 (6)	0.00 (0)	0.35 (5)
C	0.50 (65)	1.01 (45)	1.12 (30)	0.74 (17)	1.03 (57)	0.79 (16)	0.71 (38)	0.70 (26)	1.48 (4)	0.45 (6)
SIMS (wt%)										
H ₂ O	0.51 (2)	1.94 (3)	0.82 (4)	2.60 (6)	0.76 (3)	1.93 (5)	1.27 (5)	2.17 (8)	0.79 (4)	0.99 (5)
Total	97.86	87.66	96.91	85.25	95.57	85.26	98.28	79.39	93.86	85.31

Run	#6		#7		#8		#9		#10	
Pressure (GPa)	42 (4)		50 (5)		51 (5)		58 (6)		59 (6)	
Temperature (K)	3400 (170)		3450 (170)		3350 (170)		3250 (160)		3400 (170)	
Duration (sec)	10		1		30		10		10	
Phase	CaCl ₂ -SiO ₂	melt	CaCl ₂ -SiO ₂	melt	CaCl ₂ -SiO ₂	CaCl ₂ -SiO ₂	SiO ₂ -AlOOH	melt	CaCl ₂ -SiO ₂	SiO ₂ -AlOOH
FE-EPMA (wt%)										
SiO ₂	76.48 (265)	28.09 (102)	75.65 (168)	35.48 (26)	86.14 (217)	84.14	60.51 (222)	54.02	87.63 (14)	59.10 (552)
TiO ₂	0.06 (6)	2.47 (17)	0.32 (8)	1.69 (9)	0.07 (2)	0.10	0.11 (4)	1.92	0.02 (1)	0.09 (8)
Al ₂ O ₃	11.10 (58)	8.78 (28)	9.60 (19)	13.46 (29)	7.07 (44)	7.63	21.97 (130)	8.61	8.05 (22)	18.12 (267)
FeO	0.51 (16)	15.77 (87)	2.42 (29)	13.22 (31)	0.13 (5)	0.27	0.91 (11)	7.41	0.19 (8)	0.40 (7)
MgO	0.52 (12)	8.43 (7)	2.02 (30)	11.39 (25)	0.06 (0)	0.10	1.69 (28)	7.83	0.10 (3)	1.09 (34)
CaO	0.87 (26)	6.06 (23)	3.04 (18)	6.92 (15)	0.07 (5)	0.11	1.09 (47)	5.87	0.17 (3)	4.82 (163)
Na ₂ O	0.88 (46)	0.84 (8)	0.49 (6)	1.18 (17)	0.02 (2)	0.17	1.98 (92)	1.39	0.04 (1)	0.25 (7)
K ₂ O	0.01 (1)	0.21 (1)	0.03 (1)	0.09 (3)	0.01 (1)	0.00	0.06 (4)	0.07	0.01 (1)	0.07 (0)
C	0.87 (12)	1.97 (16)	1.41 (57)	0.90 (18)	0.57 (47)	1.69	1.07 (15)	1.38	1.33 (6)	0.63 (32)
SIMS (wt%)										
H ₂ O	1.51 (14)	1.73 (7)	1.13 (4)	1.74 (4)	1.18 (4)	1.40 (7)	3.38 (14)	2.13 (8)	1.85 (6)	3.48 (35)
Total	92.81	74.36	96.10	86.07	95.31	95.60	92.78	90.64	99.41	88.05

Run	#11		#12*		#13		#14		
Pressure (GPa)	100 (10)		103 (10)		121 (12)		128 (13)		
Temperature (K)	3500 (180)		3950 (200)		3450 (170)		3850 (190)		
Duration (sec)	5		5		4		4		
Phase	CaCl ₂ -SiO ₂	CaCl ₂ -SiO ₂	Bdg [†]	Ca-pv [†]	melt	CaCl ₂ -SiO ₂	melt	α-PbO ₂ -SiO ₂	melt
FE-EPMA (wt%)									
SiO ₂	79.09 (289)	79.29 (214)	46.84 (142)	49.11 (230)	21.73 (97)	78.75 (481)	28.54 (93)	74.61 (230)	22.84 (101)
TiO ₂	0.03 (1)	0.10 (4)	1.15 (8)	0.49 (8)	3.45 (6)	0.04 (3)	3.23 (4)	0.01 (1)	3.69 (15)
Al ₂ O ₃	13.07 (14)	12.46 (76)	18.23 (38)	2.92 (40)	15.10 (71)	7.60 (46)	17.48 (27)	10.89 (35)	8.20 (48)
FeO	0.22 (9)	0.33 (17)	5.60 (22)	2.20 (16)	30.05 (85)	0.29 (6)	25.52 (240)	0.09 (3)	27.13 (695)
MgO	0.42 (34)	2.18 (138)	26.27 (126)	2.52 (49)	12.59 (72)	0.50 (21)	10.50 (29)	0.35 (7)	12.29 (104)
CaO	0.28 (24)	0.59 (31)	0.81 (17)	34.98 (180)	3.26 (8)	4.91 (237)	3.11 (2)	2.42 (35)	7.21 (105)
Na ₂ O	0.10 (6)	0.25 (8)	0.18 (16)	1.72 (31)	0.12 (4)	0.25 (11)	0.18 (4)	0.02 (2)	1.06 (46)
K ₂ O	0.01 (1)	0.02 (2)	0.00 (1)	0.36 (4)	0.12 (1)	0.07 (4)	0.09 (2)	0.10 (4)	0.18 (1)
C	1.29 (10)	1.35 (9)	0.84 (32)	1.17 (46)	1.36 (10)	1.11 (23)	0.90 (22)	0.37 (67)	2.42 (7)
SIMS (wt%)									
H ₂ O	2.58 (12)	1.63 (11)			2.14 (10)	3.55 (25)	2.57 (9)	1.94 (9)	0.94 (6)
Total	97.09	98.19	99.92	95.49	89.94	97.08	92.12	90.79	85.97

Run	#15		#16		#17		#18		#19		Starting material
Pressure (GPa)	143 (14)		142 (14)		143 (14)		144 (14)		141 (14)		
Temperature (K)	3650 (180)		4100 (210)		4050 (200)		3800 (190)		3850 (190)		
Duration (sec)	4		1		10		1		10		
Phase	α-PbO ₂ -SiO ₂	melt	α-PbO ₂ -SiO ₂	melt	α-PbO ₂ -SiO ₂	melt	α-PbO ₂ -SiO ₂		α-PbO ₂ -SiO ₂		
FE-EPMA (wt%)											
SiO ₂	79.81 (126)	28.87 (200)	76.22 (26)	27.07 (62)	81.44 (11)	24.65 (44)	83.82 (99)		70.76 (1)		51.31 (91)
TiO ₂	0.02 (3)	3.01 (17)	0.11 (2)	2.89 (10)	0.02 (2)	4.15 (8)	0.08 (3)		0.07 (4)		1.24 (3)
Al ₂ O ₃	10.07 (102)	20.97 (117)	12.56 (8)	17.99 (49)	11.97 (19)	20.49 (20)	11.60 (12)		13.95 (22)		14.90 (61)
FeO	0.15 (2)	21.87 (320)	0.50 (7)	19.40 (32)	0.15 (2)	18.42 (4)	0.71 (7)		0.62 (31)		10.05 (55)
MgO	0.36 (32)	10.98 (57)	0.82 (7)	13.34 (36)	1.92 (10)	15.88 (18)	0.75 (12)		2.32 (20)		8.88 (18)
CaO	1.92 (33)	3.22 (22)	6.13 (24)	3.65 (5)	2.34 (47)	3.46 (14)	2.28 (15)		2.70 (31)		5.37 (12)
Na ₂ O	0.11 (10)	0.25 (8)	0.36 (4)	1.56 (9)	0.33 (3)	3.15 (12)	0.25 (4)		0.41 (4)		1.74 (8)
K ₂ O	0.03 (1)	0.14 (1)	-	-	-	-	-		-		0.12 (2)
C	1.29 (11)	1.07 (17)	-	-	-	-	-		-		-
SIMS (wt%)											
H ₂ O	1.99 (10)	1.84 (14)	1.74 (2)	1.32 (3)	2.17 (3)	1.49 (5)	1.97 (2)		1.84 (2)		2.66 (5)
Total	95.75	92.22	98.45	87.20	100.35	91.68	101.45		92.66		96.27

*Mass balance calculations show 31% melt, 40% SiO₂ phase, 18% Bdg, and 11% Ca-pv by weight.

[†]Obtained by subtracting the composition of coexisting SiO₂ phase by 30.1 and 23.5 wt% for Bdg and Ca-pv, respectively.

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Figure Legends

Fig. 1 | Experimental pressure-temperature conditions and lower-mantle geotherms.

Present experiments on hydrous MORB found hydrated SiO_2 phases with CaCl_2 -type (closed circles) and $\alpha\text{-PbO}_2$ -type structures (open circles) which coexisted with partial melts at temperatures much higher than normal to hot lower-mantle geotherms (yellow band)^{1,60}. Cold geotherm for subducting slabs is given by a blue band^{1,2}. Black lines show the solidus and liquidus curves for a dry MORB material⁴⁰. Melts were present even below the anhydrous solidus temperatures because the amount of H_2O added to the sample was more than the capacity of water in solid phases. The green and purple regions indicate the stabilities of phase H (ref. 9) and other hydrous phases¹, respectively, in the slab mantle. The SiO_2 phase in the slab crust is hydrated when they release water mainly at conditions marked by red stars. Recent experiments on hydrated SiO_2 were performed in simple systems; inverted open triangles¹⁶ and pluses¹⁷ in $\text{SiO}_2\text{-H}_2\text{O}$, closed triangles¹⁸ in $\text{SiO}_2\text{-Al}_2\text{O}_3\text{-H}_2\text{O}$. Black broken line shows the dehydration curve of $\delta\text{-AlOOH}$ (ref. 47). Red line indicates the phase transition boundary in MORB between the CaCl_2 - and $\alpha\text{-PbO}_2$ -type SiO_2 phases¹⁹. The present experiments demonstrate that $\alpha\text{-PbO}_2$ -type SiO_2 can hold ~ 2 wt% H_2O above melting temperatures of MORB in the CMB region. Pressure and temperature uncertainties are $\pm 10\%$ and $\pm 5\%$, respectively⁶¹.

Fig. 2 | Cross sections of partially molten samples. Samples were recovered from (a) 58 GPa and 3,250 K (run #9) and (b) 121 GPa and 3,450 K (run #13). (uppermost panels) EDS X-ray elemental maps combined for Si, Al, Ca and Mg, showing melt, hydrated SiO_2 (red), CaSiO_3 perovskite (Ca-pv, blue), $\text{SiO}_2\text{-AlOOH}$ ss. (Si-Al, light blue), bridgmanite (Bdg, orange) and the dehydrated (sm') and original starting material (sm). (middle panels) Corresponding secondary ion images for $^{28}\text{Si}^+$ and $^1\text{H}^+$ and the distribution maps of H_2O . These cross sections are slightly different from those for X-ray maps because of repolishing. As a result, the phase boundaries are slightly sifted between them. R. I., relative intensity normalized to that from sm. (lowermost panels) Line profiles of H_2O concentration along the lines A–B and C–D in the middle panels. In b, the relatively low water content in an outer melt portion was likely caused by redistribution of water upon the formation of quench crystals during quenching temperature (note that the melt is depleted in SiO_2 and has thus low viscosity). Vertical dashed lines are tie-lines between panels connecting to phase boundaries on the lines A–B and C–D. Scale bars, 10 μm .

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287 **Fig. 3 | Variations in hydrated SiO₂.** The present experiments on hydrous MORB show
 288 data for CaCl₂-type (closed circles) and α-PbO₂-type SiO₂ (open circles). Earlier data
 289 obtained in water-saturated, SiO₂-H₂O (pluses¹⁷) and SiO₂-Al₂O₃-H₂O systems
 290 (triangles¹⁸) are also given. Color indicates temperature. **a, b**, Water concentrations in
 291 SiO₂ plotted as functions of pressure and temperature, respectively. The water
 292 incorporation mechanism (Al + H coupled substitution) for Al-bearing SiO₂ (ref. 18, this
 293 study) is different from the interstitial H₂O incorporation into Al-free SiO₂¹⁷, leading to
 294 lower solubility at relatively low pressures but higher solubility under deep lower-mantle
 295 pressures. **c**, Correlation between Al and H concentrations in SiO₂, indicating the charge-
 296 coupled substitution $\text{Si}^{4+} = \text{Al}^{3+} + \text{H}^+$ is a main mechanism for water incorporation (the
 297 broken line indicates 1:1 correlation). **d**, The b/a axial ratio of the CaCl₂-type structure
 298 observed at 300 K, representing the magnitude of the orthorhombic distortion from the
 299 tetragonal structure with $b/a = 1$. Present experimental results show the modest effect of
 300 H₂O compared to theoretical calculations⁴³ at 0 K for SiO₂+1.85mol%AlOOH (red, 0.28
 301 wt% H₂O) and SiO₂+6.25mol%AlOOH (blue, 0.94 wt% H₂O). Since the b/a axial ratios
 302 were measured at room temperature, the pressures in **d** are different from those at high
 303 temperatures in **a** and **f**. **e**, Correlation between the H₂O content and the b/a ratio. **f**,
 304 SiO₂/melt partition coefficient of H₂O, demonstrating that water is partitioned into SiO₂
 305 more than into silicate melt in the lowermost mantle. The error bars for pressure and
 306 temperature are ±10% and ±5%, respectively⁶¹. Other data are presented as mean values
 307 +/- SD (+/- SEM only for the H₂O (H) content from secondary ion image).

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Methods

Starting material and high pressure-temperature experiments

We used a hydrous glass containing 2.66(5) wt% H₂O as a starting material, which was synthesized at 1.0 GPa and 1,573 K in a piston-cylinder apparatus. Its chemical composition is similar to that of a natural MORB⁶², except that the present sample includes water and less than half CaO content (Table 1) such that the SiO₂ phase appears in direct contact with partial melt (Fig. 2). High P - T conditions were generated in a laser-heated diamond-anvil cell with flat 300 μ m or beveled 120 μ m culet diamond anvils. A rhenium gasket was preindented to $\sim$ 30 μ m thick and was drilled a $\sim$ 40–120 μ m hole for a sample chamber. We loaded the hydrous MORB glass sample without pressure medium; nevertheless, sample temperature near the diamond anvils should have been low during laser-heating, and thus the glass sample remained amorphous and acted as a good thermal insulator²⁸. The sample was compressed at room temperature and then heated from both sides using a couple of 100 W single-mode Yb fiber lasers at the beamline BL10XU, SPring-8 except for runs #7 and #16–19 in which heating was made at the University of Tokyo with a similar system⁶³. The laser spot size was about 30 μ m. Temperature was measured by a spectro-radiometric method⁶⁴. Heating duration ranged from 1 to 30 s (typically 10 s). The H₂O contents in SiO₂ and melt and the resulting $D_{\text{H}_2\text{O}}$ found in run #7 with heating for 1 s at 50 GPa and 3,450 K are in good agreement with the results of other runs with longer heating durations at similar P - T range (Figs. 3a, f). Furthermore, we obtained similar results in runs #17 and #18 performed at nearly identical P - T conditions ($\sim$ 142 GPa and $\sim$ 4,100 K) with changing heating duration from 1 s to 10 s. These time-series experiments indicate that 1 s is long enough for the attainment of equilibrium water partitioning. Pressure was measured on the basis of the Raman shift of the diamond anvil⁶⁵ at 300 K after heating and corrected for thermal pressure contributions⁶⁶. We collected XRD patterns of the sample at high pressures using synchrotron X-rays with an energy of $\sim$ 30 keV. The lattice parameters and unit-cell volumes of the hydrated SiO₂ phases are shown in Extended Data Table 1 and Extended Data Fig. 3.

Subsequently we released pressure and prepared the sample cross sections for most of the experiments (runs #1–15) at the center of a heated spot parallel to the compression/laser-heating axis by using a focused ion beam (FIB) (FEI, Versa 3D DualBeam). We performed textural and compositional characterizations on these cross sections with a field-emission (FE)-type scanning electron microscope (SEM) and an energy dispersive X-ray spectrometry (EDS) (Fig. 2). After SIMS measurements (see

below), the major element compositions except H₂O of coexisting quenched melt and solid phases were determined using an FE-type electron probe micro-analyzer (FE-EPMA) (*JEOL*, JXA-8530F) (Table 1). Analyses were performed with an acceleration voltage of 12 keV and a beam current of 15 nA. We used SiO₂, TiO₂, Al₂O₃, Fe, MgO, CaSiO₃, NaAlSi₃O₈, KTiOPO₄ and Fe₃C as standards and PETJ (Si, Ti, K), TAP (Al, Mg, Na), LIFH (Fe), PETH (Ca) and LDE2H (C) as analyzing crystals. For runs #16–19, the samples were milled by ion beam in a SIMS instrument such that a wider area of SiO₂ was exposed on the surface in a plane normal to the compression/laser-heating axis. We obtained the major element compositions by EDS with a silicon drift detector (*Oxford*, X-Max 150) for these samples, which is attached with an FE-SEM (*JEOL*, JSM-7000F). The EDS measurements were performed with an acceleration voltage of 15 keV and a beam current of 1 nA.

SIMS analyses

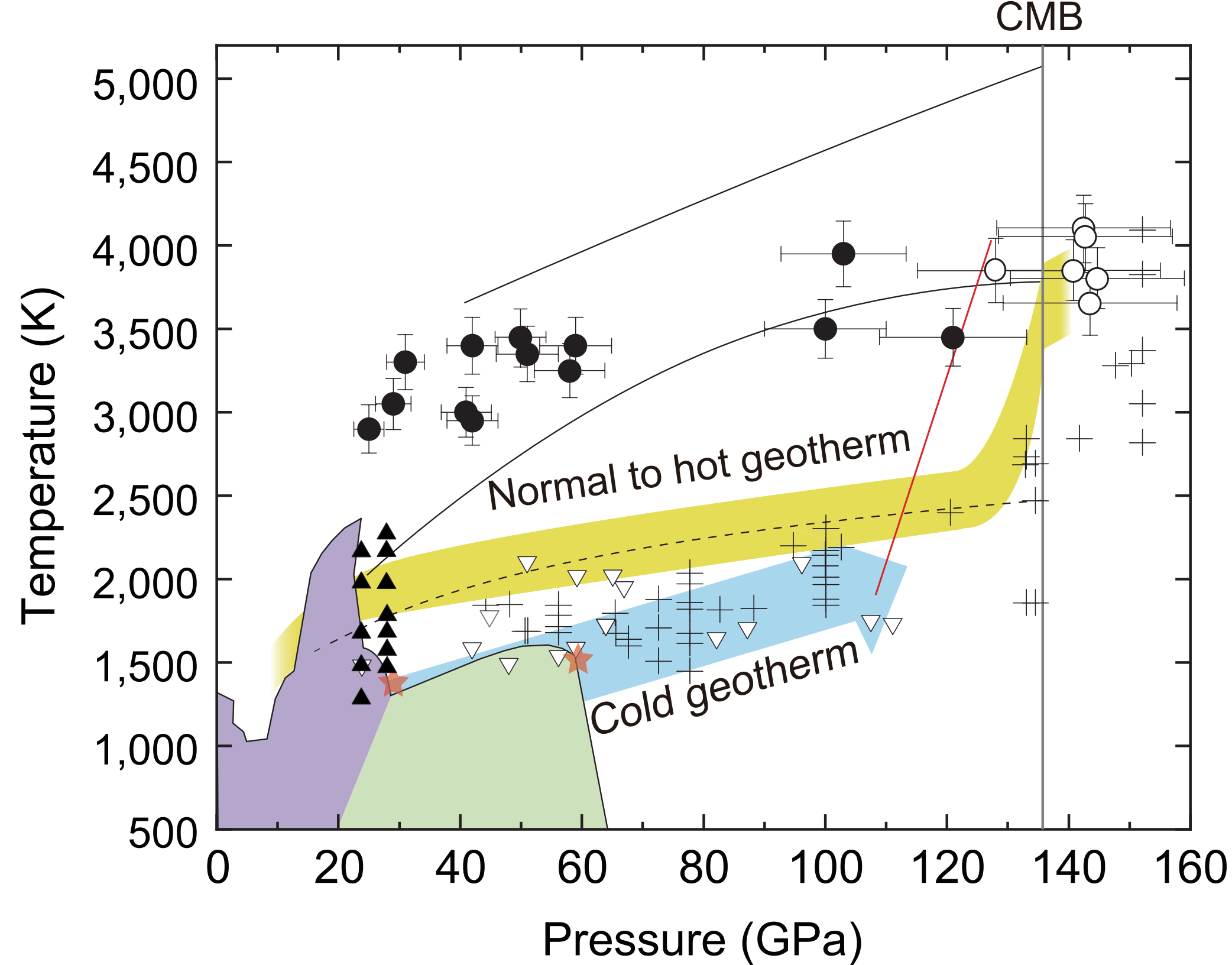
Water concentrations in SiO₂, SiO₂-AlOOH ss. and partial melt were obtained by an isotope microscope system, composed of *CAMECA* ims-1270 SIMS equipped with a two-dimensional ion detector, SCAPS (stacked CMOS-type active pixel sensor), at Hokkaido University^{61,67–69}. The sample was coated with ~70 nm thick Au layer by plasma sputtering to compensate electrostatic charging on the sample surface. We used ¹⁶O- primary beam (13 keV, 37 nA) that was focused to 20–30 μm in diameter and rastered across a 100 μm × 100 μm region on the sample surface. A contrast aperture was set to be 100 μm in diameter. In order to eliminate contributions of adsorbed water on the sample surface⁷⁰, we employed the energy slit to be 20 eV and analyzed by -100 V offsetting the sample accelerating voltage. Pressure in a sample chamber during measurements was maintained at about 4.5 × 10⁻⁹ Pa. SCAPS images of secondary ions for ¹H⁺, ²⁸Si⁺, ²⁷Al⁺, ⁴⁰Ca⁺ and ²⁴Mg⁺ were collected after pre-sputtering to obtain stable signals with accumulation time of 300, 50, 25, 50 and 25 s, respectively (Extended Data Fig. 2). Since the SCAPS exhibits a good linear relationship between an output voltage and the number of secondary ions, the abundance of each element can be quantified pixel by pixel from the intensity map. The spatial resolution of quantification is 0.5 μm × 0.5 μm. We employed three silicate glasses with known H₂O concentrations as standards (0.0 to 4.5 wt%)⁷¹ to convert the intensity ratio, ¹H⁺/²⁸Si⁺, into the mass ratio. The intensity ratios from secondary ion images and the mass ratios of H/Si of these standards were linearly correlated (correlation coefficient R² = 0.996)⁶¹. In order to reduce statistical errors, we applied a moving average method to determine the H/Si mass ratio on each region⁶⁸. Water concentration in each region was then obtained by multiplying the H/Si

ratio by the Si content acquired by major element analyses (Fig. 2). The average water contents for each phase were calculated from the SCAPS image and shown in Table 1.

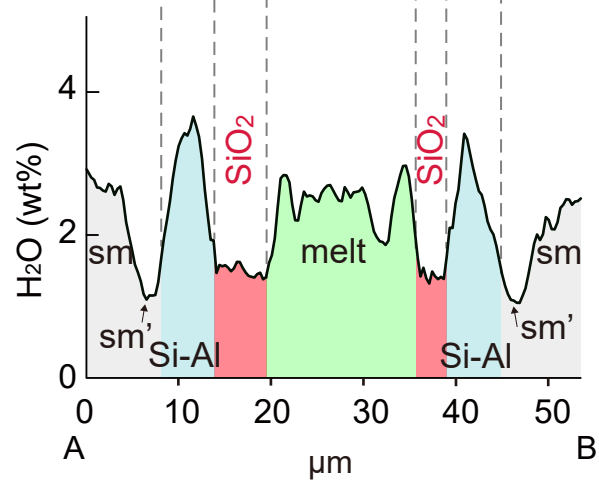
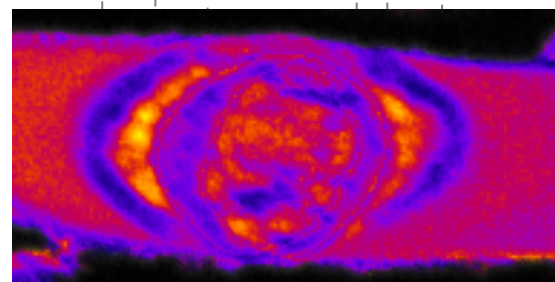
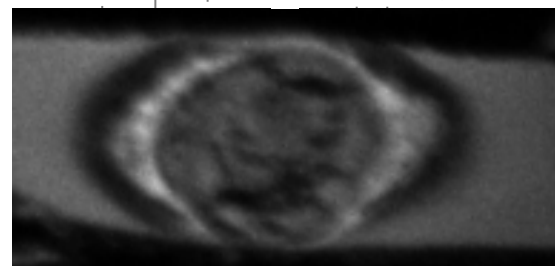
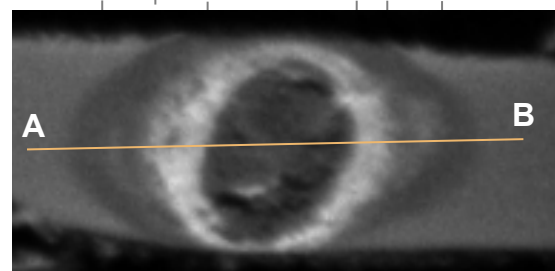
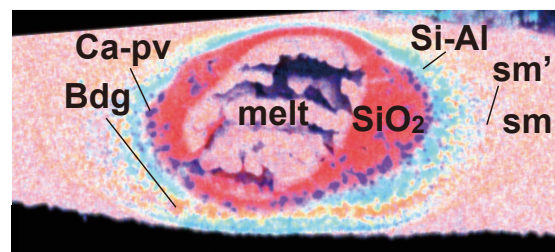
Data availability

All data supporting this study are available at <https://zenodo.org/records/10901809>.

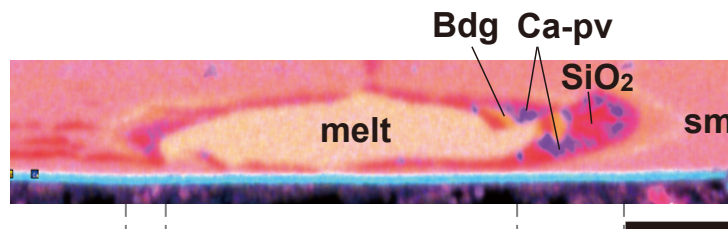
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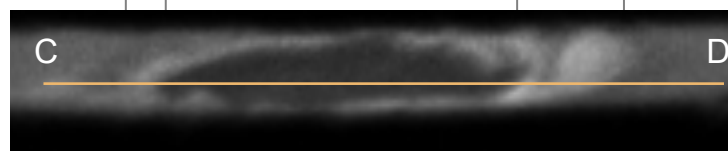
a 58 GPa, 3,250 K



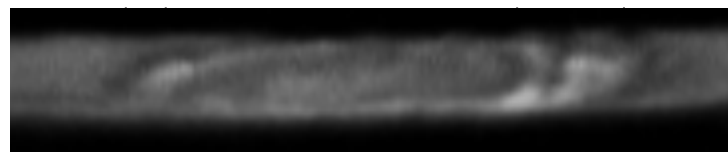
b 121 GPa, 3,450 K



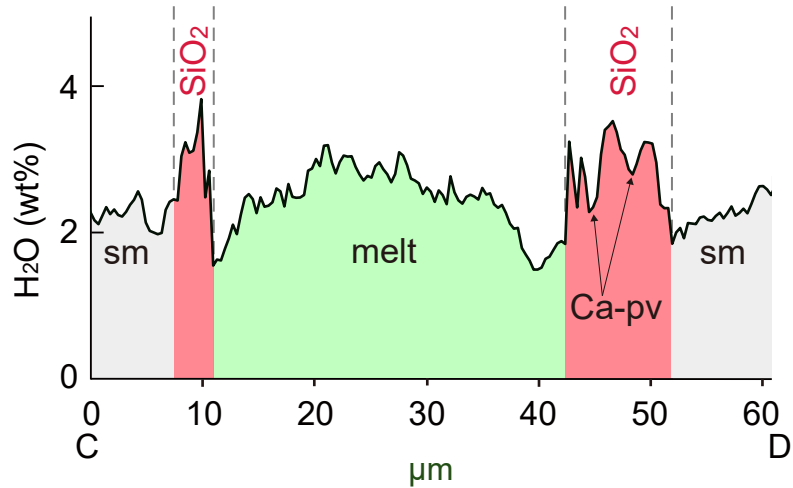
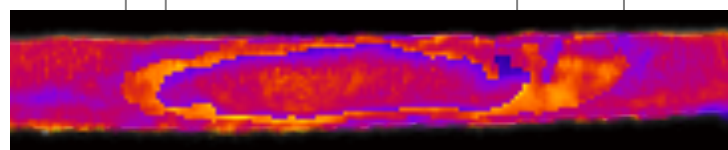
²⁸Si



¹H



H₂O



3.0

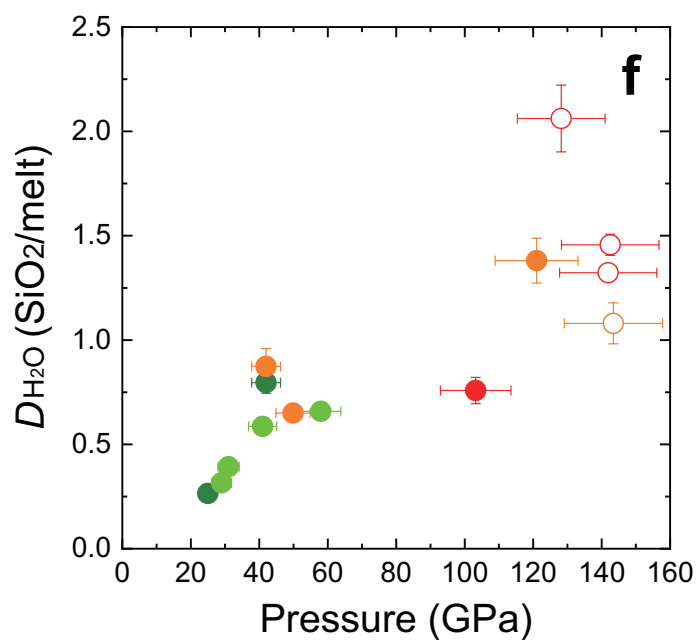
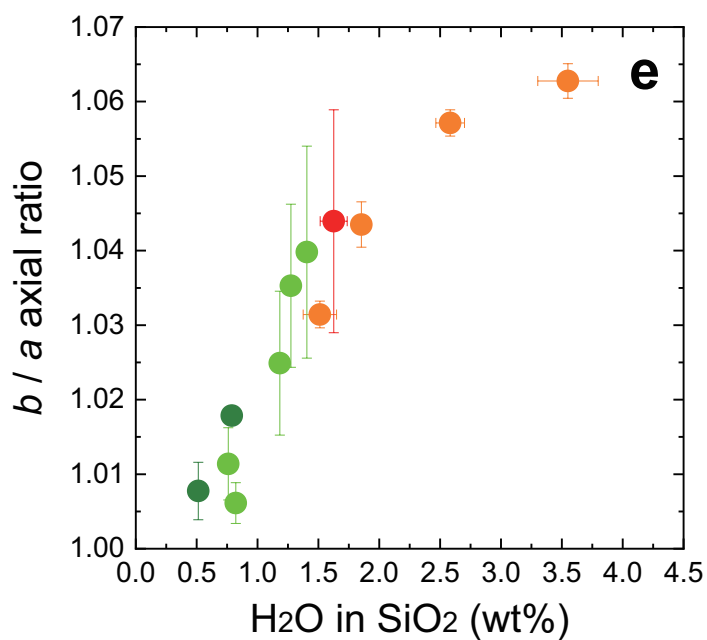
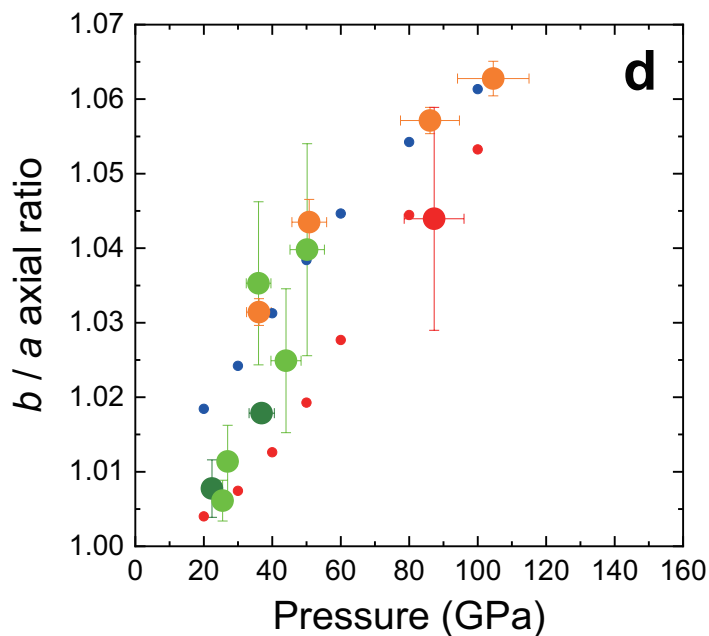
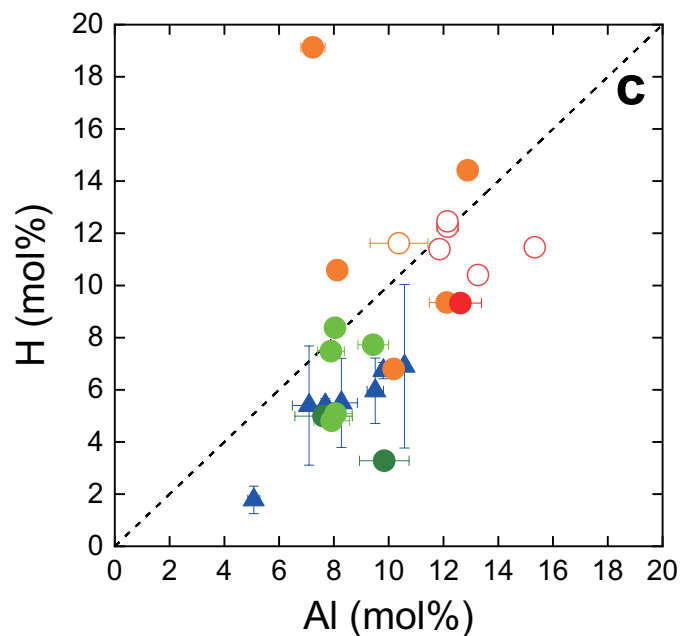
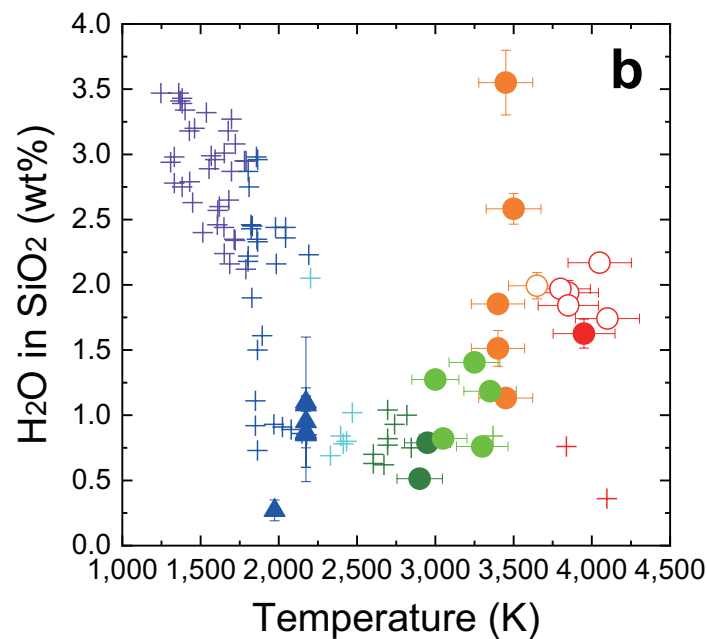
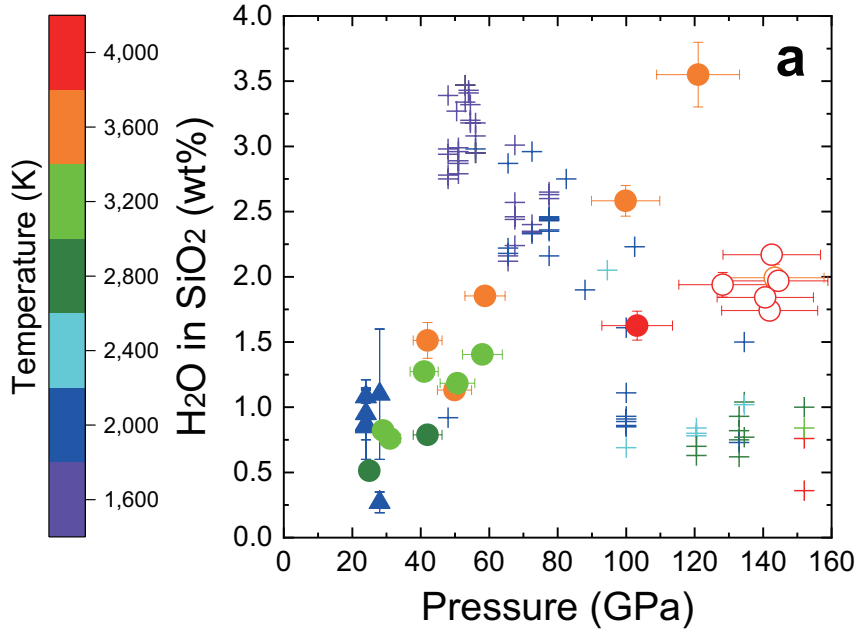
R.I.

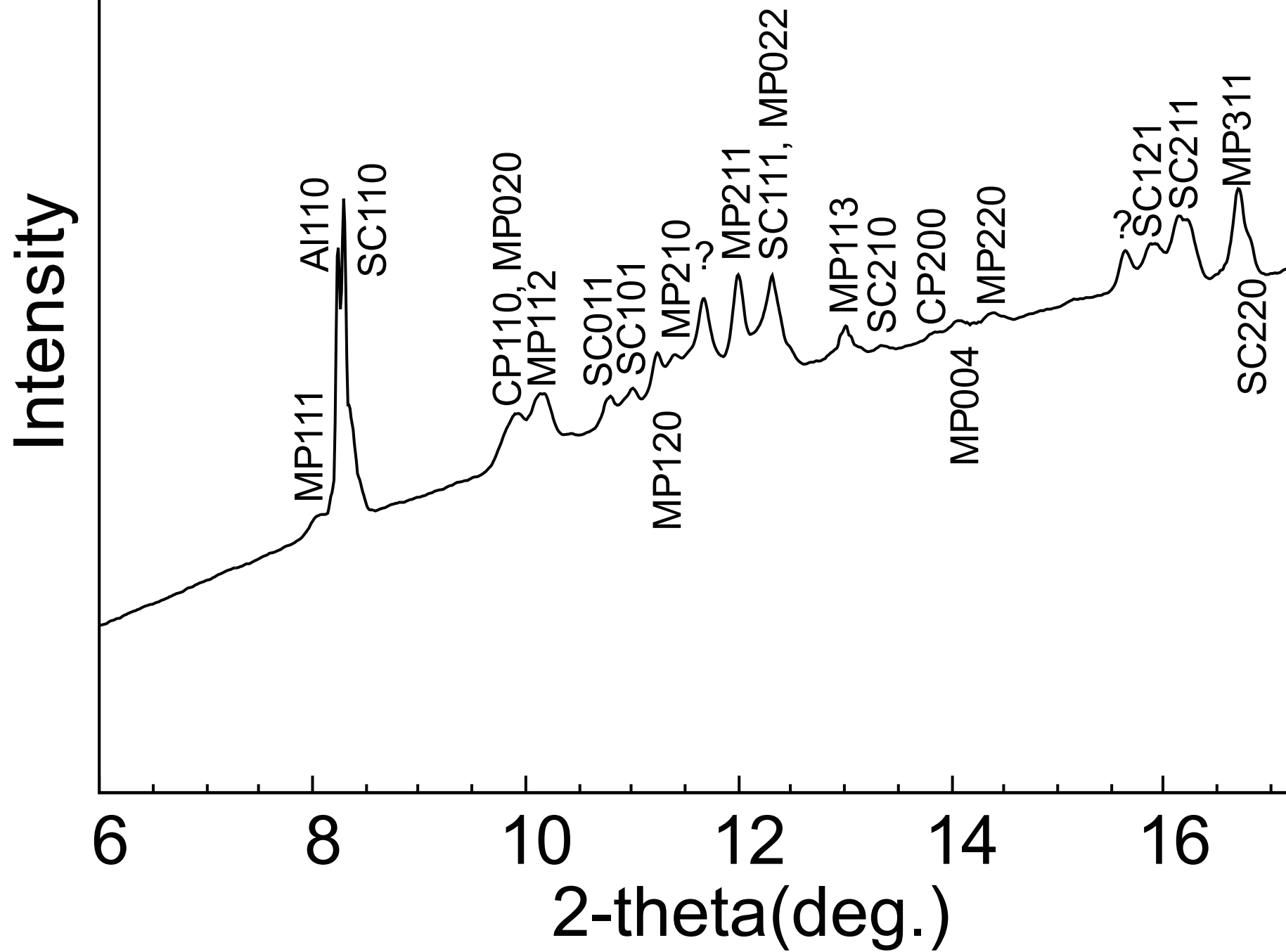
0

4

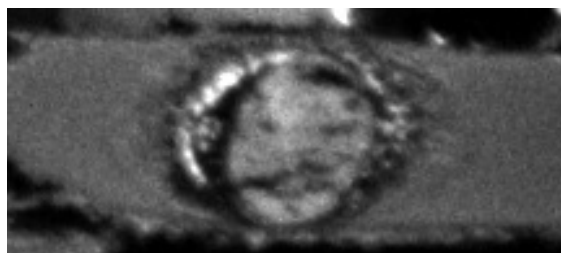
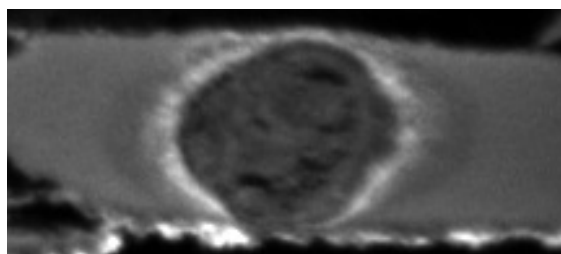
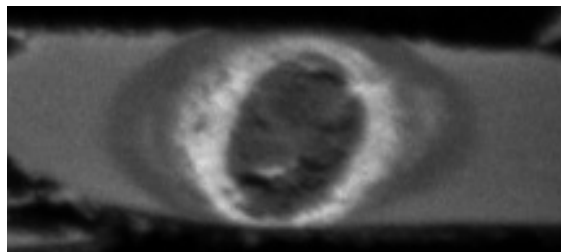
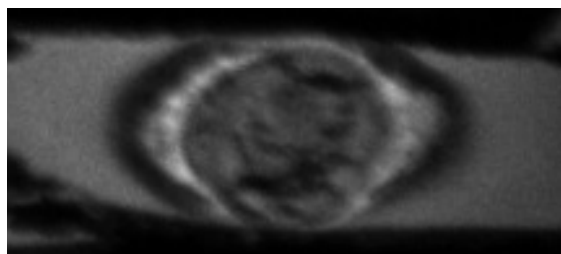
H₂O wt%

0



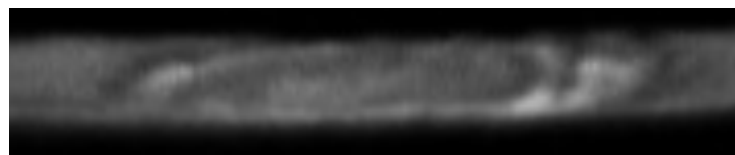


a 58 GPa, 3,250 K

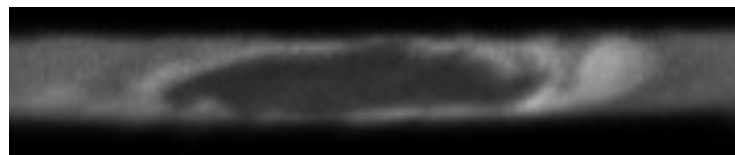


3.0 R. I. 0

b 121 GPa, 3,450 K



1H



28Si



27Al

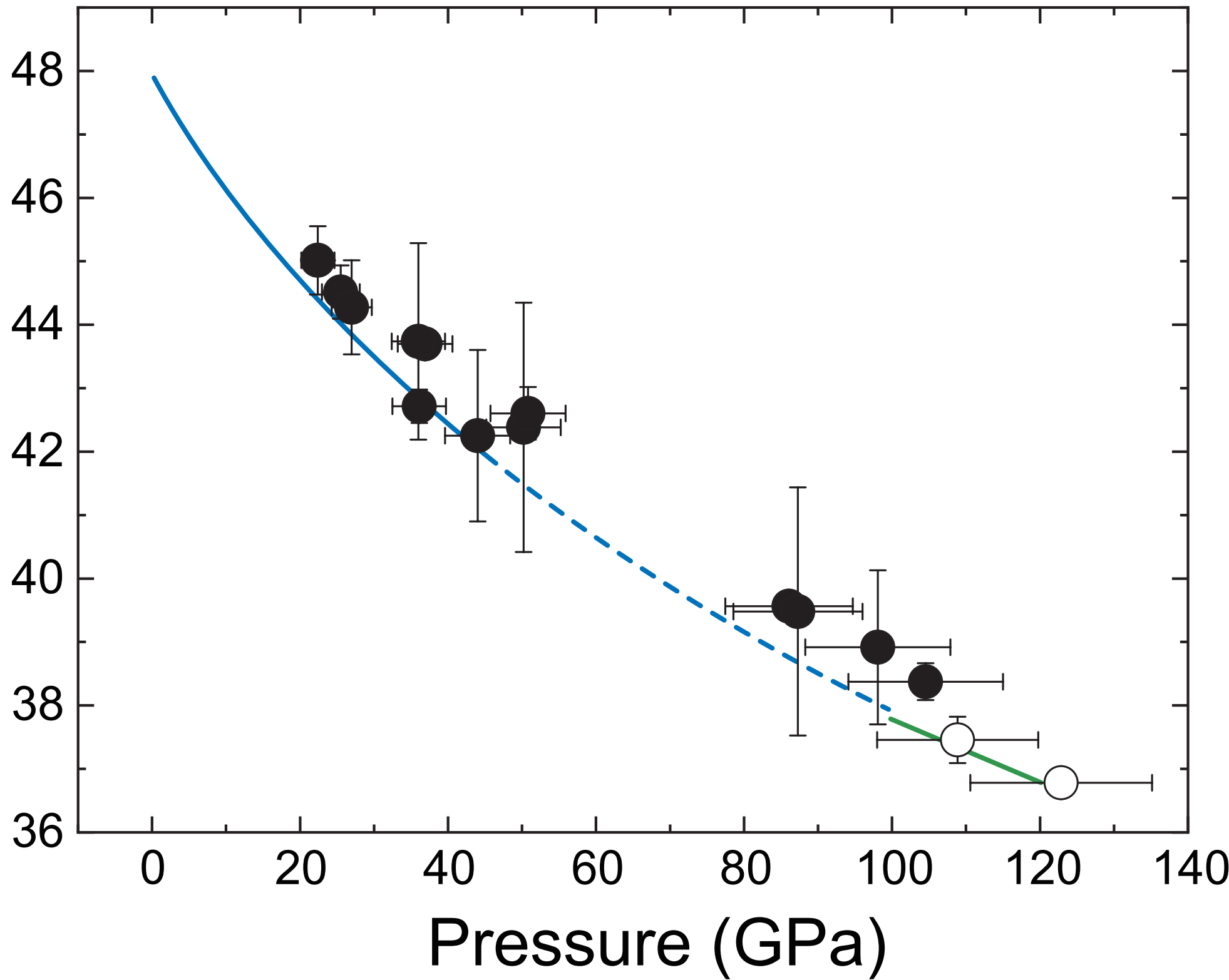


40Ca



24Mg

Volume ($\text{\AA}^3$)



Extended Data Table 1 | Lattice constants and unit-cell volumes of the SiO₂ phases

Run	Pressure (GPa)	<i>a</i> (Å)	<i>b</i> (Å)	<i>c</i> (Å)	Volume (Å ³)
CaCl ₂ -type phase					
#1	22 (2)	4.088 (12)	4.120 (11)	2.673 (4)	45.02 (54)
#2	26 (3)	4.076 (6)	4.101 (10)	2.663 (5)	44.52 (42)
#3	27 (3)	4.056 (14)	4.102 (14)	2.661 (9)	44.28 (74)
#4	36 (4)	3.996 (29)	4.137 (32)	2.646 (16)	43.74 (155)
#5	37 (4)	4.027 (1)	4.098 (1)	2.648 (1)	43.70 (8)
#6	36 (4)	3.959 (3)	4.084 (6)	2.642 (3)	42.71 (26)
#8	44 (4)	3.959 (26)	4.057 (28)	2.631 (14)	42.25 (135)
#9	50 (5)	3.940 (37)	4.097 (41)	2.625 (20)	42.38 (197)
#10	51 (5)	3.948 (7)	4.120 (9)	2.619 (4)	42.60 (42)
#11	86 (9)	3.834 (4)	4.053 (5)	2.546 (2)	39.56 (23)
#12	87 (9)	3.870 (38)	4.040 (42)	2.525 (20)	39.48 (195)
#13	105 (11)	3.784 (5)	4.022 (7)	2.522 (3)	38.38 (29)
α-PbO ₂ -type phase					
#14	109 (11)	3.781 (7)	4.709 (15)	4.208 (7)	74.91 (73)
#15	123 (12)	3.751 (3)	4.694 (2)	4.178 (1)	73.56 (15)

All data were collected at room temperature.